Synthesis and Characterization of New Organosoluble and Thermally Stable Poly(thioether-amide)s Bearing Pyridine Subunit in the Main Chain

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We report the synthesis of new polyamides containing 2,6-bis(2-thio-2-(4-carboxyphenyl)-1-oxo)pyridine subunit, under microwave irradiation using Yamasaki phosphorylation method. The solubility, thermal behavior, and viscosity of polyamides were evaluated. The structures of polymers have been characterized using IR and $^1$H NMR spectroscopy. These polyamides showed good solubility, viscosity, high thermal stability, and glass transition temperatures. Their viscosities and glass transition temperatures are in the range of 0.63–0.88 and 223–295°C, respectively. Thermal stabilities for 10% weight loss (T10) are 137–173°C and for 50% weight loss (T50) are in the range of 483–523°C. The study of surface morphology showed particle and amorphous structures.

1. Introduction

Aromatic polyamides (aramides) explore high thermal stability, good chemical resistance, excellent mechanical properties, and a series of reliable properties that have broad applications in many areas of research and engineering [1–4]. However, all of these polymers have the main problem of being difficult to process and of fabrication because of their infusibility and poor solubility in common organic solvents. The reasons are strong interchain forces and interactions, inherent macromolecular rigidity, or crystallinity.

There has been an increased attention in the preparation of polyamides with different substituents or structural irregularities to improve their process ability without lowering their other desired properties. These studies include entering flexible adducts into the polymer main chain [5–7], replacing symmetrical aromatic rings by unsymmetrical ones [8–10], introducing bulky and reliable pendent groups to minimize crystallization [11–15], and forming a noncoplanar and non-symmetrical structure [16–18].

Since the first published reports on the use of microwave irradiation to improve chemical transformations by Gedye et al. in 1986 [19], a large number of research papers have been published in this active field, referred to as microwave-assisted organic synthesis (MAOS) [20, 21]. Microwave heating compared to conventional heating procedures indicated that it could reduce reaction times, increase product yields, and enhance product purities by reducing byproducts. The advantages of this efficient technology have also been explored in the context of multistep total synthesis [22], medicinal chemistry, and drug discovery [23]. Also, these methods were used in a series of fields such as polymer synthesis [24], material sciences [25], nanotechnology [26], and biochemical processes [27]. The use of microwave irradiation in polymer chemistry has thus become such a popular technique that a series of chemists will probably use microwave energy to heat chemical reactions on a laboratory scale to prepare a large number of polymers [28]. This efficient method of organic synthesis has been used in a large number of polymer reactions such as step-growth polymerization for the preparation and synthesis of polyamides [29], polyimides [30], poly(amide-imide)s [31], polyesters [32], polyurea, and polythiourea [33]. Also, chain growth polymerization under microwave (MW) irradiation has been applied for the synthesis of a large number of polymers and macromolecules [34].
In this study, new poly(thioether-amide)s containing pyridine [35] thioether-amide subunits in the main chain were synthesized under microwave irradiation and characterized. It has been demonstrated that they are soluble in a series of common organic solvents and showed thermal resistance. Surface morphology of these polymers was studied using scanning electron microscopy (SEM).

2. Experimental

2.1. Materials, Instruments, and Physical Measurements. The reactions for the synthesis of monomer were carried out in an efficient hood. All the materials were purchased from Merck, Fluka, Across Organics, and Aldrich chemical companies. N-Methyl-2-pyrrolidinone (NMP, Merck) and pyridine (Py, Merck) were purified by distillation under reduced pressure over calcium hydride and stored over 4Å© molecular sieves. Triphenyl phosphite (TPP, Merck) was purified by fractional distillation under vacuum. Reagent grade aromatic diamines (Aldrich) including 2,6-pyridine diamine (PYDA, 6), 1,3-phenylene diamine (PHDA, 7), 3,3'-diaminodiphenylsulfone (8), and 4,4'-diaminodiphenylsulfone (9) were recrystallized from ethanol. The melting points (uncorrected) were measured with a Barnstead Electrothermal engineering LTD 9100 apparatus. Elemental analysis was performed by a CHN–O– Rapid Heraeus elemental analyzer. FT-IR spectra were recorded in potassium bromide to obtain diacid (5) in 94% yield and melting point of 223-224°C. IR (KBr) ν cm

2.2. Synthesis of 2,6-Bis (2-Thio-2-(4-phenylcarboxy)-1-oxo) pyridine (PDA, 5). To DMF (30 mL) were added 2,6-bis (2-chloro-1-oxo)pyridine [36] (3, 1 mmol, 0.259 g), K2CO3 (4 mmol, 0.56 g), and thiosalicylic acid (4, 2 mmol, 0.30 g) at room temperature as shown in Scheme 1. The reaction mixture was stirred at room temperature for 24 h. After completion of the reaction (monitored by TLC), water was added and the reaction mixture was neutralized with HCl solution. After standing for 2 h, the resulting precipitate was filtered and recrystallized from ethanol to obtain diacid (5) in 94% yield and melting point of 223-224°C. IR (KBr) ν cm

Analytical calculation for C23H19N3O6S2 (M = 497.54) was C, 55.52%; H, 3.85%; N, 8.45% and found C, 55.58%; H, 3.81%; N, 8.57%.

Scheme 1: Synthesis of monomer (5, PDA).
2.3. **Synthesis of Polyamides.** Polyamides were synthesized through the phosphorylation reaction of 2,6-bis(2-thio-2-(4-phenylcarboxy)-1-oxo)pyridine (5, PDA) with various diamines as shown in Scheme 2. A typical example for the preparation of polyamides is given. A mixture of 5 (1 mmol, 0.498 g), 1,3-phenylenediamine (1 mmol, 0.11 g), 0.3 g of CaCl$_2$, 0.6 mL of TPP, 0.5 mL of pyridine, and 4 mL of NMP was added to the quartz tube and irradiated under microwave conditions in 600W for 9 Min. (3 × 3 Min.) and for the rest time of 10 Min. (2 × 5 Min.). After cooling to
room temperature, the resulting viscose reaction mixture was poured into 300 mL of boiling methanol. The resulting crude product was precipitated and then filtered. The resulting polymer was washed with hot methanol (50 mL), hot water (twice, 50 mL), and then hot methanol (50 mL), respectively, and dried under vacuum at 100°C overnight. The yields were almost quantitative. Spectral data, thermal properties, viscosity, and solubility of these polyamides were presented in the tables and figures.

3. Results and Discussion

In this research work, we wish to report the synthesis and characterization of polyamides, which were obtained from the reaction of a new monomer (5, PDA), containing 2,6-bis(2-thio-2-(4-phenylcarboxy)-1-oxo)pyridine subunit, and aromatic diamines under microwave irradiation. Dichloroamide (3) was readily prepared by using a published method, with some modifications, from the reaction of 1 and chloroacetyl chloride [35]. Diacid monomer (5) was prepared by the reaction of 3 and thiosalicylic acid (4) in DMF at room temperature. The IR spectrum of diacid (5, PDA) is presented in Figure 1. The polymerizations were carried out via Yamazaki phosphorylation reaction, by the reaction of diacid (5) and different diamines in the presence of triphénylphosphite (TPP), pyridine (Py), N-methylpyrrolidinone (NMP), and calcium chloride (CaCl₂) under microwave irradiation almost in quantitative yields (Table 1). Polymers were precipitated in boiling methanol and washed with hot water and methanol, respectively. The inherent viscosities and the yields of polyamides were revealed in Table 1. The viscosities of polyamides were measured in DMSO at 30°C and are in the range of 0.63–0.88 (Table 1). The structures of polymers were fully characterized by their \(^1\)H NMR and IR spectra (Table 2). The IR spectrum of polyamide 13 (PPS) appears in Figure 2 and shows good agreement between absorptions and the corresponding structure (Table 2).

The thermal stability of polyamides was studied by thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) methods in a wide range of temperature. The entire polymers showed good thermal stability; the results appear in Table 3 and curves are displayed in Figure 3. Table 3. According to the effects of diamine structure on the thermal behaviour, the insertion of a sulfone group into the diamine structure decreases the overall flexibility of the polymer chains and increased the \(T_g\) value. The polymers containing diamines with no flexible groups (such as 1,4- and 1,3-phenylenediamine and 2,6-pyridinediamine derivatives) have high \(T_g\) possibly due to a less flexible polymer backbone.

The thermal stability was measured by thermogravimetric analysis (TGA) and shown that polymers are well thermally stable, their decomposition at argon atmosphere for temperature of 10% weight loss \(T_{10}\) was in the range of 137–173°C, and the temperature of 50% weight loss \(T_{50}\) was in the range of 483–523°C. The polymers decomposition temperatures for various percent of decomposition and char yield at 550°C for all polymers are presented in Table 3. According to the data obtained by thermal stability measurements, the incorporation of sulfone units into the polymers backbone enhanced the thermal stability.

The solubility behavior of polyamides is investigated qualitatively in a series of organic solvents such as N-methylpyrrolidinone (NMP), N,N-dimethylformamide (DMF), tetrahydrofuran (THF), dimethyl sulfoxide (DMSO), N,N-dimethylacetamide (DMAc), and m-cresol, and the results are summarized in Table 4. All the polymers showed reliable solubility in polar organic solvents. This might be due to the presence of aromatic and methylene subunits and the decrease of intermolecular and intramolecular hydrogen bonding. Amine, amide, and thioether units increase the intermolecular and intramolecular hydrogen bonding and
then increase the crystallinity and close packing. On the other hand, sulfone units in the aromatic diamines reduce the flexibility and increased close packing and crystallinity. According to the above discussion, therefore, the solubility of polyamides was affected by the several variables and, significantly, the structural variations in diacid monomer (4) and aromatic diamine components can be considered.

The surface morphology of polymers has been studied by scanning electron microscopy using their SEM images (Figures 4, 5, 6, and 7). According to these images, 10 and 13 showed particle structure; also 11 and 12 showed amorphous structures.

### 4. Conclusion

In summary, we have synthesized and characterized a new set of polyamides based on the pyridine and thiosalicylic acid subunits. They were successfully obtained through the direct polycondensation reaction of 2,6-bis(2-thio-2-(4-phenylcarboxy)-1-oxo)pyridine with various diamines via Yamazaki method under microwave irradiation (MW). The solubility and thermal stability polymers are high. This may be due to the presence of pyridine and methylene groups in the presence of the tetrahedral sulfide functional group. The presence of sulfone group in the diamine subunit increased

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### Table 2: Spectral data of polymers.

<table>
<thead>
<tr>
<th>Polymer</th>
<th>IR (cm⁻¹)</th>
<th>¹H NMR (500 MHz, DMSO-d₆): 𝛿 (ppm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>PPY (10)</td>
<td>3358, 3072, 2897, 1698, 1661, 1445, 1149, and 733</td>
<td>4.69 (s, 4H), 7.41 (d, J = 9 Hz, 2H), 7.54–7.75 (m, 3H), 7.68–7.72 (m, 3H), 7.93 (d, J = 8.5 Hz, 2H), 7.99 (d, J = 7.5 Hz, 2H), 8.30 (d, J = 8.5 Hz, 2H), 10.42 (s, 2H), and 10.44 (s, 2H)</td>
</tr>
<tr>
<td>PPH (11)</td>
<td>3376, 3145, 2983, 2491, 1683, and 1115</td>
<td>4.37 (s, 4H), 5.98 (m, 1H), 6.15 (m, 2H), 6.97 (m, 1H), 7.28–7.33 (m, 2H), 7.56–7.62 (m, 4H), 7.78–7.82 (m, 3H), 9.95 (dd, J = 1, 8 Hz, 2H), 10.38 (s, 2H), and 10.39 (s, 2H)</td>
</tr>
<tr>
<td>PMS (12)</td>
<td>3481, 3363, 3224, 2893, 2431, 1698, 1667, 1622, 1595, 1137, and 1045</td>
<td>3.99 (s, 4H), 6.85–6.87 (m, 2H), 7.08–7.09 (m, 2H), 7.18–7.20 (m, 2H), 7.28–7.31 (m, 2H), 7.61–7.63 (m, 4H), 7.79–7.81 (m, 3H), 9.93 (dd, J = 1, 8 Hz, 2H), 10.37 (s, 2H), and 10.38 (s, 2H)</td>
</tr>
<tr>
<td>PPS (13)</td>
<td>3366, 3281, 2985, 2433, 1663, 1593, 1138, and 1043</td>
<td>3.99 (s, 4H), 7.08 (dd, J = 2.5, 7.5 Hz, 2H), 7.41–7.43 (m, 3H), 7.45–7.47 (m, 2H), 7.68 (d, J = 9 Hz, 2H), 7.73 (d, J = 7.5 Hz, 2H), 7.78–7.79 (m, 2H), 7.95–7.98 (m, 2H), 8.86 (s, 1H), 8.92 (s, 1H), 10.92 (s, 2H), and 10.93 (s, 2H)</td>
</tr>
</tbody>
</table>

### Table 3: Thermal properties of polyamides.

<table>
<thead>
<tr>
<th>Polymer</th>
<th>T_d (°C)</th>
<th>T_10 (°C)</th>
<th>T_50 (°C)</th>
<th>Char yields (%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>PPY (10)</td>
<td>234</td>
<td>144</td>
<td>495</td>
<td>38</td>
</tr>
<tr>
<td>PPH (11)</td>
<td>223</td>
<td>137</td>
<td>483</td>
<td>32</td>
</tr>
<tr>
<td>PMS (12)</td>
<td>268</td>
<td>173</td>
<td>512</td>
<td>43</td>
</tr>
<tr>
<td>PPS (13)</td>
<td>295</td>
<td>159</td>
<td>523</td>
<td>47</td>
</tr>
</tbody>
</table>

a: Temperature of 10% weight loss determined in argon atmosphere.

b: Temperature of 50% weight loss determined in argon atmosphere.

Char yield calculated as the percentage of solid residue after heating from room temperature to 550°C under argon.

Table 4: The solubility of polyamides.

<table>
<thead>
<tr>
<th>Polymer</th>
<th>NMP</th>
<th>DMAc</th>
<th>DMF</th>
<th>DMSO</th>
<th>m-Cresol</th>
<th>THF</th>
</tr>
</thead>
<tbody>
<tr>
<td>PPY (10)</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>±</td>
<td>+</td>
</tr>
<tr>
<td>PPH (11)</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>+</td>
<td>+</td>
</tr>
<tr>
<td>PMS (12)</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>PPS (13)</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>++</td>
<td>–</td>
<td>–</td>
</tr>
</tbody>
</table>

(++): soluble at room temperature; (+): soluble upon heating; (±): partially soluble.

Solubility measured at a polymer concentration of 0.05 g/mL.
glass transition temperature, and, as a result, polymers with higher symmetric and rigid structures in diamine subunit and, as a result in overall, polymer backbone showed the highest thermal stability. Thus, we afforded polyamides with improved solubility and high heat resistance. The study of surface morphology of polyamides showed particle and amorphous structures.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

References


