

Research Article

Facile Synthesis and Photocatalytic Property of Titania/Carbon Composite Hollow Microspheres with Bimodal Mesoporous Shells

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Titania/carbon composite hollow microspheres with bimodal mesoporous shells are one-pot fabricated by hydrothermal treatment of the acidic $(\text{NH}_4)_2\text{TiF}_6$ aqueous solution in the presence of glucose at 180°C for 24 h and then calcined at 450°C . The as-prepared samples were characterized by XRD, SEM, TEM, HRTEM, UV-visible spectroscopy, and nitrogen adsorption-desorption isotherms. The photocatalytic activity of the as-prepared samples was evaluated by daylight-induced photocatalytic decolorization of methyl orange aqueous solution at ambient temperature. The effects of calcination time on the morphology, phase structure, crystallite size, specific surface area, pore structures, and photocatalytic activity of the microspheres were investigated. The results indicated that the as-obtained TiO_2/C composite hollow spheres generally exhibit bimodal mesopore size distribution with their peak intra-aggregated mesopore size in the range of 2.3–4.5 nm and peak interaggregated mesopore size in the range of 5.7–12.7 nm, depending on specific calcination time. The daylight-induced photoactivity of as-obtained hollow TiO_2/C microspheres generally exceeds that of Degussa P25. The influences of calcination time on the photoactivity are discussed in terms of carbon content, phase structures, and pore structures.

1. Introduction

Photocatalytic degradation of organic compounds for the purpose of purifying water or wastewater from industries and households has attracted great attention in the past decade [1–7]. Among various oxide semiconductor photocatalysts, titania is a very important photocatalyst due to its biological and chemical inertness, strong oxidizing power, nontoxicity, and long-term stability against photo and chemical corrosion. In order to commercialize this treatment technique, it is of great importance to improve the preparative methods of titania, because the morphology, microstructures, and photocatalytic activity of TiO_2 are significantly influenced by the preparative conditions and methods [8–11].

The fabrication of TiO_2 hollow structures attracts special attention due to their low density, high surface area, good surface permeability and larger light-harvesting capacities [9]. Especially, micro- or nanometer scale TiO_2 hollow spheres with controllable structure, composition, and properties have shown a promising perspective in many fields such as catalysts, adsorbents, sensors, light fillers, and chemical reactors [12–16]. While hollow TiO_2 structures have been synthesized without the assistance of templates by the spray-drying technique and via Ostwald ripening and chemically induced self-transformation [17–20], the templating method has been most frequently applied for the synthesis of hollow TiO_2 structures with tailored properties. For example, Hard templates (e.g., polymer latex, carbon, and anodic aluminium oxide templates) and soft

templates (e.g., supermolecules, ionic liquids, surfactant, and organogel) have been extensively employed to produce TiO₂ hollow structures either by controlled surface precipitation of inorganic molecule precursors or by direct surface reactions utilizing specific functional groups [21–27]. Very recently, a general method for the synthesis of metal oxide hollow spheres has been developed using carbonaceous polysaccharide microspheres prepared from saccharide solution as templates by Sun and coworkers and Titirici and coworkers [28, 29].

Carbon materials (e.g., amorphous carbon, activated carbon, and graphite) have attracted considerable attention due to their widespread applications as absorbents, catalyst supports, and nanocomposites [30]. Carbon is chemically inert at low temperature and a suitable support for titanium dioxide photocatalysts. Various studies have demonstrated that titanium/carbon composite photocatalysts can result in a synergistic effect of both adsorption and photocatalysis and can dramatically improve the daylight-induced photocatalytic activity of titania for different modal organic components [30–37].

However, preparation of well-crystallized titanium/carbon composite hollow microspheres with high daylight-induced photocatalytic activity keeps still a great challenge. In this study, titanium/carbon composite hollow microspheres with bimodal mesoporous shell are one-pot fabricated according to the method reported by Titirici and coworkers [29] and their visible light photocatalytic activity is investigated.

2. Experimental Section

2.1. Sample Preparation. All chemicals used in this study were reagentgrade without further purification. Distilled water was used in all experiment. (NH₄)₂TiF₆ was used as a titanium source. In a typical synthesis, 15 g (75.5 mmol) of glucose and 3.0 g (15.1 mmol) of (NH₄)₂TiF₆ were dissolved in 80 and 40 mL of distilled water under stirring, respectively. The above two solutions were mixed and the pH of the mixed solutions was adjusted to ca. 3 using 1 M HCl or NaOH aqueous solutions. Then the reaction solution was transferred into a 200 mL Teflon-lined stainless steel autoclave, followed by hydrothermal treatment of the mixture at 180°C for 24 h. After hydrothermal reaction, the black or puce precipitates were centrifuged and then washed with distilled water and absolute alcohol for 5 times. The washed precipitates were dried in a vacuum oven at 60°C for 8 h. Finally, the dried mixtures were calcined in air at 450°C for 0.5–4.0 h, the TiO₂/C composites, with the color from black to grey when the calcined time was increased, were obtained. The abbreviations of TiO₂/C composites with different calcined time are shown in Tables 1 and 2.

2.2. Characterization. The carbon content of the composite catalysts was monitored using a DTA-TG instrument (Netzsch STA 449C) in airflow of 100 mL min⁻¹ at a heating rate of 10°C min⁻¹ from room temperature to 900°C. The X-ray diffraction (XRD) patterns obtained on an X-ray

diffractometer (type HZG41B-PC) using Cu K α irradiation at a scan rate of 0.05° 2 θ s⁻¹ were used to determine the identity of any phase present and their crystallite size. The accelerating voltage and the applied current were 15 kV and 20 mA, respectively. The phase composition of TiO₂ can be calculated from the integrated intensities of anatase (101), rutile (110), and brookite (121) peaks. If a sample contained anatase and rutile two phases, the mass fraction of rutile could be calculated according to the following equation (1) [38].

$$W_R = \frac{A_R}{0.886A_A + A_R}, \quad (1)$$

where A_A and A_R represent the integrated intensity of the anatase (101) and rutile (110) peaks, respectively. The average crystallite sizes of anatase and rutilites were determined according to the Scherrer equation using the FWHM data of each phase after correcting the instrumental broadening [38]. Morphology observation was performed on a JSM-5610LV scanning electron microscope (SEM, JEOL, Japan). Crystallite sizes and shapes were observed using transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) (JEOL-2010F at 200 kV). The samples for TEM observation were prepared by dispersing the TiO₂ powders in an absolute ethanol solution under ultrasonic irradiation; the dispersion was then dropped on carbon-copper grids. The Brunauer-Emmett-Teller (BET) surface area (S_{BET}) of the powders was analyzed by nitrogen adsorption in a Micromeritics ASAP 2020 nitrogen adsorption apparatus (USA). All the samples were degassed at 180°C prior to nitrogen adsorption measurements. The BET surface area was determined by multipoint BET method using the adsorption data in the relative pressure (P/P_0) range 0.05 ~ 0.3. Desorption isotherm was used to determine the pore size distribution via the Barret-Joyner-Halender (BJH) method, assuming a cylindrical pore modal [39–41]. The nitrogen adsorption volume at the relative pressure (P/P_0) of 0.994 was used to determine the pore volume and average pore size. UV-visible diffuse reflectance spectra of as-prepared TiO₂ powders were obtained for the dry-pressed disk samples using a UV-visible spectrophotometer (UV2550, Shimadzu, Japan). BaSO₄ was used as a reflectance standard in the UV-visible diffuse reflectance experiment. The morphologies of TiO₂ powders were observed using scanning electron microscopy (SEM) (type JSM-5610LV, Japan) with an acceleration voltage of 20 kV.

2.3. Measurement of Photocatalytic Activity. The evaluation of photocatalytic activity of the prepared samples for the photocatalytic decolorization of methyl orange aqueous solution was performed at ambient temperature, as reported in our previous studies [42]. Experiments were as follows: 0.04 g of the prepared powders were dispersed in a 20 mL of methyl orange aqueous solution with a concentration of 3.1×10^{-5} mol L⁻¹ in a dish (with a diameter of ca. 7.0 cm). The solution was allowed to reach an adsorption-desorption equilibrium among the photocatalyst, methyl orange, and

TABLE 1: Effects of calcination time on phase structure, phase content, and average crystallite sizes of TiO₂/C composite microspheres at 450°C.

Time (h)	Material abbreviations	Carbon mass ratio in composites ^a (wt.%)	Anatase		Rutile	
			Crystalline Size ^b (nm)	Content ^c (%)	Crystalline Size ^b (nm)	Content ^c (%)
0.5	TiO ₂ -0.5	32.9	11 (1.0)	86.2	36	13.2
1	TiO ₂ -1	8.2	17 (1.27)	83.5	41	16.5
2	TiO ₂ -2	5.0	18 (1.39)	82.5	58	17.5
4	TiO ₂ -4	1.9	21 (1.47)	80.6	68	18.4

^aEstimated from the data of TGA-DSC measurement. ^bAverage crystalline size of TiO₂ was determined by XRD using Scherrer equation. Relative anatase crystallinity: the relative intensity of the diffraction peak from the anatase (101) plane (indicated in parentheses, reference = sample calcined at 450°C for 0.5 h). ^cDetermined by XRD method.

TABLE 2: Effects of calcination time on surface areas and pore parameters of TiO₂/C composite microspheres at 450°C.

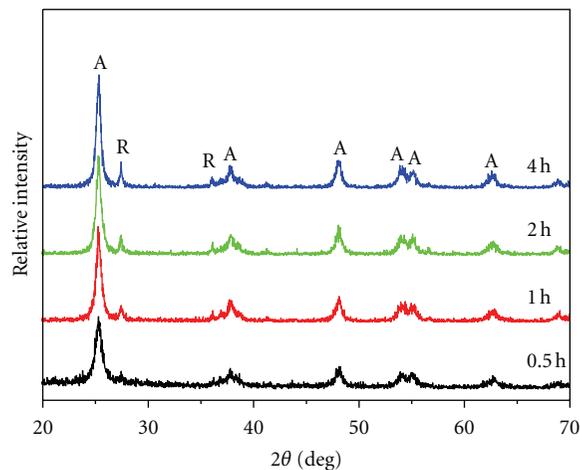
Time (h)	Material abbreviations	S _{BET} ^a (m ² /g)	Pore volume ^b (cm ³ /g)	Average pore Size ^b (nm)	Porosity ^c (%)
0.5	TiO ₂ /C-0.5	210.5	0.129	2.4	33.5
1	TiO ₂ /C-1	55.3	0.076	5.5	22.8
2	TiO ₂ /C-2	46.4	0.073	6.3	22.2
4	TiO ₂ /C-4	20.8	0.044	8.5	14.6

^aThe BET surface area was determined by multipoint BET method using the adsorption data in P/P_0 range from 0.05 to 0.3. ^bPore volume and average pore size were determined by nitrogen adsorption volume at $P/P_0 = 0.994$. ^cThe porosity is estimated from the pore volume determined using the desorption data at $P/P_0 = 0.994$.

water before daylight irradiation. An 18-W daylight lamp (3 cm above the dish) was used as a light source. The integrated daylight intensity was 0.46 ± 0.01 mW/cm², as measured by a UV radiometer (made in the photoelectric instrument factory of Beijing Normal University) with the peak intensity of 420 nm. After visible-light irradiation for 60 min, the reaction solution was filtrated, and the concentration of methyl orange aqueous solution was determined by a UV-visible spectrophotometer (UV-2550, SHIMADZU, Japan). As for the methyl orange aqueous solution with low concentration, its photocatalytic decolorization is a pseudo-first-order reaction and its kinetics may be expressed as $\ln(c_0/c) = kt$, where k is the apparent rate constant, and c_0 and c are the adsorption-desorption equilibrium and reaction concentrations of methyl orange, respectively. Each set of photocatalytic measurements were repeated for three times, and the experimental error was found to be within $\pm 5\%$.

3. Results and Discussion

3.1. Crystal Structure. Figure 1 shows the XRD pattern of the composites catalysts prepared at 450°C for different calcined time. The phase content, crystal size, and relative anatase crystallinity of TiO₂/C samples are shown in Table 1. At 450°C for 0.5 h, anatase and rutile phases appear, and their mass percentages are 86.2% and 13.8%, respectively. With increasing calcined time, crystallite size, relative anatase crystallinity, and rutile content increases gradually. This may be ascribed to the fact that the longer calcined time enhance grain growth and favor the phase transformation of anatase to rutile. On the other hand, with increasing calcined time

FIGURE 1: XRD patterns of TiO₂/C composite hollow spheres after calcination at 450°C for 0.5 (a), 1 (b), 2 (c), and 4 h (d).

from 0.5 to 4 h, the carbon content of TiO₂/C samples changes from 32.9 to 1.9%. The presence of carbon layer may suppress the phase transformation of anatase to rutile to a certain extent. Similar results were reported by Shanmugam et al. and Tsumurae al. [33, 43]. The carbon layers are acting as barriers for phase transformation of anatase to rutile.

3.2. SEM and TEM Studies. SEM and TEM were used to characterize the morphology and crystal structure of the as-prepared samples. Calcination of the TiO₂/C composites in air resulted in the formation of hollow TiO₂/C

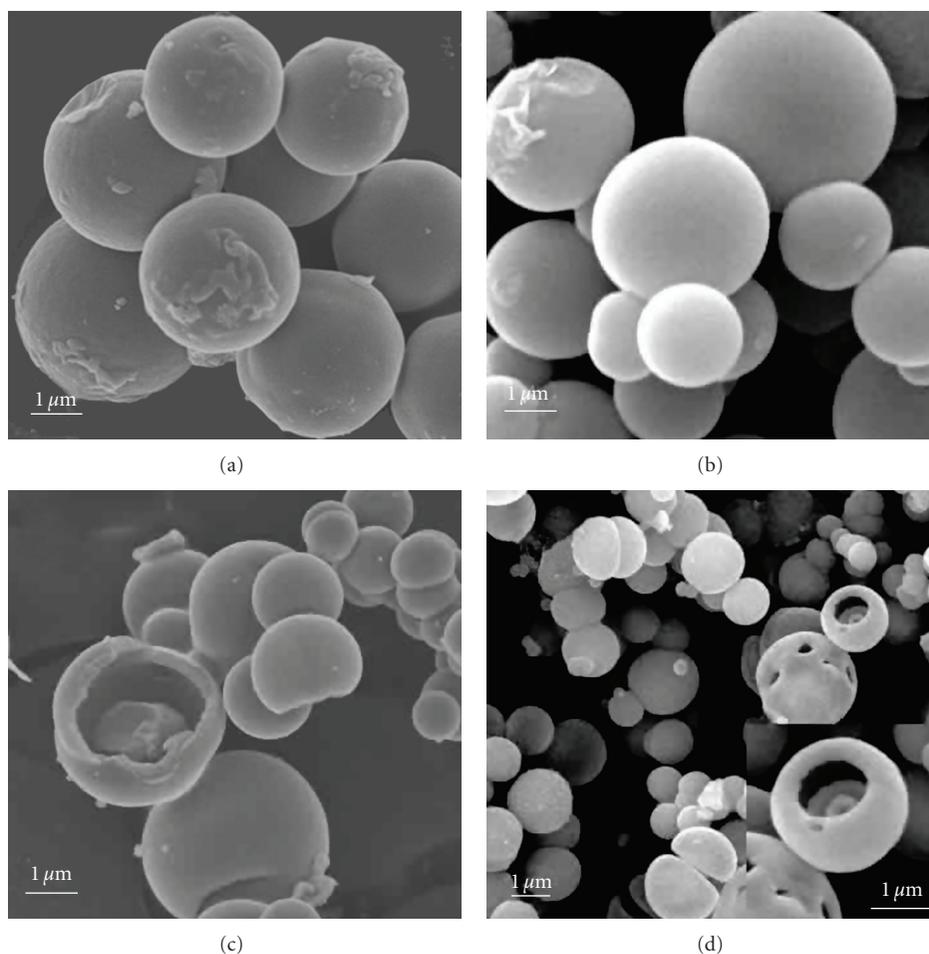


FIGURE 2: SEM image of the TiO_2/C composite hollow spheres calcined at 450°C for 0.5 (a), 1 (b), 2 (c), and 4 h (d).

microspheres. Figure 2 displays the SEM images of hollow TiO_2/C microspheres obtained after hydrothermal reaction and calcination in air at 450°C for different time. It can be seen from Figure 2(a) that after calcination for 0.5 h, the microspheres with surface wrinkles have a uniform diameter of ca. $4\ \mu\text{m}$. It is interesting to note that with increasing calcination time, the diameter of microspheres decrease drastically probably due to a large amount of carbon removed during calcination, resulting in drastic shrinkage of corresponding hollow spheres. At 400°C for above 2 h, the samples are composed of hollow spheres with a diameter range from 0.5 to $2.0\ \mu\text{m}$ and cavities are occasionally found in some broken microspheres with a shell thickness $\sim 400\ \text{nm}$ (Figure 2(c) and inset in Figure 2(d)).

The morphology and microstructures of TiO_2/C hollow spheres are further investigated by TEM analysis. Figure 3(a) shows a typical TEM image of the samples calcined at 450°C for 2 h. There is a strong contrast difference observed for all microspheres with dark edge and bright center, clearly confirming their hollow structures. Further observation indicates that the prepared hollow microspheres appear unique sphere-in-sphere superstructures. Li and coworkers

reported the similar sphere-in-sphere superstructures prepared solvothermally in glycerol, alcohol, and ethyl ether, which allow multireflections of electromagnetic waves, such as ultraviolet and visible light, within their interior cavities, endowing these spheres with greatly enhanced properties [44]. Figure 3(b) presents a typical HRTEM lattice image of the TiO_2 nanoparticles in the shell of TiO_2/C hollow microspheres. The selected area electron diffraction (SAED) patterns (inset in Figure 3(b)) reveal the polycrystalline nature of the anatase and rutile phases for the TiO_2/C hollow microspheres. By measuring the lattice fringes, the resolved interplanar distances are ca. 0.35 and 0.33 nm, corresponding to the (101) planes of anatase and the (110) planes of rutile, respectively. This further confirms the mixed biphasic structures of hollow microspheres.

3.3. Pore Structure and BET Surface Areas. Figure 4 shows the nitrogen adsorption-desorption isotherms of the TiO_2/C composite hollow spheres calcined at 450°C for 0.5–4 h. It can be seen that all the samples show a type IV isotherm with two hysteresis loops. The shapes of two hysteresis loops are different from each other. At low relative pressures between

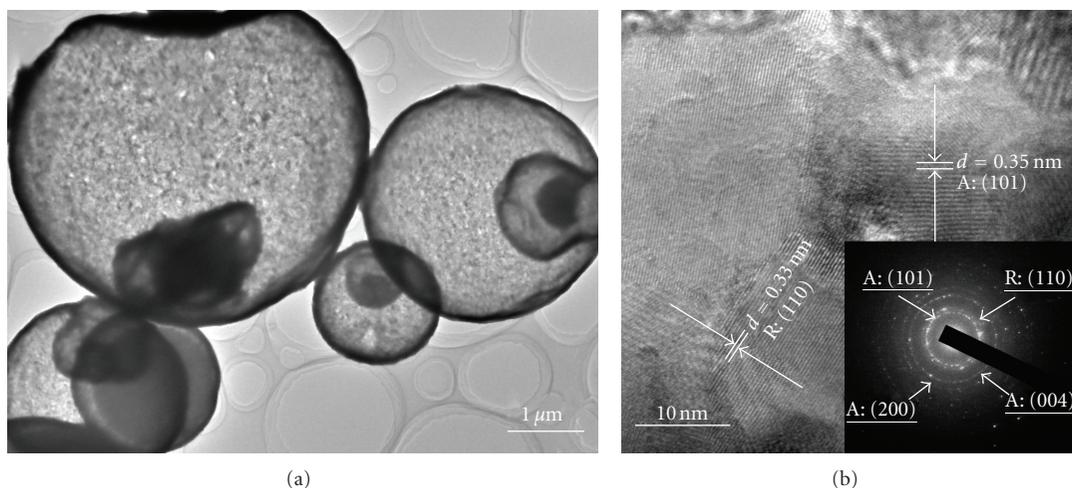


FIGURE 3: TEM (a) and corresponding HRTEM (b) images of the TiO_2/C composite hollow spheres calcined at 450°C for 2 h.

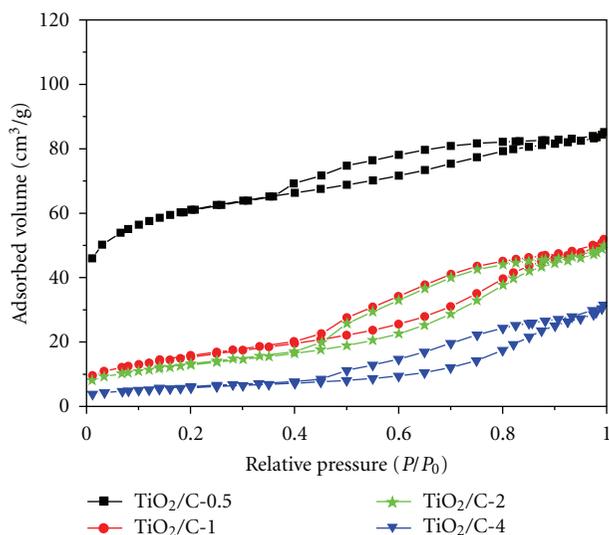


FIGURE 4: Nitrogen adsorption-desorption isotherms of the TiO_2/C hollow spheres calcined at 450°C for 0.5–4 h.

0.4 and 0.8, the hysteresis loops are of type H2, which can be observed in the pores with narrow necks and wider bodies (ink-bottle pores) [45, 46]. However, at high relative pressures between 0.8 and 1.0, the shape of the hysteresis loop is of a type H3, associated with plate-like particles giving rise to narrow slit-shaped pores [39, 46]. Further observation indicates that with increasing calcination time, the hysteresis loops shifted to a higher relative pressure region. The isotherms of the TiO_2/C composites prepared at 450°C for 0.4–4 h having two hysteresis loops indicate bimodal pore-size distributions in the mesoporous regions existed in the shells.

Figure 5 shows the corresponding pore-size distributions of the TiO_2/C composite hollow spheres calcined at 450°C

for 0.5–4 h. All samples show bimodal pore-size distributions, consisting of finer intra-aggregated pores and larger interaggregated pores. For example, the sample calcined at 450°C for 0.5 h has small mesopores (peak pore: ca. 2.3 nm) and larger mesopores (peak pore: ca. 5.8 nm), which are related to finer aggregated pore formed between small anatase crystallites 11 nm in size, and larger aggregated pore produced by large rutile crystallites 36 nm in size (see Table 1), respectively. Further observation indicates that with increasing calcination time, the corresponding maximum peaks shift to the right, indicating the increase of pore size. There are two possible factors resulting in the increase of pore size. One is that the aggregation of greater crystallites forms bigger pores. The other is that the content of carbon in the shells decreases, which probably inserts into the pore of TiO_2 to cause the decrease in pore size. The effects of calcination time on physical properties of TiO_2/C hollow microspheres are shown in Table 2. With increasing calcination time, the BET-specific surface areas, pore volumes, and porosity steadily decrease. However, the average pore size increases. This is ascribed to the fact that increase of calcination time resulted in decrease of the carbon content in the TiO_2/C composites.

3.4. UV-Vis Spectrum. Usually, the carbon content obviously influences light absorption characteristics of TiO_2/C composites [32–35]. Figure 6 shows the UV-visible absorption spectra of the TiO_2/C composite hollow spheres calcined at 450°C for 0.5–4 h. A significant increase in the absorption at wavelengths shorter than 400 nm can be assigned to the intrinsic band gap absorption of TiO_2 [3]. It is noticeable that there is an obvious correlation between the calcination time and the UV-vis spectrum change. With increasing calcination time, the absorption in the near UV and visible-light region gradually decreases and the absorption edge of the samples shows an obvious blue shift. The differences in adsorption are attributed to the change of carbon content in the TiO_2/C composites. This clearly indicates an increase in the band gap

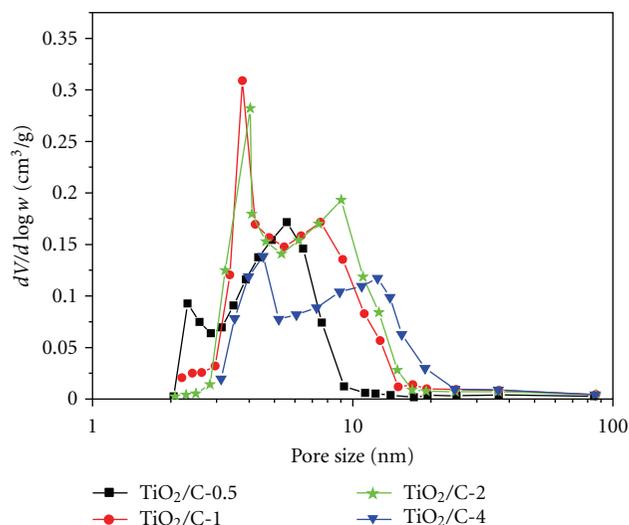


FIGURE 5: Pore-size distribution curves of the TiO_2/C hollow spheres calcined at 450°C for 0.5–4 h.

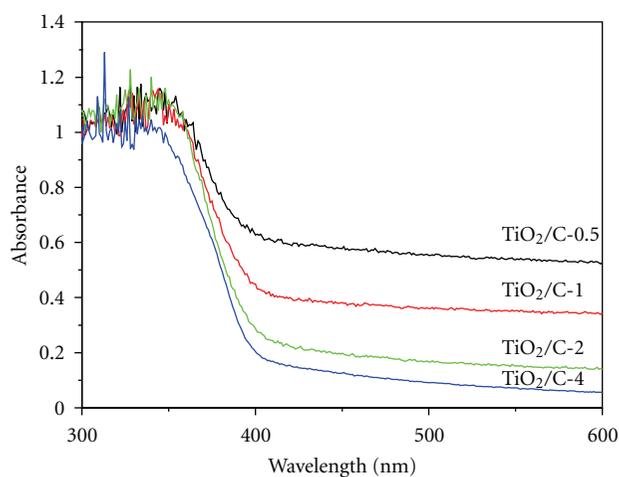


FIGURE 6: UV-vis spectra of TiO_2 hollow spheres calcined at 450°C for 0.5–4 h.

energy of TiO_2 . Band gap energy could be estimated from Figure 6. The intercept of the tangent to the plot would give an approximation of the band gap energy for indirect band gap materials such as TiO_2 [47, 48]. The band gap energies were estimated to be about 3.05, 3.09, 3.15, and 3.20 eV for the TiO_2/C hollow spheres calcined at 450°C for 0.5, 1.0, 2.0, and 4 h, respectively. Obviously, the longer calcination time is, the bigger value of the band energies. This is due to the difference in carbon content of TiO_2/C composites [34, 49].

3.5. Photocatalytic Activity. The photocatalytic activity of the as-prepared TiO_2/C composite hollow spheres was evaluated by the daylight-induced photocatalytic decolorization of methyl orange aqueous solution at ambient temperature. For comparison, the daylight-induced photocatalytic activities of the commercial TiO_2 powder Degussa P25 (P25) were also

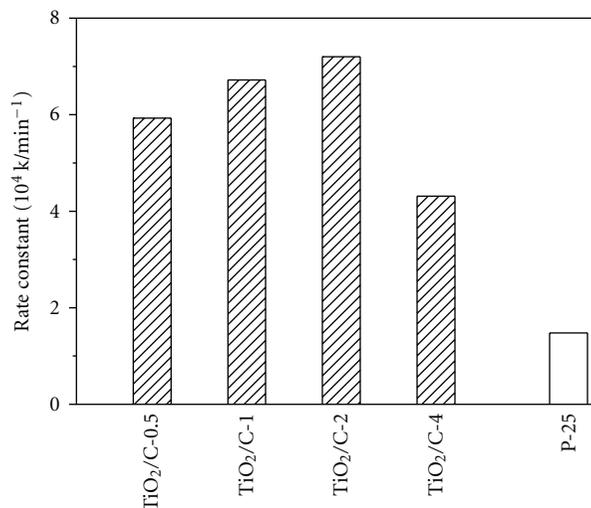


FIGURE 7: Effects of calcination temperatures on the apparent rate constants of the TiO_2/C composite hollow microspheres.

tested. Figure 7 shows the comparisons of the apparent rate constants of P25 and the TiO_2/C composite hollow spheres at various calcination time. It can be seen that the TiO_2/C -0.5 sample has a high daylight-induced photocatalytic activity. This is due to the fact that the TiO_2/C -0.5 sample has the large carbon content of about 32.9% and surface area of $210.5 \text{ m}^2/\text{g}$ (see Table 1). With increasing calcination time, the daylight-induced photocatalytic increases. At 450°C for 2 h, the photocatalytic activity of the TiO_2/C -2 sample reaches a maximum value, and its activity exceeds that of Degussa P25, which is recognized as an excellent photocatalyst [3, 43–45]. On the one hand, this is attributed to the former having bimodal mesoporous structures, which is more beneficial in enhancing the adsorption and desorption of reactants and products, respectively [45]. On the other hand, the high photocatalytic activity of the TiO_2/C -2 sample is due to its large surface area and high adsorption ability in the near UV and visible-light region. With further increasing calcination time, the photocatalytic activity of the powders decreases. The highest daylight-induced photocatalytic activity of the TiO_2 -2 sample is due to the following factors.

Usually, TiO_2/C composite photocatalysts can result in a synergistic effect of both TiO_2 and carbon on the photocatalytic degradation of different modal organic components. Carbon can dramatically improve the photocatalytic activity of titania as coadsorbent due to its very high surface area. Therefore, the carbon content in the TiO_2/C composite catalysts plays an important role in its visible-light photocatalytic activity. It can be seen from Figure 7 that the TiO_2/C -2 sample with higher carbon content shows the superior photocatalytic activity compared with the TiO_2/C -4 sample. This may be ascribed to the fact that higher carbon content results in a intense increase in absorption in the near UV and visible-light region and a red shift in the absorption edge of the TiO_2 -2 sample (as shown in Figure 6), implying that the TiO_2 -2 sample can be easier activated by visible

light and more photo-generated electrons and holes can be generated and participate in the photocatalytic reactions. However, adsorption of carbon is not only factor for the enhancement of the photocatalytic activity of TiO_2 . For example, compared with the TiO_2 -2 sample, the two TiO_2/C -0.5 and TiO_2/C -1 samples with the larger carbon content have lower daylight-induced photocatalytic activity. This result can be interpreted in two aspects. On one hand, the surplus carbon of the two TiO_2/C -0.5 and TiO_2/C -1 samples can scatter the photons in the photoreaction system [35]. On the other hand, the lower carbon content of the TiO_2/C -2 sample enables the methyl orange aqueous solution to get the active site easier and faster [33]. Therefore, the TiO_2/C -2 sample with an optimum carbon content of about 5.0% has the highest daylight-induced photocatalytic activity.

Apart from the above carbon content, another possible understanding is that the TiO_2 -1 sample possesses a relative large surface area and good anatase crystallinity (as shown in Tables 1 and 2). Usually, a large specific surface area may enhance the rate of photocatalytic degradation reactions, as a large amount of adsorbed organic molecules promote the photocatalytic reaction [50, 51]. However, the specific surface areas and crystallinity usually appear to be two conflicting intrinsic properties for TiO_2 nanoparticles [52]. The powders with a large surface area are usually associated with large amounts of crystalline defects or weak crystallization, which favor the recombination of photo-generated electrons and holes, leading to a poor photoactivity [53–55]. For example, with increasing calcination time from 0.5 to 2 h, the surface area decreases greatly, but the photocatalytic activity increases. This may be due to the increase of the relative anatase crystallinity (as shown in Table 1). So a balance between specific surface area and crystallinity is a very important factor in determining the photocatalytic activity of TiO_2 powders. According to the above results and discussion, the TiO_2/C -2 sample possesses a relative large surface area and good anatase crystallinity (as shown in Tables 1 and 2), resulting in a highest daylight-induced photocatalytic activity.

4. Conclusions

Titania/carbon composite hollow microspheres with bimodal mesoporous shells are one-pot fabricated by hydrothermal treatment of the acidic $(\text{NH}_4)_2\text{TiF}_6$ aqueous solution in the presence of glucose at 180°C for 24 h and then calcined at 450°C for 2–4 h. Calcination time obviously influenced on the morphology, phase structure, crystallite size, specific surface area, pore structures, and daylight-induced photocatalytic activity of the as-obtained hollow microspheres. All TiO_2/C composite hollow spheres generally exhibit bimodal mesopore size distribution with their peak intra-aggregated mesopore size in the range of 2.3–4.5 nm and peak interaggregated mesopore size in the range of 5.7–12.7 nm. The TiO_2 -2 sample showed the highest daylight-induced photocatalytic activity and greatly exceeded that of Degussa P25.

Acknowledgments

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