



Post Deposition Annealing Effects on Optical, Electrical and Morphological Studies of ZnTTBPc Thin Films

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Abstract: Phthalocyanines (Pcs) act as efficient absorbants of photons in the visible region, specifically between 600 and 700nm. It will produce an excited triplet state. In this paper we report the annealing effects of optical, electrical and surface morphological properties of thermal evaporated Zinc-tetra-tert-butyl-29H, 31H phthalocyanine (ZnTTBPc) thin films. The optical transmittance measurements were done in the visible region (400-800 nm) and, films were found to be absorbing in nature. From spectral data the absorption coefficient α , dielectric constant ϵ and the extinction coefficient k were evaluated and, results discussed. Also the optical band gap of the material was estimated. The activation energies were measured. Scanning electron microscopic studies was carried out to determine surface uniformity of films.

Key words: Thermal evaporation, phthalocyanines, band gap energy, dielectric constant, conductivity, surface morphology.

Introduction

Metal phthalocyanines have properties such as their thermal and chemical stability, ability to form well-ordered thin films and wide absorption band within optical region. This makes them very attractive for applications in optical logic display devices [1], electrophotography, solar cells [2, 3], sensitizes and colour filters [4] and optical recording [5]. They can easily be sublimed to get pure thin films without decomposition. Due to their strong absorption in the ultraviolet and visible region, phthalocyanine and its metal complexes are considered as

electrophotographic material [6], optical recording media [7], non linear optical materials [8], organic light emitting diodes [9] and Schottky diodes [10].

Various workers have studied the basic properties of many of the metallo-phthalocyanines (MPcs) in detail [11-13]. Zinc phthalocyanine is the most interesting one and has been intensively studied. This material is reported to be superior to other metal substituted phthalocyanines in terms of sensitivity, reproducibility, response and recovery times in detection of gases. The better recovery characteristics of ZnTTBPC relates to its loosely packed structure (larger lattice spacing), which makes easier for the deeply adsorbed gas molecules to desorb and discharge. A few years ago, metal porphyrin complexes have been reported for applications to gas sensing materials [14-15].

Peripheral substitution in Pc, which is rather insoluble in organic solvents, provides a method of enhancing solubility of Pc. Four *tert*-butyl (tBu) groups are added to four periphery benzene rings of Zinc(II) phthalocyanine, which gives the planar molecules a three-dimensional structure *tert*-Butyl groups. This is widely used as a substituent to design functional molecules, enhance solubility, improve chemical stability, and light fastness of organic molecules [16]. Figure 1 shows the molecular structure of ZnTTBPC thin films.

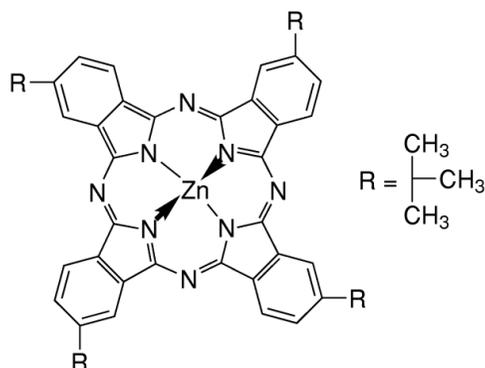


Figure 1. Molecular structure of ZnTTBPC thin films.

It is known that *tert*-butyl substitutes are very bulky. They can prevent coplanar π - π -interaction between phthalocyanine molecules and influence on crystalline polymorphic state of the metallophthalocyanine molecules.

Materials and Measurements

ZnTTBPC thin film was prepared by vacuum sublimation of ZnTTBPC powder procured from Aldrich USA. Thin film deposition was done at room temperature onto pre-cleaned glass plates kept at a base pressure of 4×10^{-5} Torr using the Hind Hivac 12 A4 coating unit. The evaporation was carried out by resistive heating of ZnTTBPC powder from a molybdenum boat and the deposition rate was kept constant between 10 and 12 nm/min. The films were annealed at four different annealing temperatures- 323, 373, 473 and 523 K for

one hour in air. Thickness of the film was determined by Tolansky's multiple beam interference technique [17].

Optical characterization was done using UV-Vis-NIR absorption and reflection spectroscopy (model: JASCO V570) over a range of 300-900 nm. The electrical study was done using a programmable Keithley electrometer (model no.617) for all the annealed samples. For ohmic contact vacuum deposited silver, with an inter-electrode distance of 1cm was used. To avoid any possible contamination, conductivity measurements were performed in vacuum and in a dark chamber to reduce the photoconductive contribution of the molecules. The morphological studies were done by scanning electron microscopy (SEM, model: JEOL 5400).

Results and Discussion

Optical characterization

Figure 2 shows the absorbance versus wavelength of ZnTTBPC thin film. It was found that all films showed identical absorption coefficient patterns and there was no systematic change with respect to the annealing temperature. The film annealed at 323K exhibited lowest absorbance while the ZnTTBPC film annealed at 373K showed the highest absorbance. This was due to the different percentage loss of the unstable morphology layer on the ZnTTBPC film during the annealing process. In absorption spectra we got two bands. The Q-band was obtained in the region 550 to 712nm while the B band is in the region 300 to 400 nm. The Q-band showed two shoulder peaks while the B-band showed only one. This is in a good agreement with the results obtained by other researchers [17]. The Q-band corresponds to the excitation between the ground state π (HOMO) to the π^* (LUMO) [18]. Optical band gap values were obtained from analysis of the absorption spectrum in high energy region.

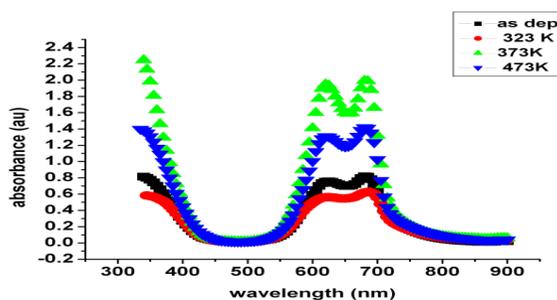


Figure 2. (Optical absorption spectra of ZnTTBPC thin film with different annealing temperature.

Intense band in the blue end of the UV region called B band positioned at 334nm is attributed to the deeper π levels \rightarrow LUMO transitions, which gave the fundamental absorption edge, while broad band that appeared in the visible region called Q band, positioned at 681nm, has been attributed to the $\pi \rightarrow \pi^*$ electronic transition from the HOMO (highest occupied molecular orbital) to the LUMO (lowest unoccupied molecular orbital) of

the Pc^{2-} ring which gives the excitonic energy. In order to find the optical band gap of ZnTTBPc thin films, the fundamental absorption edge was analyzed using the relation:

$$\alpha = \alpha_0 (h\nu - E_g)^n$$

n determines the type of transition, n can take values 1/2 and 3/2 for allowed and forbidden transitions respectively [19]. A linear fit is obtained for $n=1/2$, indicating the existence of a direct gap. A graph is drawn with α^2 vs. $h\nu$ from which E_g can be obtained. The plots of α^2 vs. $h\nu$ for ZnTTBPc thin films of thickness 250nm is given in figure 3. The extrapolation of the graphs to $\alpha^2=0$ yields the optical band gap E_g . The values of optical band gap of ZnTTBPc are collected in Table. 1

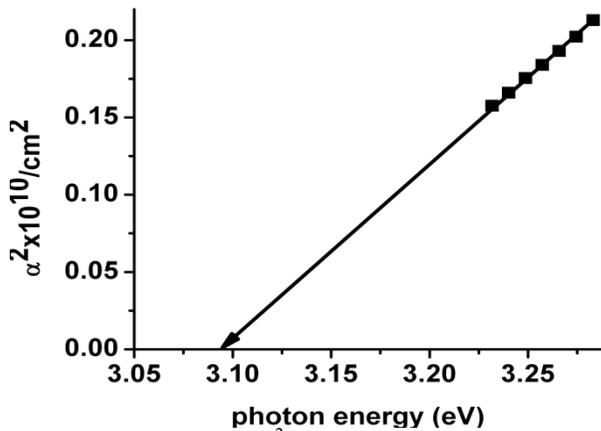


Figure 3. Plot of α^2 versus $h\nu$ of ZnTTBPc thin film.

The optical band gap was determined as 3.09 eV within the accuracy of ± 0.01 eV. The value of band gap energy does not change considerably for higher annealing temperature. Values of the energy gap for as deposited and annealed films are depicted in Table 1.

Table 1. Band gap energy of as deposited and annealed ZnTTBPc thin films.

Annealing temperature (K)	Band gap energy (eV)
As deposited	3.09
323	3.1
373	3.1
423	3.09
523	3.1

Due to weak Van der Waal's interaction between the molecules in the film, their optical properties are not significantly changed when compared with free molecules.

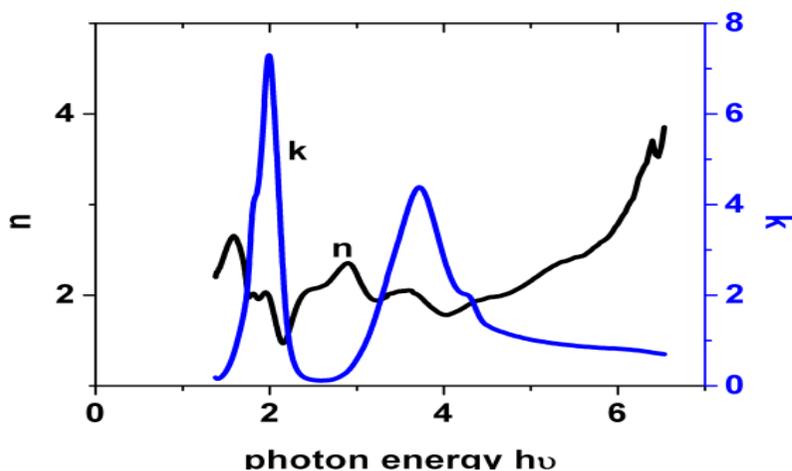


Figure 4. The refractive index and the extinction coefficient of ZnTTBPc as a function of photon energy.

The most accurate method for determining the energy band structure of semi conductors is one based on investigating the spectral distribution of both the refractive index n and the absorption index k . The reflectivity R of an absorbing medium of indices n and k in air for normal incidence is given by

$$R = \frac{(n-1)^2 + k^2}{(n+1)^2 + k^2}$$

Where n is the refractive index, k is the extinction coefficient ($k = \alpha\lambda/4\pi$), λ is the wavelength.

The variation of the refractive index and the extinction coefficient with photon energy for ZnTTBPc films is shown in figure 4. The rise and fall in the extinction coefficient and refractive index is due to the variation in the absorbance.

The optical properties of the film are characterized by complex refractive index ($N = n - ik$) and complex dielectric constant ($\epsilon = \epsilon_1 - i\epsilon_2$). If the refractive index n and extinction coefficient k are known, the real and imaginary parts of dielectric constants (ϵ_1 and ϵ_2) of the film can be also estimated.

$$\begin{aligned}\epsilon_1 &= n^2 - k^2 \\ \epsilon_2 &= 2nk\end{aligned}$$

The real part ϵ_1 generally relates to the dispersion, while the imaginary ϵ_2 gives a measure of the dissipation rate of the wave in the medium.

The dependence of photon energy ϵ_1 and ϵ_2 is plotted in Figure 5.

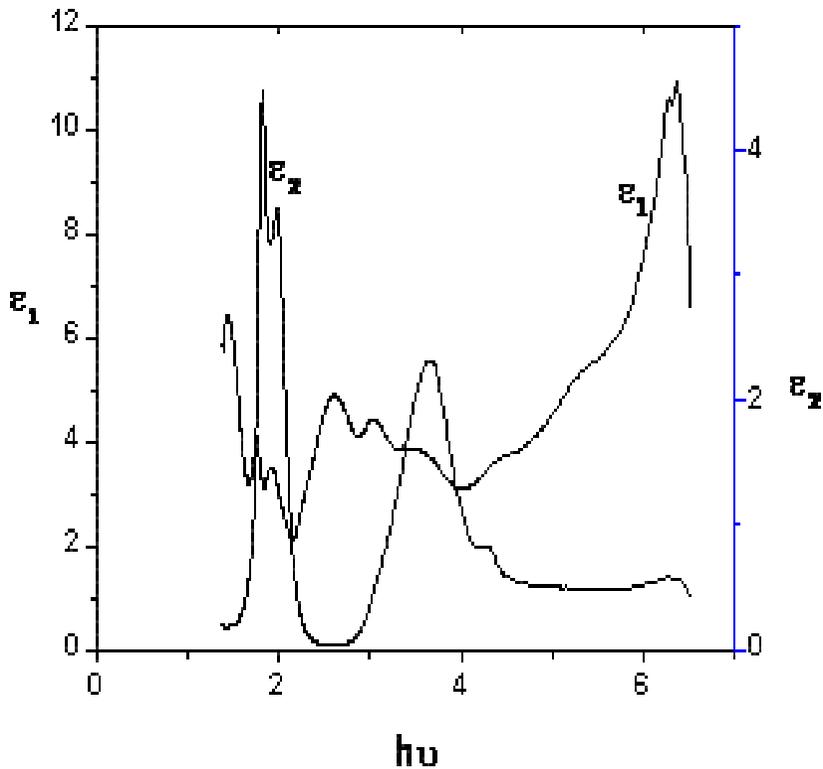


Figure 5. The real and imaginary parts of the dielectric constant of ZnTTBPC as a function of its photon energy.

Electrical studies

Figure (6) gives the plot of $\ln \sigma$ vs. $1000/T$ for the ZnTTBPC thin films at different annealing temperature. Hot probe method was used to find the type of charge carriers. The charge carriers in ZnTTBPC are found to be p-type. The activation energy of the films were calculated from the slope of the Arrhenius plot of conductivity (Figure 6) using the relation.

$$\sigma = A \exp\left(\frac{-E_1}{kT}\right) + B \exp\left(\frac{-E_2}{kT}\right) + \dots$$

Where σ is conductivity, E_1 is the intrinsic energy gap and E_2 is the activation energy needed to excite the carriers from the corresponding trap levels to the conduction band[20], k is the Boltzmann constant, T is the absolute temperature A and B are constants.

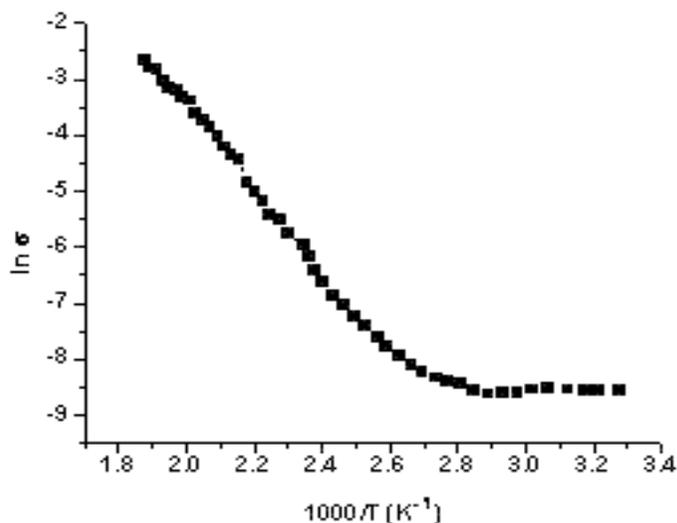


Figure 6. $\ln \sigma$ vs. $1000/T$ of ZnTTBPc thin films.

It is seen from the plot that for all the films, the conductivity is a linear function of the reciprocal of temperature. Linearity of the plot is nearly same for all the films.

The resistance of the film was measured using a Keithley programmable electrometer. The electrical conductivity σ of the film of resistance R , length l , breadth b and thickness t are given by

$$\sigma = \frac{l}{Rbt}$$

The activation energies in the intrinsic region (E_1) and impurity scattering regions (E_2) are given in Table 2.

Table 2. Activation energies of ZnTTBPc thin films.

Annealing temperature(K)	Activation energy(eV)	
	E_1	E_2
As deposited	0.66	0.08
373	0.62	0.17
423	0.63	0.02
473	0.64	0.05
523	0.62	0.12

From the table we can see that as the annealing temperature increases the activation energy E_1 increases from 0.62 to 0.64, there after it decreases. Also E_2 is found to increases 423K.

Surface morphology

Surface morphological studies were carried out using the scanning electron microscopy. From these images it is clear that, the surface morphology is different. Figure 7(a) shows that the film deposited at room temperature had smooth featureless surface with small crystal grain structure. However when the film is annealed at higher temperature we get the fibre like structure seen in figure7(c).

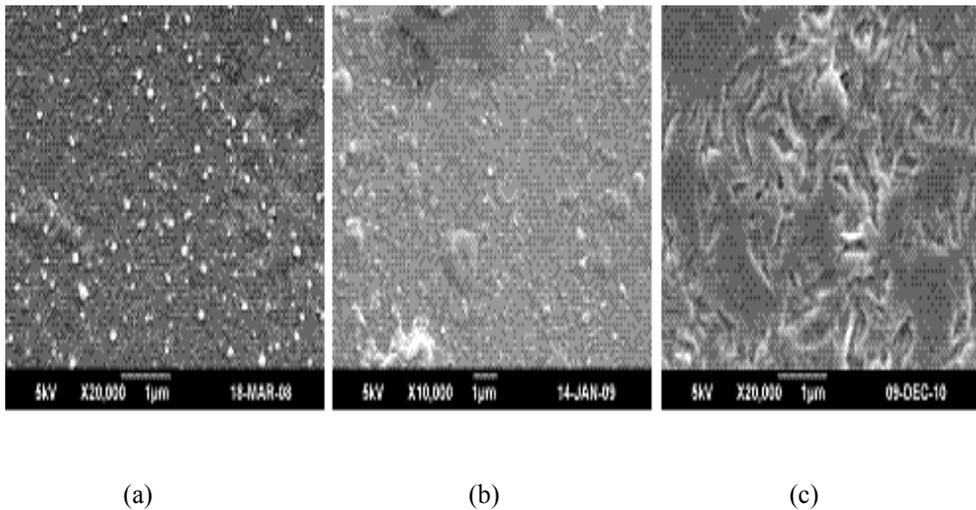


Figure 7. Surface morphology of as deposited film (a), 373K annealed film (b) and 523K annealed film(c).

Summary and Conclusions

The effects of air annealing on the optical, electrical and surface morphology of the ZnTTBPc thin films were investigated. The optical properties of ZnTTBPc thin films deposited at room temperatures was studied in the spectral range 300-900nm. The two main absorption bands of metal phthalocyanine are ascribed to the π to π^* transition. The plots of the refractive index n , and the extinction co-efficient k , with photon energies were studied. The variation of ϵ_1 & ϵ_2 , the real and imaginary parts of the dielectric constant, with the photon energy was also described. From the slope of linear portions of the graph, $\ln \sigma$ vs. $1000/T$, the activation energies were calculated. The activation energy E_1 corresponding to the high temperature region was due to intrinsic carriers. The activation energy E_2 corresponds to the extrinsic conduction due to impurity states. Surface morphology is also studied.

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