Research Article

Tunable Upconversion Luminescence and Energy Transfer Process in BaLa$_2$ZnO$_5$:Er$^{3+}$/Yb$^{3+}$ Phosphors

Lefu Mei, Jing Xie, Libing Liao, Ming Guan, and Haikun Liu

Beijing Key Laboratory of Materials Utilization of Nonmetallic Minerals and Solid Wastes, National Laboratory of Mineral Materials, School of Materials Science and Technology, China University of Geosciences, Beijing 100083, China

Correspondence should be addressed to Lefu Mei; mlf@cugb.edu.cn and Libing Liao; clayl@cugb.edu.cn

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Ba$_2$ZnO$_5$:Er$^{3+}$/Yb$^{3+}$ has been synthesized via a high temperature solid-state method, and the tunable upconversion luminescence and energy transfer process between Yb$^{3+}$ and Er$^{3+}$ in this system have been demonstrated. Upon 980 nm laser excitation, the intense green and red emission around 527, 553, and 664 nm were observed for BaLa$_2$ZnO$_5$:Er$^{3+}$/Yb$^{3+}$, which can be assigned to the characteristic energy level transitions of $^2$I$_{11/2} \rightarrow ^4$S$_{3/2}$, $^4$F$_{9/2} \rightarrow ^4$I$_{15/2}$, and $^4$I$_{15/2}$ of Er$^{3+}$, respectively. The critical Er$^{3+}$ quenching concentration (QC) was determined to be about 5 mol%, and the power studies indicated that mixture of 2- and 3-photon process was responsible for the green and red upconversion luminescence.

1. Introduction

It is believed that the appearance of photon upconversion (UC) luminescence phenomenon has attracted numerous attentions focused on the UC phenomena and UC luminescent materials because of their potential applications in medical labels, multicolor displays [1–4]. The upconversion (UC) luminescence process is achieved through the sequential absorption of two or more excitation photons, which is accorded with the anti-Stokes emission phenomenon. Consequently, UC processes can be induced by low power, continuous wave lasers, obviating the need for high-cost, high-intensity pulsed lasers that are required for simultaneous multiphoton absorption experiments such as simultaneous two-photon absorption and second harmonic generation [5, 6]. Up to now, efficient photon UC process has been observed to occur primarily in the rare-earth elements, namely, those of the Ln series. Lanthanide ions are very suitable to be used in UC process as they have the rich energy level structures, but allows for efficient spectral conversion. Among all the lanthanides, Er$^{3+}$ ion has abundant energy level structures, and always acting as the luminescent center can emit intense green and red light. In contrast, Yb$^{3+}$ ion has a strong and broad near-infrared absorption cross section around 980 nm with relatively simple electronic structure of two energy level manifolds: $^2$F$_{7/2}$ ground state and $^2$F$_{5/2}$ excited state around 10,000 cm$^{-1}$ in NIR region [7]. Additionally, the similar value of energy level of excited $^2$F$_{5/2}$ state of Yb$^{3+}$ is close to the $^4$I$_{11/2}$ levels of Er$^{3+}$ ions. Accordingly, cooperative UC process has also been reported for Tm$^{3+}$/Yb$^{3+}$, Ho$^{3+}$/Yb$^{3+}$, and Tm$^{3+}$/Ho$^{3+}$/Yb$^{3+}$ couples in many host materials [8–11].

The ternary oxides XY$_2$ZO$_5$(X = Ba, Y = rare-earth, and Z = Cu, Zn) are receiving much attention because of their very interesting structural, excellent physical, and chemical stability and special magnetic, optical, and superconducting properties. As a member of these compounds, BaY$_2$ZnO$_5$ and BaGd$_2$ZnO$_5$ have been proved to be efficient UC hosts [12–15]. However, the UC properties of BaLa$_2$ZnO$_5$ based phosphor have not been investigated. Based on the effective ionic radii and charge balance of cations with different coordination number (CN), the rare-earth Er$^{3+}$/Yb$^{3+}$ ions are expected to occupy the La$^{3+}$ sites randomly in the BaLa$_2$ZnO$_5$ host. Therefore, BaLa$_2$ZnO$_5$ can be an excellent host doped with various ions, and they are promising candidates for practical applications. In this paper, Er$^{3+}$/Yb$^{3+}$ codoped BaLa$_2$ZnO$_5$ UC materials are synthesized via a solid-state reaction process, and the structure and UC luminescent...
characteristics of these phosphors have been discussed in detail.

2. Experimental

A series of polycrystalline phosphors BaLa$_2$ZnO$_5$::Er$^{3+}$/Yb$^{3+}$ were synthesized by a solid-state reaction technology. The raw materials were Ba$_2$CO$_3$(AR), La$_2$O$_3$ (99.99%), ZnO (99.99%), Er$_2$O$_3$ (99.99%), and Yb$_2$O$_3$ (99.99%), which were used directly without any treatment. The selected starting materials were mixed and ground thoroughly. The homogeneous mixtures were calcined at 1250°C for 3 hours, with the heating rate of 5°C/min, and then the samples were cooled to room temperature naturally. After that, the samples were washed three times by the deionized water and dried for the following measurement.

The phase and crystal structure of the samples were recorded by X-ray diffraction (XRD, D8 Advance diffractometer, Bruker Corporation, Germany) with Cu-Kα radiation ($\lambda = 0.15406$ nm, 40 kV, 30 mA). The morphology of the as-prepared samples was characterized by a field emission scanning electron microscopy (FE-SEM, JSM-7001F). The UC luminescence spectra were recorded on a Hitachi F-4600 spectrophotometer (Hitachi High Technologies Corporation, Tokyo, Japan) equipped with an external power-controllable 980 nm semiconductor laser (Beijing Viasho Technology Company, China) as the excitation source. All the measurements were carried out at room temperature.

3. Results and Discussion

The crystallization and morphology of the as-prepared samples were checked by XRD and SEM measurements. Figure 1 shows the XRD patterns of the BaLa$_{1-x}$ZnO$_5$::x%Er$^{3+}$/0.10Yb$^{3+}$ ($x = 0.2\%$, 0.5\%, 1\%, 2\%, 3\%, 4\%, 5\%, 7\%, and 10\%) and the standard PDF diffraction lines of BaLa$_2$ZnO$_5$ as a reference. It can be seen that all of the diffraction peaks are matched well with the standard data of BaLa$_2$ZnO$_5$ (JCPDS number 52-1670) indicating the introduction of Er$^{3+}$/Yb$^{3+}$ ions into the BaLa$_2$ZnO$_5$ lattice is completely dissolved in the BaLa$_2$ZnO$_5$ host lattice by substitution for the La$^{3+}$ ion owing to their similar ionic radii and properties. Moreover, the diffraction peaks of the as-prepared samples shift toward the larger 2θ side owing to the small size of Yb$^{3+}$ ion and Er$^{3+}$ ion substituting for La$^{3+}$ in the compound.

Figure 2 shows the SEM micrographs of the typical BaLa$_2$ZnO$_5$:0.05Er$^{3+}$/0.10Yb$^{3+}$ powders prepared at 1250°C with different plotting scale. SEM result shows sheet-like phosphor grains with an average diameter of about 30 μm. Compared to the as-prepared samples via high temperature solid-state technology, we can find that the current samples have smooth surface and better crystallinity in the form of two-dimensional flaky states, which indicate that they should own better luminescence properties because of the decreased particle surface defects [16].

Upon the 980 nm laser excitation, strong visible emission was observed in codoped crystals due to the result of the upconversion process. Figure 3(a) shows the comparison of UC luminescence spectra of the as-prepared BaLa$_2$ZnO$_5$:xEr/Yb10% ($x = 0.2\%$, 0.5\%, 1\%, 2\%, 3\%, 4\%, 5\%, 7\%, and 10\%) samples. The emission consists of two strong bands: the red one peaked at 664 nm associated with $^{4}F_{9/2}$ → $^{4}I_{15/2}$ transition and the green band centered at 553 nm assigned to the mixed transition $^2H_{11/2} + ^4S_{3/2} → ^4I_{15/2}$ of the acceptor Er$^{3+}$ ion [17]. To date, a wide variety of Er$^{3+}$ doped BaLa$_2$ZnO$_5$ have been generated that are capable of emitting a wide range of colors within the visible spectral region. The above result testified the UC process in the Er$^{3+}$/Yb$^{3+}$ doped BaLa$_2$ZnO$_5$. Additionally, as shown in Figure 3(b), it can be seen that either the green

![Figure 1: XRD patterns of the BaLa$_{1-x}$ZnO$_5$:x%Er$^{3+}$/0.10Yb$^{3+}$ ($x = 0.2\%$, 0.5\%, 1\%, 2\%, 3\%, 4\%, 5\%, 7\%, and 10\%) and the standard PDF diffraction lines of BaLa$_2$ZnO$_5$ as a reference: (a) $10^\circ \leq 2\theta \leq 70^\circ$ and (b) $29.5^\circ \leq 2\theta \leq 32^\circ$.](image-url)
Figure 2: SEM images of the BaLa$_2$ZnO$_5$:0.05Er$^{3+}$/0.10Yb$^{3+}$ plotting scale ×1000 (a) and plotting scale ×5000 (b).

Figure 3: Comparison of UC luminescence spectra of the BaLa$_2$ZnO$_5$:xEr$^{3+}$/10%Yb$^{3+}$ ($x = 0.2\%, 0.5\%, 1\%, 2\%, 3\%, 4\%, 5\%, 7\%, \text{and} 10\%$) under 980 nm laser excitation (a), and the variation of green and red emission intensities corresponds to different Er$^{3+}$ doping concentration (b).

Er$^{3+}$ ions concentration (%)

<table>
<thead>
<tr>
<th>Er$^{3+}$ mol%</th>
<th>0.2%</th>
<th>0.5%</th>
<th>1%</th>
<th>2%</th>
<th>3%</th>
<th>5%</th>
<th>7%</th>
<th>10%</th>
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553 nm: $^2H_{11/2} \rightarrow ^4I_{15/2}$
527 nm: $^4S_{3/2} \rightarrow ^4I_{15/2}$
664 nm: $^4F_{9/2} \rightarrow ^4I_{15/2}$

The physical mechanism involved in the upconversion processes can be elucidated by analyzing the dependence of the integrated upconverted intensity ($I_{em}$) as a function of the pumping intensity ($P_{pump}$), which is suggested to obey the following empirical equation [19–21]:

$$I_{em} \propto (P_{pump})^n,$$

where $n$ is the number of pump photons required for the transition from ground state to the upper emitting state. A plot of log $I_{em}$ versus log $P_{pump}$ yields a straight line with slope $n$. Figure 4(a) shows the UC emission spectra of BaLa$_2$ZnO$_5$:5%Er$^{3+}$/10%Yb$^{3+}$ with different pumping power, and dependence of green and red UC luminescence intensities upon pumping power is shown in Figure 4(b). With the increasing pumping power, UC emission intensities...
of BaLa$_2$ZnO$_4$:5%Er$^{3+}$/10%Yb$^{3+}$ increased. The calculated slopes were 2.14 for the red emission (664 nm: $^{4}F_{9/2} \rightarrow ^{4}I_{15/2}$) and 2.10 for the green emission (553 nm: $^{2}S_{3/2} \rightarrow ^{4}I_{15/2}$), indicating that both red and green emission are the mixture of 2- and 3-photon process which were responsible for the green and red upconversion luminescence. The corresponding energy levels scheme for the infrared excitation and upconversion emission is demonstrated in the schematic energy level diagram of Er$^{3+}$ and Yb$^{3+}$ ions, as shown in Figure 5. According to the above-mentioned three-photon process in BaLa$_2$ZnO$_4$:Er$^{3+}$/Yb$^{3+}$, the Yb$^{3+}$ ions act as sensitizer and the Er$^{3+}$ ions as activators. Under 980 nm laser excitation, a Yb$^{3+}$ ion can be excited by one of the near-infrared photons and transited from the ground state of $^{2}F_{7/2}$ to the only excited state of $^{2}F_{5/2}$, and then transfers the energy to Er$^{3+}$ ion and promotes the transition of $^{4}I_{15/2} \rightarrow ^{4}I_{11/2}$ of Er$^{3+}$ ion. Furthermore, Er$^{3+}$ in $^{4}I_{11/2}$ level is easy to reach a lower excited level of $^{4}I_{13/2}$ by a nonradiative relaxation as the similar value of energy level. Then, the second step of ET$_{2}$ from Yb$^{3+}$ can promote an excited state absorption (ESA) of Er$^{3+}$ from $^{4}I_{13/2}$ to the $^{2}F_{9/2}$ level. In this case, the emission band located in the red region which associated with the transition of $^{4}F_{9/2} \rightarrow ^{4}I_{15/2}$. The third energy transfer step (ET$_{3}$) are followed as: Er$^{3+}$ ions at $^{4}F_{9/2}$ level relax nonradiatively again and back to another lower excited level of $^{4}I_{9/2}$, and then the Er$^{3+}$ are excited from $^{4}I_{9/2}$ to $^{4}F_{7/2}$ by another ESA process. In this case, the emission band located in the green region which associated with the $^{2}H_{11/2} \rightarrow ^{4}I_{15/2}$ and $^{4}S_{3/2} \rightarrow ^{4}I_{15/2}$ transitions of Er$^{3+}$ ions, respectively. Accordingly, green emission concerted at 527 and 549 nm was detected in the UC spectra.

4. Conclusions

UC phosphors of Er$^{3+}$/Yb$^{3+}$ codoped BaLa$_2$ZnO$_4$ were synthesized by a traditional solid-state reaction method. Under 980 nm near-infrared laser excitation, both green (527 nm and 553 nm) and red (664 nm) emission bands have been found in the UC spectra, and these emission peaks are assigned to the characteristic level transition of $^{2}H_{11/2} \rightarrow ^{4}I_{15/2}$, $^{4}S_{3/2} \rightarrow ^{4}I_{15/2}$, and $^{4}F_{9/2} \rightarrow ^{4}I_{15/2}$ of Er$^{3+}$, respectively. The influence of Er$^{3+}$ doped concentration on UC luminescence intensities has been studied, which
depicts that the optimum Er$^{3+}$ doped concentration is 5%. The dependence of the UC luminescence on pumping power indicates that the energy transfer from Yb$^{3+}$ to Er$^{3+}$ in the BaLa$_2$ZnO$_4$ host is a three-photon process. The mechanisms for the green and red UC luminescence were discussed in detail.

Conflict of Interests
The authors declare that there is no conflict of interests regarding the publication of this paper.

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References


[9] J. Y. Sun, B. Xue, G. C. Sun, and D. P. Cui, “Yellow upconversion luminescence in Ho$^{3+}$/Yb$^{3+}$ co-doped Gd$_{0.5}$Mo$_{0.5}$O$_4$ phosphor,” Journal of Rare Earths, vol. 31, no. 8, pp. 741–744, 2013.


