SOME INVESTIGATIONS ON THE ANISOTROPY OF THE CHEMICAL ETCHING OF (hk0) AND (hhl) SILICON PLATES IN A NaOH 35% SOLUTION. PART I: 2D ETCHING SHAPES

C. A. HODEBOURG\textsuperscript{a} and C. R. TELLIER\textsuperscript{b,\*}

\textsuperscript{a}Laboratoire de Chronométrie Electronique et Piezoelectricité, Ecole Nationale Supérieure de Mécanique et des Microtechniques, 26 chemin de l'Epitaphe, 25030 Besançon cédex France;

\textsuperscript{b}Institut des Microtechniques de Franche-Comté, Avenue de l'Observatoire, 25030 Besançon cédex France

(Received 1 February 2001; In final form 19 March 2001)

In this paper a study of the anisotropic dissolution of (hk0) and (hhl) silicon plates in a NaOH 35% solution is undertaken. Effects of orientation on firstly, the geometrical features of etched surfaces and secondly, on the cross-sectional shape of starting circular plates are systematically investigated. Conclusions of practical interest on the roughness of etched (hk0) and (hhl) planes are drawn. 2D etching shapes are then analysed in terms of the tensorial model for the anisotropic dissolution and of dissolution criteria. Finally a comparative analysis of results related on the one hand, to 2D surface profiles and on the other, to out-of-roundness profiles is made. This comparison shows that shapes observed for profilometry traces agree with theoretical shapes as derived when we use the resemblance in shapes between out-of-roundness profiles and polar diagrams of the dissolution slowness.

Keywords: Anisotropic etching; Silicon; NaOH etchant

\*Corresponding author. Tel.: 03-81-40-28-30, Fax: 03-81-88-57-14, e-mail: ctellier@ens2m.fr
1. INTRODUCTION

In the last decade a great lot of works [1–9] was devoted to the study of various etching baths dedicated to the micromachining of silicon crystal. To obtain with reproducibility mechanical microstructures of definite shape and size it is advantageous to use highly anisotropic etchants. So it is usual to work with etching baths which possess the particularity to dissolve very slowly \{111\} silicon planes [1–7]. EDP and KOH based solutions [1–5] are currently used but in the past few years new etching baths such as TMAH [5–7], Cesium hydroxide [8] and NaOH + IPA solutions [9] were investigated. This paper continues on this line and reports some results on the anisotropy of a pure NaOH 35% solution. The paper is divided into two parts, the first part is mainly concerned with 2D etching shapes such as surface profiles and out-of-roundness profiles corresponding to cross-sections of etched thick silicon plates whereas the second part deals with 3D structures micromachined in various (hk0) and (hhl) silicon planes. A practical point of view is adopted in the two parts of this paper and emphasis is placed on technical problems encountered in wet micromachining processes such as the degradation of etched surfaces, the identification of facets limiting microstructures, the importance of concave or convex undercutting and the formation of terraces on edges bounding membranes and mesa.

2. EXPERIMENTAL DETAILS

To study the changes in starting 2D shapes induced by chemical etching differently oriented thick circular silicon plates (10 mm in diameter, initial thickness \(d_0 = 1.5\) mm) were cut in a silicon ingot. The angles of cut as defined (Fig. 1) in the IEEE Standard on Piezoelectricity [10] are, for the various \(\{hk0\}\) and \(\{hhl\}\) plates, listed in Table I. The two sides and the contour of plates were firstly mechanically lapped and then polished (initial arithmetic roughness \(Ra \approx 0.01\ \mu m\)). All plates were immersed in a 35% in weight NaOH solution maintained at a constant (±0.5°) temperature of 60°C. After
isothermal etchings of duration $\Delta t = 1$ h following investigations were performed:

(1) Decrements in thickness $\Delta d$ were measured using a Palmer instrument.

(2) Modifications of the contour shape of starting circular plates were investigated using a Talyrond analyser which furnishes an out-of-roundness profile superimposed to the least mean square circle.
Geometrical features of etched surfaces were observed by Scanning Electron Microscopy. In addition a computer based mechanical profilometer gives 3D topographies of surfaces and 2D profilometry traces made along two specific directions. For the \{hk0\} plates the two directions are the rotated \(x'_1\) axis (Fig. 1) and the [001] direction. For the \{hhl\} plates traces lie parallel to the primed \(x'_2\) axis and to the (110) direction. Roughness parameters such as the centre-line average \(Ra\), the root-mean-square roughness \(Rq\) and the peak-to-valley height \(R_{max}\) were evaluated.

3. EXPERIMENTAL RESULTS

3.1. The Etch Rate

For the silicon crystal which belongs to the class \(m\overline{3}m\) the dissolution rates for the two faces of a \{hk0\} or of a \{hhl\} plate are similar. Consequently the dissolution rate \(R\) for a face is simply given by:

\[
R(\varphi_o, \theta_o) = \frac{1}{2} \Delta d
\]

where \(\Delta d\) is the decrement in thickness.

Variations in the etch rate \(R\) with the angle of cut \(\varphi_o\) (\{hk0\} plates) or with the angle \(\theta_o\) (\{hhl\} plates) are drawn in Figure 2. Turning attention to \{hk0\} plates we observe that \{100\} plates (\(\varphi_o=0^\circ\)) dissolve more slowly than \{110\} plates (\(\varphi_o=45^\circ\)). Moreover the maximum of the etch rate \(R(\varphi_o)\) occurs for \(\varphi_o \approx 26^\circ\) that is to say for \{120\} planes. So for \{hk0\} planes the anisotropy is crudely similar to those observed for EDP [1, 2, 3], TMAH [6] and KOH [1, 3] etchings. The variations in the etch rate \(R(\varphi_o = -45^\circ, \theta_o)\) with the angle of cut \(\theta_o\) show as expected a marked minimum for the (111) plane and a minor minimum for the (001) plane (\(\theta_o=90^\circ\)). Moreover for (hhl) planes, the etch rate presents two maxima: a major maximum for \(\theta_o \approx 65^\circ\) and a slightly less accentuated minimum for \(\theta_o = 0^\circ\) (i.e., for a \{110\} plane). In terms of the dissolution slowness [11, 12] we thus conclude that certainly the dissolution slowness surface exhibits in the one hand, valleys associated with planes close to \{113\} planes and in the second hand, two types of protuberances. Elongated protuberances
correspond to \{111\} planes whereas small peaks appear for \{100\} planes. Comparison of changes in \(R(\phi_o)\) and in \(R(\phi_o = -45^\circ, \theta_o)\) for cuts in the vicinity of the (110) plane indicates that \{110\} planes do not correspond to a minor peak in the dissolution slowness as observed for various etching baths such KOH 35\% [3] and TMAH 25\% [6].

3.2. The Topography of Etched Surfaces

SEM images of \{hk0\} and \{hhl\} etched surfaces are displayed in Figures 3 to 6. Etched surfaces are covered by dissolution figures.
characteristic of the cut. For all the \{hk0\} plates investigated here the dissolution figures are aligned along the rotated axis $x'_1$, but the more elongated dissolution figures develop on plates close to \{100\} plates or to \{120\} plates. When the angle of cut is in the range $34^\circ - 45^\circ$ the dissolution figures seem to be partly limited by \{111\} facets and partly composed of small hillocks. Moreover contrary to other etchings the dissolution of \{110\} and \{980\} plates in a NaOH 35% solution causes the formation of somewhat pyramidal figures. Typical features of etched \{hk0\} plates are listed in Table II.
Let us now turn attention to (hhl) plates and recall that when the angle of cut $\theta_0$ increases from 0° ((110) plane, Fig. 6D) to 90° ((001) plane, Fig. 5a) the (hhl) plane passes through the (111) plane. Examination of SEM images of Figures 5 and 6 reveals that:

1. Etching causes dissolution figures to appear whose shape changes progressively with the angle of cut $\theta_0$. For convenience Table II summarises this evolution in shape.
2. Dissolution figures which develop on (hhl) plates seem to be more ((551) plane) and less ((112) plane) elongated in a direction $x'_3$, which is perpendicular to the direction (110).
(3) Most of dissolution figures (see Figs. 5b, 5c, 5d for example) appear to be symmetrical with respect to the cross-sectional \{110\} plane.

(4) \((hhl)\) plates in the vicinity of the \((001)\) plane \((90^\circ \geq \theta_o > 62^\circ)\) show the less degraded surfaces. In particular smooth elongated bumps are formed on etched \((114)\) and \((113)\) surfaces. When \(\theta_o\) decreases from \(55^\circ\) to \(10^\circ\) \(\{111\}\) terraces with steps partly covered by small "pyramidal" hillocks are generated yielding surfaces of poor quality.

3.3. Profilometry Traces

Since the dissolution figures are aligned along specific directions which for \((hk0)\) and \((hhl)\) plates correspond to the rotated \(x'_1\) axis and \(x'_3\).
FIGURE 6 SEM images of etched (hhl) plates, A, B, C and D are for (100), (117), (114) and (112) planes respectively. Arrows indicate the direction of the rotated axis $x'_1$ ((110) direction).

TABLE II Geometrical features of dissolution figures formed on (hk0) and (hhl) planes

<table>
<thead>
<tr>
<th>Planes</th>
<th>Geometrical features</th>
<th>Planes</th>
<th>Geometrical features</th>
</tr>
</thead>
<tbody>
<tr>
<td>(010)</td>
<td>Flat pits</td>
<td>(551)</td>
<td>Sharp hillocks</td>
</tr>
<tr>
<td>(610)</td>
<td>Somewhat concave terraces</td>
<td>(441)</td>
<td>Sharp hillocks</td>
</tr>
<tr>
<td>(410)</td>
<td>Terraces</td>
<td>(331)</td>
<td>Terraced hillocks</td>
</tr>
<tr>
<td>(730)</td>
<td>Elongated bumpy hillocks</td>
<td>(221)</td>
<td>Terraces</td>
</tr>
<tr>
<td>(740)</td>
<td>Bumpy hillocks</td>
<td>(112)</td>
<td>Terraced surface</td>
</tr>
<tr>
<td>(320)</td>
<td>Terraced hillocks</td>
<td>(113)</td>
<td>Very flats elongated hillocks</td>
</tr>
<tr>
<td>(430)</td>
<td>Terraced hillocks</td>
<td>(114)</td>
<td>Very flats elongated hillocks</td>
</tr>
<tr>
<td>(990)</td>
<td>Some pyramidal hillocks present</td>
<td>(115)</td>
<td>Somewhat convex-concave pits</td>
</tr>
<tr>
<td>(110)</td>
<td>Pyramidal hillocks present</td>
<td>(117)</td>
<td>Elongated pits</td>
</tr>
</tbody>
</table>

respectively it is very interesting to study changes in shape of profilometry traces made along these directions and along the directions which are perpendicular to directions of alignment (i.e., (110) direction for (hhl) planes and [001] direction for (hk0) plates).
Figures 7 to 10 illustrate the changes in shapes of these traces with the angles of cut. Several features emerge from these figures:

1. The evolution of the shape for $x'$ traces appears to be progressive. As the plate is rotated from (110) plate to the (001) plate the shape is progressively modified from convex for the (110) plate to concave for the (001) plate. For (hhl) plates corresponding to intermediate orientations ($6^\circ < \theta_o < 35^\circ$ and $36^\circ < \theta_o < 81^\circ$) the $x'$ trace exhibits an alternate concave-convex shape ((111) plates) and convex-concave shape ((hhl) plates).
Let us recall that to analyse the shape of a 2D surface profile we have to consider the polar diagram of the dissolution slowness \( L \) [12, 13] lying in the cross-sectional \( \{110\} \) plane which intersects the \( (hhl) \) plane along the \( x'_{3} \) axis. Consequently the observed changes in shape can be easily explained if we keep in mind that when the angle of cut \( \theta_{o} \) varies from 0° to 90° the dissolution slowness vector \( L \) of an \( (hhl) \) plane moves along the polar diagram corresponding to the \( (110) \) plane.

2) An unlike behaviour can be depicted (Fig. 8) for \( (110) \) traces made on \( (hhl) \) planes. Effectively in this case various cross-sectional planes concern us so that no progressive changes in shape can occur. In fact all \( (110) \) traces show symmetrical shapes with respect to the vertical axis. Hence these traces satisfy the mirror symmetry specific to \( \{110\} \) planes since the cross-sectional section of a given \( (hhl) \) plane is necessary an other \( (h'h'l') \) plane.

3) Turning now attention to traces made on \( (hk0) \) plates it clearly appears that \( x'_{1} \) traces follow also a progressive evolution of shape according to the scheme (Fig. 9):

Concave(\( \{100\} \)planes) \( \rightarrow \) concave - convex \( \rightarrow \) convex

(\( \{120\} \)planes) \( \rightarrow \) convex - concave

(\( \{320\} \)to\( \{980\} \)planes)

As in remark (1) we attribute this evolution to the fact that for \( (hk0) \) plates the final etching shapes of the 2D \( x'_{1} \) traces are all determined from the \( (001) \) polar diagram of the dissolution slowness [2].

4) The behaviour for \( [001] \) traces made on \( (hk0) \) plates (Fig. 10) is close to that observed in (2) for \( (110) \) traces on \( (hhl) \) planes with firstly, not any changes in shape with the angle of cut \( \varphi_{o} \) and secondly, with profiles that verify the mirror symmetry associated now with the \( (001) \) plane.

3.4. Surface Roughness

Several works [14, 15] have given evidence that four-pole piezoresistive elements diffused on \( (hhl) \) planes may be conveniently used to measure
three in-plane stresses (uniaxial stresses $T'_{11}$ and $T'_{33}$, shear stress $T'_{13}$). For such applications mechanical devices must be micromachined in (hhl) planes. Consequently it is interesting to study the roughness of etched surfaces which can significantly limit the metrological performances of mechanical sensors.

For this purpose roughness parameters $R_a$, $R_q$, and $R_{max}$ were evaluated from $x_1$ and (110) traces made on (hhl) planes. Variations in the roughness parameters $R_a$ and $R_q$ (in dotted lines) as a function of
FIGURE 9  $\chi'_{1}$ profilometry traces made on various (hk0) plates. Values of the angle of cut $\varphi_0$ are indicated on diagrams. For convenience arrows show the $\chi'_{1}$ direction with respect to the direction of the trace.

the angle of cut $\vartheta_0$ are drawn in Figure 11. Note that these plots which are fitted on an average behaviour give evidence for a marked degradation of etched surfaces when (hhl) planes approach the (110) plane. We effectively observe in the one hand, that roughness parameters decrease when the (hhl) plane is rotated toward the (001) plane and in the other hand, that (111) surfaces become rougher and rougher as the miller index 1 takes higher values. In particular for the
etched (110) surface a maximum value of 1 \( \mu m \) is obtained for the roughness parameter \( R_a \). For (hhl) cuts in the vicinity of the (110) plane the peak-to-valley parameter \( R_{max} \) reaches values of about 5 \( \mu m \), so these cuts must be avoided for numerous micromachining applications.

Figure 12 gives complementary results on the roughness of etched (hk0) plates. We observed that for etched \{hk0\} plates the most degraded surfaces are obtained for cuts close to (320) plate for which the \( R_{max} \) parameter (curve in dotted line) takes values higher than 4 \( \mu m \).
FIGURE 11 Variations in the roughness parameters $R_a$ and $R_q$ as a function of the angle of cut $\theta_0$ of (hhl) planes. A and B are for traces made along the (110) direction and the $x'_3$ axis respectively. Full lines and dotted lines are for the parameters $R_a$ and $R_q$ respectively. In C is shown the evolution of the parameter $R_{\text{max}}$ where here full lines and dotted lines are for the $x'_3$ and (110) $x'_3$ traces respectively.

FIGURE 12 Changes in the average roughness parameters $R_a$, $R_q$ and $R_{\text{max}}$ with the angle of cut $\varphi_0$ of etched (hk0) plates.
4. CROSS-SECTIONAL SHAPES OF STARTING THICK CIRCULAR PLATES

In this section we use out-of-roundness profiles to study the modification of the cross-section of starting circular plates. First consider the out-of-roundness profiles related to etched (hk0) plates (Fig. 13). The arrows on Figure 13 indicate the [001] direction, which are common to all (hk0) plates. The most significant and rapid changes (Figs. 13a to 13c) in the shape of out-of-roundness profiles are obtained when the angle of cut \( \phi_o \) varies from 0° ([010] plane) to 14° ([140] plane). In contrast the out-of-roundness profiles look crudely similar (Figs. 13d to 13f) for etched (hk0) plates whose angle of cut \( \phi_o \) is in the [34°–45°] range. For the (110) out-of-roundness profile the four more pronounced peaks lie in the \( \{111\} \) directions. Hence as observed [1–9] for other etching baths (EDP, KOH, TMAH) the \( \{111\} \) planes are certainly associated with large protuberances of the dissolution slowness surface. It is also convenient to examine

![Figure 13](image-url)

**Figure 13** Orientation effects in the final out-of-roundness profiles related to various (hk0) sections. a, b, c, d, e and f are for \( \phi_o = 5^\circ, \phi_o = 10^\circ, \phi_o = 14^\circ, \phi_o = 34^\circ, \phi_o = 37^\circ, \) and \( \phi_o = 45^\circ \) respectively.
out-of-roundness profiles (Fig. 14) as obtained after a prolonged etching of the circular (hhl) plates. Arrows now correspond to the (110) direction. On Figure 14 we again depict two different behaviours for the evolution of the profiles with the angle of cut $\theta_o$. A little more rapid evolution is observed for (111) out-of-roundness profiles whereas the shape of out-of-roundness profiles remains quite unchanged for (hhl) cross-sections. In fact the (111) plane constitutes a boundary plane and the more marked changes in shape occur when the (hhl) plane passes through the (111) plane (compare the (112) and (221) out-of-roundness profiles for example). Moreover (hhl) out-of-roundness profiles exhibit shapes quite similar to those (Figs. 13d to 13f) obtained for (hk0) plates in the vicinity of the (110) plate.

FIGURE 14 Orientation effects in the final out-of-roundness profiles related to various (hhl) sections as indicated on diagrams.
5. ANALYSIS OF EXPERIMENTAL RESULTS

5.1. Theoretical Basis of the Analysis

In the following we analyse the experimental etching shapes in terms of the tensorial model for the anisotropic dissolution of crystals [11, 12]. Hence let me recall that in the framework of this model to each surface element $ds$ of orientation $(\phi_0, \theta_0)$ we associate a dissolution slowness vector $L$ whose orientation is that to the unit normal $n$ to the surface element and whose magnitude $L(\phi_0, \theta_0)$ is equal to the reciprocal of the etch rate $R$. When the angles of cut vary the vector $L$ describes in space a representative surface called the dissolution slowness surface. In the tensorial model the equation of the dissolution slowness surface involves dissolution constants which in fact are the independent coefficients of tensors [11]. The interest of this model lies in the two following properties:

(1) It is possible [16] to calculate the displacement $P$ of each element $ds$ within the crystal during the dissolution from the equation of the dissolution slowness surface. Thus it is possible to construct numerically 2D and 3D etching shapes from a simulation tool called TENSOSIM [17–19] provided the database (composed of dissolution constants) was determined from experiments.

(2) The tensorial model satisfies the dissolution criteria as stated several years ago by Irving [20]. So if the starting shape is concave the final etching shape must be composed of surface elements corresponding to maxima in the dissolution slowness. Conversely for a starting convex shape minima of $L$ determine the final etching shape. Note that in the case of a 2D etching shape such as a 2D surface profile or a 2D cross-section of a cylindrical crystal we only consider extrema in the polar diagram of $L$ lying in the corresponding cross-section [13, 19]. But for a 3D etching shape we have to consider extrema associated with peaks and valleys in the dissolution slowness surface [17, 18]. It should be also pointed out that approximate 2D etching shapes can be easily deduced from geometrical constructions [3] involving only profile elements associated with extrema in the polar diagram (minima for a convex shape, maxima for a concave shape).
5.2. Analysis of 2D Etching Shapes

For a comprehensive analysis of 2D etching shapes the dissolution slowness must be known. Fortunately previous works [12, 13, 19] have shown that an out-of-roundness profile constitutes a crude image of corresponding the cross-sectional polar diagram of \( L \). In particular very small departures are obtained between angular positions \( \Psi_e \) and \( (\Psi_v, \Psi_p) \) related respectively to extrema \( L_e \) in the polar graph of \( L \) and to valleys (v) and peaks (p) of the out-of-roundness profile. So we can conveniently use the out-of-roundness profiles of Figures 13 and 14 to analyse the experimental traces. But first of all since out-of-roundness profiles offer images of successive sections of the dissolution slowness surface careful examination of Figures 13 and 14 allow us to drawn some primarily conclusions that agree with remarks made in Section 3.1:

1. Prolate protuberances are aligned along \( \langle 111 \rangle \) directions. The influence of these accentuated protuberances extend on large angular sectors that is to say on the ranges \( 19^\circ < \theta_o \leq 0^\circ \) for \( \langle 111 \rangle \) planes and \( 30^\circ \leq \varphi_o \leq 45^\circ \) for \( \langle h k 0 \rangle \) planes.

2. Minor maxima in the dissolution slowness surface occur for \( \{100\} \) planes. Effectively all \( \langle h k 0 \rangle \) out-of-roundness profiles show peaks in the \( [001] \) direction.

3. If we turn attention to the \( \langle 110 \rangle \) plane it is difficult to conclude because in the corresponding \( [110] \) direction the \( \{100\} \) and \( \{110\} \) out-of-roundness profiles exhibit respectively peaks and valleys.

4. Minima which on the \( \langle 110 \rangle \) profile (Fig. 13f) appear for directions close to \( \langle 113 \rangle \) directions seems to be associated with \( \{113\} \) valleys of the dissolution slowness surface because minima are also observed on out-of-roundness profiles when we move away from the \( \langle 110 \rangle \) section (see Figs. 13d and 13e).

If we now apply the previous model to the analysis of 2D surface profiles it is clear that we have to consider the cross sectional polar diagrams indicated in Table III and that on these polar diagrams we use only an angular sector \( (-\alpha_{\text{max}}, +\alpha_{\text{max}}) \) centred on the dissolution slowness of the reference plate. From Table III it appears that \( x'_3 \) and \( x'_1 \) traces must show a progressive evolution as pointed out in Section 3.3. The experimental shapes fulfil this requirement. Now can use the
TABLE III Identification of cross-sectional polar diagrams of L

<table>
<thead>
<tr>
<th>(hk0) planes</th>
<th>(hhl) planes</th>
</tr>
</thead>
<tbody>
<tr>
<td>x'₁ trace: (001) plane</td>
<td>x'₁ trace: (110) plane</td>
</tr>
<tr>
<td>[001] traces: The same (hk0) plane*</td>
<td>(110) traces: specific (h'h'l') planes for (hhl) and (111) planes**</td>
</tr>
</tbody>
</table>

*Owing to the symmetry.
**For example the correspondences (118) ↔ (441), (111) ↔ (112), (114) ↔ (221) can be easily established.

out-of-roundness profiles of Figures 13 and 14 to discuss the effect of the anisotropic etching on experimental traces. At this point it is convenient to recall that the final shape of a profilometry trace [12, 19] depends on the nature and on the number of extrema of L present in the angular sector \(-\alpha_{\text{max}}, +\alpha_{\text{max}}\) of the polar diagram. Table IV summarises the most significant results of previous studies.

A comparison of experimental results for \(x'_1\) and \(x'_2\) traces displayed in Figures 9 and 7 and of theoretical predictions of Table IV can be undertaken. It appears that this table furnishes comprehensive explanation for the formation of surface profiles with alternate shape \(0^\circ < \varphi_0 < 26^\circ\) and \(30^\circ < \varphi_0 < 45^\circ\) for (hk0) planes and \(65^\circ < \theta_0 < 81^\circ\), \(36^\circ < \theta_0 < 64^\circ\) and \(6^\circ < \theta_0 < 20^\circ\) for (hhl) planes.

We can also justify the observed shapes for [001] and \(\langle 110 \rangle\) traces (Figs. 10 and 8 respectively) on the basis of theoretical results shown in Table IV. Effectively let us recall that as indicated in Table III expected shapes for [001] traces can be deduced from (hk0) out-of-roundness profiles using an angular sector centred on the dissolution slowness \(L(hk0)\) of the reference (hk0) surface. So for [001] traces we work with the sector centred on the direction which is perpendicular to the [001] direction. It is thus possible to draw some conclusions on the expected shapes of [001] traces. These conclusions are listed in Table V where for convenience the number \(N\) and the type of extrema are also indicated. Examination of the Table V reveals that the adequation between predicted shapes and theoretical shapes is very good for angles of cut in the range \((0^\circ - 26^\circ)\). The interpretation is a little more complicated for cuts with \(\theta_0 > 30^\circ\). For these cuts it is clear that when the inclination \(\alpha\) of the surface element is in the range \((\pm 10^\circ)\) the shape of traces is only determined by the presence of a minimum in the polar diagram which occurs for an orientation corresponding to
TABLE IV  Correlation between final shapes of etched surface profiles and the number and the nature of extrema present in the angular sector. Corresponding plots of the dissolution slowness $L$ as a function of the slope $\alpha$ of starting surface profile elements are also given.

<table>
<thead>
<tr>
<th>$L(\alpha)$</th>
<th>Shape of profile</th>
<th>$L(\alpha)$</th>
<th>Shape of profile</th>
</tr>
</thead>
<tbody>
<tr>
<td><img src="image1" alt="Diagram" /></td>
<td>convex</td>
<td><img src="image2" alt="Diagram" /></td>
<td>Concave</td>
</tr>
<tr>
<td><img src="image3" alt="Diagram" /></td>
<td>Concave-convex</td>
<td><img src="image4" alt="Diagram" /></td>
<td>Convex-concave</td>
</tr>
<tr>
<td><img src="image5" alt="Diagram" /></td>
<td>Concave-concave</td>
<td><img src="image6" alt="Diagram" /></td>
<td>Convex-convex</td>
</tr>
</tbody>
</table>
TABLE V  Prediction of theoretical shapes for [001] traces on (hk0) plates. N is the number of extrema present in the angular sector. Note that for all the reference (hk0) plate the dissolution slowness L(hk0) passes through a maximum or through a minimum (α = 0°)

<table>
<thead>
<tr>
<th>Angle θ₀ (°)</th>
<th>N</th>
<th>Type of extrema</th>
<th>Expected shape for the [001] trace</th>
<th>Experimental shape</th>
</tr>
</thead>
<tbody>
<tr>
<td>0</td>
<td>1</td>
<td>maximum</td>
<td>concave</td>
<td>concave</td>
</tr>
<tr>
<td>5</td>
<td>1</td>
<td>Maximum</td>
<td>Concave</td>
<td>concave</td>
</tr>
<tr>
<td>10</td>
<td>1</td>
<td>Maximum</td>
<td>Concave</td>
<td>concave</td>
</tr>
<tr>
<td>14</td>
<td>1</td>
<td>maximum</td>
<td>concave</td>
<td>concave</td>
</tr>
<tr>
<td>34</td>
<td>1</td>
<td>Minimum (α = 0°)</td>
<td>Convex</td>
<td>Convex (elements with small slopes) and concave-concave</td>
</tr>
<tr>
<td>3* 37</td>
<td>1</td>
<td>Convex curvature-minimum -convex curvature</td>
<td>Concave-concave</td>
<td>Convex (elements with small slopes) and concave-concave</td>
</tr>
<tr>
<td>3*</td>
<td>1</td>
<td>Minimum (α = 0°)</td>
<td>Convex</td>
<td>Convex (elements with small slopes) and concave-concave</td>
</tr>
<tr>
<td>42</td>
<td>1</td>
<td>Convex curvature-minimum -convex curvature</td>
<td>Concave-concave</td>
<td>Convex (elements with small slopes) and concave-concave</td>
</tr>
<tr>
<td>3*</td>
<td>1</td>
<td>Convex curvature-minimum -convex curvature</td>
<td>Concave-concave</td>
<td>Convex (elements with small slopes) and concave-concave</td>
</tr>
</tbody>
</table>

*For a sector of ± 30°.
**TABLE VI**  Correspondences between (hkl) plates and out-of-roundness (O-of-R) profiles used for the prediction of theoretical (110) traces. N is the number of extrema present in the angular sector. Note that for all the reference (hk0) plate the dissolution slowness passes through a maximum ($\alpha = 0^\circ$)

<table>
<thead>
<tr>
<th>Plate (hkl)</th>
<th>O-of-R profile</th>
<th>N</th>
<th>Type of extrema</th>
<th>Expected shape for the (110) trace</th>
<th>Experimental shape</th>
</tr>
</thead>
<tbody>
<tr>
<td>118</td>
<td>441</td>
<td>1</td>
<td>maximum</td>
<td>concave</td>
<td>concave</td>
</tr>
<tr>
<td>115</td>
<td>331</td>
<td>1</td>
<td>Maximum ($\alpha = 0^\circ$)</td>
<td>Concave</td>
<td>Concave</td>
</tr>
<tr>
<td>112</td>
<td>111</td>
<td>1</td>
<td>Maximum-minimum</td>
<td>Convex-convex</td>
<td>Convex-convex</td>
</tr>
<tr>
<td>221</td>
<td>114</td>
<td>1</td>
<td>Maximum ($\alpha = 0^\circ$)</td>
<td>Concave</td>
<td>Concave (elements with small slopes) and convex-convex</td>
</tr>
<tr>
<td>331</td>
<td>115</td>
<td>3*</td>
<td>Minimum-maximum-minimum</td>
<td>Convex-convex</td>
<td>Convex-convex</td>
</tr>
<tr>
<td>551</td>
<td>118*</td>
<td>3*</td>
<td>Minimum-maximum-maximum</td>
<td>Convex-convex</td>
<td>Convex-convex</td>
</tr>
</tbody>
</table>

* For a sector of ± 30°.

* Indicates that we use the out-of-roundness profile which is close to the cross-sectional profile.
the reference surface $(\alpha = 0^\circ)$). But the influence of protuberances associated with \{111\} planes covers a large sector of the dissolution surface. So portions of polar diagram for $|\alpha| > 20^\circ$ exhibit a convex curvature. The effect of such convex curvatures on the final shape of trace must be certainly close to that due to maxima in $L$. So on these traces concave curvatures (with large slope) can be observed in accord with experiments.

A similar analysis can be performed for \langle110\rangle traces made on \langle hhl \rangle planes taking now into account correspondences between \langle hhl \rangle planes and cross-sectional planes (see Tab. III). Thus we can establish in Table VI the more probable shape for \langle110\rangle traces taking attention in this case that the angular sector into consideration is centred on the direction perpendicular to the \langle110\rangle direction (indicated on the out-of-roundness profiles of Fig. 14). Here again we obtain a good agreement between theoretical predictions and experimental observations.

6. DISCUSSION AND CONCLUSION

The chemical etching of silicon crystal in a 35% NaOH solution is anisotropic. The anisotropy is characterised by the presence of prolate protuberances in the dissolution slowness surface related to \{111\} minor maxima in the dissolution slowness surface occurs for \{100\} planes. This anisotropy is not far from those observed for other etchants \[1-8\] even if for the NaOH solution changes in the dissolution slowness with orientation appear to be different for planes close to the \{110\} planes.

If we turn attention to the degradation of surfaces caused by a prolonged etching we see that as for other etchants \[1, 2, 4-6, 9\] planes in the vicinity of \{100\} planes exhibit a relatively good smoothness. It is also the case for \{hhl\} planes close to \{113\} planes. So these orientations for which the NaOH etching yields surfaces of good quality merit to be considered for the micromachining of new mechanical and optical micro-devices. In the contrary plates close to \{110\} planes which are covered by hillocks appear to be inappropriate for the micromachining of not opened structures such as membranes.

No deviating behaviour results from the comparison of the variations in the etch rate with orientation (Figs. 13d) with the
out-of-roundness profiles related to (100) and (110) cross-sections. It is in particular the case when we consider the angles of cut for which in the one hand, minima and maxima in the etch rate and in the other hand converse extrema in the polar graph of the dissolution slowness (i.e., in the out-of-roundness profiles) occur. The comparative analysis of etching shapes with predictions deduced from shapes of out-of-roundness profiles show that the tensorial model combined with dissolution criteria can suitably explain final shapes of traces made on (hk0) and (hhl) plates.

At this point of the analysis it is clear that only the micromachining of 3D structures could provide useful additional information on the anisotropy of the chemical attack of silicon in a NaOH 35% solution. So results on micromachining will be reported in the second part of this paper.

References


