Research Article

XRD-Rietveld Method for Evaluating the Leaching Characteristics of Hardened Cement Paste in Flowing Water

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1. Introduction

Concrete structures such as dams, bridges, ports, wharves, and tunnels are vulnerable to erosion by flowing water [1, 2], which affects the durability and service life of them [3, 4]. Calcium ion is the main component of hydration products of Portland cement, and its content affects the stability of hydration products. Flowing water will cause the diffusion and leaching of Ca(OH)2 in concrete and the decalcification of C-S-H gel [5, 6]. The leaching of calcium ion in concrete is largely responsible for the reduction of concrete strength [7], and the dynamic leaching rate is higher than the static leaching rate.

The leaching test methods include the direct method and indirect method. Direct method is usually used to test the leaching of specimens directly with pure water or deionized water [8]. The indirect method is to use chemical solution to accelerate the test, such as ammonium nitrate solution [9], ammonium chloride solution, and electrochemical theory [10]. In terms of test efficiency, the leaching process of specimens in the direct method is relatively slow [11, 12]. The indirect method has the characteristic of fast erosion speed [13, 14], but the chemical reaction is complex and there are many influencing factors.

The leaching of concrete is mainly caused by the leaching of hardened cement paste, accompanied by the change in crystal content of hydration products, such as Ca(OH)2, CaCO3, C-S-H, and SiO2. There are many parameters to evaluate the leaching effect, such as the permeability of concrete, the change in pH value of leachate, the quantity of dissolved chemical substances, and the loss of mass and strength [15]. XRD can be used to identify and quantify the mineral composition of cement-based materials, discriminate hydration products, and study the hydration process of cement, and determine the hydration rate of cement clinker minerals [9, 16]. The method based on full-spectrum...
2.1. Materials and Mix Proportions. The cement used in the test was ordinary Portland cement (OPC) and superfine cement (SMC) with 5% silica fume, and the strength grade was 42.5. The physical properties of the materials are shown in Tables 1 and 2. Ultrafine cement was obtained by grinding OPC. Its average particle size was 4.25 μm, and the specific surface area was 5770 m²/kg. Tap water was used as mixing water and leaching medium. The pH value of tap water was 7.5–8.0.

2.2. Preparation of the Specimens

2.2.1. Specimens for Leaching and Mechanical Tests. The specimens were prisms of size 10 mm × 10 mm × 60 mm. The water-binder ratio was 0.55. After the cement paste was molded, the standard curing time was 24 h, and then, the mold was removed and the leaching test was carried out. There were 3 specimens in each group in the flexural strength test. The strength value was taken as the average of the measured values of three specimens.

2.2.2. XRD Testing. Sampling positions were 0–2 mm (surface), 2–4 mm (middle), and >4 mm (core) depth, respectively (shown in Figure 1). The samples were placed in an agate bowl and ground manually until they passed through 200 meshes (0.074 mm) sieve, and then, 5 g samples were randomly taken. In order to prevent carbonization during grinding, anhydrous ethanol was added to the grinder. The samples were dried to constant weight at 40°C in the oven, and then, the XRD spectra were collected in X-ray diffractometer.

2.3. Tests and Methods. After 24-hour standard curing, the specimens were dismantled and soaked in water tanks, respectively. The experimental flow velocities were 0 mm/s, 2.8 mm/s, and 11.2 mm/s, respectively. The experimental flow velocities were calculated according to the groundwater flow field of a reservoir project. The actual average flow velocity was 2.8 mm/s and 0 mm/s meant that groundwater did not flow. The test ages were 3 d, 7 d, 28 d, 56 d, and 90 d, respectively.

As shown in Figure 2, the flow field test device consisted of an upper water tank, a lower water tank, a flow control valve, a flow tank, a support platform for the upper water tank, a support platform for the flow tank, a sand cushion, a water pump, and a water pipe. After the water in the upper tank flowed out from the flow control valve, it entered the flume to erode the specimens on the sand cushion. There was an opening at the end of the flume. The water flowed out from the opening and entered the lower tank. The upper water tank was a high-strength plastic water tank with a capacity of 200 L. The water in the tank was controlled intelligently to maintain a uniform height. The flow tank was a PVC rectangular tank with a net width of 92 mm. Six specimens could be arranged evenly along the width direction. The specific operation steps were as follows:

1. Appropriate amount of water was injected into the upper water tank, and then, the flow control valve was opened; the flow rate was calibrated by measuring cylinder and stopwatch, and the opening degree of the valve was adjusted.

2. The test blocks were put into the flow tank, and a gap was kept between the test blocks for the flow to pass through, while a part of the test blocks was reserved for curing in still water under the same conditions.

3. When reaching the required erosion time, the specimens were tested for flexural strength and XRD. Six specimens in each group were tested, of which three were for flexural strength, and the other three for XRD. The testing age of XRD was 28 d, 56 d, and 90 d. Among them, the 56 d samples were also tested for XRD with different depths.

2.3.1. Leaching Resistance Coefficient Tests. The ratio of the flexural strength of specimens leached with different flow rates to that of hydrostatically leached specimens at the same age was defined as the leaching resistance coefficient, and the leaching resistance coefficient was used to evaluate the leaching effect of cement-based materials. The larger the coefficient was, the better the leaching resistance of the material was. The flexural strength of the specimens was tested by three-point bending-tension test, and $L$ was 40 mm:
where $K_f$ is the leaching resistance coefficient; $f_e$ is the flexural strength of the test piece under flowing water erosion, MPa; and $f_0$ is the flexural strength of the test piece under static water, MPa.

2.3.2. X-Ray Powder Diffraction (XRD). When the multi-phase system is irradiated by monochrome X-ray, the diffraction patterns of each phase in the diffraction space superimpose each other to form a one-dimensional diffraction pattern. In the process of weighted overlapping addition of powder diffraction spectra of each phase, the positions of the diffraction lines of each phase will not change, and the intensity of the diffraction lines varies with the percentage of the phase in the mixture (volume or mass), the scattering force, and the absorption force of other phases, and the scale factor is the reflection of the intensity change.

The obtained dry hydration samples were subjected to X-ray diffraction (XRD) analysis using a X-ray diffractometer (D8 ADVANCE, Bruker AXS Corporation, GER) employing Cu-Kα radiation ($\lambda = 0.15418$ nm, 40 kV, 50 mA) over scanning range $2\theta = 15^\circ \sim 65^\circ$ at step width 2° per min.

### Table 1: Physical properties of OPC.

<table>
<thead>
<tr>
<th>Analysis item</th>
<th>CaO (%)</th>
<th>SiO₂ (%)</th>
<th>Al₂O₃ (%)</th>
<th>Fe₂O₃ (%)</th>
<th>MgO (%)</th>
<th>SO₃ (%)</th>
<th>Na₂O + K₂O (%)</th>
<th>LOI (%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>OPC</td>
<td>55.72</td>
<td>23.42</td>
<td>7.64</td>
<td>2.93</td>
<td>3.15</td>
<td>2.41</td>
<td>0.41</td>
<td>2.7</td>
</tr>
</tbody>
</table>

### Table 2: Physical properties of silica fume.

<table>
<thead>
<tr>
<th>Analysis item</th>
<th>SiO₂ (%)</th>
<th>Na₂O (%)</th>
<th>K₂O (%)</th>
<th>R₂O (%)</th>
<th>Cl (%)</th>
<th>LOI (%)</th>
<th>Activity index (%)</th>
<th>Specific surface area (m²/kg)</th>
<th>Average particle size (µm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Silica fume</td>
<td>93.1</td>
<td>0.32</td>
<td>0.46</td>
<td>0.48</td>
<td>0.02</td>
<td>2.9</td>
<td>107</td>
<td>18954</td>
<td>0.24</td>
</tr>
</tbody>
</table>

### Figure 1: Diagram of sampling position of XRD sample: (a) sampling position of cross section (unit: mm); (b) sample of hardened cement paste and XRD sampling location.

### Figure 2: Schematic diagram of the flowing water erosion test device: (a) vertical view; (b) lateral view (1, upper water tank; 2, lower water tank; 3, water flow control valve; 4, water channel; 5, upper water tank support platform; 6, water channel support platform; 7, specimen; 8, sand cushion; 9, water pump; 10, water pipe).
the mechanical properties deteriorating [7, 23]. The three-point flexural tensile strength of specimens at different ages and velocities was tested. The flexural strength is shown in Figure 3(a), and the results of calculation of leaching resistance coefficient by formula (1) are shown in Figure 3(b).

From Figure 3(a), it can be seen that the flexural strength of the specimens increased with the increase in age, the growth rate was relatively high before 28 days, and the OPC is more obvious. The flexural strength of OPC decreased slightly after 28 days, and the decrease in flexural strength of OPC was more obvious than that of SMC. As can be seen from Figure 3(b), the leaching resistance coefficient of OPC decreased gradually, and the boundary point was about 28 days later. The leaching resistance coefficient of SMC increased and approached 1 with the increase in time. The designed flow rate had little effect on the flexural strength within 90 days.

From the change in flexural strength, the leaching resistance of SMC was better than that of OPC. This was related to the addition of silica fume. Active SiO$_2$ in silica fume could produce pozzolanic reaction and filling effect. Free Ca(OH)$_2$ from cement hydration reacted with active SiO$_2$ to form stable low alkalinity C-S-H gel. The C-S-H gel with low alkalinity was more compact and denser than that without silica fume, and its porosity was lower. Its strength was higher than that of Ca(OH)$_2$ crystal. Therefore, the addition of silica fume reduced the content of Ca(OH)$_2$ in the paste, and the macro pore was replaced by the small pore, which reduced the most probable pore size and specific surface area, reduced the porosity, and improved the compactness of the structure. Macroscopically, the strength, impermeability, and frost resistance of cement with the addition of silica fume were improved [24].

**Figure 3:** Mechanical properties of leaching specimens at different ages and flow rates: (a) flexural strength; (b) leaching resistance coefficient.
3.2. Analysis of XRD-Rietveld. Hydration of Portland cement mainly generates Ca(OH)$_2$ and C-S-H, as follows:

\[
2(3\text{CaO} \cdot \text{SiO}_2) + 6\text{H}_2\text{O} \rightarrow 3\text{CaO} \cdot 2\text{SiO}_2 \cdot 3\text{H}_2\text{O} + 3\text{Ca(OH)}_2
\]  

(2)

However, in the water environment, Ca(OH)$_2$ in the hardened cement slurry will interact with CO$_2$ in the air to form CaCO$_3$, as shown in the following equations:

\[
\text{Ca(OH)}_2 + \text{Ca(HCO}_3)^+ \rightarrow \text{CaCO}_3 + \text{H}_2\text{O}
\]  

(3)

\[
\text{CO}_2 + \text{H}_2\text{O} + \text{Ca(OH)}_2 \rightarrow \text{CaCO}_3 + 2\text{H}_2\text{O}
\]  

(4)

\[
\text{CO}_2 + \text{H}_2\text{O} + \text{CaCO}_3 = \text{Ca(HCO}_3)^+
\]  

(5)

In addition, after the hydration reaction of cement, the relative concentration of hydration product reaches a certain value and keeps an equilibrium. Among them, Ca(OH)$_2$ has the highest limit concentration and is most easily dissolved by permeable water. Once the equilibrium is broken, the amount of Ca(OH)$_2$ will be adjusted, resulting in calcium ion precipitation. So, like the solution is continuously diluted, the hardened cement paste constantly dissolves out calcium hydroxide under the action of current, which can lead to the decalcification of hydrated calcium silicate and the decomposition of Ca/Si after lowering. The strength of hardened cement paste reduces, and its performance deteriorates.

3.2.1. XRD Testing. In order to further analyze the leaching characteristics of SMC and OPC from the microscopic scale, XRD analysis was carried out on the samples. The intensity of XRD diffraction line varied with the percentage (volume or mass) of phase in the mixture, the scattering force, and the absorptive force of the phase. The variation trend reflected the content of the phase.

The main hydration products of ordinary Portland cement are calcium hydroxide (about 20%), C-S-H gel (about
70%), calcium sulphoaluminate hydrate (about 7%), calcium aluminohydrate, and calcium ferrite hydrate (<3%) [25]. C-S-H gel was amorphous colloid with poor crystallinity, so Ca(OH)₂, CaCO₃, and SiO₂ crystals were mainly selected for analysis.

The XRD results of the powder samples sampled at the depth of 0–2 mm at different ages and velocities are shown in Figures 4–6. XRD results of 56 d samples with different sampling depths are shown in Figure 7.

Figure 4 shows that Ca(OH)₂ was the hydrated product of each sample at 28 d, and its diffraction peak was the highest. In addition, the diffraction characteristic peaks of CaCO₃ and SiO₂ can be seen, and the diffraction peaks of SiO₂ were weak. From Figure 4(a), it can be seen that the diffraction peak of Ca(OH)₂ and CaCO₃ on the surface of OPC increased with the increase in water velocity. From Figure 4(b), it can be seen that the Ca(OH)₂ diffraction peak of SMC decreased with the increase in water velocity, which indicates that less and less Ca ions can be leached. The diffraction peaks of CaCO₃ increased gradually, indicating that more and more stable CaCO₃ was formed by calcium ions.

As the curing age increased to 56 d, it can be seen from Figure 5 that the diffraction peaks of Ca(OH)₂ and CaCO₃ in OPC increased with the increase in flow rate. SMC specimens were the opposite. The diffraction peaks of SiO₂ in the samples were still weak, and there was little difference among the samples at different flow velocities.

When the age reached 90 days, the diffraction peaks of Ca(OH)₂ and CaCO₃ in the sample atlas decreased with the increase in flow rate, which indicates that the leaching of calcium ions tends to be stable and slow. The diffraction peak of SiO₂ was still weak.

According to the effect of flow rate on leaching in Figures 4–6, the effect of flow rate set in this experiment on erosion was not obvious. With the increase in age, the effect

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**Figure 5:** XRD of 56 d leaching of specimens at different flow rates: (a) OPC; (b) SMC.
of flow velocity was greater, which indicates that, in the actual environment, the calcium leaching of concrete structure is also a slow process, and the longer the leaching time is, the more obvious the calcium leaching is. The diffraction peaks of Ca(OH)\(_2\) and CaCO\(_3\) in each sample increased first and then decreased gradually, indicating that the leaching of calcium ions was more before 56 days. The leaching of the specimens proceeded from the surface to the interior gradually. With the increase in the leaching depth, the diffusion and leaching rate of calcium ions slowed down. When the SMC sample was mixed with silica fume, the diffraction peak of SiO\(_2\) had little change compared with that of OPC. The main component of silica fume was active SiO\(_2\), which had a pozzolanic reaction with cement hydrates, but it was consumed because of its small amount \([26]\). The decrease in Ca(OH)\(_2\) diffraction peak in SMC should be attributed to the reaction with SiO\(_2\), and the conversion of hydration products to low alkaline hydration products was also the main reason for the leaching resistance of hardened cement paste with silica fume.

On the one hand, the increase in the diffraction peak of CaCO\(_3\) indicates that Ca(OH)\(_2\) dissolved into CaCO\(_3\) gradually, while CaCO\(_3\) was more stable and filled the pore; on the other hand, it would be beneficial to reduce the leaching of Ca ions. The leaching rate was related to the porosity of hardened cement paste. Combined with the change rule of flexural strength of specimens, this phenomenon was verified. The porosity of ordinary Portland cement paste was larger than that of superfine cement. So OPC was easier to erode than SMC.

XRD at different sampling depths for 56 d is shown in Figure 7. The diffraction peak values of Ca(OH)\(_2\) and CaCO\(_3\) of OPC increased with the depth of sampling. It shows that the leaching of hardened cement paste starts from the surface and deepens gradually. From the point of view of flow rate, the intensity of diffraction peaks of each sample
Figure 7: XRD at different sampling depths for 56d. 0, 2.8, and 11.2 represent the flow velocity, respectively. (a) OPC; (b) SMC.
had little change, so the effect of flow rate on leaching was not significant. The intensity of phase diffraction peak of SMC varied little at different depths, which indicates that SMC has good leaching resistance.

### 3.2.2. Rietveld Quantitative Analysis

From Figures 4~7, it is not easy to quantify and compare the mass change laws of the main phases. Rietveld quantitative phase analysis is based on the relationship between the scaling factor and the reference intensity ratio to obtain the relationship between the relative content of the phase and the scaling factor \[27, 28\]. Quantitative calculation of phase was carried out by the Rietveld refinement method with Jade software. Quantitative analysis of the phases of different age samples is shown in Figures 8 and 9.

According to Figure 8(a), the relative mass fractions of Ca(OH)\(_2\) and CaCO\(_3\) in different depths of OPC specimens changed little with the increase in age, which indicates that the effect of flow rate set in the experiment on calcium leaching was not obvious. However, on the 90\(^{th}\) day, the larger the flow rate was, the smaller the amount of Ca(OH)\(_2\) was, which indicates that more Ca ions leached or transformed into CaCO\(_3\), and the leaching phenomenon was more obvious. Figure 8(b) shows that, with the increase in age, the relative mass fraction of Ca(OH)\(_2\) decreased gradually, while the relative mass fraction of CaCO\(_3\) increased gradually, but the change was not obvious. This

![Figure 8](image)

Figure 8: Quantitative analysis of phase mass fraction by XRD-Rietveld for specimens of different ages: (a) OPC; (b) SMC.

![Figure 9](image)

Figure 9: Mass fraction of phase for XRD-Rietveld quantitative analysis of 56d samples at different depths: (a) OPC; (b) SMC.
indicates that the reaction between active SiO$_2$ and Ca(OH)$_2$ in silica fume was basically completed on the 28$^{th}$ day. This is consistent with the conclusions of the literature [29, 30]. With the increase in day, the leaching of calcium ions gradually slows down. This is related to CaCO$_3$ filling pore and reducing calcium ion diffusion.

For 56 days of OPC, the relative mass fraction of Ca(OH)$_2$ and CaCO$_3$ was not affected by flow rate. With the increase in sampling depth, the relative mass of Ca(OH)$_2$ increased gradually, while the relative mass of CaCO$_3$ decreased gradually, as shown in Figure 9(a). This indicates that the calcium leaching of OPC was gradual from outside to inside. Figure 9(b) shows that the relative mass fractions of Ca(OH)$_2$ and CaCO$_3$ were not affected by the flow rate and sampling depth of SMC at 56 d. The reaction between active SiO$_2$ and Ca(OH)$_2$ in silica fume was basically completed before 56 days. SMC promoted the hydration of C$_3$S due to the reduction of Ca$^{2+}$ and OH-$\cdot$ concentration in the early stage of cement-water system, shortening the induction period. Fine silica fume particles also acted as crystallization nucleus of hydration products, thus accelerating the early hydration of cement and making the concrete with silica fume have early strength. This is consistent with the results of flexural strength of SMC mentioned above. In addition, due to the pozzolanic reaction, Ca(OH)$_2$ in cement paste was absorbed with a large amount, forming a low alkalinity C-S-H gel. The amount of Ca(OH)$_2$ decreased with the increase in silica fume and hydration age. From Figures 8 and 9, it can be seen that the relative content of SiO$_2$ in OPC and SMC was very small.

In fact, in running water environment, the calcium dissolution of hardened cement paste occurs at the same time with the hydration of the cement. After 28 days, the hydration rate of the cement slows down, and the dissolution is related to the ion concentration of the solution of the pore. As shown in Figure 3(a), the bending performance of hardened cement paste shows a decreasing trend at different ages and flow rates. From the perspective of XRD, the phase Ca(OH)$_2$ decreases and CaCO$_3$ increases, as shown in Figure 8, which is more obvious at different sampling depths, as shown in Figure 9(a). According to equations (2)~(5), as the age increases, Ca(OH)$_2$ dissolves more and reacts with CO$_2$ to generate CaCO$_3$. Therefore, the leaching of hardened cement paste may be correlated with the mass gradient of Ca(OH)$_2$.

In Figures 8 and 9, the mass fraction of the main phases determined by quantitative analysis is actually a relative quantity, and the mass fraction of Ca(OH)$_2$, CaCO$_3$, and SiO$_2$ are selected as variables. In order to facilitate the comparison with the change trend of mechanical indexes,
the ratio of CaCO₃ to Ca(OH)₂ mass fraction was taken as the change quantity, and the change curve of this ratio was drawn, as shown in Figure 10. It can be seen from Figure 10(a) that the mass ratio of OPC samples does not change much in different ages and tends to increase, indicating that the relative content of CaCO₃ increases, the leaching of calcium is more obvious, and the greater the water velocity is, the greater its influence on Ca ions erosion is. For the SMC sample, the changing trend was similar to that of OPC, but the water velocity had little effect on the leaching effect. According to Figure 10(b), the leaching of the OPC sample took place from the outside in, while the leaching difference between the inside and outside of SMC sample was not significant, which shows that the leaching of hardened cement paste is related to the changes in the relative quantity of CaCO₃ and Ca(OH)₂. The higher the relative content of CaCO₃ is, the denser the hardened cement paste is, and the better its erosion resistance is. The erosion resistance of hardened cement paste is related to the mass gradient of Ca(OH)₂. The smaller the gradient is, the better the erosion resistance is.

Compared with mechanical properties test and XRD phase analysis, Rietveld phase quantitative calculation shows that the conclusions of three analytical methods are consistent.

4. Conclusions

Using tap water as the leaching medium, the leaching tests of three different hardened cement pastes with flow velocities of 0 mm/s, 2.8 mm/s, and 11.2 mm/s were carried out. The flexural strength, phase composition, and mass fraction of the paste after leaching were tested and analyzed. The following conclusions were drawn:

(1) Within 90 days, the effect of flow rate on the leaching of hardened cement paste is not obvious.

(2) Tap water acts as the erosion medium at test flow rate. Before 28 d, the leaching of OPC and SMC is not obvious, and the flexural strength increases gradually. After 28 d, the leaching of OPC is higher than that of SMC, and the flexural strength of OPC decreases, while the strength of SMC does not change much.

(3) XRD-Rietveld quantitative method was used to analyze the leaching characteristics of hardened cement paste. It is found that the leaching of hardened cement paste is related to the increase in CaCO₃ content and the gradient of Ca(OH)₂ content.

(4) By comparing the change in flexural strength and XRD diffraction intensity, the XRD-Rietveld method can be used to evaluate the leaching characteristics of hardened cement paste under flowing water.

Data Availability

The data used to support the findings of this study are available from the corresponding author upon request.

Conflicts of Interest

The authors declare that there are no conflicts of interest.

Acknowledgments

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