

Research Article

Synthesis and Characterization of Vanadium Doped Zinc Oxide Thick Film for Chemical Sensor Application

Rayees Ahmad Zargar,¹ Manju Arora,² Masroor Ahmad,³ and Aurangzeb Khurram Hafiz¹

¹Department of Physics, Jamia Millia Islamia, New Delhi 110025, India

²CSIR-National Physical Laboratory, Dr. K.S. Krishnan Marg, New Delhi 110012, India

³Superconductivity Research Laboratory, Department of Physics, Barkatullah University, Bhopal 462026, India

Correspondence should be addressed to Manju Arora; marora@nplindia.org

Received 14 June 2015; Revised 22 September 2015; Accepted 7 October 2015

Academic Editor: Achim Trampert

Copyright © 2015 Rayees Ahmad Zargar et al. This is an open access article distributed under the Creative Commons Attribution License, which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly cited.

Zinc oxide and vanadium pentoxide nanoparticles derived by chemical coprecipitation route were used to cast $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film by screen printing method. The structural, morphological, optical, and electrical properties of the film were characterized by powder XRD, SEM, Raman, UV-VIS, and DC conductivity techniques. XRD pattern, SEM image, and Raman spectrum of the film confirm the single phase formation of Wurtzite structure with preferential orientation along [101] plane, minor variation in lattice parameters, and vanadium ions substitution at zinc sites. $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ pellet has been used for sensing ammonia vapor concentrations in 20–50°C temperature range which exhibits maximum responsiveness and sensitivity at 30°C. The minor variations in resistance are observed with ammonia vapor concentration. The adsorption of ammonia vapors through weak hydrogen bonding and its insertion into lattice by nitrogen lone pairs donation at vacant/defect sites in lattice caused by vanadium doping are considered to explain gas sensing mechanism.

1. Introduction

The metal oxide semiconductors have been extensively explored by researchers in pure and doped form. Doping introduces remarkable changes in the structural, optical, electrical, magnetic, and semiconducting properties which encourages their use in various applications, namely, sensors and piezoelectric, photovoltaic, electrooptic, and microelectromechanical devices [1–5]. Zinc oxide (ZnO) is a multifunctional material with wide band gap (3.37 eV), n-type hexagonal structured, high electron mobility, large piezoelectric constants, high nonlinear optical coefficients, radiation hardness, biocompatibility, and large 60 meV exciton binding energy characteristics. It is suitable for small wavelength optoelectronic devices. The stability of ZnO film has been found to be an alternative substitute to tin oxide and indium tin oxide films whose electrical and optical properties degrade at high temperatures. Nowadays pure and mixed oxide films of CuO, Al_2O_3 , CdO, NiO, and SnO_2 are used in various photonic devices by optimizing their band gap and activation energy.

The doping of ZnO with vanadium caused marked changes in electrical and magnetic properties [6, 7]. Single phase vanadium doped ZnO belongs to the diluted magnetic semiconductor (DMS) category. The ferromagnetism in V doped ZnO was predicted both theoretically and experimentally [8, 9]. The enhancement in electrical properties of V doped ZnO films has been reported by Jin et al. [10] due to increased electron concentration resulting from the electric charge compensation by the substitution of vanadium ions at Zn^{2+} ions site in ZnO. In addition to the better ferromagnetic and electrical properties, V doped ZnO also exhibits ferroelectricity at and above ambient temperature [6, 7]. The R&D work on vanadium doped zinc oxide ($\text{Zn}_{1-x}\text{V}_x\text{O}$) is still going on to explore its utilization in more promising applications in the field of optoelectronics, sensors, and ferroelectric memory devices.

Doped ZnO nanoparticles were synthesized by different physical and chemical methods such as spray pyrolysis, RF sputtering, electrochemical deposition, chemical vapor deposition, chemical coprecipitation, and sol-gel [11–22]. Out

of these, the simple and economic chemical coprecipitation method has been adopted for the synthesis of zinc oxide and vanadium pentoxide nanoparticles from zinc acetate dihydrate and ammonium metavanadate reactants. The thick films were cast from prepared ZnO and V_2O_5 nanoparticles in 96 wt.% : 4 wt.% ratio, respectively, to cast $Zn_{0.96}V_{0.04}O$ thick films by screen printing method, and characterized for their structural, morphological, and optical properties. This film is tested for ammonia vapor sensor application at different operating temperature and ammonia concentration. The ammonia vapor sensing has been chosen because it is most extensively used in the chemical, textile, and pharmaceutical industries worldwide. Its toxic vapor causes many health issues and even death on long exposure.

2. Experimental Measurements

2.1. Materials Used. The as-obtained Sigma Aldrich zinc acetate dihydrate, ammonium metavanadate, and sodium hydroxide analytical reagent (AR) grade chemicals without any further purification are used for synthesis of nanoparticles (NPs). AR grade anhydrous zinc chloride and ethylene glycol, Merck, India, Make, were used as adhesive and binding agents in screen printing of thick film. Analytical reagent (LR) grade ammonia solution of Merck, India, was used for ammonia vapor sensing.

2.2. Synthesis of Vanadium Doped Zinc Oxide Thick Films. First, ZnO and V_2O_5 nanoparticles were prepared by chemical coprecipitation route by using zinc acetate dihydrate and ammonium metavanadate as reactants. 1.00 M NaOH was added to 0.5 M zinc acetate dihydrate and 0.5 M ammonium metavanadate aqueous solutions separately for precipitating zinc oxyhydroxide and vanadyl oxyhydroxide. These precipitated nanoparticles were filtered and dried at 70°C and then sintered at 450°C to obtain to oxides of respective nanoparticles. Thick film paste was prepared by thoroughly mixing 4 wt.% V_2O_5 NPs with 96 wt.% ZnO NPs to produce $Zn_{0.96}V_{0.04}O$ composition with anhydrous $ZnCl_2$ as an adhesive agent and ethylene glycol as a binder. The prepared paste was used for screen printing on pre-cleaned glass substrates. The details of film casting procedure are described in [19, 20] and setup schematic is shown in Figure 1. The as-deposited films were dried at 60°C for two hours and then further annealed in a muffle furnace in air at 550°C for ten minutes for proper adherence, stability, and decomposition of organic and inorganic compounds for achieving the desired stoichiometry of vanadium doped zinc oxide film. The thickness of the film 5 μm was measured by employing a profilometer (SurfTest SJ-301).

2.3. Characterization Techniques Used. X-ray diffraction pattern was recorded on advanced Rigaku diffractometer in 2θ range of 10°–80° using $Cu-K\alpha$ X-ray radiation source. The surface morphological information was derived by using scanning electron microscope (SEM, Leo-440, UK) for recording micrographs. Perkin Elmer Fourier Transform

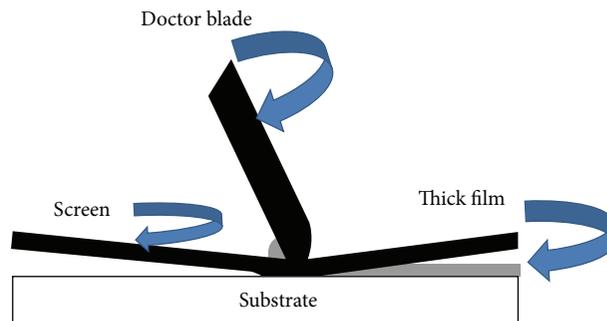


FIGURE 1: Thick film casting by screen printing setup schematic.

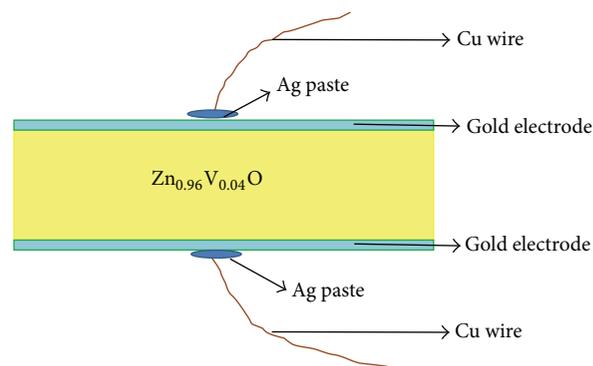


FIGURE 2: Schematic of $Zn_{0.96}V_{0.04}O$ pellet based sensing element.

Raman spectrometer was used for recording Raman spectrum of the as-deposited film in 3500–100 cm^{-1} region at room temperature. The optical transmission spectrum was measured on Hitachi Make UV-VIS spectrometer-3900 in the 350–650 nm range. DC resistivity measurement was done by using standard four-probe technique.

2.4. Ammonia Vapor Sensor Setup. For ammonia vapor detection, the sensing element is shown in Figure 2 which consists of 5 mm \times 3 mm \times 1 mm $Zn_{0.96}V_{0.04}O$ pellet with gold evaporated electrodes on the edges. For electrical connection, copper wires were pasted by silver paste. It acts like a parallel plate condenser as shown in the following schematic diagram.

A micro pipette was used to inject ammonia (50, 75, 100, 125, and 150 μL) or (45, 67.5, 90, 112.5, and 135 ppm) into a sealed glass chamber with a syringe through the inlet. The resistance of the film was measured before and after being exposed to ammonia vapors using Keithley 2000 multimeter. The series of experiments were carried out to check repeatability of results. Further, similar measurements were done at different temperatures using a constant temperature water bath to check the responsiveness of sensor at different temperatures from 20°C to 50°C.

3. Results and Discussion

In doped nanocrystalline materials, the variation in electrical, electronic, optical, and magnetic properties generally resulted from modification in composition and lattice network by

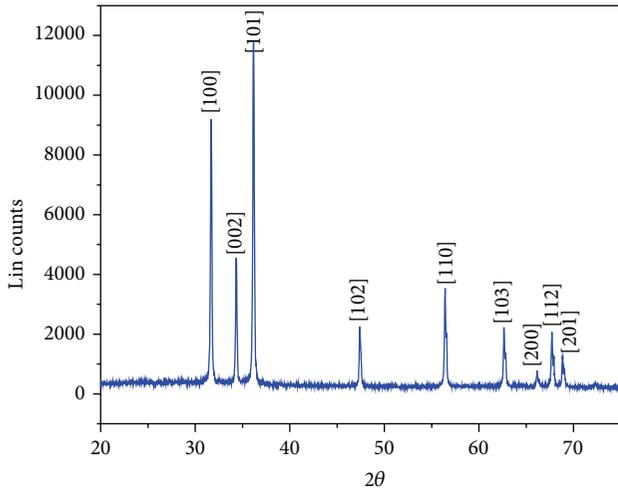


FIGURE 3: XRD pattern of $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film.

dopant atoms/molecules in the host crystal lattice. The formation vanadium doped ZnO thick film was confirmed from powder X-ray diffraction (XRD) pattern as shown in Figure 3 with peaks plane indexing. This shows the polycrystalline nature of film with preferential orientation along (101) plane, wurtzite crystal structure, and lattice parameters $a = 3.2601 \text{ \AA}$ and $c = 5.2113 \text{ \AA}$ (JCPDS number 36-1451). As there is no other crystalline phase seen in the XRD pattern, hence this indicates the high purity of the as-synthesized nanoparticles. The lattice parameter calculations [21, 22] reveal that c lattice parameter value of V: ZnO film increases linearly from 5.18 (for pure ZnO) to 5.21 \AA (for 4 wt.% V doped ZnO thick film). This increase in the c parameter value confirms the substitution of bigger $\text{V}^{2+}/\text{V}^{5+}$ ions (ionic radius 0.93 $\text{\AA}/0.88 \text{ \AA}$) in the hexagonal wurtzite ZnO structure at smaller Zn ions (ionic radius 0.74 \AA). This may lead to an increase in the lattice parameter values. The average crystallites size is $\sim 30 \text{ nm}$ and lattice strain: 0.0005.

SEM image recorded at 20 kV energy and $\times 33,000$ magnification exhibits the uniform deposition of films uniformly distributed large sized grains consisting of hexagonal shaped tips as marked in image, agglomeration of particles of rectangular to hexagon shaped ones. At some regions the fusion of agglomerated tubes is also observed. The recorded micrograph is presented in Figure 4.

This micrograph showed that this small concentration of V^{2+} ions doping in ZnO lattice causes growth of large sized grains which results in porous surface. The bigger ionic radii of vanadium ions distort crystal structure due to ionic radii mismatch which increases grains/particles growth activity of ZnO and formation of bigger particles [23].

ZnO belongs to the space group C_{6v}^4 with two molecules per unit cell in which A_1 and E_1 branches have both Raman and infrared active modes and E_2 branches are only Raman active, whereas B_1 branches have both Raman and infrared inactive (silent) modes. The nonpolar E_2 phonon modes have two frequencies: E_2 (high) is associated with the vibration of oxygen atoms and E_2 (low) is associated with the Zn

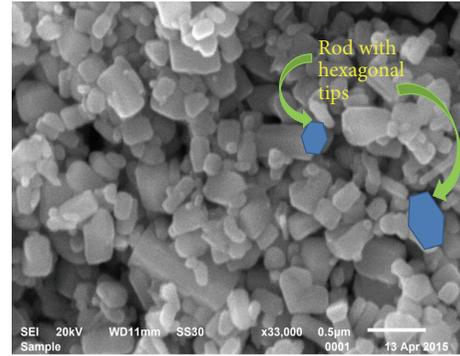


FIGURE 4: SEM micrograph of $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film.

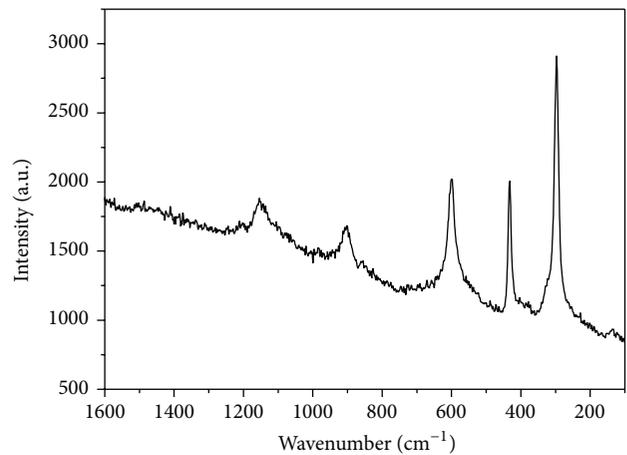


FIGURE 5: Raman spectrum of $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film in 1600–100 cm^{-1} region.

sublattice. Raman spectrum in Figure 5 of V doped ZnO thick film showed peaks at 296, 435, 578, 908, and 1156 cm^{-1} . In low wavenumber range, a very strong peak observed at 296 cm^{-1} is attributed to V-O lattice vibration mode [24], while the peak at 435 cm^{-1} is attributed to ZnO E_2 (high) phonon mode. The peaks at 578 and 1156 cm^{-1} are assigned as $E_1(\text{LO})$ and $2 E_1(\text{LO})$ modes, respectively. The medium intensity peak at 908 cm^{-1} in 1000–800 cm^{-1} region arises from the Zn-O-V lattice vibration. The deviation in observed peak positions as compared to pure ZnO has been explained in terms of the difference in masses and ionic radii of the V and Zn ions in lattice. The presence of strong E_2 mode and weak $E_1(\text{LO})$ mode confirms good crystal quality with less structural defects and impurities.

Figure 6 transmittance spectrum of $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ film measured in 200–700 nm region shows that films are highly visible with 81% transmission value.

The energy band gap value is reflected in the inset of Figure 6 which comes out to be 3.21 eV, that is, less than pure ZnO (3.37 eV). The band gap energy of V_2O_5 is 2.3 eV and ZnO is 3.37 eV; when V_2O_5 is doped in ZnO, alloy is formed which causes decrease in band gap. So the band gap

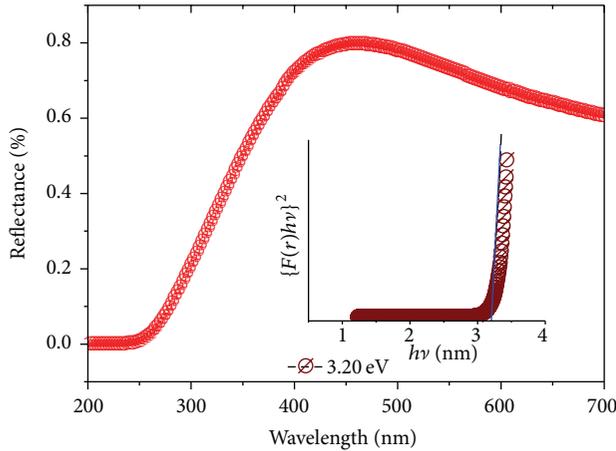


FIGURE 6: Optical reflection spectrum of $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film in 200–700 nm region.

narrowing in vanadium doping in ZnO lattice is an outcome of the alloying effect between V_2O_5 and ZnO.

DC electrical conductivity measurements have been carried out in the temperature range 300–400 K. The electrical resistivity (ρ) was calculated by using [25]

$$\rho = \frac{\pi t}{\ln 2} \left(\frac{V}{I} \right), \quad (1)$$

where ρ is the resistivity ($\Omega\text{-cm}$), t is the sample thickness (cm), V is the applied voltage, and I is the source current (A). The temperature dependency of the DC resistivity can be shown by the well-known Arrhenius equation [26].

Micheletti and Mark [27] estimated the intergranular barrier height (ϕ_b) from the following equation:

$$\mu = \mu_0 \exp\left(-\frac{\phi_b}{KT}\right), \quad (2)$$

where all the terms have their usual meanings. The value of (ϕ_b) obtained from plot Figure 7 $\log(\rho)$ versus $1/T$ is 0.65 eV. Charge carrier mobility μ is determined from the relation given in

$$\mu = \frac{\sigma}{ne}, \quad (3)$$

where n is electron density and σ is the conductivity.

It implies the semiconducting nature of the sample because conductivity increases with increase in operating temperature due to increase of electron carriers.

3.1. Ammonia Vapor Sensing Mechanism. Ammonia vapor sensing is considered as a surface adsorption phenomenon of gas-solid interactions and the electrical conductivity at ambient temperature. In the adsorption process, ammonia vapors are chemically adsorbed on the active sites which gradually form the stable hydrogen bond stabilized chemisorbed NH_3 group on the surface. When this first layer is formed then physical adsorption of ammonia vapors takes place as shown in Figure 8. The chemical absorption of

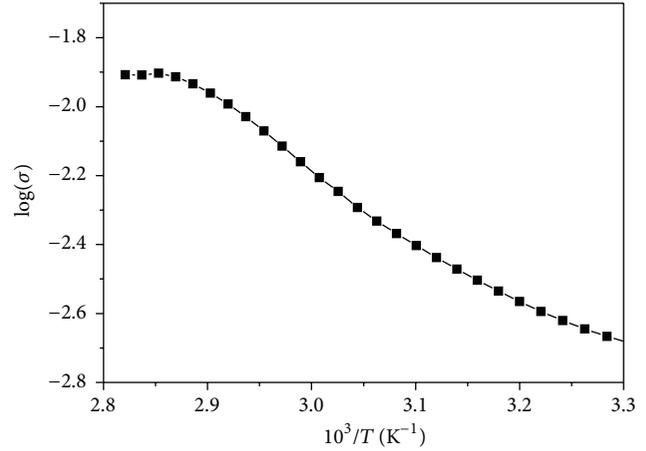


FIGURE 7: $1000/T$ versus $\log(\sigma)$ of $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film.

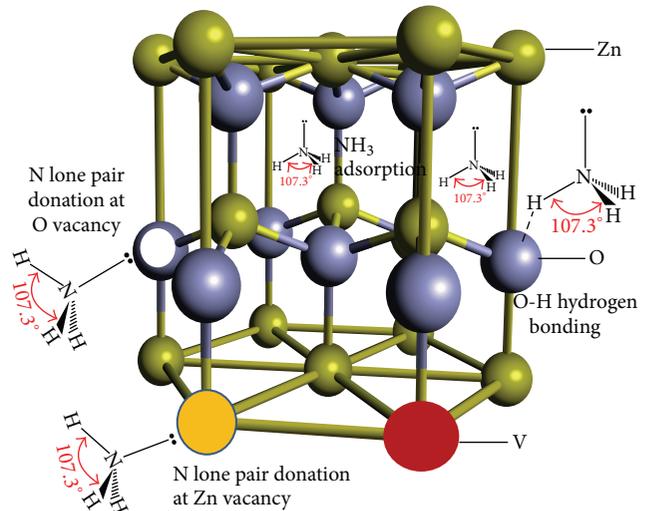


FIGURE 8: Chemisorption and adsorption of ammonia vapors on $\text{Zn}_{0.96}\text{V}_{0.04}\text{O}$ thick film.

ammonia vapors takes place between ZnO (O) and ammonia (H) via weak hydrogen and nitrogen lone pair donation at oxygen/vacancy/Zn sites. These inherent vacancies are formed in ZnO lattice during growth process. Ammonia gas is physically adsorbed on the surface pores of film. The high electrostatic field in the chemisorbed OH layer and the proton conduction takes place in the physisorbed layers. The condensation of ammonia vapor takes place in the mesopores and predominates in 30°C measurements. These results are in agreement with the resistance measurements. Vanadium doped zinc oxide nanoparticles exhibit enhancement in ammonia vapor sensing sensitivity and responsiveness with increase in ammonia vapor concentration. The change in electrical resistance has been used as the measure of ammonia vapor response at different temperatures.

Vanadium doped zinc oxide pellet prepared from nanoparticles with gold evaporated electrodes on the surface was used as sensing element for ammonia vapor detection in the laboratory. These nanostructure based sensors are

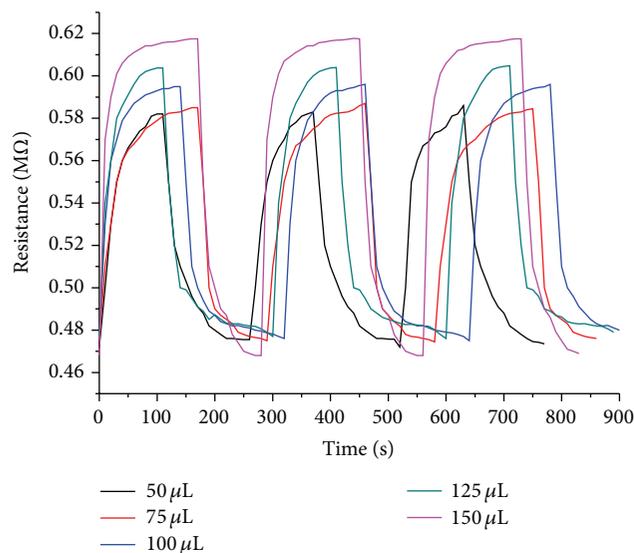


FIGURE 9: Resistance (MΩ) versus time (sec.) at 30°C for 50, 75, 100, 125, and 150 μL concentration of ammonia solution.

superior as compared to conventional gas sensors due to high sensitivity, selectivity, fast response, and recovery due to large number of active centers. In this work, ammonia vapors are sensed by vanadium doped zinc oxide sensing element in 20°C to 50°C at an interval of 5°C. The response and recovery curve of $Zn_{0.96}V_{0.04}O$ thick film based semiconductor gas sensor at 30°C operating temperature under ammonia vapor concentration of 50, 75, 100, 125, and 150 μL is presented in Figure 9. It is seen that the sensor resistance is higher in ammonia vapor atmosphere than in air. The sensor resistance decreases as the operating temperature increases.

In our studies the largest resistance increase was observed at 30°C. On varying ammonia vapor concentration, the resistance varies slightly in a particular temperature operating range.

The quality of a gas sensor, responsiveness, sensitivity, and response time are the important parameters. The responsiveness R_s is defined as $R_s = (R_f - R_o)/R_o$ where R_f is the electrical resistance of the sensor in air and R_o is its resistance in ethanol vapor. The sensitivity of sensor is defined as R_f/R_o . The response time is defined as the time required for the sample resistance variation to reach 90% of equilibrium value following a step increase in the concentration of the testing gas. The resistance, responsiveness sensitivity, and response time of vanadium doped zinc oxide based ethanol vapor sensor for 50, 75, 100, 125, and 150 μL concentration of ammonia at 30°C are obtained from the response and recovery curves listed in Table 1.

Similarly, resistance, responsiveness, sensitivity, and response time of vanadium doped zinc oxide based ammonia vapor sensor are obtained for other temperatures and concentrations from the response and recovery curves. These results show that responsiveness, sensitivity, and time constant are practically constant with small deviations. This work reports the use of vanadium doped zinc oxide films as sensing element for the detection of ammonia vapor. The

TABLE 1

R_o	R_f	R_s ($R_f - R_o$)/ R_o	Sensitivity (R_f/R_o)	Response time (τ_{90})
0.547	0.600	0.0969	1.0969	16
0.543	0.602	0.1087	1.1087	11
0.555	0.627	0.1297	1.1297	42
0.548	0.633	0.1551	1.1551	25
0.566	0.650	0.1484	1.1935	37

sensing mechanism is the interaction of ammonia vapors on the surface of vanadium doped zinc oxide nanoparticles via weak hydrogen bonding through hydrogen and nitrogen lone pair donation at zinc/oxygen vacant sites as shown in Figure 8. The substitution of bigger sized V^{2+} ions at Zn^{2+} site in ZnO lattice may be the reason for resistance increase during ammonia vapor sensing.

4. Conclusion

In the present work, zinc oxide and vanadium oxide nanoparticles were prepared by chemical coprecipitation route. The thick films from these as-synthesized nanoparticles were cast by simple, cost effective screen printing technique followed by drying and sintering of the film at 550°C temperature. The structural, optical, and morphological characteristics of these films were investigated. The ammonia vapor sensing properties of sample in the rectangular pellet form with different concentrations and temperature of ammonia vapors were studied. The responsiveness, sensitivity, and response time of the sensor were calculated from their response and recovery curves which show minor variations at a particular temperature with concentration of ammonia vapors. This material is found to be suitable candidate for ammonia vapor sensor in the selected temperature range.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

Acknowledgment

Rayees Ahmad Zargar is thankful to the CSIR-NPL, New Delhi, for giving permission to use characterization facilities required for analysis.

References

- [1] C.-L. Hsu and S.-J. Chang, "Doped ZnO 1D nanostructures: synthesis, properties, and photodetector application," *Small*, vol. 10, no. 22, pp. 4562–4585, 2014.
- [2] I. G. Dimitrov, A. O. Dikovska, P. A. Atanasov, T. R. Stoyanchov, and T. Vasilev, "Al doped ZnO thin films for gas sensor application," *Journal of Physics: Conference Series*, vol. 113, no. 1, Article ID 012044, 5 pages, 2008.

- [3] M. L. Grilli, A. Sytchkova, S. Boycheva, and A. Piegari, "Transparent and conductive Al-doped ZnO films for solar cells applications," *Physica Status Solidi A: Applications and Materials Science*, vol. 210, no. 4, pp. 748–754, 2013.
- [4] K. Khun, Z. H. Ibupoto, C. O. Chey, J. Lu, O. Nur, and M. Willander, "Comparative study of ZnO nanorods and thin films for chemical and biosensing applications and the development of ZnO nanorods based potentiometric strontium ion sensor," *Applied Surface Science*, vol. 268, pp. 37–43, 2013.
- [5] R. A. Zargar, S. Chackrabarti, S. Joseph, M. S. Khan, R. Husain, and A. Hafiz, "Synthesis and characterization of screen printed ZnO films for solar cell applications," *Optik*, vol. 126, no. 23, pp. 4171–4174, 2015.
- [6] T. Naydenova, P. Atanasov, M. Koleva et al., "Influence of vanadium concentration on the microstructure and magnetic properties of V-doped ZnO thin films," *Thin Solid Films*, vol. 518, no. 19, pp. 5505–5508, 2010.
- [7] E. Schlenker, A. Bakin, B. Postels et al., "Magnetic characterization of ZnO doped with vanadium," *Superlattices and Microstructures*, vol. 42, no. 1–6, pp. 236–241, 2007.
- [8] K. Sato and H. Katayama-Yoshida, "First principles materials design for semiconductor spintronics," *Semiconductor Science and Technology*, vol. 17, no. 4, pp. 367–376, 2002.
- [9] N. Tahir, S. T. Hussain, M. Usman, S. K. Hasanain, and A. Mumtaz, "Effect of vanadium doping on structural, magnetic and optical properties of ZnO nanoparticles," *Applied Surface Science*, vol. 255, no. 20, pp. 8506–8510, 2009.
- [10] Z. Jin, M. Murakami, T. Fukumura et al., "Combinatorial laser MBE synthesis of 3d ion doped epitaxial ZnO thin films," *Journal of Crystal Growth*, vol. 55, pp. 214–215, 2000.
- [11] E. A. Meulenkamp, "Synthesis and growth of ZnO nanoparticles," *Journal of Physical Chemistry B*, vol. 102, no. 29, pp. 5566–5572, 1998.
- [12] P. S. Kumar, A. D. Raj, D. Mangalaraj, and D. Nataraj, "Growth and characterization of ZnO nanostructured thin films by a two step chemical method," *Applied Surface Science*, vol. 255, no. 5, pp. 2382–2387, 2008.
- [13] Z. Wang, H. Zhang, L. Zhang, J. Yuan, S. Yan, and C. Wang, "Low-temperature synthesis of ZnO nanoparticles by solid-state pyrolytic reaction," *Nanotechnology*, vol. 14, no. 1, pp. 11–15, 2003.
- [14] S. C. Erwin, L. Zu, M. I. Haftel, A. L. Efros, T. A. Kennedy, and D. J. Norris, "Doping semiconductor nanocrystals," *Nature*, vol. 436, no. 7047, pp. 91–94, 2005.
- [15] L. Wang and M. Muhammed, "Synthesis of zinc oxide nanoparticles with controlled morphology," *Journal of Materials Chemistry*, vol. 9, no. 11, pp. 2871–2878, 1999.
- [16] S. Vijayalakshmi, S. Venkataraj, and R. Jayavel, "Characterization of cadmium doped zinc oxide (Cd:ZnO) thin films prepared by spray pyrolysis method," *Journal of Physics D: Applied Physics*, vol. 41, no. 24, Article ID 245403, 7 pages, 2008.
- [17] S. Mondal and P. Mitra, "Preparation of cadmium-doped ZnO thin films by SILAR and their characterization," *Bulletin of Materials Science*, vol. 35, no. 5, pp. 751–757, 2012.
- [18] J. A. Najim and J. M. Rozaiq, "Effect Cd doping on the structural and optical properties of ZnO thin films," *International Letters of Chemistry, Physics and Astronomy*, vol. 15, pp. 137–150, 2013.
- [19] R. A. Zargar, S. U. Khan, M. S. Khan, M. Arora, and A. K. Hafiz, "Synthesis and characterization of screen printed Zn_{0.97}Cu_{0.03}O thick film for semiconductor device applications," *Physics Research International*, vol. 2014, Article ID 464809, 5 pages, 2014.
- [20] R. A. Zargar, S. Chackrabarti, M. Shahabuddin, J. Kumar, M. Arora, and A. K. Hafiz, "Novel composites of Zn_{1-x}Cd_xO (x = 0, 0.05, 0.1) thick films for optoelectronic device application," *Journal of Materials Science: Materials in Electronics*, 2015.
- [21] B. D. Cullity, *Elements of X-Ray Diffraction*, Addison-Wesley, Reading, Mass, USA, 1978.
- [22] S. Mahamuni, K. Borgohain, B. S. Bendre, V. J. Leppert, and S. H. Risbud, "Spectroscopic and structural characterization of electrochemically grown ZnO quantum dots," *Journal of Applied Physics*, vol. 85, no. 5, pp. 2861–2865, 1999.
- [23] H. Çolak and O. Türkoğlu, "Synthesis, crystal structural and electrical conductivity properties of Fe-doped zinc oxide powders at high temperatures," *Journal of Materials Science and Technology*, vol. 28, no. 3, pp. 268–274, 2012.
- [24] K. Samanta, P. Bhattacharya, R. S. Katiyar, W. Iwamoto, P. G. Pagliuso, and C. Rettori, "Raman scattering studies in dilute magnetic semiconductor Zn_{1-x}Co_xO," *Physical Review B*, vol. 73, no. 24, Article ID 245213, 5 pages, 2006.
- [25] J. Chan, *Four-Point Probe Manual, EECS143, Micro Fabrication Technology*, 1994, Modified by Paul Friedberg, 2002.
- [26] P. K. Ghosh, R. Maity, and K. K. Chattopadhyay, "Electrical and optical properties of highly conducting CdO:F thin film deposited by sol-gel dip coating technique," *Solar Energy Materials and Solar Cells*, vol. 81, no. 2, pp. 279–289, 2004.
- [27] F. B. Micheletti and P. Mark, "Effects of chemisorbed oxygen on the electrical properties of chemically sprayed cds thin films," *Applied Physics Letters*, vol. 10, no. 4, pp. 136–138, 1967.



Hindawi

Submit your manuscripts at
<http://www.hindawi.com>

