Research Article

Generation of Porous Alumina Layers in a Polydimethylsiloxane/Hydrogen Peroxide Medium on Aluminum Substrate in Corona Discharges

A. Groza,1 A. Surmeian,1 C. Diplasu,1 C. Luculescu,1 C. Negrila,2 and M. Ganciu1

1 National Institute of Lasers, Plasma and Radiation Physics, Magurele, 077125 Bucharest, Romania
2 National Institute for Materials Physics, Magurele, 077125 Bucharest, Romania

Correspondence should be addressed to A. Groza; andreea@infim.ro

Received 2 July 2014; Accepted 1 August 2014; Published 3 September 2014

1. Introduction

The porous alumina (Al2O3) layers are aluminum oxide films consisting of cylindrical pores arranged parallel to each other. By conventional methods, these layers are prepared by an anodization process, namely, the electrochemical oxidation of aluminum films in acid reactive medium under direct current conditions. Although the chemical and physical aspects of this process are not completely understood, it is generally accepted that the porosity of the alumina layer is produced in two steps: (a) growth of an aluminum oxide at the alumina/Al interface due to the counter migration of Al3+, OH−, and O2− ions and (b) dissolution of aluminum oxide at alumina film/solution interface [1].

Recent research studies in the field of anodization methods of aluminum were directed toward alternative methods for the generation of alumina layers, namely, corona discharge treatments [2]. Running of these kinds of discharges in air at atmospheric pressure has the unique advantage of simultaneous production of charged (negative or positive ions) and neutral (e.g., O3) species of oxygen known for their reactive character in the interaction with materials.

Over the time, alumina layers have been intensely studied due to their widespread application in corrosion protection for aerospace, automotive, architectural, or packaging [2]. Their nontoxic and biocompatible properties make them also attractive for biotechnological applications [3]. Alumina nanoparticles can be trapped and studied in electromagnetic fields [4, 5] and are of great interest for their antimicrobial activity and their possible clinical applications [6].

As substrates, porous anodic Al oxide layers can improve the adherence of different types of thin layers [7–9]. On the other hand, a PDMS layer deposited on an anodized porous Al oxide layer [10] could act as an effective barrier preventing the contact of water based corrosive media with the underlying metallic surfaces [11, 12].

In our previous papers, we showed that, during the polymerization of a polydimethylsiloxane (PDMS) liquid film lying on the surface of an aluminum substrate in corona discharge, an Al oxide layer is generated at polymer/substrate
interface [10]. The OH and Si-OH groups present at polymer/substrate interface determine the formation of Si-O-Al bonds announcing the possibility of counter migration of Si and Al atoms. The aluminum oxide layer has porous structure, the pores having a nonuniform distribution on the alumina layer [13]. The dependence of the pores structure on the corona discharge current values [14, 15] and the formation of the Si-O-Al bonds evidenced the existence of an anodization process at polymer/Al substrate interface.

Responsible for oxidation of the Al surface in negative corona discharge are the negative ions of oxygen (O$_2^-$, O$_3^-$, O$_4^-$, CO$_2^-$, and CO$_3^-$), the water, and consequently the OH radicals present either in liquid or in solid film further formed on the substrate due to the PDMS polymerization process [10]. Even these species are able to generate an anodization process of the Al in a similar way with the classical one; the nonuniform distribution of the pores structure in the alumina layer suggests that the experimental conditions need further improvements.

In this paper, we report for the first time a uniform distribution of the pores in the alumina layer generated in corona discharge operating conditions by mixing the hydroxyl terminated PDMS liquid precursor with hydrogen peroxide (H$_2$O$_2$). In this way, the water molecules and the density of OH radicals at Al substrate interface are enhanced, knowing that hydrogen peroxide can be involved in the following reactions:

$$\text{H}_2\text{O}_2 \rightarrow \text{H}_2\text{O} + 1/2\text{O}_2 \, [16],$$

$$\text{OH} + \text{H}_2\text{O}_2 \rightarrow \text{H}_2\text{O} + \text{HO}_2 \, [17],$$

$$\text{HO}_2 + \text{H}_2\text{O}_2 \rightarrow \text{O}_2 + \text{H}_2\text{O} + \text{OH} \, [17],$$

$$2\text{O}_3 + \text{H}_2\text{O} \rightarrow \text{OH} + \text{O}_2 + \text{HO}_2 \, [17].$$

The morphology, the elemental composition, and the processes which determine the porous alumina layer formation in negative corona discharge in the presence of a PDMS/H$_2$O$_2$ chemical medium are investigated by scanning electron microscopy (SEM), electron dispersion spectroscopy (EDS), and X-ray photoelectron spectroscopy (XPS).

2. Materials and Methods

2.1. Description of the Experimental Conditions Used for the Generation of the Porous Alumina Layers. The experimental setup used for the operating of the negative corona discharge in a Trichel pulse regime is presented in Figure 1. It consists in a point to plane corona discharge electrode configuration, the distance between electrodes being 10 mm. The cathode is a 20 mm tungsten wire placed perpendicular to the centre of the anode (plane electrode) in a glass cage in air at atmospheric pressure. As anode, we used an Al disk electrode with a diameter of 10 mm. A high voltage of 11 kV was applied between electrodes, through a resistor of $5\, \Omega$, the mean value of the discharge current being 40 $\mu$A.

The negative ions of oxygen O$_2^-$, O$_3^-$, O$_4^-$, CO$_2^-$, and CO$_3^-$ produced in the Trichel regime of the negative corona discharge in the vicinity of the cathode [18] travel to the anode in regular packages. The transport of these particles through the interelectrode gap have a pulsating nature due to the periodical character of positive/negative ions production respectively destruction processes occurring at the cathode. Thus, the corona discharge current, Figure 2, is formed by regular pulses with a period of about 1.5 $\mu$s.
Figure 3: SEM images of (a) porous anodic Al oxide layer and (b) transversal section of the alumina layer.

Due to the uniformity of the current density distribution on the anode surface and to the $\text{O}_2^{-}$, $\text{O}_3^{-}$, $\text{O}_4^{-}$, $\text{CO}_2^{-}$, and $\text{CO}_3^{-}$ negative ions production, the Trichel pulse regime of the corona discharge is proper to be used for thin film generation [10]. It was shown [19] that, in a Trichel pulse regime, the distribution of the current density on the plane electrode is uniform on a circular surface with a radius given by the relation $r = \sqrt{3d}$, where $d$ is the interelectrode gap. In our experimental conditions $r = 17.2$ mm.

Previously [10], we showed that a drop of 0.1 $\mu$L hydroxyl terminated polydimethylsiloxane (PDMS) liquid precursor stretched on an Al substrate in negative corona discharge is transformed into a solid polymer layer after 2 hours. As a result of the polymerization process, water molecules are generated in the PDMS layer. They can be decomposed under the influence of corona electric fields and associated charges injection generating OH radicals in the polymer. The Si-OH groups present also in the PDMS layer have proved their utility in the generation of Si-O-Al bonds at polymer/Al substrate interface. In the same time, at polymer/Al substrate interface an anodization process occurs. The negative ions of oxygen produced in corona discharge, the water molecules and the OH groups resulting as byproducts of the polymerization process [10] are mainly responsible for the alumina layer formation.

In experimental conditions described above, we placed a 10 $\mu$L of liquid precursor on an optical polished Al substrate (anode). This liquid precursor contains a proportion of 33% PDMS with hydroxyl end groups and 67% hydrogen peroxide ($\text{H}_2\text{O}_2$). The addition of $\text{H}_2\text{O}_2$ to the PDMS liquid precursor does not change the electrical regime of the corona discharge and its optical characteristics. Under negative corona charge injection, the stretch of the PDMS liquid assures the uniform covering of the substrate with the $\text{H}_2\text{O}_2$. The further decomposition of $\text{H}_2\text{O}_2$ in negative corona discharge produced in air gives an enhanced infusion of water molecules and OH radicals at the surface of the Al substrate.

The solid layer formed after 5 hours of irradiation in negative corona discharge of the PDMS/$\text{H}_2\text{O}_2$ has been chemically removed. Thus, the anodization processes of the Al substrate were revealed through the morphology and elemental composition analysis of porous anodic Al oxide layer.

2.2. SEM Measurements. The morphology of the alumina layers has been investigated using a FEI Inspect S scanning electron microscope in both high- and low-vacuum modes.

2.3. EDS Measurements. Elemental composition analysis was performed by energy dispersive X-ray spectroscopy using an EDAX Inc. SiLi detector attached inside the scanning electron microscope. The measurements were performed at a voltage of 10 kV.

2.4. XPS Measurements. The X-ray photoelectron spectroscopy (XPS) measurements were performed using a VG ESCA 3 MK II XPS installation ($E\alpha = 1486.7$ eV). The vacuum analysis chamber pressure was $p = 3 \times 10^8$ torr. The XPS recorded spectrum involved an energy window $w = 20$ eV with the resolution $R = 50$ eV with 256 recording channels. The XPS spectra were processed using Spectral Data Processor v 2.3 (SDP) software. The measured binding energy (BE) scale was referenced to a C 1s peak at the BE value of 284.8 eV [20]. The accuracy for BE assignments is $\pm 0.2$ eV.

3. Results and Discussions

3.1. SEM Investigations. In our previous papers [10, 14, 15] we showed that the pores formed into the alumina layer during the polymerization process of the PDMS liquid precursor on Al substrate in negative corona discharge do not present uniform and regular structures.

In the present paper, by matching the current intensity (40 $\mu$A) with the liquid precursor's composition (33% PDMS with hydroxyl end groups and 67% $\text{H}_2\text{O}_2$) and with the operating time of the negative corona discharge (5 hours), we succeed to generate a uniform distribution of the pores.
in the anodic Al oxide layer, Figure 3(a). The columnar structure of these pores is revealed in Figure 3(b). Thus, under the experimental conditions presented above, the uniform injection of the negative oxygen ions into a liquid/layer rich in water molecules and OH groups is proper for the generation in a controllable manner of an advanced anodization process (columnar structure formation [20]) of the Al substrate.

3.2. EDS Investigations. The establishment of the elements contained in the porous alumina layer is important in order to determine the mechanisms responsible for the anodization of the Al substrate.

In Figure 4, the EDS spectrum of the porous anodic Al oxide layer displayed in Figure 3 is presented. It can be observed that, besides the peaks of Al and O, a peak characteristic to Si atoms appears in this spectrum. As in the SEM image from Figure 3(a), there are no residual pieces of polymer on the surface of porous Al oxide layer; it means that during the anodization process of the substrate the Si atoms are incorporated into the anodic Al oxide layer.

3.3. XPS Investigations. The chemical processes induced by the PDMS/H₂O₂ into the Al substrate were investigated by X-ray photoelectron spectroscopy (XPS) measurements.

The XPS spectrum of the porous anodic Al oxide layer is displayed in Figure 5.

In order to better reveal the processes that take place during the anodization of Al substrate in corona discharge in the presence of PDMS/H₂O₂, we applied a peak fitting analysis for the experimental bands specific to the Al 2p, Si 2p, and O 1s XPS peaks. For each peak, we used as fitted curve a Gaussian type profile curve.

The experimental XPS bands characteristic to Al 2p, Si 2p, and O 1s together with the fitted curves are presented in Figure 6. The intensity of the deconvoluted XPS band from 73.6 eV belonging to Al₂O₃ [21] is almost the same with that one located at 74.9 eV and assigned to aluminosilicates (Al₅SiO₅), Figure 6(a) [22].

The deconvolution of the Si 2p XPS peak reveals the nature of the Si incorporated into the anodic Al oxide porous layer. The presence of the 99.8 eV (Si), 101.3 eV (SiO) [23], 102.8 eV (Al₅SiO₅) [24], and 104.2 eV (SiO₂) [25] XPS bands in the Si 2p spectrum from Figure 6(b) indicates how Si atoms are linked in the porous anodic Al oxide layer.

Two fitted peaks located at 530.7 eV and 532.1 eV were observed in the high resolution XPS spectrum of O 1s shown in Figure 6(c). This certified the formation of Al-O-Si bonds [26], respectively, of the Al₂O₃ [27].

The XPS and EDS measurements presented above are in good agreement with one another showing that Si atoms (coming from PDMS/H₂O₂ medium) can be incorporated into the porous alumina layer. This result can be understood if we consider that in a classical electrochemical anodization process there are two stages in the pores formation due to the competing oxidation and dissolution processes: (a) the growth of aluminum oxide due to the counter migration of Al³⁺, OH⁻, and O₂⁻ ions and (b) the dissolution of aluminum oxide at the interface between the alumina film and solution [1, 20].

4. Conclusions

In this paper, we report for the first time the formation of a porous anodic Al oxide layer in negative corona discharge in the presence of the PDMS/H₂O₂ lying on the surface of the Al substrate. This process is favored by the enhanced infusion of H₂O and OH radicals, respectively, by the negative ions of oxygen and neutral species (O₃) produced in negative corona discharge in air at atmospheric pressure. The OH radicals are produced at PDMS/H₂O₂/substrate interface as a result of the H₂O₂ and H₂O decomposition.
Figure 6: High resolution XPS spectra for (a) Al 2p, (b) Si 2p, and (c) O 1s peaks. The experimental curve is plotted in black and the theoretical fitted curve is in red dot line.

**Conflict of Interests**

The authors declare that there is no conflict of interests regarding the publication of this paper.

**Acknowledgment**

This work was supported by the PN-II-ID-PCE-2011-3-0958 and PT-PCCA-2011-3.1-1136 UEFISCDI projects.

**References**


