Research Article

Properties of GH4169 Superalloy Characterized by Nonlinear Ultrasonic Waves

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Received 26 December 2014; Revised 28 January 2015; Accepted 28 January 2015

Academic Editor: Jainagesh A. Sekhar

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The nonlinear wave motion equation is solved by the perturbation method. The nonlinear ultrasonic coefficients $\beta$ and $\delta$ are related to the fundamental and harmonic amplitudes. The nonlinear ultrasonic testing system is used to detect received signals during tensile testing and bending fatigue testing of GH4169 superalloy. The results show that the curves of nonlinear ultrasonic parameters as a function of tensile stress or fatigue life are approximately saddle. There are two stages in relationship curves of relative nonlinear coefficients $\beta^* \delta^*$ versus stress and fatigue life. The relative nonlinear coefficients $\beta^*$ and $\delta^*$ increase with tensile stress when tensile stress is lower than 65.8% of the yield strength, and they decrease with tensile stress when tensile stress is higher than 65.8% of the yield strength. The nonlinear coefficients have the extreme values at 53.3% of fatigue life. For the second order relative nonlinear coefficient $\beta^*$, there is good agreement between the experimental data and the comprehensive model. For the third order relative nonlinear coefficient $\delta^*$, however, the experiment data does not accord with the theoretical model.

1. Introduction

GH4169 superalloy has good integrate performance in 253–400°C. It is manufactured as complex components and widely used in aerospace, nuclear energy, petroleum industry, and extrusion dies. Under the tensile loading and the cyclic loading, the metal structural components gradually occur as plastic deformation and induce the damages. They induce the components to crack or fracture.

Fatigue is one of damage mechanisms of components. According to statistics, more than 90% of the parts failures result from fatigue damages. The fatigue life of structural components can be divided into three stages as follows: early dislocation generation and lattice deformation, microcrack formation, and fracture failure. For structural components, the first and second stages are 80%–90% in fatigue life. Therefore, the detection of early fatigue damage is critical to evaluation of component life [1–3]. The ultrasonic method is believed to be the most promising nondestructive testing technique compared with other destructive and nondestructive testing. Many researches show that the linear ultrasonic parameters such as ultrasonic velocity and attenuation are effective in detection and evaluation in second stage and third stage of fatigue life [4–6]. But the ultrasonic linear parameters are not sensitive to nonlinear behavior and early fatigue state in metal components [5–8]. Therefore it is a focus issue that the nonlinear ultrasonic parameters evaluate the early fatigue state or stress in metal components and predict the fatigue life.

In 1755, Euler proposed the concept of nonlinear acoustic. Researchers such as Lagrange (1760) [9], Stokes (1848) [10], and Rayleigh (1910) [11] studied the theory of nonlinear acoustic [9]. In the 1960s, researchers began to study the phenomenon of nonlinear acoustic in solid [12]. In 1963, Hikata et al. observed that the harmonic waves appeared in aluminum. And Breazeale et al. detected third order elastic constants of single crystal aluminum and copper in 1963 and 1968 [13, 14]. Buck et al. studied the acoustic harmonic generation at unbonded interfaces and fatigue cracks [15]. Nagy used nonlinear ultrasonic waves to assess fatigue damage in plastics [16]. Kim et al. used nonlinear ultrasonic waves to characterize fatigue damage in a nickel-base superalloy [17]. Metya et al. studied the generation of second harmonic wave in grade maraging steel [18]. Shui et al.
used nonlinear longitudinal waves to evaluate plastic damage in metal materials [19]. Ruiz et al. used nonlinear acoustic parameter to assess early detection of thermal degradation of mechanical properties in 2205 duplex stainless steel [20]. Punnose et al. used nonlinear ultrasonic waves to evaluate fatigue in austenitic stainless steel in low cycle regime [21]. The results show that the second order nonlinear coefficient $\beta$ is related to the fatigue life or fatigue damage of materials. But few researchers studied the relationship of the third order nonlinear coefficient versus stress or fatigue life.

The one-dimensional nonlinear ultrasonic wave motion equation is solved by perturbation method. During tensile testing and bending fatigue testing, the relationships of nonlinear ultrasonic parameters and tensile stress or fatigue cycles are studied. The nonlinear ultrasonic testing system is used to detect the received signal. RETIC advanced measurement system generates a tone burst signal and receives the signal propagated in GH4169 superalloy plate. The received signals are transformed by FFT. The fundamental and harmonic amplitudes are obtained, and the parameters are calculated. The relationship curves of nonlinear parameters as the functions of tensile stress or fatigue life are obtained to detect or predict stress or fatigue life of GH4169 superalloy.

2. Nonlinear Acoustic Properties

2.1. Nonlinear Acoustic Wave Theory. When one-dimensional longitudinal wave propagates in nonlinear medium, the wave motion equation is as follows [22]:

$$\frac{1}{\rho} \cdot \frac{\partial \sigma}{\partial x} = \frac{\partial^2 u}{\partial t^2}, \quad (1)$$

where $u$ is the displacement along $x$ direction, $\rho$ is the density of medium, and $\sigma(x,t)$ is the positive stress along $x$ direction. The perturbation method is used to solve (1). The solution is

$$u(x,t) = A_1 \cos(kx - \omega t) - \frac{\beta_1}{8} k^2 A_1^2 x \cos(2(kx - \omega t))$$

$$+ \frac{\beta_2}{24} k^3 A_1^3 x [\cos(3(kx - \omega t)) + 3 \cos(kx - \omega t)], \quad (2)$$

where $\omega$ is the circular frequency and $k$ is the wave number. If the amplitude of $\cos(2(kx - \omega t))$ and $\cos(3(kx - \omega t))$ are, respectively, assumed as $A_2$ and $A_3$, they can be obtained:

$$\beta = \frac{8A_2}{k^2 x A_1^2}, \quad (3)$$

$$\delta = \frac{24A_3}{k^3 x A_1^3}. \quad (4)$$

With the fixed frequency and the fixed propagation distance, the nonlinear coefficients $\beta'$ and $\delta'$ are, respectively, proportional to $A_2/A_1^2$ and $A_3/A_1^3$. In the measurements, the relative nonlinear parameters can be defined as

$$\beta' = \frac{A_2}{A_1^2} = \frac{k^2}{8} \beta x, \quad (4)$$

$$\delta' = \frac{A_3}{A_1^3} = \frac{k^3}{24} \delta x. \quad (5)$$

The $\beta'$ and $\delta'$ are the functions of propagation distance $x$. The relationship of $\beta'$ and $x$ is a good linearity for propagation distance in the far field. In current experiments, $\beta'$ is measured over $x$ [23]. And in the optimized distance, $\beta'$ and $\delta'$ are measured over stress and fatigue life.

2.2. Dislocation Model for Second Order Nonlinear Coefficient. Nonlinearity of solid material mainly comes from two aspects: (1) nonharmonic property of particles interaction force of the material and (2) internal microdefects of the material, such as crystal dislocation, lattice slip, and microcrack. Many researches show that the ultrasonic nonlinearity induced by dislocations is the main factor leading to the variation of ultrasonic nonlinearity. Cantrell et al. promoted two dislocation models [15, 24, 25].

Hikata et al. [12], Buck et al. [15], and Cantrell [24] proposed the dislocation monopole model (shown in Figure 1). The generation of the second harmonic wave depends on the motion of the dislocation between the dislocation pinning points. The nonlinear acoustic coefficient is written as [24]

$$\beta_{mp} = -E_3 \frac{E_2}{E_3} \frac{\Lambda_{mp}}{G^2 b^2} \frac{L^4 R^3 E_1^2}{\sigma}, \quad (6)$$

where $\Lambda_{mp}$ is the dislocation density, $b$ is Burgers vector, $E_2$ and $E_3$ are, respectively, the second or third order elastic coefficients, $G$ is shear modulus, $\Omega$ is the transform coefficient from shear strain to principle strain, and $R$ is the transform coefficient from shear stress to principle stress.

The first part of (4) is relative to second and third order elastic coefficients $E_2$ and $E_3$. It is caused by lattice and is
negligible in tensile testing and fatigue bending testing. The second part of (4) is relative to dislocation density, string length, and stress. It is caused by dislocation and is the main factor to the ultrasonic coefficient.

Cantrell [24] proposed dislocation dipole model. With the increasing of stress and fatigue state, the dislocations increase and slip and interact. They form the dislocation dipole model (shown in Figure 2). Cash and Cai [25] improved the dipole model and the nonlinear acoustic coefficient is written as

$$\beta_{dp} = -\frac{E_3}{E_2} + \frac{16\pi^2\Omega\Lambda_{dp}h^2R^2(1-\nu)^2E_2^2}{G^2b} + \frac{384\pi^3\Omega\Lambda_{dp}h^3R^3(1-\nu)^3E_2^2}{G^3b^2}$$  \hspace{1cm} (7)

where ν is Poisson ratio and h is the distance of dipoles. The second part of (5) is relative to dislocation density and the distance of dipoles and is the main factor to the ultrasonic coefficient.

The monopole model and the dipole model have their limits. The two models are, respectively, according to the measurement in some experiments. In the monopole model, the nonlinear coefficient is related to the length of the dislocation string and stress. But in the dipole model, the nonlinear coefficient is related to the dislocation density and the distance of dipoles. Cantrell proposed a comprehensive model [26]. The total nonlinear coefficient is the summation of elastic anharmonicity and the acoustic nonlinearity contributed by the monopole and the dipole, which can be expressed as [26]

$$\beta = a_1 \cdot \frac{24}{5} \cdot \frac{\Omega\Lambda_{mp}L^4R^3E_2^2}{G^3b^2} \sigma + a_2 \left( \frac{16\pi^2\Omega\Lambda_{dp}h^2R^2(1-\nu)^2E_2^2}{G^2b} + \frac{384\pi^3\Omega\Lambda_{dp}h^3R^3(1-\nu)^3E_2^2}{G^3b^2} \right)$$  \hspace{1cm} (8)

where $a_1$ and $a_2$ are weight coefficients.

2.3. Dislocation Model for Third Nonlinear Coefficient. The total strain in material is as follows [25, 26]:

$$\varepsilon = \left( \frac{1}{E_2} + \frac{4\pi\Omega\Lambda_{dp}h^2R(1-\nu)}{G} \right) \sigma - \left( \frac{E_3}{2E_2} + \frac{8\pi^2\Omega\Lambda_{dp}h^3R^2(1-\nu)^2}{G^2b} \right) \sigma^2 + \frac{64\pi^3\Omega\Lambda_{dp}h^4R^3(1-\nu)^3}{G^3b^2} \sigma^3 + O(\sigma^4).$$  \hspace{1cm} (9)

Assuming specimen under a quasistatic loading, the exciting transducer creates a small oscillatory stress $\Delta\sigma$ of amplitude to the static stress $\sigma$. An additional strain $\Delta\varepsilon$ is created in response to the change in stress $\Delta\sigma$. It can be written as

$$\Delta\sigma = \frac{\partial\sigma}{\partial\varepsilon} \Delta\varepsilon + \frac{\partial^2\sigma}{\partial\varepsilon^2} (\Delta\varepsilon)^2 + \frac{\partial^3\sigma}{\partial\varepsilon^3} (\Delta\varepsilon)^3 + \cdots$$

$$= \left( \frac{\partial\varepsilon}{\partial\sigma} \right)^{-1} \Delta\varepsilon + \frac{1}{2} \frac{\partial^2\varepsilon}{\partial\sigma^2} \left( \frac{\partial\varepsilon}{\partial\sigma} \right)^{-3} (\Delta\varepsilon)^2 + \frac{1}{6} \frac{\partial^3\varepsilon}{\partial\sigma^3} \left( \frac{\partial\varepsilon}{\partial\sigma} \right)^{-4} (\Delta\varepsilon)^3 + \cdots$$  \hspace{1cm} (10)

The third order nonlinear coefficient $\delta_{dp}$ is

$$\delta_{dp} = \frac{\partial^3\varepsilon}{\partial\sigma^3} \left( \frac{\partial\varepsilon}{\partial\sigma} \right)^{-3}$$

$$= \frac{384\pi^3\Omega\Lambda_{dp}h^3R^3(1-\nu)^3}{\left(1/E_2 + 4\pi\Omega\Lambda_{dp}h^2R(1-\nu)/G \right) G^3b^2} + O(\sigma).$$

In most cases

$$\frac{1}{E_2} \gg \frac{4\pi\Omega\Lambda_{dp}h^2R(1-\nu)}{G}.$$  \hspace{1cm} (12)

Equation (11) can be written as

$$\delta_{dp} = \frac{384\pi^3\Omega\Lambda_{dp}h^4R^3(1-\nu)^3E_2^3}{G^3b^2}.$$  \hspace{1cm} (13)

3. Experimental Materials and Methods

3.1. Materials. The experimental material is GH4169 super-alloy. The hardness is 81HRB, and the yield limit is 365 MPa. The dimension of specimens is shown in Figure 3 in tensile testing. The dimension of specimens is 200 mm × 50 mm × 2 mm in fatigue bending testing.

3.2. Nonlinear Experimental System. Figure 4 shows the ultrasonic nonlinear testing system. It concludes RITEC advance measurement system, Tektronix DPO 4104B digital phosphor oscilloscope, exciting and received transducers, and computer. The center frequency of exciting transducer is 2.5 MHz and the one of received transducer is 5 MHz.
Table 1: Parameters for theoretical models.

<table>
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<th>1</th>
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<th>3</th>
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<tr>
<td>Stress (MPa)</td>
<td>0</td>
<td>50</td>
<td>100</td>
<td>150</td>
<td>200</td>
<td>250</td>
<td>300</td>
<td>350</td>
<td>400</td>
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<tr>
<td>(A_{mp} (\times 10^{14} m^{-2}))</td>
<td>8.64</td>
<td>2.14</td>
<td>1.87</td>
<td>1.56</td>
<td>1.28</td>
<td>1.01</td>
<td>0.72</td>
<td>0.44</td>
<td>0</td>
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<tr>
<td>(A_{dp} (\times 10^{14} m^{-2}))</td>
<td>0.45</td>
<td>0.45</td>
<td>0.52</td>
<td>0.83</td>
<td>1.11</td>
<td>1.39</td>
<td>1.68</td>
<td>1.96</td>
<td>2.40</td>
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The experimental curves of relative nonlinear coefficients \(\beta'\) and \(\delta'\) versus stress are shown in Figure 9. The error bars are obtained by repeating the measurements three times for each specimen at a certain stress level. The relative nonlinear coefficients \(\beta'\) and \(\delta'\) increase with stress when tensile stress is lower than 240 MPa. And they both decrease with stress when tensile stress is higher than 240 MPa. Therefore the plastic deformation begins to exist when tensile stress reaches about 240 MPa which is 65.8% of the yield strength. For the GH4169 superalloy at room temperature, when the plastic deformation occurs (\(\sigma > 240\) MPa), the dislocation pins break and the chord length increases. The dipole dislocation density increases with the stress [19, 26]. They cause the nonlinear coefficients decrease or changes irregularly with the tensile stress when the tensile stress is higher than 240 MPa. So the nonlinear coefficients are closely related to the plastic deformation.

Figure 9(a) shows the relationship of the second order relative nonlinear coefficient \(\beta'\) versus stress of the experimental data and that predicted by theoretical models. The models, elastic coefficient \(E = 200\) MPa, shear modulus \(G = 77.2\) MPa, Burgers vector \(b = 25\) nm, Poisson ratio \(\nu = 0.3\), the distance of dislocation \(h = 20\) nm, the dislocation length \(L = 25\) nm, \(a_1 = 0.85\), \(a_2 = 0.15\), and other parameters are shown in Table 1. The variation trend of experimental data is approximately in agreement with the predicted data by both the monopole model and the comprehensive model. But the experiment data is much closer by the comprehensive model than that by the monopole model.

Figure 9(b) shows the relationship of the third order relative nonlinear coefficient \(\delta'\) versus stress of the experimental data and that predicted by theoretical model. Because there is no stress value in (13), the theoretical model is not in good agreement with the experimental data.

4.3. Fatigue Bending Testing. The variation trends of ultrasonic nonlinear coefficients and harmonic amplitudes are observed in different fatigue cycles during fatigue bending testing. In order to search the influence of fatigue cycles on nonlinear phenomenon, measurements are taken at varying fatigue cycles. Measurements consist of many individual measurements at fatigue cycles that vary from 0 N to fracture in increments of 1000 N. The error bars are obtained by repeating the measurements three times for each specimen at certain fatigue cycles. The experimental curves of relative nonlinear coefficient \(\beta'\) and \(\delta'\) versus fatigue life are shown in Figure 10. The relative nonlinear coefficients \(\beta'\) and \(\delta'\) increase when the fatigue life is lower than 53.3%.
Figure 4: Block diagram of ultrasonic nonlinear testing system.

Figure 5: Time domain and amplitude frequency diagrams of exciting signal.

Figure 6: Time domain and amplitude frequency diagrams of received signal.
Table 2: Parameters for theoretical models.

<table>
<thead>
<tr>
<th>Number</th>
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<td>Fatigue life (%)</td>
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<td>12</td>
<td>24</td>
<td>36</td>
<td>48</td>
<td>60</td>
<td>72</td>
<td>84</td>
<td>100</td>
</tr>
<tr>
<td>$\Lambda_{mp}$ ($\times 10^{-4}$ m$^{-2}$)</td>
<td>8.24</td>
<td>2.04</td>
<td>1.67</td>
<td>1.51</td>
<td>1.08</td>
<td>0.91</td>
<td>0.72</td>
<td>0.42</td>
<td>0</td>
</tr>
<tr>
<td>$\Lambda_{dp}$ ($\times 10^{-4}$ m$^{-2}$)</td>
<td>0.48</td>
<td>0.50</td>
<td>0.54</td>
<td>0.93</td>
<td>1.31</td>
<td>1.59</td>
<td>1.76</td>
<td>2.01</td>
<td>2.52</td>
</tr>
<tr>
<td>Stress (MPa)</td>
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<td>16.35</td>
<td>28.71</td>
<td>35.02</td>
<td>42.03</td>
<td>44.37</td>
<td>46.7</td>
<td>49.50</td>
<td>51.37</td>
</tr>
</tbody>
</table>

Figure 7: Variation in relative nonlinear coefficient $\beta'$ as a function of propagation distance.

Figure 8: Stress versus time.

nonlinear coefficients $\beta'$ and $\delta'$ decrease when the fatigue life is higher than 53.3%.

Figure 10(a) shows the relationship of the second order relative nonlinear coefficient $\delta'$ versus fatigue life of the experimental data and predicted by theoretical model. $\delta'$ predicted by the theoretical data increases with the fatigue life, but there is an extreme value in the experimental data. The reason is that only dislocation is considered and no stress value is considered in theoretical model.

In elastic stage, the interior microstructure change with the fatigue cycles, the changes of monopole, and dipole dislocation cause the rising of ultrasonic nonlinearity. When the plastic deformation emerges, the monopole density decreases and veins and cells increase. They induce the decrease of nonlinear coefficient [25]. Therefore the relationship between nonlinear coefficient and fatigue life can be used to predict the fatigue state of metal components if the relationship is known.

5. Conclusions

During tensile testing and fatigue bending testing, microstructures change in superalloy. The monopole and dipole densities change with tensile stress or fatigue cycles. They induce the change of the nonlinear coefficients during the ultrasonic propagation.

Measurements are taken at varying stress. The results show that ultrasonic nonlinear coefficients increase with tensile stress in elastic stage, and they decrease in plastic stage. The relationship of the second order nonlinear coefficient as a function of tensile stress can be explained by the comprehensive model.

Measurements are taken at varying fatigue cycles. The results show that ultrasonic nonlinear coefficients increase with the fatigue cycles in elastic stage, because the monopole and dipole densities change with fatigue cycles. The ultrasonic nonlinear coefficients decrease in plastic stage because the monopole density decreases and veins and walls appear. The relationship of the second order nonlinear coefficient as a function of fatigue life can be explained by the comprehensive dislocation model. Therefore, the variation trend of nonlinear coefficient during fatigue testing can be used to predict the fatigue state of materials if the relationship between the nonlinear coefficient and the fatigue cycles is known.

We promote a model of the third order nonlinear coefficient $\delta$. It is used to predict the relationship between the third order nonlinear coefficient $\delta$ and tensile stress or fatigue life. Having no regard for the stress, the theoretical model does not accord with the experimental data. The future work will engage in finding better models to explain the experimental data.
Advances in Materials Science and Engineering

Conflict of Interests
The authors declare that there is no conflict of interests regarding the publication of this paper.

Acknowledgment
The research is supported by elastic solid residual stress field in situ acoustic energy control mechanism (no. 51275042), which is financed by NSFC of China.

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