The Geometry Variation of As-Grown Carbon Coils with Ni Layer Thickness and Hydrogen Plasma Pretreatment

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Carbon coils could be synthesized using \( \text{C}_2\text{H}_2/\text{H}_2 \) as source gases and \( \text{SF}_6 \) as an incorporated additive gas under thermal chemical vapor deposition system. Ni layer on \( \text{SiO}_2 \) substrate was used as a catalyst for the formation of carbon coils. Ni powder was evaporated to form Ni layer on the substrate. The characteristics (formation densities, morphologies, and geometries) of as-grown carbon coils on the substrate were investigated as a function of the evaporation time for Ni catalyst layer formation. By hydrogen plasma pretreatment prior to carbon coils synthesis reaction, the dominant formation of the nanosized wave-like geometry of carbon coils could be achieved. The characteristics of as-grown carbon coils with or without hydrogen plasma pretreatment process were investigated. The cause for the control of the carbon coils geometries from the microsized type to the nanosized wave-like one by \( \text{H}_2 \) plasma pretreatment was discussed in association with the stress of Ni catalyst layer on the substrate.

1. Introduction

Since the first report by Davis et al. carbon coils have been noticed because of their unique shape [1]. Spring-like helix-shaped geometry may induce an electrical current and consequently generate a magnetic field. So, the predicted electrical, magnetic, and mechanical properties of carbon coils are attractive to be used in electromagnetic absorbers in the GHz to THz regions, high sensitive nano/microsized detectors, resonators, mechanical springs, essential building blocks for the assembly of nanodevices, and so forth [2–4]. In addition, carbon coils are also predicted to have a potential application as an effective fillers for nanocomposites because their spring-like geometry would make them better anchored in embedding matrix and consequently favor a better load transfer to the matrix [5].

The production of coils-type geometry was accidental and lacks reproducibility. Furthermore, the electrical properties of helically coiled geometry may have metallic, semiconducting, or semimetallic characteristics depending on their geometry including diameter [6]. Therefore, the controlled geometry (diameter, pitch, length, and turning direction) of carbon coils would be the urgent problem to be preferentially addressed in order to achieve the controlled characteristics of carbon coils.

For the synthesis method of carbon coils, catalytic chemical vapor deposition (CCVD) technique has been noticed for an effective choice to produce carbon coils because it has the flexibility in controlling growth conditions to achieve the preferred geometry. Basically, carbon coils growth involves a decomposition—diffusion—precipitation mechanism that is initiated on the surface of the catalyst particles [7, 8]. In this respect, the characteristics of the used metal catalyst would be the important factor to control the geometry of carbon coils besides CVD system parameters. Namely, the catalyst chemical composition, size, thickness, shape, and crystallographic orientation would play a critical role in determining the geometry of carbon coils. Up to the present, various kinds of catalysts have been widely investigated [9–23]. Among the metal catalysts currently used in the growth of carbon coils, Ni was regarded as an effective material for the growth of carbon coils [10, 12]. For Ni catalyst study, Ni (100) was known to give the highest coil yield followed by Ni (111) and then Ni (110) among Ni single crystal planes [13]. So, the driving force for the formation of carbon coils would be the anisotropic deposition rate of a carbon according to Ni single
Table 1: Ni layer thickness for the samples according to the evaporation time of Ni powders and without or with H\textsubscript{2} plasma pretreatment.

<table>
<thead>
<tr>
<th>Samples</th>
<th>Evaporation time of Ni powders</th>
<th>Resulting Ni layer thickness</th>
<th>5 minutes H\textsubscript{2} plasma pretreatment</th>
</tr>
</thead>
<tbody>
<tr>
<td>A</td>
<td>10 sec</td>
<td>(~50) nm</td>
<td>None</td>
</tr>
<tr>
<td>B</td>
<td>20 sec</td>
<td>(~80) nm</td>
<td>None</td>
</tr>
<tr>
<td>C</td>
<td>40 sec</td>
<td>(~150) nm</td>
<td>None</td>
</tr>
<tr>
<td>D</td>
<td>1 minute</td>
<td>(~250) nm</td>
<td>None</td>
</tr>
<tr>
<td>E</td>
<td>3 minutes</td>
<td>(~350) nm</td>
<td>None</td>
</tr>
<tr>
<td>F</td>
<td>1 minute</td>
<td>(~250) nm</td>
<td>Yes</td>
</tr>
</tbody>
</table>

Crystal planes. The catalyst particle size was also known to be the critical factor to determine the carbon coils geometry. Tang et al. [14] observed that helical carbon structure grew from Fe\textsubscript{2}O\textsubscript{3} catalyst particles with diameters <150 nm. At the larger size of the catalyst particle, the straight carbon structure appeared. Zhang et al. [15] reported that the coiled carbon fibers were only obtained when the size of Cu catalyst particles was between 30 and 60 nm in diameter. Hokushin et al. [16] showed that carbon nanocoils (CNCs) were only observed for Fe/In/Sn catalyst particle sizes ranging between 50 and 150 nm. Despite these abundant efforts, the study of the geometry control of carbon coils according to the thickness of Ni catalyst layer is few up to the present.

In this work, we investigated the geometries of the as-grown carbon coils with Ni layer thickness. In addition, we introduce the method to control the geometry of carbon coils through altering the characteristics of nickel catalyst layer by H\textsubscript{2} plasma pretreatment prior to carbon coils synthesis reaction. With or without H\textsubscript{2} plasma pretreatment process, the synthetesis of carbon coils using thermal chemical vapor deposition were performed and the geometries of as-grown carbon coils were compared and discussed.

2. Experimental Details

For silicon oxide substrate, about 300 nm SiO\textsubscript{2} layer on Si substrate was employed. They were prepared by the thermal oxidation of 2.0 \(\times\) 2.0 cm\textsuperscript{2} p-type Si (100) substrates. For Ni catalyst layer deposition on the substrates, about 0.01 g Ni powder (99.7%) was evaporated to form Ni catalyst layer on the substrate using thermal evaporator. The Ni layer thickness was manipulated by controlling the evaporation time of Ni powders. It was measured by the cross-sectional image of field emission scanning electron microscopy (FESEM) for the as-grown film. The evaporation time of Ni powders and the resulting Ni layer thickness for each sample were shown in Table 1.

For H\textsubscript{2} plasma pretreatment, Ni-coated substrate was placed in radiofrequency (13.56 MHz, 25 W) plasma enhanced chemical vapor deposition (PECVD) system prior to carbon coils deposition. H\textsubscript{2} gas was introduced into PECVD chamber. The flow rate for H\textsubscript{2} was fixed at 17 standard cm\textsuperscript{3} per minute (sccm). The substrate was pretreated for 5 minutes using H\textsubscript{2} plasma at 0.5 Torr total pressure.

For carbon coils deposition, thermal chemical vapor deposition (TCVD) system was employed. C\textsubscript{2}H\textsubscript{2} and H\textsubscript{2} were used as source gases. SF\textsubscript{6}, as an incorporated additive gas, was injected into the reactor during the initial reaction stage. The flow rate for C\textsubscript{2}H\textsubscript{2}, H\textsubscript{2}, and SF\textsubscript{6} was fixed at 15, 35, and 35 sccm, respectively. The reaction conditions were shown in Table 2. Detailed morphologies of carbon coils-deposited substrates were investigated using FESEM. The qualities of carbon coils according to the samples were investigated in the range of 800~2000 cm\textsuperscript{-1} by a micro-Raman spectrometer (Renishaw 2000) with about 50 µm spot size of Ar-ion laser.

3. Results and Discussion

Five samples (samples A–E) having the different Ni catalyst layer thickness were prepared. At samples A and B, namely less than 100 nm Ni catalyst layer thickness case, the formation of as-grown carbon coils could be observed here and there on the surface of the sample.

At \(~150\) nm Ni layer thickness, sample C, both the microsized carbon coils and the nanosized carbon coils were well developed on the most part of the sample surface as shown in Figures 1(a) and 1(c). Most of the nanosized carbon coils were gathered along the side of the microsized carbon coils as shown in Figures 1(b) and 1(d). The deposition aspect for the nanosized carbon coils and the microsized carbon coils seems to be almost similar irrespective of the position on the surface of sample C. Indeed, in case of more than \(~150\) nm Ni layer thickness, the deposition aspect for the nanosized carbon coils and the microsized carbon coils hardly depends on the position on the surface of the sample.

At \(~250\) nm Ni layer thickness, sample D, the microsized carbon coils were mostly observed on the surface of the sample as shown in Figure 2. The length of the microsized carbon coils is more than hundred micrometers as shown in Figure 2(a). The diameters of the microsized carbon coils are in the range from several micrometers to ten micrometers as shown in Figure 2(b). Compared with those of sample C, a relatively few amount of the nanosized carbon coils seems to exist along the side of the microsized carbon coils as shown in Figure 2(c). In general, many types of carbon coils-related geometries could be observed on the sample surfaces [24]. Particularly, the nanosized carbon coils could be classified into two geometrical categories, namely, wave-like nanosized carbon coil (w-NC) and knot-type nanosized coil (k-NC) (see Figure 2). In this case, most of the nanosized carbon coils have w-NC type geometry. Representative FESEM images for the microsized carbon coils are shown in Figure 3. The magnified FESEM image clearly indicates the well-developed double helix geometry for these microsized carbon coils. Ni catalyst grain seems to be incorporated as the form of Ni\textsubscript{3}C in the bright spot in Figure 3(b) [25].

At \(~350\) nm Ni layer thickness, sample E, the density of the nanosized carbon coils is higher than that of sample D (compare Figures 4(a) with 2(a)). The high-magnified images (Figures 4(b) and 4(c)) clearly reveal that the formation of
Table 2: Experimental conditions for the deposition of carbon coils.

<table>
<thead>
<tr>
<th>Samples</th>
<th>C$_2$H$_2$ flow rate (sccm)</th>
<th>H$_2$ flow rate (sccm)</th>
<th>SF$_6$ flow rate (sccm)</th>
<th>Total pressure (Torr)</th>
<th>Total deposition time (min)</th>
<th>C$_2$H$_2$ injection time (min)</th>
<th>H$_2$ injection time (min)</th>
<th>SF$_6$ injection time (min)</th>
<th>Substrate temperature (°C)</th>
</tr>
</thead>
<tbody>
<tr>
<td>A~F</td>
<td>15</td>
<td>35</td>
<td>35</td>
<td>100</td>
<td>90</td>
<td>90</td>
<td>90</td>
<td>5</td>
<td>750</td>
</tr>
</tbody>
</table>

Figure 1: FESEM images for sample C at the edge area under the magnification of (a) 1,000 and (b) 10,000 and at the center area of the substrate under the magnification of (c) 1,000 and (d) 10,000.

the microsized carbon coil was initiated by two k-NC-type nanosized carbon coils. Except for k-NCs in the microsized carbon coils, most of the nanosized carbon coils existed as a form of w-NC geometry. Representative FESEM images for the k-NC-type nanosized carbon coils are shown in Figure 4(d). This regularly grown nanosized carbon coil has the single helix geometry. It has a coil pitch of ~300 nm with little coil gap and a coil diameter of ~400 nm. The carbon nanofibers that built up this coil have the circular type shape.

Using the square graph papers and FESEM images, we measured the average ratio of the occupied areas by the different-type carbon geometries, namely, the linear and the microsized geometries from several 1,000 magnified FESEM images. Under the assumption of the monolayer-grown carbon materials on the samples, the ratio of the occupied areas of carbon geometries were measured in equal-sized FESEM images. Figure 5(a) shows the variation of the ratio of the occupied areas in FESEM images for the linear and the microsized geometries as a function of the sample. The ratio of the occupied areas by w-NC or k-NC was also measured using several 5,000 magnified FESEM images. Figure 5(b) shows the variation of the ratio of the occupied area for w-NC or k-NC as a function of the sample. For samples A and B, this ratio could not be measured due to the unevenness according to the position on the surface of the sample. The results of Figures 5(a) and 5(b) reveal that the dominant geometry of the carbon coils would be the nanosized carbon coils with increasing Ni catalyst layer thickness from 250 to 350 nm. In addition, it is believed that the geometry control for carbon coils could not be completely achieved merely by the manipulation of Ni catalyst layer thickness.

To fully control the geometry of carbon coils, altering the characteristics of nickel catalyst layer was performed by H$_2$ plasma pretreatment prior to carbon coils synthesis reaction. Figure 6 shows the surface morphologies of as-grown carbon coils on the surface of the 5 min H$_2$ plasma pretreated sample
Figure 2: FESEM images for sample D under the magnification of (a) 1,000, (b) 5,000, and (c) 10,000, and (d) the schematic images for the typical geometries of the nanosized carbon coils.

Figure 3: Representative FESEM images for the microsized carbon coils under the magnification of (a) 10,000 and (b) 50,000.

The nanosized carbon coils entirely existed on the whole surface of the sample as shown in Figure 6(a). Comparing the results of sample D (see Figure 2) with this result, it is understood that the control of carbon coils geometry from the microsized type to the nanosized one is possible merely by H₂ plasma pretreatment on Ni layer of the substrate. Most of the nanosized carbon coils have a form of w-NC-type geometry and they existed on the entire surface of the substrate. The length seems to be more than several micrometers. Their coil diameter was in the range from several nanometers to two hundred nanometers (see Figure 6(c)).

Due to the stress between the metal layer and the substrate, H₂ plasma pretreated-Ni layer would be more easily peeled off and broken into very tiny nanosized Ni pieces and eventually scattered in surrounding area. Basically,
Figure 4: FESEM images for sample E under the magnification of (a) 1,000, (b) 5,000, (c) 10,000, and (d) 50,000.

Figure 5: The variation of the ratio of the occupied areas in FESEM images for linear type coil, microsized coil, wave-like nanosized coil (w-NC), and knot-type nanosized coil (k-NC) as a function of the sample. The ratio of the occupied area by the linear-type and microsized coils was measured in the 1,000 magnified FESEM images and those by w-NC and k-NC in 5,000 high magnified FESEM images.
the mechanism of carbon coils growth depended on the metal size and shape [15]. The nanosized carbon coils were formed from the nanosized Ni pieces and they would deposit on the whole surface of the substrate. This is the reason why H$_2$ plasma pretreatment gives rise to the dominant existence of the nanosized carbon coils on the entire surface of the substrate. Figure 7 shows FESEM images indicating the different situation of the peeled-Ni layers from the substrate without H$_2$ plasma pretreatment (sample D, Figure 7(a)) and from the substrate with H$_2$ plasma pretreatment (sample F, Figure 7(b)) after cooling down the substrate from 750°C under vacuum. As shown in these images, the substrate with H$_2$ plasma pretreatment gives rise to the more flaky state for Ni layer, which may form the nanosized geometry for carbon coils.

Meanwhile, the qualities of carbon coils for the samples without or with H$_2$ plasma pretreatment were also investigated in the range of 800−2000 cm$^{-1}$ by a micro-Raman spectrometer as shown in Figure 8. The D and G peaks in all the different samples were observed around 1350 cm$^{-1}$ and 1600 cm$^{-1}$, respectively. Curve fitted values of the area intensity ratios of I(D)/I(G) for those samples are around 2.0, which indicates the existence of the nanocrystalline aromatic $\pi$-bonded clusters in the carbon coils [26]. I(D)/I(G) value for sample F is slightly higher than that for sample D, indicating a larger presence of the disordered carbon phase in sample F. This tendency indicates the well-developed polycrystalline of the carbon coils for sample D, namely, for the microsized carbon coils on the substrate [27–30].
4. Conclusions

In case of Ni catalyst layer thickness less than 100 nm, the carbon coils were formed sporadically according to the position on the surface of the sample. At more than ~150 nm Ni catalyst layer thickness, the deposition aspect for the nanosized carbon coils and the microsized carbon coils hardly depends on the position on the surface of the sample. At ~250 nm Ni layer thickness, the microsized carbon coils were mostly observed on the surface of the sample. However, the nanosized carbon coils became dominant by increasing Ni catalyst layer thickness from 250 to 350 nm.

Although the geometry of carbon coils was varied according to the different Ni layer thickness, the geometry control for carbon coils could not be completely achieved merely by the manipulation of Ni catalyst layer thickness. However, H$_2$ plasma pretreatment prior to carbon coils synthesis reaction gives rise to the complete geometry change from the microsized type to the nanosized one on the entire surface of the substrate. The main cause for the geometry change of the carbon coils by H$_2$ plasma pretreatment seems to be due to the more easily peeled-off Ni catalyst layer and then scattered in surroundings as tiny nanosized Ni pieces.

References


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