A Novel Inward Gradient Self-Lubrication Layer with Soft Alloys and Its Lubricating Mechanism

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A novel ceramic composite inward gradient distribution layer has been developed. The layer is a lubricating layer in which soft-metal lubricants are compounded into the ceramic matrix by high frequency induction infiltrating method. The design of the layer and its lubricating mechanism are investigated in the paper. The results show that the property of the layer greatly depends on the wetting angle of the soft-metal lubricants on the matrix and the proportion of Ag, Cu, Sn, and Pb as well as the infiltrating parameters. Based on a lot of experiments, a novel inward gradient layer with Pb28Sn19Ag6Cu has been developed. The layer has an excellent lubricating property (friction coefficient about 0.2∼0.3 at 600°C). The research reveals the lubricating mechanism, observing the phenomenon that the soft-metal in the matrix diffuses out of the frictional surface, and measures the lubricating film thickness as about 20μm on the worn surface.

1. Introduction

Self-lubricating coatings are widely introduced to solve wear and lubrication problems in extreme abrasion conditions [1–3]. However, the unsoundness of the bonding strength between coating and the substrate always puzzles the engineering fields and limits its application range, especially in extreme abrasion conditions. Therefore, besides lubricating properties of coating, studies on composite technology, novel lubrication theories, and preparation processes are laying increasing emphasis on solving the bonding strength problem [4, 5].

Recently, Liu and Wang have developed a kind of novel porous metal cermet with orderly micropores structure [6, 7]. The interpenetrating network structure of the cermet allows lubricants, such as soft metal, to be infiltrated into the ceramic matrix and form an inward gradient lubricating layer, so that the tribological property of the ceramic composite can be greatly improved due to the combination of high strength at high temperature, wear resistance of the ceramic matrix, and self-lubricating characteristics. Particularly, Liu studied the diffusing mechanisms of the porous cermet infiltrating the single soft-metal and effect of some operating variables on its tribological properties [7]. However, when the soft alloys with different elements are mixed and infiltrated into the porous cermet at elevated temperature, a new material structure and some interesting lubricating phenomenon also existed. In fact, it can be observed that the orderly micropores structure of material can be used to develop a new inward gradient lubricating layer by design of the mixed soft alloys and infiltrated them into the porous cermet; a new inward gradient lubricating layer can be formed. In the paper, its design of infiltration agent and lubricating mechanisms are discussed.

2. Experiments

2.1. Preparation of Porous Matrix. To fabricate the metal ceramic matrix with orderly micropores structure, a mixture of 74.1 vol.% Fe-Cr-W-Mo-V metal alloy powder and
18.5 vol.% TiC powder together with 7.4 vol.% complex pore-forming agent was milled for 2 h and then was cold-compressed into a mold at 600 MPa and was sintered at about 1230 °C in a vacuum furnace [8]. The specimen of the porous matrix and its microstructure are shown in Figure 1.

2.2. Wetting Experiments. A specimen of the ceramic matrix with diameter of 58 mm and thickness of 8 mm dealing with ultrasonic cleaness is used as a substrate. Then a small molten alloy block with different proportion of soft metals Ag, Cu, Sn, and Pb is put on the surface of the specimen. After the alloy block was melted again in a vacuum high frequency induction furnace at 700 °C for 30 minutes and cooled down, the wetting angle between the alloy and the ceramic matrix was measured.

2.3. Infiltration Experiments. To fabricate the inward gradient lubricating layer, the porous cermet, which was soaked in the B$_2$O$_3$-KF infiltration agent solution, and the lubricating alloy were put into a graphite crucible and then were rapidly heated to 700 °C in a vacuum high frequency induction furnace. After the lubricants completely melted, the pressure of the furnace was raised to 0.3 MPa by filling nitrogen. Holding the temperature and pressure for 30 mins, the composite was formed.

2.4. Friction Experiments. A test rig was developed for purpose of hot wear up to 600 °C. The experiment was carried out in the test rig (as shown Figure 2). The pin (4) and disc (5) specimens are placed at the center of a wire-wound tube furnace (3) temperature range of which can be changed from the room to 800 °C by a heating instrument (2). The pin and disc are mounted in holders at the end of stainless sinter tube (1) filled with insulating ceramic fiber. The disc specimens have the shape of washers and are secured to their holder (6) by a bolt through their central hole. The friction force and dynamic loading force are measured directly by force sensors (7) and (8), respectively. Based on measured statistics, frictional coefficient can be calculated and printed out in real time by using a microprocessor controlled data acquisition system.

The pins of the composite infiltrated with different lubricating alloy were tested and an alumina (Al$_2$O$_3$) ceramic disc was used as counter face of these pins. The pins, 15 mm long, of 12 mm diameter, were rounded to hemispherical shape with diameter of 8 mm at one end. The disc was 58 mm diameter and 8 mm thick with an average hardness of HV 19.42 GPa (digital microhardness tester HVS-1000, 5 times) and a final surface roughness of 0.32 μm on the counter surface. The friction tests were conducted at 600 °C with a sliding velocity of 0.2 m/s and a load of 50 N.

A scanning electron microscope (SEM, HITACHI X-650) equipped with an energy dispersive spectrocope (EDS, Kevex Super Quantum) was used to observe the wear surfaces of the pin and disc and the cross section of the self-lubrication layer to clarify the lubrication mechanisms.

3. Results and Discussion

For the purpose of obtaining an excellent tribological inward gradient lubrication layer, a series of research have been done,
mainly on the design of wetting property between the soft alloys and the ceramic composite and lubricating design of the soft alloys.

3.1. Design of Infiltration Agent. When the ceramic matrix containing Cr, Mo, W, and Ti was exposed to air at elevated temperature, these elements can easily be oxidized forming an oxide film on the ceramic. The oxide film will deteriorate the wetting property of the molten soft alloy on the matrix surface. To solve the problem, two gases and infiltration agents were used in the design.

Figures 3(a), 3(b), and 3(c) show, respectively, the wetting test results of molten Pb30Sn20Ag on the ceramic matrix surface in N2, H2, and B2O3-KF agent. It is observed that the alloy has poor wetting property on the ceramic matrix surface in N2 (as shown in Figure 3(a)) and H2 (as shown in Figure 3(b)) at 550°C, but when the B2O3-KF agent is used in the experiment, the wetting property is markedly improved (as shown in Figure 3(c)).

Obviously, the addition of infiltration agent in the preparation process of the gradient lubrication layer effectively optimizes the wetting property of the lubricants on the matrix, so that the soft alloys can easily be infiltrated into the micropores of the ceramic matrix [9].

3.2. Wetting Design of the Soft Alloys. The melting points of soft alloys such as Ag and Pb are lower. When they are used in inward gradient layer, the infiltrating temperature should be much lower than the phase-transition temperature of the matrix, so that the strength of the ceramic matrix does not decrease in the infiltration process. On the other hand, oxides of certain soft alloys and their metal salts reacting with W and V also have excellent high temperature properties. Thus, when they are used to infiltrate into the micropores of the ceramic matrix, the inward gradient layer could have excellent high temperature lubricating properties.

However, the wetting design of the soft alloys is complicated. Some of them such as Pb and Ag have poor wetting property and solubility with Fe, which is the major component of the matrix material, and others have better reactivity with Fe such as Sn [10, 11]. Therefore, it is necessary to design the soft alloys based on phase diagrams theory to improve the wetting property when they are infiltrated into the matrix materials. In the research, the design of the soft alloys component is based on phase diagrams of Fe-Pb, Fe-Ag, and F-Sn and lots of experiments. For the purpose of making the Sn element effectively improve the wetting ability of the soft alloys and Fe, and limiting its high SnO2 resulting from high temperature (SnO2 has a poor lubricating properties), the content of Sn element is always controlled lower than 30 wt%. On the other hand, Cu is added to improve the wetting property of the soft alloys due to its high mutual solubility with Fe. Thus, several main designs are considered in the research: only Pb, Pb30Ag, Pb30Sn20Ag, and Pb28Sn19Ag6Cu. Their wetting properties with the ceramic matrix tested at 550°C are as shown in Figure 3. The spreading appearance of the soft alloys designed in Figure 3 shows that Sn and Cu can greatly improve wetting ability of the soft alloys with the ceramic matrix and the design of the Pb28Sn19Ag6Cu presents an excellent wetting property of which the wetting angle is only about 5°. Based on the discussion above, a series of combinations of the soft alloys, the wetting angle measurements and lubricating experiments of them have been done with the results as shown in Figure 4.

3.3. Structural Characteristics of the Lubricating Layer. Figure 5(a) is the cross section of the inward gradient lubricating layer. It reflects the structural characteristics of the layer. The white points in the section indicate the soft alloy phase and the gray area represents the matrix phase of the ceramic composite. Obviously, an inward gradient lubrication layer is formed when the soft alloys are infiltrated into the porous matrix. As a typical example, the distribution of element Sn of soft alloys in the layer is shown in Figure 5(b).

This indicates that the lubricating elements formed an inward gradient distribution in the material and the composition design of the soft alloys is very important in controlling the thickness of the gradient lubricant layer. Therefore, it is necessary to design the thickness of the layer or control the diffusing rate of the soft alloys so that there are enough elements of soft alloys for consumption during the operating life of the frictional pairs. Fortunately, the work has been done in another paper through controlling the composition.
3.4. *Lubrication Mechanism of the Composite.* Figure 6 shows the friction curve of the ceramic composite with inward gradient layer at 600°C. The curves (A–D) in Figure 6 present the friction behaviors and time of the composites infiltrated with Pb28Sn19Ag6Cu, Pb30Sn20Ag, Pb20Sn20Ag10Sb, and Pb13Au20Cu15Sn. Obviously, the ceramic composite with Pb28Sn19Ag6Cu layer has a better lubrication performance of which the friction coefficient is lower than 0.3. Comparing
Figure 6: Friction coefficients of ceramic composite with gradient layer infiltrated with different soft alloys on A_2O_3 disk.

Figure 7: Friction coefficient curve of the gradient layer infiltrated with Pb28Sn19Ag6Cu sliding against Al_2O_3.

Figure 8: The diffusing process of the soft alloys from the inward gradient layer.
Figures 4 and 6, it is evident that the excellent wetting and mutual solubility of Pb28Sn19Ag6Cu with matrix material plays an important role in formation of the lubricating film and its spreading out on the frictional surface.

In order to investigate the lubricating mechanism, the typical friction curve of ceramic composite with Pb28Sn19Ag6Cu sliding against Al2O3 disc at 600°C is presented in Figure 7. In the beginning stage of the friction process, the friction coefficient is very low due to the fact that elements of soft alloys in the layer are drove out to the worn surface at high temperature. In fact, the frictional couple is actually in quasi-liquid lubrication in the case. Then, as the elements spread out but not to the extent that is enough to form a perfect lubricating film on the frictional surface, the friction coefficient increases over time. With more and more lubricant coming out in the sliding process, a perfect lubricating film is formed on the worn surface and the friction coefficient decreases and keeps on a stable value about 0.2.

Figures 8(a), 8(b), and 8(c) show the diffusing process of the soft alloys from the ceramic composite to the inward gradient layer at high temperature friction test. It can be observed that the beads shape particles sweat out to the worn surface from the pores of the gradient layer. The major components of the particle include Ag, Sn, and Pb. Obviously, these soft alloy elements are driven out to the friction surface by the heat expend factor and frictional-thermal stress at the high temperature sliding condition. Thus
the diffusing mechanism provided excellent lubrication for fictional surface [13].

Figure 9 is a typical example of friction of the ceramic composite with inward gradient layer on the Al₂O₃ disc. In the experiment, the soft alloys formed a perfect lubricating film on the frictional surface (as shown in Figure 9(a)) and elements of the soft alloys in the layer such as Ag, Cu, Sn, and Pb are present in the film (as shown in Figure 9(b)). It is evident that the soft alloys have melted at high temperature friction and mixed together to form a lubricating film because of their low melt points.

Figures 9(c), 9(d), and 9(e) show the transferring lubrication film forming on the disk worn surface. The thickness (about 20 μm) of the film is measured by coordinate measuring instrument with the results shown in Figure 9(d). It is obvious that the film has a strong adhesive force on the disk surface and an excellent anti-high temperature property. Obviously, the formation of lubrication film on the worn surface both of the composite and the disc gives the material an excellent lubricating property at elevated temperature.

4. Conclusions

The conclusions were as follows:

(1) By using B₂O₃-KF infiltration agent and adding Sn and Cu, the wetting property of the molten lubricants on the ceramic matrix is greatly improved. For example, the wetting angle of the molten Pb₂₈Sn₁₉Ag₆Cu on the matrix composite decreased to about 5°.

(2) Based on the research above, a novel inward gradient lubricating layer with Pb₂₈Sn₁₉Ag₆Cu is made which has an excellent lubricating property at 600°C. The friction coefficient of the ceramic composite is about 0.2 at 600°C.

(3) The lubrication mechanism of the composite is investigated by using SEM and EDX techniques in the research. In the high temperature sliding process, the lubricants in the lubricating layer diffusing out under the thermal and contact stress and forming lubricating film on the worn surface both of the composite and disc are the key contribution to the excellent high temperature lubricating performance of the composite.

Competing Interests

The authors declare that they have no competing interests.

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