Research Article

XRD and DTA Analysis of Developed Agglomerated Fluxes for Submerged Arc Welding

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A unique study of structural and chemical analysis of crystalline phases in developed agglomerated fluxes was carried out. Thirty-two fluxes were developed by using a mixture of oxides, halides, carbonates, silicates, and ferroalloys for submerged arc welding. The present paper focuses on only ten (out of thirty-two) fluxes which were analyzed by X-ray diffraction (XRD) to know the different types of oxides formed and changed in oxidation number of metallic centers after sintering process at around 850°C. To know the effect of temperature over phase transformation and melting of different compounds, differential thermal analysis (DTA) was carried out from 1000 to 1400°C. This study aims to know the quantity of ions present (percentage) and melting behavior of developed agglomerated fluxes for submerged arc welding process.

1. Introduction

The agglomeration method is used for flux preparation by using a mixture of oxides, halides, carbonates, silicates, and ferroalloys [1]. The chemistry of weld metal is always governed by the electrochemical changes at the weld pool flux interface because the important properties of submerged arc welding fluxes are the function of their physiochemical properties [2–5]. The important chemical reactions take place during heating of ceramic oxide which is the chemical reaction among the different phases [6], phase transformation of minerals [7], formation of crystalline phases [8], and chemical and structural characterization of crystalline phases in agglomerated fluxes for submerged arc welding were observed [9]. Therefore, the developed agglomerated fluxes can be analyzed to know their crystal structure observed during the rise of temperature. The objective of this analysis is to know the crystalline phases as well as to quantify the different types of ions and their distribution in developed agglomerated fluxes (AGF1101–AGF1110) by X-ray diffraction

(XRD) and differential thermal analysis (DTA) is used to know the peaks of endothermic temperatures.

2. Experimentation

In the present study, the fluxes have been designed on the basis of binary and ternary phase diagrams for different oxide and fluoride systems [10] and formulated by using response surface methodology technique [11, 12]. In these fluxes, varying amount of oxides and fluorides were added to know the crystalline phases as well as to quantify the different types of ions and their distribution in developed agglomerated fluxes (AGF1101–AGF1110) by X-ray diffraction (XRD) and differential thermal analysis (DTA) was carried out to know the peaks of endothermic temperatures.

CaO-Al₂O₃-SiO₂-based flux systems have been selected for the study as these are the most widely used fluxes at the commercial level. The details of basic constituents, alloying constituents, and their percentage in the developed agglomerated fluxes are reported in Table 1.

TABLE 1: Constituents and their ranges.

Basic constituents and alloying constituents	CaO	SiO ₂	Al ₂ O ₃	MnO	CaF ₂	MgO	NiO	Fe-Cr
Amount (wt %)	20	35	10	5–10	10–20	5–15	0–10	0–10

TABLE 2: Crystalline phases and crystal structure in developed agglomerated fluxes.

Flux number	Crystalline phase	Crystal structure
AFG1101	CaAl ₂ SiO ₈ (Dmisteinbergite)	Hexagonal
AFG1101	Mg ₇ Fe ₇ Si ₈ O ₂₂ (OH) ₂ (Protoanthrophyllite)	Orthorhombic
AFG1101 AFG1104	CaAl ₂ Si ₆ O ₁₆ · 5(H ₂ O) (Epistilbite)	Triclinic
AFG1101	Mg ₂ Al ₂ Si ₄ O ₁₀ (OH) · 4(H ₂ O) (Palygorskite)	Monoclinic
AFG1101	Ca ₂ Fe ⁺⁺⁺ (Fe ⁺⁺⁺ , Al) ₂ (SiO ₄)(Si ₂ O ₇)(O, OH) ₂ · H ₂ O (Julgoldite)	Monoclinic
AFG1102	CaAl ₂ Si ₂ O ₈ (Svyatoslavite)	Monoclinic
AFG1102	K ₂ CaMg ₂ (Al, Si) ₃₆ O ₇₂ · 28H ₂ O (Mazzite)	Hexagonal
AFG1102 AFG1103 AFG1104	Mg ₂ (Al, Fe ⁺⁺⁺) ₃ Si ₃ AlO ₁₀ (OH) ₈ (Sudoite)	Monoclinic
AFG1102	Mn ₇ Si ₃ O ₁₂ (OH) ₂ (Leucophoenicite)	Monoclinic
AFG1102	Ca ₃ Al ₂ (SiO ₄) _{3-x} (OH) _{4x} (x = 0.2–1.5) (Hibschite)	Isometric
AFG1104	KAl ₂ (Si ₃ Al)O ₁₀ (OH, F) ₂ (Muscovite)	Monoclinic
AFG1104 AFG1107 AFG1108	Ca ₂ (Mn, Fe ⁺⁺)Fe ⁺⁺⁺ Si ₅ O ₁₄ (OH) (Manganbabingtonite)	Triclinic
AFG1103 AFG1110	Ca ₁₀ Mg ₄ Al ₂ Si ₁₁ O ₃₉ · 4(H ₂ O) (Juanite)	Orthorhombic
AFG1105 AFG1109	Ca ₅ Si ₂ O ₇ (CO ₃) ₂ (Tilleyite)	Monoclinic
AFG1104 AFG1108	Mg ₉ (SiO ₄) ₄ (OH, F) ₂ (Hydroxyl clinotumite)	Monoclinic
AFG1108	Ca ₇ (SiO ₄) ₃ (OH) ₂ (Chegemite)	Orthorhombic
AFG1110	Ca ₁₄ (Mn ⁺⁺) ₁₄ Si ₂₄ O ₅₈ (OH) ₈ · 2(H ₂ O) (Truscottite)	Triclinic
AFG1110	CaO · 3(Fe ⁺⁺ , Mg, Fe ⁺⁺⁺) ₃ Si ₄ Al ₄ O ₁₀ (OH) ₂ · 4(H ₂ O) (Ferroaponite)	Monoclinic
AFG1110	Ca ₄ Si ₂ O ₆ CO ₃ (OH, F) ₂ (Fukalite)	Orthorhombic
AFG1105	Mg ₅ (Fe ⁺⁺⁺) ₅ (SiO ₄) ₂ (OH, F) ₂ (Chondrodite)	Monoclinic

TABLE 3: Quantity of oxides in percentage weight for developed agglomerated fluxes.

Serial number	Oxides	Quantity of oxides present (wt.%) in developed agglomerated fluxes									
		AFG1101	AFG1102	AFG1103	AFG1104	AFG1105	AFG1106	AFG1107	AFG1108	AFG1109	AFG1110
1	CaO	8.29	12.48	28.10	5.76	11.47	9.33	4.89	17.21	14.65	24.53
2	SiO ₂	58.80	31.16	42.38	5.71	21.19	38.57	21.79	31.27	12.65	33.12
3	Al ₂ O ₃	10.50	22.94	14.30	18.29	24.18	16.66	11.32	23.39	20.00	3.27
4	MgO	7.80	3.37	8.12	14.47	29.78	6.59	5.78	17.34	18.52	3.37
5	MnO	0.02	14.29	—	1.86	—	—	2.32	1.85	—	21.20
6	NiO	0.02	—	—	—	—	—	—	—	—	—
7	FeO	0.91	0.26	0.44	0.89	—	3.91	0.78	0.63	—	4.20
8	Fe ₂ O ₃	6.47	0.58	0.64	3.37	—	—	3.48	2.79	—	1.73
9	Na ₂ O	0.04	—	—	—	—	—	—	—	—	0.48
10	CO ₂	—	—	—	—	5.41	7.82	6.31	—	13.57	2.15
11	MnO ₂	—	—	—	—	—	16.66	—	—	—	—

- (1) SiO_2
- (2) $\text{CaAl}_2\text{Si}_2\text{O}_8$
- (3) $\text{Mg}_7\text{Fe}_7\text{Si}_8\text{O}_{22}(\text{OH})_2$
- (4) $\text{CaAl}_2\text{Si}_6\text{O}_{16} \cdot 5\text{H}_2\text{O}$
- (5) $\text{Mg}_2\text{Al}_2\text{Si}_4\text{O}_{10}(\text{OH}) \cdot 4\text{H}_2\text{O}$
- (6) $\text{Ca}_2\text{Fe}^{+++}(\text{Fe}^{+++}, \text{Al})_2\text{SiO}_4(\text{Si}_2\text{O}_7(\text{O}, \text{OH})_2 \cdot \text{H}_2\text{O})$

FIGURE 1: XRD graph of AGF1101 flux.

- (1) CaSiO_3
- (2) $\text{Ca}_{10}\text{Mg}_4\text{Al}_2\text{Si}_{11}\text{O}_{39} \cdot 4(\text{H}_2\text{O})$
- (3) $\text{Mg}_2(\text{Al}, \text{Fe}^{+++})_3\text{Si}_3\text{AlO}_{10}(\text{OH})_8$

FIGURE 3: XRD graph of AGF1103 flux.

- (1) $\text{CaAl}_2\text{Si}_2\text{O}_8$
- (2) $\text{K}_2\text{CaMg}_2\text{Al}_{36}\text{Si}_{36}\text{O}_{72} \cdot 28(\text{H}_2\text{O})$
- (3) $\text{Mg}_2(\text{Al}, \text{Fe}^{+++})_3\text{Si}_3\text{AlO}_{10}(\text{OH})_8$
- (4) Fe, Ni
- (5) $\text{Mn}_7(\text{SiO}_4)_3(\text{OH})_2$
- (6) $\text{Ca}_3\text{Al}_2(\text{SiO}_4)_3 \cdot X(\text{OH})_{4X} \text{ (X = 0.2-1.5)}$

FIGURE 2: XRD graph of AGF1102 flux.

- (1) $\text{CaAl}_2\text{Si}_6\text{O}_{16} \cdot 5(\text{H}_2\text{O})$
- (2) $\text{KAl}_2(\text{Si}_3\text{Al})\text{O}_{10}(\text{OH}, \text{F})_2$
- (3) $\text{Ca}_2(\text{Mn}, \text{Fe}^{++})\text{Fe}^{+++}\text{Si}_5\text{O}_{14}(\text{OH})$
- (4) $\text{Mg}_2(\text{Al}, \text{Fe}^{+++})_3\text{Si}_3\text{AlO}_{10}(\text{OH})_8$
- (5) $\text{Mg}_9(\text{SiO}_4)_4(\text{OH}, \text{F})_2$
- (6) Mn_5Si_3

FIGURE 4: XRD graph of AGF1104 flux.

- (1) $\text{Mg}_4\text{Al}_2(\text{OH})_{12}\text{Co}_3 \cdot 4(\text{H}_2\text{O})$
 (2) $\text{Ca}_5\text{Si}_2\text{O}_7(\text{Co}_3)_2$
 (3) Mg_2SiO_4
 (4) $\text{Mg}_9(\text{SiO}_4)_4(\text{OH}, \text{F})_2$
 (5) Al_2O_3

FIGURE 5: XRD graph of AGF1105 flux.

- (1) SiO_2
 (2) SiO_2
 (3) CaCo_3
 (4) $(\text{Mg}, \text{Fe}^{++})_5(\text{SiO}_4)_2(\text{F}, \text{OH})_2$
 (5) Al_2O_3
 (6) MnO_2

FIGURE 6: XRD graph of AGF1106 flux.

TABLE 4: DTA analysis for endothermic temperatures.

Flux	Peaks of endothermic temperature ($^{\circ}\text{C}$)
AGF1101	1239.89
AGF1102	1234.25
AGF1103	1233.58
AGF1104	1240.77
AGF1105	1226.99
AGF1106	1334.85
AGF1107	1244.58
AGF1108	1191.77
AGF1109	1245.84
AGF1110	1248.37

- (1) $\text{MgCo}_3 \cdot 5(\text{H}_2\text{O})$
 (2) $\text{Al}_2\text{O}_3 \cdot (\text{SiO}_2)_{1.3-2} \cdot (\text{H}_2\text{O})_{2.5-3}$
 (3) $\text{Ca}_2(\text{M}, \text{Fe}^{++})\text{Fe}^{+++}\text{Si}_5\text{O}_{14}(\text{OH})$
 (4) Ni_2Fe to Ni_3Fe

FIGURE 7: XRD graph of AGF1107 flux.

After ascertaining the compositions and constituents of fluxes, they were prepared by agglomeration method by taking small batches (2 kg) of weighed quantities of powdered chemicals except binder and thoroughly mixed in a ball miller. The prepared granular flux was dried and baked in an electrically heated furnace for 3 hrs at 850°C . The dried flux was allowed to cool down to room temperature before storing it in a moisture free box. The XRD analysis was carried out by diffractometer with the monochromatic light wavelength $1.540600 \text{ \AA}$ ($\text{Cu K}\alpha$) for all developed agglomerated flux. The DTA was carried out in an Al_2O_3 crucible from 28 to 1400°C at the heating rate of $10^{\circ}\text{C}/\text{min}$. Alumina was used as a standard reference for differential thermal analysis.

- (1) $\text{Al}_2\text{O}_3 \cdot (\text{SiO}_2)_{1.3-2} \cdot (\text{H}_2\text{O})$
- (2) $\text{Ca}_2(\text{Mn}, \text{Fe}^{++})\text{Fe}^{+++}\text{Si}_5\text{O}_{14}(\text{OH})$
- (3) MgAl_2O_4
- (4) Mg_5Si_3
- (5) $\text{Ca}_7(\text{SiO}_4)_3(\text{OH})_2$

FIGURE 8: XRD graph of AGF1108 flux.

- (1) $(\text{Ca}, \text{Mn}^{++})_{14}\text{Si}_{24}\text{O}_{58}(\text{OH})_8 \cdot (\text{H}_2\text{O})$
- (2) $(\text{Mn}^{++})_2\text{SiO}_3(\text{OH})_2 \cdot (\text{H}_2\text{O})$
- (3) $\text{Ca}_4\text{SiO}_2\text{O}_6(\text{CO}_3)(\text{OH}, \text{F})_2$
- (4) $\text{Ca}_{10}\text{Mg}_4\text{Al}_2\text{Si}_{11}\text{O}_{39} \cdot 4(\text{H}_2\text{O})$
- (5) MnO_2

FIGURE 10: XRD graph of AGF1110 flux.

- (1) $\text{Ca}_5\text{Si}_2\text{O}_7(\text{CO}_3)$
- (2) $\text{CaMg}_3(\text{CO}_3)_4$
- (3) $\text{Mg}_9(\text{SiO}_4)(\text{OH}, \text{F})_2$
- (4) Al_2O_3

FIGURE 9: XRD graph of AGF1109 flux.

FIGURE 11: DTA graph of AGF1101 flux.

3. Result and Discussion

3.1. XRD and DTA Analysis of Developed Agglomerated Fluxes.

The XRD results of developed agglomerated fluxes are shown in Figures 1, 2, 3, 4, 5, 6, 7, 8, 9, and 10 and by these graphs, it has been observed that a number of crystalline phases are formed due to chemical interaction among ions with potassium silicate binder. The oxides like MnO, MgO, SiO₂, Al₂O₃, and CaO were found in compound form, which means they do not react with one another. The names of different crystalline phases observed in XRD analysis along

with their crystal structure were given in Table 2. The quantity of different oxides shown in Table 3 was based on XRD peaks with high intensity [13]. The development of different types of silicates and oxides is mainly due to mixing of potassium silicate during the agglomeration process.

The curves of differential thermal analysis (DTA) are shown in Figures 11, 12, 13, 14, 15, 16, 17, 18, 19, and 20 and it shows that most of the oxides were stable oxides up to 1000°C. The ranges of DTA curves were found between 1000

FIGURE 12: DTA graph of AGF1102 flux.

FIGURE 14: DTA graph of AGF1104 flux.

FIGURE 13: DTA graph of AGF1103 flux.

FIGURE 15: DTA graph of AGF1105 flux.

to 1400°C. With a few exceptions, most of the reactions were endothermic in nature and the endothermic peaks were observed at different temperatures for different fluxes given in Table 4. The endothermic peaks were different for different fluxes because of their different chemical compositions. These endothermic peaks were comparable with the melting reaction of different crystalline phases for different fluxes. In some DTA curves, a zigzag behavior was observed due to gassing or blistering of the glass formation [14]. The anions generally present in the fluxes are O^{2-} and F^{1-} . The cations from the flux like Ca^{2+} , Mg^{2+} , Al^{3+} , Fe^{2+} , Fe^{3+} , Si^{4+} , Mn^{2+} , Mn^{3+} , and Mn^{4+} can react with oxygen and tetrahedral unit SiO_4^{4-} from the silicates formed in the fluxes. Since, calcium and magnesium ions have the largest negative heat of formation energy ΔG_f , therefore, they react quickly with oxygen in the welding arch.

There were changes in oxidation number in some cationic constituents like Fe^{2+} , Fe^{3+} , Mn^{2+} , and Mn^{4+} . In most of the fluxes, calcium was present which increases the stability of electric arc during welding. The viscosity of fluxes increases by the presence of corundum and SiO_2 which are present in most of the fluxes. In most of the fluxes, it has been observed that the MnO and SiO_2 are jointly present and increase the thermomechanical behavior of weld joint.

4. Conclusion

The study identified the quantity of different ions present in the weld pool, and it also helps to know the different types of oxides and probable crystalline phases in developed

FIGURE 16: DTA graph of AGF1106 flux.

FIGURE 17: DTA graph of AGF1107 flux.

FIGURE 18: DTA graph of AGF1108 flux.

FIGURE 19: DTA graph of AGF1109 flux.

FIGURE 20: DTA graph of AGF1110 flux.

agglomerated fluxes during submerged arc welding. Such types of studies are very much helpful to know the effect of compositions of fluxes on thermomechanical behavior of weld metal.

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