

## Research Article

# Aluminum-Doped SnO<sub>2</sub> Hollow Microspheres as Photoanode Materials for Dye-Sensitized Solar Cells

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Al doped SnO<sub>2</sub> microspheres were prepared through hydrothermal method. As-prepared SnO<sub>2</sub> microspheres were applied as photoanode materials in dye-sensitized solar cells (DSCs). The properties of the assembled DSCs were significantly improved, especially the open-circuit voltage. The reason for the enhancement was explored through the investigation of dark current curves and electrochemistry impedance spectra. These results showed that the Al doping significantly increased the reaction resistance of recombination reactions and restrained the dark current. The efficient lifetime of photoexcited electrons was also obviously lengthened.

## 1. Introduction

Dye-sensitized solar cells (DSCs) have been actually promoted by the development of nanocrystal materials, especially the preparation of TiO<sub>2</sub> nanocrystals with different morphology [1, 2]. High photoelectric conversion efficiency up to 14.3% has been obtained [3]. However, the band gap of TiO<sub>2</sub> is about 3.2 eV which causes obvious catalyzing properties under UV light. The organic solvent in DSCs might be slightly decomposed by TiO<sub>2</sub> nanoparticles on the photoanode under sunlight. As an alternative to TiO<sub>2</sub>, SnO<sub>2</sub> has been extensively investigated as a photoanode material in DSCs. SnO<sub>2</sub> has a wider band gap (about 3.6 eV) than that of TiO<sub>2</sub> which was inactive to the organic solvent in DSCs. And SnO<sub>2</sub> has high electron mobility (about 150 cm<sup>2</sup>V<sup>-1</sup>s<sup>-1</sup>) which is a benefit for the collection of the photoexcited electrons in the photoanode [4, 5]. However, the band edge of conduction band edge of SnO<sub>2</sub> is -4.5 eV (vacuum level) which causes serious back reaction between the electrons in the conduction band and the oxide ions in the electrolyte. It is an efficient way to inhibit these back reactions by doping SnO<sub>2</sub> with other metal elements. Duan et al. doped SnO<sub>2</sub> nanoparticle with Al and found the tuning of the conduction band and suppression of charge recombination [6]. Li et al.

prepared Zn-doped SnO<sub>2</sub> nanocrystals to obtain longer electron lifetimes and higher dye loading [7].

In this work, we prepared Al doped SnO<sub>2</sub> hollow microspheres. The electron recombination was efficiently restrained and the photoelectrical conversion efficiency was significantly enhanced compared with that of pure SnO<sub>2</sub>.

## 2. Experimental

**2.1. Preparation of Pure SnO<sub>2</sub> Power and Al Doped SnO<sub>2</sub> Power.** SnO<sub>2</sub> microspheres were synthesized using the hydrothermal method as follows. For the preparation of SnO<sub>2</sub> power, 0.8 g of stannous chloride dihydrate (SnCl<sub>2</sub>·2H<sub>2</sub>O) was dissolved in 80 mL deionized water. D-Glucose was used as the soft template. The content of D-glucose was 7.024 g. The former mixture was stirred for 30 min. at room temperature. The resulting well-distributed mixture was transferred into a 100 mL autoclave for hydrothermal reaction at 180°C. The hydrothermal reaction time was 16 h. After the autoclave was cooled to room temperature, the product was collected by centrifugation and washed with distilled water and ethanol several times. The obtained sample was dried at 60°C overnight. The resulting brown power was calcined at 550°C for 1 h in air to obtain the final product. For preparing the

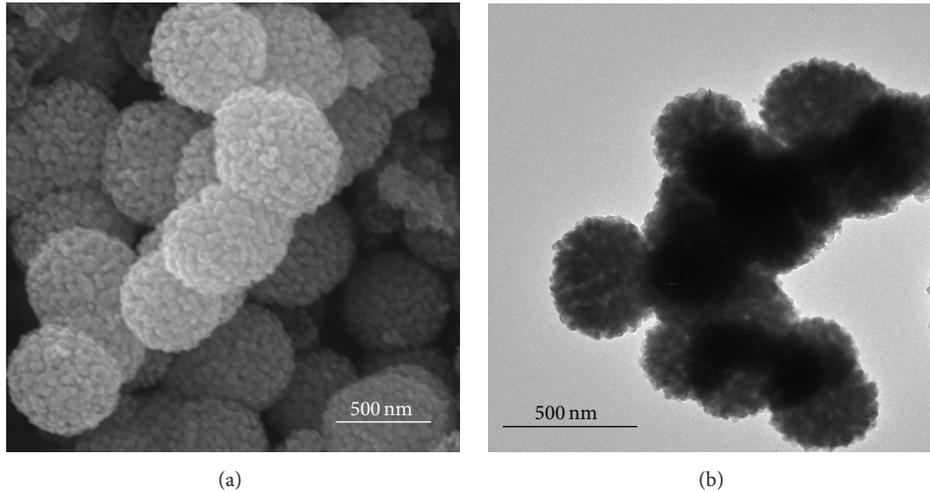


FIGURE 1: SEM (a) and TEM (b) of  $\text{SnO}_2$  hollow microspheres doped with 1.5% Al.

Al doped  $\text{SnO}_2$  sample, aluminum(III) chloride ( $\text{AlCl}_3$ ) was added to the precursor solution. The addition amount of Al was controlled to be 0.5%, 1.0%, 1.5%, and 2.0% (at%) of Sn content in the solution. The preparation processes of Al doped  $\text{SnO}_2$  powder were controlled to be the same as that described above except for the addition of  $\text{AlCl}_3$ . The obtained samples of pure  $\text{SnO}_2$  and Al doped  $\text{SnO}_2$  powder were denoted as pure  $\text{SnO}_2$ , 0.5% Al doped  $\text{SnO}_2$ , 1.0% Al doped  $\text{SnO}_2$ , 1.5% Al doped  $\text{SnO}_2$ , and 2.0% Al doped  $\text{SnO}_2$ , respectively.

**2.2. Fabrication of DSSC Based on Pure  $\text{SnO}_2$  and Al Doped  $\text{SnO}_2$ .** To prepare the working electrode,  $\text{SnO}_2$  or Al doped  $\text{SnO}_2$  slurry was covered on fluorine-doped tin oxide (FTO) glass ( $1 \times 2 \text{ cm}^2$ ,  $15 \Omega \text{sq}^{-1}$ , Opvtech) using a doctor blade technique and then sintered at  $450^\circ\text{C}$  for 30 min. After cooling to  $80^\circ\text{C}$ , the samples were immersed in  $5 \times 10^{-4} \text{ mol L}^{-1}$  ethanol of N719 dye for 24 h. Pt counter electrode was prepared by spreading 5 mM  $\text{H}_2\text{PtCl}_6$  aqueous solution on an FTO glass substrate, followed by pyrolyzation at  $390^\circ\text{C}$  for 15 min. The mixture of 0.6 M dimethylpropylimidazolium iodide, 0.1 M iodine, 0.5 M 4-*tert*-butylpyridine, and 0.1 M lithium iodide in methoxyacetonitrile was prepared as the electrolyte of DSCs.

**2.3. Characterization and Optical Measurements.** The crystalline phase of the samples was characterized by DX-2700 X-ray diffractometer (XRD) with monochromatized Cu K irradiation. The morphology was studied using a JSM-7001F field emission scanning electron microscope (FE-SEM) and JEM 2100 transmission electron microscope (TEM). XPS measurements were performed in Thermo Scientific ESCALAB 250 station (Thermo Fisher Scientific, Massachusetts, USA). Photocurrent density-voltage ( $J$ - $V$ ) characteristics were measured using a Keithley 2440 Source Meter under AM 1.5 G illumination from a Newport Oriel Solar Simulator with an intensity of one sun. The incident light intensity was calibrated with a standard Si solar cell provided by Newport

Oriel. The active cell area of the assembled DSCs was  $0.25 \text{ cm}^2$ . An electrochemistry workstation (IM6) was used to investigate the electrochemical impedance spectra (EIS) of DSCs. This measurement was also carried out with the same structured DSCs as that used in the former experiments. The impedance measurement of DSCs was recorded under dark condition at the bias potential of  $-0.6 \text{ V}$  over a frequency range of 0.1–1 MHz with an AC amplitude of 10 mV.

### 3. Results and Discussion

Figure 1(a) shows the morphology of the prepared Al doping  $\text{SnO}_2$  microspheres. The diameter is 300–500 nm. These  $\text{SnO}_2$  microspheres were piled up with homogeneous nanoparticles. The size of the particles is 20–40 nm. There are some broken microspheres which indicate that the prepared  $\text{SnO}_2$  is hollow microspheres. This structure is a benefit for the absorbing of dye and the diffusion of the electrolyte in DSCs. TEM was also carried out to confirm the hollow spheres structure of  $\text{SnO}_2$ . The TEM of  $\text{SnO}_2$  microspheres is shown in Figure 1(b). The whole  $\text{SnO}_2$  spheres show almost the same darkness which indicates that the thickness at the center of  $\text{SnO}_2$  spheres is almost the same as that of edge. Therefore, the as-prepared  $\text{SnO}_2$  should be hollow spheres.

Figure 2 shows the XRD spectra of  $\text{SnO}_2$  microspheres doped with different Al content. The prepared  $\text{SnO}_2$  microspheres correspond to the cassiterite structured  $\text{SnO}_2$  (JCPDS database card number 41-1445). There is almost no change in the XRD spectra with the addition of Al element which might be because the content of added Al element is too little to change the structure of  $\text{SnO}_2$ . The crystalline particle size ( $D$ ) could be estimated from the 110, 101, and 211 diffraction peak using the Scherrer equation [8]:

$$D = \frac{0.89\lambda}{\beta \cos \theta}, \quad (1)$$

where  $\lambda$  is the wavelength of the X-ray,  $\beta$  is the full-width at half-maximum (FWHM), and  $\theta$  is the Bragg angle in the

TABLE 1: Detailed photovoltaic parameters of the DSCs based on the SnO<sub>2</sub> hollow microspheres doped with different Al content.

Al content/at%	$V_{OC}$ mV	$J_{SC}/\text{mA cm}^{-2}$	FF/%	Efficiency/%
0	179	8.05	32.2	0.46
0.5	347	9.40	51.7	1.69
1.0	412	10.71	53.4	2.36
1.5	474	11.48	55.8	3.04
2.0	442	9.56	55.2	2.34

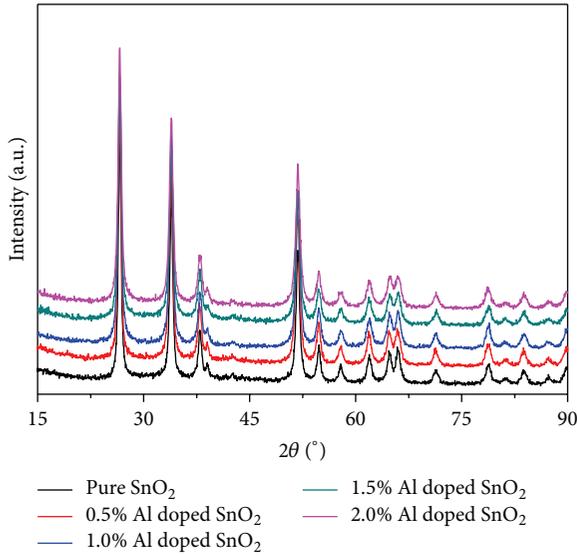


FIGURE 2: XRD of SnO<sub>2</sub> hollow microspheres doped with different Al content.

diffraction pattern. The particle size was estimated to be about 17.0 nm for Al doped SnO<sub>2</sub>.

XPS were carried out to confirm the introduction of Al into the SnO<sub>2</sub> hollow spheres. The survey scan spectra and the element narrow scan of 1.5% (at%) Al doped SnO<sub>2</sub> are shown in Figure 3. The XPS of pure hollow microspheres were shown in the insert of the corresponding spectra. The binding energies of O 1s and Sn 3d show no obvious shift after Al doping. However, a new weak peak appears at about 75 eV which corresponds to Al 2p. The content of Al was also estimated to be about 0.61% (at%). The content of Al is so little that the O 1s and Sn 3d have no change which is in accordance with the result of XRD.

Figure 4(a) shows the  $J$ - $V$  curves of the DSCs assembled with Al doped SnO<sub>2</sub> microspheres photoanode. The specific parameters of these  $J$ - $V$  curves were summarized in Table 1. The photoanode prepared with pure SnO<sub>2</sub> microspheres shows low open-circuit voltage ( $V_{OC}$ , 179 mV) due to the low conduction band edge.  $V_{OC}$  increased from 179 mV to 474 mV when the addition of Al increased from 0 to 1.5% (at%). At the same time, the short circuit current density ( $J_{SC}$ ) was also enhanced significantly from 8.05 mA cm<sup>-2</sup> to 11.48 mA cm<sup>-2</sup>. The photoelectric conversion efficiency ( $\eta$ )

increased from 0.46% to 3.04%. However,  $V_{OC}$  and  $J_{SC}$  both decreased when the addition of Al further increased to 2.0% (at%). The increase of  $V_{OC}$  should be due to the inhibition of back reaction after the doping of Al. Figure 4(b) shows the dark current-voltage curves. The dark current density becomes weaker and weaker with the increase of Al doping content. The depression of dark current should be one key reason for the increase of  $V_{OC}$ .

Electrochemical impedance spectroscopy (EIS) is an efficient method to investigate the recombination process of the photoexcited electrons. EIS was carried out on the SnO<sub>2</sub> and Al doped SnO<sub>2</sub> under dark condition. The bias potential is -0.6 V. The Nyquist plots are shown in Figure 5(a). The Nyquist plots were fitted using the equivalent circuit shown in the insert of Figure 5(a). In the equivalent circuit,  $R_S$  represent the series resistors during the transport of electrons.  $R_{ct1}$  correspond to the resistance during the charge-transfer processes occurring at the counter electrode/electrolyte interface.  $R_{ct2}$  should correspond to the recombination of electron at the SnO<sub>2</sub>/electrolyte interface. From the fitting result of Nyquist plots, it can be seen that  $R_S$  and  $R_{ct1}$  show few changes before and after Al doping. However,  $R_{ct2}$  increases from 35  $\Omega$  cm<sup>-2</sup> to 127  $\Omega$  cm<sup>-2</sup> after Al doping. This indicates that the photoelectron recombination is efficiently inhibited. Figure 5(b) shows the Bode phase plots of the pure SnO<sub>2</sub> sphere film and the Al (1.5%, at%). There are two electrochemical processes,  $\omega_1$  and  $\omega_2$ , which occur at high frequency ( $10^3$  Hz– $10^5$  Hz) and low frequency (1 Hz to  $10^3$  Hz), respectively.  $\omega_1$  corresponds to the charge-transfer processes occurring at the counter electrode/electrolyte interface [9, 10].  $\omega_2$  should correspond to the charge-transfer processes occurring at the SnO<sub>2</sub>/electrolyte interface. The characteristic frequency of  $\omega_2$  may reflect the electron lifetimes ( $\tau_e$ ) of the injected electrons [11]. The lifetimes ( $\tau_e$ ) of the photoexcited electron in the photoanodes were determined using the following equation:

$$\tau_e = \frac{1}{2\pi f_{\max}}. \quad (2)$$

The characteristic frequencies of these photoanodes, SnO<sub>2</sub> and Al doped SnO<sub>2</sub>, are 3.1 and 1.2 Hz, respectively. According to (2), the electron lifetimes ( $\tau_e$ ) were calculated to be about 51 ms and 132 ms for the SnO<sub>2</sub> and Al doped SnO<sub>2</sub> electrodes, respectively. It can be seen that Al doping can enhance the efficient electron lifetime of SnO<sub>2</sub> electrodes. This result is in accordance with that of dark current experiments (shown in Figure 4).

## 4. Conclusions

Al doped SnO<sub>2</sub> microspheres were prepared through hydrothermal method using glucose as template. The SnO<sub>2</sub> microspheres were piled up with SnO<sub>2</sub> nanoparticles. As-prepared SnO<sub>2</sub> microspheres were applied as photoanode materials in dye-sensitized solar cells. The results showed that the Al doping significantly restrained the dark current and improved the open-circuit voltage of the cells. Electrochemistry impedance spectra showed that the reaction resistance of recombination reactions increased sharply after doping of Al and the efficient

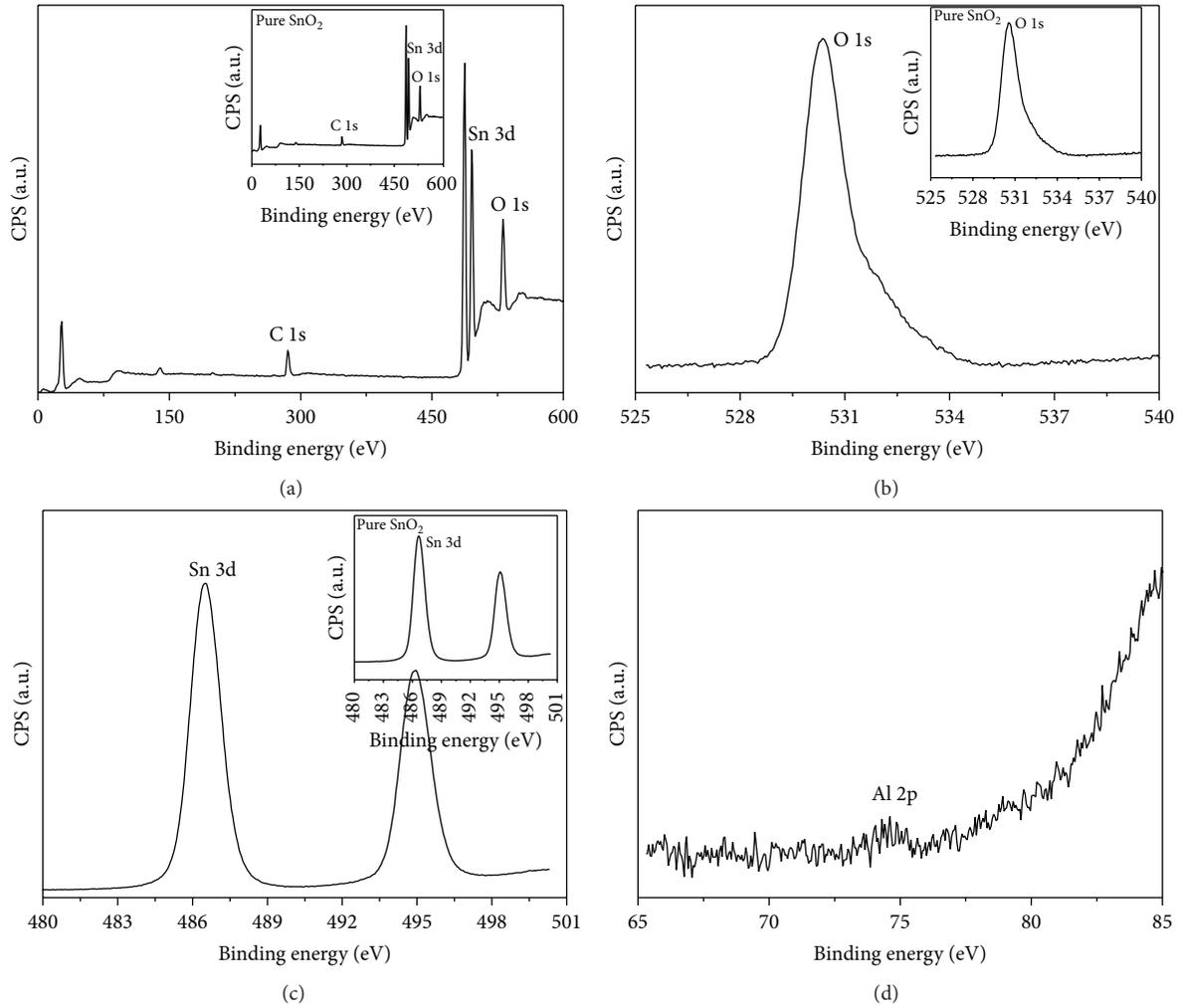


FIGURE 3: XPS of SnO<sub>2</sub> hollow microspheres doped with 1.5% Al and pure SnO<sub>2</sub> hollow microspheres (given in the insert): (a) full spectra; (b) high resolution of O 1s; (c) high resolution of Sn 3d; (d) high resolution of Al 2p.

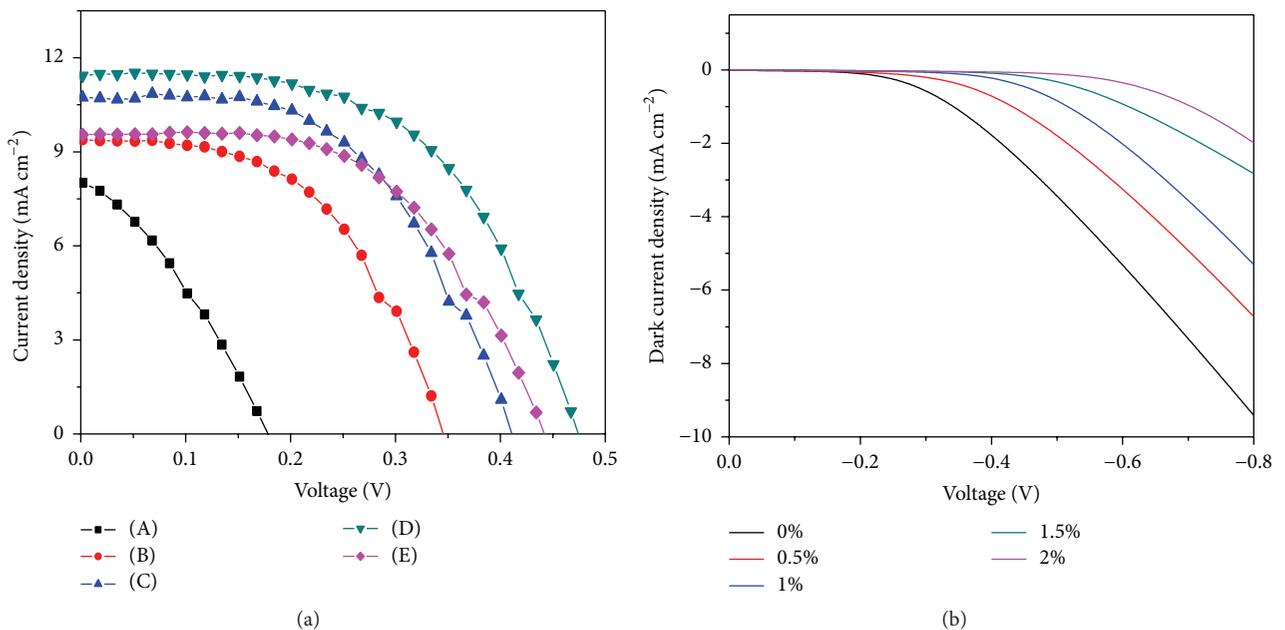


FIGURE 4:  $J$ - $V$  characteristic of the DSCs based on the SnO<sub>2</sub> hollow microspheres doped with different Al content (a) and the corresponding dark current-voltage curves (b).

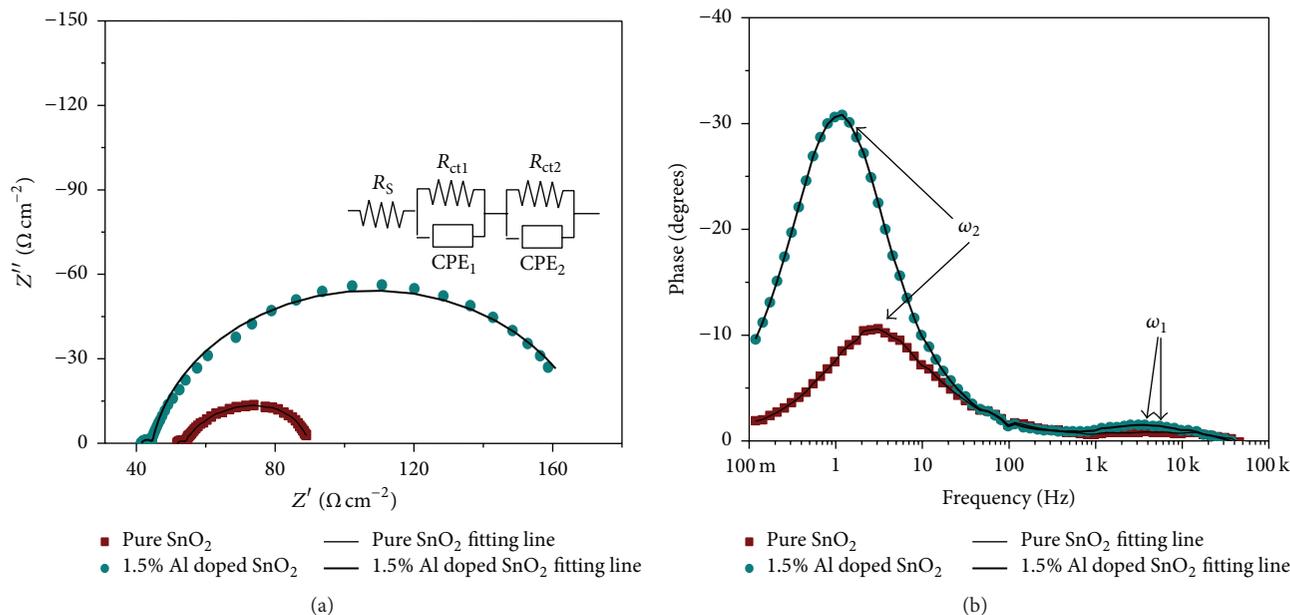


FIGURE 5: Nyquist plots (a) and Bode phase plots (b) of the DSCs based on the pure SnO<sub>2</sub> and 1.5% Al doped SnO<sub>2</sub> hollow microspheres.

lifetime of photoexcited electrons was lengthened. The total photoelectrical conversion efficiency was improved from 0.46% to 3.04% after Al doping.

## Competing Interests

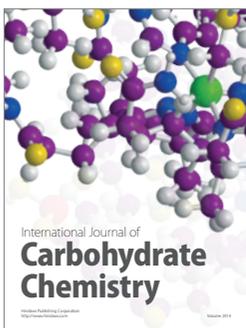
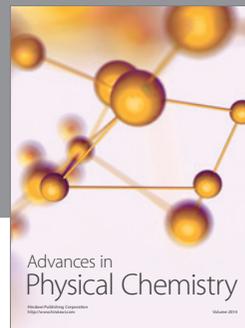
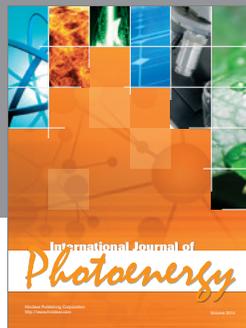
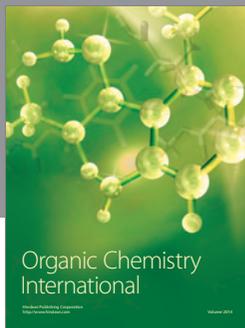
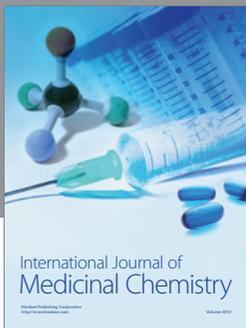
The authors declare that they have no competing interests.

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