Research Article

Two-Step Self-Assembly and Lyotropic Liquid Crystal Behavior of TiO₂ Nanorods

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Several self-assembly structures of anatase TiO₂ nanorods were obtained by a two-step assembly process, and these structures formed differently lyotropic liquid crystal in solution. Primary self-assembly occurred in synthesis process and formed two structures, in the morphology of ribbon and honeycomb, respectively. Secondary-assembly took place when the products were placed at lower temperature, where unique structures were obtained as the relative amount of ribbon and honeycomb changed with the increase of TiO₂ concentration. These structures showed nematic, spherulites, and lamellar phases. The mechanism of the two-step self-assembly was clarified. The driving force of primary assembly is deduced to be anisotropic attractive force, for NRs can assemble at any concentrations, while gravity is the driving force of the secondary assembly. It is worth mentioning that this paper is the first report about spherulites composing of anatase TiO₂ nanorods. The spherulites obtained were negative or of tangential type, and its structure, growth process, and temperature influences were also investigated. The spherulites may have promising application in temperature microsensor.

1. Introduction

For more than a hundred years, the new liquid crystal materials have acquired increasing attraction of scientists who focus on practical application and physical principles. It is a major breakthrough when mineral crystals were first observed by Zocher [1] in 1925 and since then many reports had focused on inorganic liquid crystal (LC) phase with nanometer-scale objects [2, 3]. Inorganic nanocrystals may be viewed as an emerging new class of macromolecules [4, 5] and enable self-assembly techniques to be extended to inorganic materials, which possesses a diverse range of optical, electrical, and magnetic properties [6]. Many types of inorganic LC phases have already been observed, such as nematic, lamellar, columnar, and spherulites [7–9]. The development of liquid crystal composing of different morphology nanoparticles has been explored, including nanosphere, nanorods, nanodisk, graphene, and nanotubes [10–14], among which the study of rod-shape lyotropic LC behavior is under the spotlight.

The anisotropy of the structural units is necessary for formation of lyotropic LC. Moreover, these units were found to exhibit orientational order in a certain range of concentrations. Recently, the study of oriented nanorods attracts a lot of attention due to its unique properties [15, 16]. There are two basic approaches to the assembly of nanoscale rigid-rod, while the assembly of nanorods (NRs) is necessary for the formation of lyotropic liquid crystals. The first one relies on the Onsager mechanism [17], driving the parallel alignment of NRs when their volume fraction and their aspect ratio are high enough. The second approach utilizes anisotropic attractive forces and thus allows one to produce assembled structures at practically any NRs concentration [18]. The formation of LC phase also offered the potential to orient anisotropic nanoparticles and obtained a large scale of self-assembly superlattic. Some groups had obtained various alignment of anisotropic nanoparticles by self-assembly in liquid crystal, including nanowire [19] and nanorods [20, 21]. Meanwhile oriented NRs self-assembly materials might be obtained by
applying the external field to the LC. Zorn and coworkers had reported on switching behavior under an applied AC (Alternating Current) electric field of ZnO nanorods [22]. In this respect the orientation of functional semiconducting nanoparticles is interesting for materials science [23, 24]. In order to obtain the stable nanorods LC in organic solvent, the particles have to be surface-functionalized to overcome the strong adhesion forces among them and achieve good dispersion [25]. Therefore, the concept of “hairy” rods, which was originally developed for stiff main-chain LCs, is promising [26]. In this model, a stiff insoluble core is solubilized by long alkyl chains (the hairs) on its surface. Other semiconductor NRs were functionalized to give the liquid-crystalline phase in organic solvent, like CdSe [27].

Our group has obtained several self-assembled TiO$_2$ nanorods structures and tried to study the mechanism of the assembly process [28]. However, some of the critical problems are still waiting to be solved, such as diverse structure of self-assembly and the control of the assembly process. Work presented here introduces a controllable preparation of self-assembly and the control of the assembly process.

### 3. Result and Discussion

#### 3.1. Synthesis and Primary-Assembly of Nanorods

We synthesized TiO$_2$ NRs solution with three different concentrations by keeping the molar ratio of TBT : OA : OLA = 2 : 10 : 3 and varying the amount of TBT, and the concentration of NRs were 0.188 M, 0.231 M, and 0.262 M, respectively (At this point, the volume of TBT was 0.48 mL/0.72 mL/0.96 mL). Figure 1(a) shows typical powder XRD result and TEM image (inset) of the 0.231 M products (other concentrations show similar results), which clearly demonstrates that the products are composed of pure anatase TiO$_2$ without any other crystalline phase and there is an obvious indication of the crystal growth direction. The [001] diffraction peak exhibits a relatively narrow peak width, which illustrates that the anatase TiO$_2$ nanorods have a preferred growth orientation along the [001], the c-axis direction of the anatase crystal. It can also be proved that the diameter and length of the TiO$_2$ are 2.3–3.5 nm and 25–30 nm, respectively.

As mentioned before, the final products were obtained after two-step assembly. The primary self-assembly of the TiO$_2$ nanorods solution occurred spontaneously during the synthesis process [30], and two structures (ribbon and honeycomb) were obtained. Ribbon structure (Figure 1(a), inset) was assembled through side-by-side attraction of NRs to 0.1–1 μm, and honeycomb (Figure 2(a)) is a high packing density structure consisting of perpendicularly aligned rods in hexagonal superlattice. After the sample was kept at 21–25°C
Figure 1: (a) The XRD pattern of anatase TiO$_2$ nanorods and corresponding TEM image (inset). (b) The products after primary and secondary assembly, and resulting gel-like products.

Figure 2: (a, b) TEM images of 0.188 M products after synthesis and being placed one week. (c) POM image of 0.188 M.

3.2. Secondary Self-Assembly and LC Phases on Increasing Concentration. The following discussion will focus on the structures and properties of LC phases after secondary assembly at different concentrations. The possible reason for those differences is that the relative amount of two secondary building units (ribbon and honeycomb). As shown in Figure 2(a), at low NRs concentration such as 0.188 M, both structures’ concentrations are too low to trigger the secondary assembly, so the secondary-assembly process does not happen. Moreover, the amount of ribbon structure is still much more than honeycomb (Figure 2(b)), and they will enrich and form nematic phase after being placed for a week. The corresponding POM image (Figure 2(c)) also shows a typical nematic texture.

It is worth mentioning that the aggregates in 0.231 M show a Maltese-cross extinction pattern under polarizing microscope (Figure 3(c)), which indicates that they form spherulites structure. It also can be found that the first and third quadrants are yellow, while the second and fourth quadrants are blue in spherulites texture, inferring that we obtain negative or tangential spherulites [32, 33] whose...
crystallites are suggested to lie along the circumference. Figure 3(a) shows a representative TEM image of the products after primary assembly, as the secondary building unit, it is obvious that the amount of honeycomb is almost equal to that of ribbon structure and its diameter is more than two hundreds nanometer. The assembly is completely different from the aggregation of NRs in 0.188 M; this time the secondary-assembly occurred. The honeycomb becomes the key factor, because it was used as nuclei to induce nucleation when its content and diameter reach the threshold. After that, the ribbon structure assembles along the circumference of nuclei and grows into spherulites with dozens or even hundreds of microns in diameter under certain temperature. Figure 3(b) shows a TEM image of not so perfect TiO₂ spherulite whose structure is consistent with the theory and experiments. Figure 3(d) shows the schematic illustration of the formation of spherulites. This is the first report about spherulites formed from anatase TiO₂ NRs, although other systems are known to have formed spherulites, including organic material [34, 35] and inorganic CdSe nanorods [36].

The growth processes of TiO₂ nanorods spherulite at 0.231 M were determined by examining different time at an appropriate temperature. According to a series of experiments, the most suitable temperature for the formation of spherulites is 23°C (other range of temperatures will form subperfect spherulites). Figures 4(a)–4(e) show the POM images of the growth of NRs spherulites at 23°C. Apparently the growth process starts from the center, while many of the honeycomb structures assemble gradually, which forms the spherulites core of hundreds of nanometer, and then the nanoribbons continue to assemble around the core along the circumferential direction to form 20–30 μm spherulites. As shown in Figure 4(f), a well-defined Maltese cross was observed and the TiO₂ nanorods spherulites formed have the opposite negative-type crystal structure.

A series of other experiments about temperature range in which spherulite can exist have been carried out for the applications. The initial temperature was 23°C, and both the heating and cooling rates were set at 1°C/min and maintained for 5 min at every testing temperature (to prevent errors because of the slow growth of spherulites and there is no other changes happen given more time). The spherulites disappeared when the temperature reached 30°C and re-appeared when they were cooled down to 26–28°C (Figure 5), but the spherulites have a high chemical stability at a certain temperature. According to features of spherulites, we can develop a kind of temperature microsensor, which works under 23–40°C. Of course the operating temperature range can be regulated by the NRs concentrations, solvent types, or other factors. Based on the growth process, spherulites also may be used in drug detoxification [37].

**Figure 3:** (a) TEM image of structures when primary assembly of TiO₂ NRs was complete; inset is the honeycomb structure. (b) TEM image of spherulite after secondary assembly at 0.231 M. (c) POM image of spherulites of anatase TiO₂. (d) The schematic illustration for the formation of spherulites.
Figure 4: (a–e) Images of the growth process of spherulites from nucleation under crossed polarizing microscope. (f) Representative POM image of spherulites and a Maltese-cross extinction pattern. All scale bars are 5 μm.

Figure 5: The influence of temperature variation to the spherulites under crossed polarizing microscope.

When the concentration increases to 0.262 M, the NRs solution will assemble into lamellar (or smectic) phase which is a more ordered, lower energy of lyotropic liquid crystal phase. Figure 6(a) shows a typical TEM image after the primary self-assembly, which demonstrates that there is no sign of honeycomb structure. As the amount of TiO$_2$ ribbon structure is very high, it is arranged in a perfect layer structure by end-by-end contact type, and the distance between the layer is approximately twice of the length of surface ligands of the NRs (Figure 6(b)). This is a novel method to obtain an ordered array of nanorods via regulation of lyotropic liquid crystal. Figure 6(c) is a corresponding POM image at 0.262 M, which also reveals a typical texture of smectic phase. Figure 6(d) shows schematic illustration for the formation of lamellar phase.

3.3. Assembly Mechanism and the Driving Force. Based on the TEM and POM experimental observations, the process and self-assembly mechanism of the NRs is hypothesized (Figures 5(d) and 6(d)). The final products were obtained by a two-step assembly, that is, primary and secondary self-assembly. In the synthesis process, the primary assembly occurred and formed two structures named as ribbon and honeycomb, respectively. These two structures appeared in various concentrations, including the lowest one, which is lower than what is needed to satisfy the Onsager theory. Those facts reveal that the main driving force for primary-assembly should be the anisotropic attractive force and thus allow the NRs to assemble at any concentrations. Only ribbon structure is found at high NRs concentration, and the most possible reason is that higher concentrations mean wider size distribution, so the NRs intend to form ribbon structure, which is less ordered than the honeycomb. We explain the assembly by Van der Waals attraction between surface ligands of the TiO$_2$ NRs. Because the affinity of
OA for the lateral surfaces of NRs is higher than that for the end facet, the Van der Waals attraction among laterals is more powerful, which makes rods form side-by-side structure. Meanwhile, the repulsion force will increase with the decrease of distance between the NRs to keep system in balance. In the case of having closer parameters and uniform aspect ratio, the NRs form honeycomb structure to decrease the interfacial energy effectively, because honeycomb is more compact compared with ribbon. The multilayer honeycomb can be obtained as the pairing of the dipole moment along the longitudinal by the antiparallel [38] of NRs, while the two ends whose polarity is opposite cannot be infinitely close to each other because the free H+ will be absorbed to the negative end, thus the system is also in equilibrium. As long as the major forces for this process are the intermolecular/colloidal forces, further assembly is possible as secondary and tertiary self-assembly [31]. When the product was placed at 21–25°C for about one week, secondary self-assembly occurred. The main driving force for this assembly is gravity. Several structures are formed because the relative amount of honeycomb and ribbon structures is different at various concentrations. The enrichment of ribbon structure causes the formation of nematic phase in 0.188 M. Each structure shows unique LC behavior after secondary-assembly, namely, nematic, spherulite, and lamellar on increasing concentration (see Figures 2, 3, and 6).

4. Conclusion

In conclusion, various TiO₂ NRs structures were obtained by a two-step self-assembly. In primary assembly occurring in the synthesis process, two structures appeared, that is, ribbon and honeycomb, respectively. However, the relative amount of these two structures is different at various concentrations: only a small amount of both two structures were found in 0.188 M, approximately equal amount of honeycomb and ribbon were found in 0.231 M, and only ribbon appeared in 0.262 M concentration. Since these two structures obtained by primary assembly were used as the secondary building unit and the amount of them are different, we obtained three secondary assembly structures after secondary assembly with the increase of concentration. Each structure has unique LC behavior. We investigate the growth process of spherulites and conceive the possible applications such as spherulites in temperature microsensor and drug detoxification. The mechanism of the two-step self-assembly was clarified. Further experiments will be carried out to characterize the behavior and properties of TiO₂ nanorods LC for its possible applications.

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