Research Article

Effect of Subgrains on the Performance of Mono-Like Crystalline Silicon Solar Cells

Su Zhou, Chunlan Zhou, Wenjing Wang, Yehua Tang, Jingwei Chen, Baojun Yan, and Yan Zhao

The Key Laboratory of Solar Thermal Energy and Photovoltaic System, Institute of Electrical Engineering, Chinese Academy of Sciences, Beijing 100190, China

Correspondence should be addressed to Wenjing Wang; wangwj@mail.iee.ac.cn

Received 4 June 2013; Accepted 11 September 2013

1. Introduction

Most current industrial solar cells are made of Czochralski (Cz)-grown monocrystalline silicon material and cast multicrystalline silicon substrates [1]. However, Cz monocrystalline silicon material has a high cost and undergoes serious light-induced degradation. The use of cast multicrystalline silicon is also hindered by its high dislocation densities and high surface reflectance after texturing. Mono-like crystalline silicon is a promising material because it has the advantages of both mono- and multicrystalline silicon. However, when mono-like wafers are made into cells, the efficiencies of a batch of wafers often fluctuate within a wide range of >1% (absolute). In this work, mono-like wafers are classified by a simple process and fabricated into laser doping selective emitter cells. The effect and mechanism of subgrains on the performance of mono-like crystalline silicon solar cells are studied. The results show that the efficiency of mono-like crystalline silicon solar cells significantly depends on material defects that appear as subgrains on an alkaline textured surface. These subgrains have an almost negligible effect on the optical performance, shunt resistance, and junction recombination but significantly affect the minority carrier diffusion length and quantum efficiency within a long wavelength range. Finally, an average efficiency of 18.2% is achieved on wafers with hardly any subgrain but with a small-grain band.

2. Experimental

All wafers used in this study were commercial grade, 1 Ω·cm, p-type mono-like crystalline Si wafers with an area of 6 square inches (243.4 cm²) and thickness of ∼200 μm. Wafers
are typically optically perfect and appear like square Cz monowafers. After texturing in a mixture of 1% NaOH and 4% isopropyl alcohol, some subgrains appeared on some parts of the mono-like wafers, as shown in Figure 1. Figure 1(a) shows an alkaline textured surface like monocrystalline wafer. However, when the image was taken at a tilt angle of 45°, subgrains with an estimated size of 3–7 mm can be observed as shown in Figure 1(b).

According to the content of these subgrains, the wafers were divided into three classes. Wafers with ratios of subgrains area to total wafer area of <10%, ~50%, and >90% on their surface were defined as grades A, B, and C, respectively. All wafers after alkaline texturing were phosphorus diffused to a sheet resistance of 80 Ω/□ with POCl₃ liquid source. An industrial wet chemical etching process was then performed to achieve edge junction isolation. A SiNₓ antireflection coating (ARC) was deposited onto the front surface of the wafers using an industrial remote plasma-enhanced chemical vapor deposition system. Aluminum (Al) paste was then screenprinted on the rear surface of the wafers and fired in a belt furnace at 900°C to form the back surface field of the cells. Diluted phosphoric acid was spin coated on the SiNₓ film. A 532 nm Q-switched Nd:YAG laser was used to remove the dielectric layer and pattern laser-doping n+ finger patterns on the n-type surface simultaneously. Nickel (Ni) and silver (Ag) were then plated onto the patterned fingers by light-induced plating and sintered to form Ni silicate, which provided low-resistance contact.

The surface reflectance of textured and passivated samples and internal quantum efficiency (IQE) of cells within the range of 300 nm to 1200 nm were measured using a solar cell spectral response/quantum efficiency measurement system (QEX7, PV Measurement). The electroluminescence (EL) images, light beam-induced current (LBIC), and diffusion length of fabricated solar cells were characterized using an infrared defect inspection tool (ELT C02, ASIC) and tabletop PV measurement system (WT-2000, Semilab). A current-voltage (I–V) tester was used to obtain both dark and illuminated I–V curves and to assess the electrical performances of the laser-doped p-type mono-like crystalline Si solar cells with different sub-grain amounts.

3. Results and Discussion

3.1. Optical Performances and IQE. Figure 2 shows an experimental comparison of the percentage reflectance of textured and ARC-coated wafers with the IQE of cells with sub-grain and monocrystalline areas within a wavelength range of 300 nm to 1200 nm. The overall reflectance of sub-grain areas is similar to that of monocrystalline areas, except for a slight increase within the short wavelength range of 300 nm to 400 nm. The weighted reflectances of sub-grain and monocrystalline areas after ARC coating are 4.03% and 4.04%, respectively. The similar weighted reflectance means that the subgrains have minimal effect on the light trapping of the pyramid texture. However, as shown in Figure 2, the IQE of cells with sub-grain areas is significantly decreased within the range of 600 nm to 1100 nm wavelength compared with the result of monocrystalline areas. According to the deep penetration of long wavelength light in silicon, the reduction of IQE may be attributed to the recombination in silicon substrate. Thus, the recombination rate is likely to be higher in sub-grain areas than in monocrystalline areas, which decreases the IQE of fabricated cells within the long wavelength range under the same light condition.

3.2. Analysis on Solar Cell Parameters. Table 1 compares various cell parameters fabricated by wafers with different sub-grain contents. For each type of silicon wafer, 10 solar cells were fabricated and measured. With increased sub-grain content from <10% to >90%, the cell efficiency decreases from 17.6% to 16.0%. With increased sub-grain content, the open-circuit voltage (V_oc) of the cell decreases from 628.9 mV to 615.9 mV, and the short-circuit current density (I_sc) of the cell decreases from 36.87 mA/cm² to 34.35 mA/cm². The decrease in the efficiency can be attributed to the significant decrements in V_oc and I_sc.
The local ideality factor, \( m \), of a solar cell in the dark is given by

\[
m = \frac{1}{V_T} \left( \frac{dV}{d \ln(I)} \right),
\]

where \( V \) and \( I \) are the measured dark voltage and current, respectively, and \( V_T = 0.026 \) eV. The local ideality factors in different voltage regions indicate different mechanisms which may have an effect on cell performance [13, 14]. Figure 3 shows the local ideality factor curves derived from the dark \( I-V \) curves of mono-like crystalline laser-doping (LD) solar cells fabricated on different substrates. The local ideality factors of different-grade wafers are similar (\( > 2 \)) around the low-voltage region (\( < 0.4 \) V), indicating that shunting may have occurred in all these cells. This shunting problem may be caused by insufficient edge isolation. The local ideality factors of different-grade wafers around the medium-voltage region (near the maximum power point) are also similar, indicating that the subgrains hardly affect the formation of localized Schottky contacts. The result shows that the shunting property and metal contact of mono-like crystalline LD solar cells are independent of the subgrains.

### Table 1: Average electrical parameters of 10 solar cells fabricated on different substrates.

<table>
<thead>
<tr>
<th>Grade</th>
<th>( V_{oc} ) (mV)</th>
<th>( J_{sc} ) (mA/cm²)</th>
<th>FF (%)</th>
<th>Eta (%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>A</td>
<td>628.9 ± 1.2</td>
<td>36.9 ± 0.1</td>
<td>75.8 ± 0.6</td>
<td>17.6 ± 0.2</td>
</tr>
<tr>
<td>B</td>
<td>621.6 ± 2.3</td>
<td>35.9 ± 0.2</td>
<td>75.2 ± 0.4</td>
<td>16.8 ± 0.2</td>
</tr>
<tr>
<td>C</td>
<td>615.9 ± 3.0</td>
<td>34.4 ± 0.2</td>
<td>75.7 ± 0.6</td>
<td>16.0 ± 0.3</td>
</tr>
</tbody>
</table>

3.3. EL Image. Figure 4 shows the EL images of cells fabricated with different sub-grain contents. With increased sub-grain content, dark line clusters spread from a small part to almost the entire surface. These dark line clusters indicate a low EL intensity which can be caused by series resistance variations or locally enhanced recombination. These dark line clusters are also closely correlated with subgrains observed on the wafer surface. This finding indicates that these subgrains, which can represent material defects such as grain boundaries and dislocations, play an important role in enhancing series resistance or recombination of minority carriers. Grain boundaries and dislocations are known to be easily generated during casting, and the dislocation density increases from bottom to top of the ingots [15, 16]. The dislocation density, especially dislocation clusters correlated with subgrains, has been found to affect the recombination of light-generated minority carriers significantly and thus the solar cell efficiencies of cast multicrystalline silicon [17, 18]. Therefore, dislocations caused by sub-grain formation can be inferred to result in the recombination of minority carriers and the dark line clusters found in the EL images.

3.4. Minority Carrier Diffusion Length. Figure 5 shows various spatial distributions of cells with different sub-grain contents. Red represents the regions with a minority carrier diffusion length (MCDL) of \( \sim 130 \mu m \), and black represents those with an MCDL of \( \sim 420 \mu m \). Strong local variations in MCDL can be clearly seen in different regions from Figures 5(a) to 5(c). With increased sub-grain content, the diffusion length significantly decreases and the diffusion length distribution changes from uniform to nonuniform. Moreover, these regions of low diffusion length in Figures 5(b) and 5(c) also correspond with subgrains observed on the wafer surface. These reduced diffusion length regions, which indicate locally enhanced recombination, may lead to...
decreased $V_{oc}$. Those longitudinal lines in figures are resulted from fingers of solar cells by blocking the testing laser spot light during the test.

3.5. LBIC Measurement. Further characterizations were performed to determine the influence of subgrains on recombination. The IQE distribution, which was measured by the two-dimensional LBIC method, can represent recombination activity in different regions at different wavelengths of light [19–21]. The IQE distributions measured at 405 and 979 nm are shown in Figures 6 and 7. The uniform and similar IQE distributions of cells with different sub-grain contents in Figure 5 indicate that the locally enhanced recombination caused by subgrains has hardly any effects on the light-induced current at 405 nm. The penetration depth of 405 nm wavelength light is about 500 nm in crystalline silicon. Therefore, the IQE measured at this wavelength indicates the information of emitter and PN junction. The strong field passivation effect caused by the PN junction and the dominating auger recombination caused by the highly doped $n$-type emitter may result in uniform light-induced current and IQE at 405 nm.

However, the result measured at 979 nm shows a significant difference. With increased sub-grain content, the IQE significantly decreases, and the IQE distribution changes from uniform to non-uniform. These non-uniform areas in Figure 7 correspond with sub-grain regions. The penetration depth of 979 nm wavelength light is about 99 $\mu$m in crystalline silicon. Therefore, the IQE measured at this wavelength indicates information on the bulk material. The locally enhanced recombination caused by defects and grain boundaries shown as subgrains significantly affects the light-induced current in the bulk silicon.

3.6. Solar Cell Performance of Wafers without Subgrains. To illustrate the effect of subgrains on cell performance, wafers with hardly any sub-grain but with a small-grain band, which means an area with many small grains, were chosen to prepare LDSE cells. Figures 8(a) and 8(b) show optical and EL images of fabricated solar cells, respectively. Hardly any
Figure 5: Spatial distribution of minority carrier diffusion length of solar cells fabricated on different substrates: ((a)–(c)) grades A–C, respectively.

<table>
<thead>
<tr>
<th>$V_{oc}$ (mV)</th>
<th>$J_{sc}$ (mA/cm²)</th>
<th>FF (%)</th>
<th>Eta (%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>634.1 ± 1.4</td>
<td>37.2 ± 0.1</td>
<td>76.9 ± 0.5</td>
<td>18.2 ± 0.1</td>
</tr>
</tbody>
</table>

This study showed that cell efficiency decreases with increased sub-grain content. The reflectance of monocrystalline and sub-grain areas after texturing and ARC coating is the same. However, the IQE of cells with sub-grain areas significantly decreases within 600 nm to 1100 nm compared with monocrystalline areas. The local ideality factor result shows that the shunting property and metal contact of mono-like crystalline LD solar cells are independent of subgrains. However, subgrains shown as dark line clusters in the EL image significantly decrease the diffusion length and the IQE at long wavelengths in these regions. This finding indicates that defects caused by subgrains act as the recombination center for minority carriers, degrading the IQE in middle- and long-wavelength range and further affecting the solar cell performance. Finally, an average efficiency of 18.2% was achieved on wafers with hardly any sub-grain but with a small-grain band to indicate that the negative effect of sub-grain cell performance would be more serious than that of small-grain band. This average efficiency also illustrates the promising application potential of mono-like wafers. It can be inferred that higher efficiency would be obtained on the mono-like wafers without any sub-grain areas and small-grain bands. Thus, defects appeared as subgrains significantly affect cell performance, and dislocations must be eliminated by optimizing the casting process for the industrial application of mono-like crystalline silicon.

4. Conclusions

The application of optically perfect mono-like wafers can be challenging because of abnormal efficiency fluctuations.
Figure 6: IQE distribution of solar cells fabricated on different substrates at 405 nm wavelength: ((a)–(c)) grades A–C, respectively.

Figure 7: IQE distribution of solar cells fabricated on different substrates at 979 nm wavelength: ((a)–(c)) grades A–C, respectively.
Figure 8: Optical (a) and EL (b) images of solar cells fabricated on wafers with hardly any sub-grain but with a small-grain band.

Acknowledgments

This work was supported by the National High Technology Research and Development Program of China (Grant no. 2011AA050515) and National Basic Research Program of China (Grant no. 2012CB934204).

References


