Research Article
Characterization of Class F Fly Ash Using STXM: Identifying Intraparticle Heterogeneity at Nanometer Scale

J. Ha,1 S. Chae,2 K. W. Chou,3,4 T. Tyliszczak,3 and P. J. M. Monteiro2

1School of Environmental and Sustainability Sciences, Kean University, Union, NJ 07083, USA
2Department of Civil and Environmental Engineering, University of California-Berkeley, Berkeley, CA 94720, USA
3Advanced Light Source, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, USA
4Henkel Ibérica S. A., Edificio Eureka, Campus UAB, Bellaterra, 08193 Barcelona, Spain

Correspondence should be addressed to J. Ha; haj@kean.edu

Received 13 October 2015; Accepted 25 February 2016

Academic Editor: Ping Xiao

Copyright © 2016 J. Ha et al. This is an open access article distributed under the Creative Commons Attribution License, which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly cited.

Chemical and physical characterization of fly ash particles were conducted using scanning transmission X-ray microscopy (STXM). Compositional and spatial investigation and correlation among the main elemental constituents of fly ash (Al, Si, and Fe) were conducted based on microscopic and NEXAFS spectral analysis. Homogeneous oxidation and coordination state of Al and Fe were observed whereas Si shows spatial variation in its chemical state. We also identified that Si and Al are spatially correlated at nanometer scale in which high concentration of Si and Al was concurrently and consistently observed within the 30 nm resolution whereas Fe distribution did not show any specific correlation to Al and Si. Results of this study indicate that fly ash chemical composition has heterogeneous distribution depending on the elements which would determine and result in the differences in the reactivity.

1. Introduction

The manufacturing process for Ordinary Portland Cement (OPC) emits a large amount of green-house gas, producing between 800 and 1000 kg of carbon dioxide per ton of OPC. In fact, the manufacturing of OPC is estimated to be responsible for around 8% of all carbon dioxide emissions worldwide [1]. Because of the chemical composition of fly ash, which contains quartz and amorphous silica, fly ash has emerged as a good substitute for the raw materials in cement production since fly ash can serve as pozzolan in producing cementitious materials [2]. In addition to its compositional advantage, the spherical shape as well as particle size distribution of fly ash further makes it a good construction material because of the low water demand and high particle packing [3, 4]. Hence, in an effort to recycle the industrial waste byproducts and to reduce global CO2 production, numerous studies have been done on fly ash application to construction materials [5–8]. With the annual production of fly ash in the United States alone mounting to approximately 63 million tons per year from coal combustion in power plants, more opportunities are awaiting for using fly ash for different applications including cement production [9, 10]. In particular, recent development and advancement in cement industry has identified that high-volume fly ash (HVFA) substituted materials (a.k.a. geopolymers or green cement) have potential to serve as useful and effective construction materials due to their durability and strengths without releasing carbon dioxide in the air during their manufacturing process [11, 12].

Despite the wide applications and increasing demands of recycling fly ash, relatively a small number of studies have been done on characterization of fly ash due to the inherent chemical and physical complexity of the materials. In the past, compositional heterogeneity has been identified as the spherical glass phase and the crystalline phase are simultaneously present in fly ash. Extensive X-ray diffraction (XRD) studies, compositional analysis such as X-ray fluorescence (XRF) measurements, and thermal analysis using thermogravimetric analyzer (TGA) were carried out to study correlation between the glassy phases and crystalline phases.
Table 1: Percent (%) of elemental composition of Class F fly ash based on XRF analysis.

<table>
<thead>
<tr>
<th>wt. %</th>
<th>Na₂O</th>
<th>MgO</th>
<th>Al₂O₃</th>
<th>SiO₂</th>
<th>P₂O₅</th>
<th>K₂O</th>
<th>CaO</th>
<th>TiO</th>
<th>MnO</th>
<th>Fe₂O₃</th>
</tr>
</thead>
<tbody>
<tr>
<td>Fly ash</td>
<td>0.54</td>
<td>1.21</td>
<td>16.62</td>
<td>63.24</td>
<td>0.359</td>
<td>1.39</td>
<td>3.13</td>
<td>0.86</td>
<td>0.05</td>
<td>6.41</td>
</tr>
</tbody>
</table>

in determining the reactivity of fly ash [13–18]. However, most of the studies to date on fly ash properties lack the sensitivity to probe the intrinsic heterogeneous nature of fly ash because they are done mostly on bulk materials. Conventional analytical techniques have limitations on providing structural and chemical details such as coexistence of different glassy phases in a single particle of fly ash. The spatial heterogeneity of the elemental composition on fly ash particles has been especially neglected due to the lack of proper experimental and analytical techniques.

However, it is important to identify the spatial correlation between the amorphous and crystalline phase as well as the elemental compositional distribution in fly ash particles at nanometer scale in order to identify the governing factor of determining the reactivity of fly ash. For example, when fly ash reacts with alkaline solution in cement production, surface chemistry of the fly ash determines the dissolution pathways which will then ultimately determine the following precipitation reaction pathways and reaction byproducts [19–21]. It has been shown that the key parameter in determining the overall reactivity of fly ash has been the glass contents in the materials [22–25]. Having the amorphous glassy phase buried within the spherical morphological feature of the fly ash can possibly slow down the kinetics of the reactions compared to the ones with the amorphous phase exposed on the surface of the fly ash particles. It is also possible that fly ash with more iron atoms at the surface of particles has higher reactivity compared to the ones with more silicon atoms at the surface. Therefore, identifying the glass phases and the intercalated chemical states of the mixed phases is important in characterizing fly ash since the distribution and concentration of different phases ultimately control the reactivity of the materials and their applicability to construction materials.

Studies have suggested that most of the glassy constituents in fly ash are aluminosilicates but no further details with respect to the concomitant existence of glassy phases and the crystalline phases have been reported to date [26–28]. Current model for fly ash particles is expected to be the holocrystalline matter such as mullite while the glass phase itself is amorphous [29–33]. Despite the efforts and recent advances in experimental approaches by using a combination of vibrational spectroscopy, chemical separation methods, NMR, and XPS, challenges still remain in identifying the structure of fly ash particles [34–36].

In this study, we used a novel approach in characterizing fly ash particles by using synchrotron based scanning transmission X-ray microscopy (STXM) to investigate the spatial distribution of different elements and phases at nanometer scale. Two-dimensional map at spatial resolution of 30 nm was obtained by STXM to get the compositional and morphological information on fly ash particles while concurrent measurements of near edge X-ray absorption fine structure (NEXAFS) provided chemical speciation of the samples. To our knowledge, this is the first work on characterizing fly ash particles using STXM work to resolve the current challenge in studying fly ash particles.

2. Experimental Approach

2.1. Materials and Sample Preparation. Class F fly ash particles are obtained from Bridger plant (Northern California) of Headwaters Resources. The American Society for Testing and Materials (ASTM) specifies two types of fly ash, Class F ash and Class C ash, for use as a pozzolan or mineral admixture in concrete. Class F ash is normally produced from burning anthracite or bituminous coal. Samples of Class F fly ash from combustion of bituminous coal at electric power plants were examined in this study. XRF analysis was conducted at University of California at Berkeley, XRF Laboratory, to identify the chemical composition using Phillips PW2400 X-ray fluorescence (XRF) spectrometer. The major and minor element compositions are given in Table 1. The sample contains Si, Al, and Fe as the major constituents with approximately 63% Si, 17% Al, and 6% Fe. The Ca concentration was approximately 3%. XRD analysis indicates that hematite (α-Fe₂O₃), mullite (Al₆[Si₆O₁₈]•O₁₀−2x), and quartz (SiO₂) are the main crystalline phases occurring in the bulk samples of Class F fly ash.

For STXM measurements, samples were studied in polished cross section in order to examine the internal structure and composition of individual particles. A representative portion of fly ash sample was impregnated with epoxy, allowed to harden, and then polished using ion milling techniques. Sample was characterized by randomly selecting 3–4 fields of view and examining all the fly ash particles observed within the selected fields to obtain a statistical representation of the sample.

2.2. STXM Measurements. STXM is a transmission X-ray microscopy using a monochromated X-ray beam produced by synchrotron radiation. X-ray absorption edges occur when an incident photon energy exceeds the binding energy needed to remove an electron completely from an orbital shell. Different near edge absorption bands are observed depending on the absorbing atoms and these bands are sensitive indicators of the local chemical bonding environment surrounding the atom of interest. In addition to obtaining the chemical speciation information, STXM allows identifying heterogeneity within the samples on a particle-by-particle basis. Using the scanning transmission X-ray microscopy (STXM) together with concurrent measurements of near edge X-ray absorption fine structure (NEXAFS) spectra, it is possible to examine sample compositions and spatial heterogeneities.

Two different types of results were collected at the STXM beamline. First, image contrast data is obtained from
differential absorption of X-rays depending on the chemical composition of the sample. Second, image stacks or line scan data are obtained by scanning the sample in the x-y direction (image stack) energy increments of 0.1 eV over the energy range of interest. Normalization and background subtraction of the spectra were performed by dividing each spectrum from the sample by the spectrum of the sample-free location. Elemental maps of the samples can be obtained by subtracting the image obtained from below the absorption energy level and the image obtained above the absorption energy. Axis 2000 software (version 2.1) [37] was used to align image stacks and extract NEXAFS spectra from image stack or line scan measurements. In this study, the STXM results were collected at the Advanced Light Source (ALS) branch lines 11.0.2.2. and 5.3.2. with the synchrotron storage ring operating at 1.9 GeV and 200–400 mA stored current. More details on experimental approach and data analysis can be found in previous STXM work on cement samples [38, 39].

3. Results and Discussions

Figure 1 shows STXM image of Class F fly ash particles at below and above Al K-edge. As described in the experimental section, image contrast measurements, a.k.a. image mapping, identify a concentrated area of the element of interest by taking the pixel intensity before the absorption edge of the element and subtracting it from the same image taken above the absorption edge. Figure 1(a) is the area of clustered fly ash particles taken at 1555 eV, which is below the absorption edge of Al. This image is subtracted from image of the same location taken above the Al absorption edge (1570 eV) as shown in Figure 1(b). The final results, that is, image mapping, are shown in Figure 1(c) where the high concentrated Al area is identified as bright white locations. Once it was confirmed that there was element of interest, then a smaller area of the image is selected and magnified to observe fine features of samples and to measure NEXAFS spectra from various locations to identify possible heterogeneity in chemical species within the samples at spatial resolution up to 25~30 nm. 10 μm × 10 μm of smaller region was chosen for detailed investigation shown as a small rectangular region in Figure 1(c).

Al K-edge NEXAFS spectra are taken from various locations within the single particle of fly ash as shown in Figure 2(a). As shown in Figure 2(b), no significant changes in the peak positions and features in the Al K-edge spectra were observed suggesting that no changes in oxidation state or coordination environments occur with

Figure 1: STXM images of Class F fly ash for Al K-edge taken at (a) 1550 eV; (b) 1570 eV; (c) image map of samples by taking a contrast between (a) and (b); (d) image map of the smaller area shown as a white rectangular region in (c) for spectral analysis.
respect to Al atoms in this sample. Concentration variation was observed since areas with relatively less Al contents were found within the nm range of the spatial resolution (see spectra 1–3 in Figure 2(b)). Studies have shown that a clear distinction can be made between the XANES spectra of tetrahedrally coordinated Al and octahedrally coordinated Al. Absorption band occurring at 1566 eV is characteristic of tetrahedral compounds whereas peaks at 1568 and 1572 eV are characteristic of octahedral compounds [40]. Comparing the sample spectra to the reference Al containing mineral spectra as shown in Figure 2(c), we conclude that the single peak at 1566 eV in our fly ash sample is due to the tetrahedrally coordinated Al. Our sample spectra are similar to previously reported Al K-edge NEXAFS spectra of mordenite-type aluminosilicate containing Al atoms in the zeolite framework or mullite [41]. Based on our STXM and NEXAFS analysis, we conclude that only tetrahedrally coordinated Al is present and its concentration varies within a single particle of fly ash. Other studies have also identified tetrahedrally coordinated Al in cementitious materials using various other experimental approaches [42, 43], but our work is the first of the kind in suggesting the special heterogeneity of the Al coordination in fly ash sample.

Figures 3(a) and 3(b) show Fe L_{II,III}-edge and Si K-edge image maps of the same sample site as Al K-edge image map was taken. We observed spatial heterogeneity with different elements by comparing the image maps of all three different elements. For example, areas 4, 5, 6, and 7 have relatively high concentration of both Si and Al based on the NEXAFS spectral analysis. Areas 1, 2, and 3 all contain relatively low
Figure 3: STXM image map of Class F fly ash at (a) Fe L\textsubscript{II,III}-edge; (b) Si K-edge. NEXAFS spectra of sample taken from various regions as shown in the image map for (c) Fe L\textsubscript{II,III}-edge; (d) Si K-edge.

Based on the spectral features and their comparison to previous Fe L\textsubscript{II,III}-edge NEXAFS spectra, we conclude that it is dominantly oxidized Fe (i.e., Fe\textsuperscript{3+}) in the fly ash sample as shown in Figure 3(c). Unlike Al and Fe, Si K-edge NEXAFS spectra showed spatial variations as shown in Figure 3(d). For example, spectrum taken from area 4 has its major peak at 1846.8 eV whereas the one from area 7 has its peak at 1858.3 eV. Previous work has demonstrated that Si K-edge NEXAFS spectrum can serve as fingerprints of local structural variations in silicate. For example, a four-oxygen coordinated Si, such as in quartz, has a Si K-edge absorption edge at around 1846.8 eV whereas six-oxygen coordinated Si, such as in stishovite, has its absorption edge at around 1858.3 eV [44]. Hence, we conclude that Si has intra- and interparticle spatial variation with respect to its coordination environment.

concentration of both Si and Al. This confirms that there is fairly strong chemical correlation between Si and Al. STXM also confirms that spatial variation in terms of elemental concentration exists within the fly ash particles. On the other hand, Fe concentration and spatial distribution did not have strong correlation with Si nor Al. For example, relatively low concentration of both Si and Al was found in area 2 whereas high concentration of Fe was found in area 2. Area 7 was found to contain high concentration of Si and Al, yet it had relatively low amount of Fe. We cannot conclude that there is an inverse correlation either since area 6 contains relatively high concentration for all three elements. We would like to note that this conclusion is based on examination of at least three different sites on different particle samples to enhance our statistical confidence in conclusion.
4. Conclusions

In summary, we have observed homogeneous oxidation and coordination state of Al as shown in Al K-edge NEXAFS spectra with heterogeneity in its spatial distribution. We also identified that Si and Al are spatially correlated at nanometer scale in which high concentrations of Si and Al were concurrently and consistently observed within the 30 nm resolution. However, Fe distribution did not show any specific correlation to Al and Si and mostly oxidized form of Fe based on Fe LIII-edge NEXAFS spectra. Unlike Al and Fe where no variation in chemical oxidation and coordination was observed, Si showed spatial inter- and intraparticle variations in polymerization. Our work suggests that spatial heterogeneous distribution of elements exists in the fly ash sample. Future work on examining the overall reactivity of the fly ash and relating it to nanometer range elemental distribution and chemical compositions will further enhance our understanding of fly ash.

Competing Interests

The authors declare that there are no competing interests regarding the publication of this paper.

Acknowledgments

This research is funded by the Republic of Singapore's National Research Foundation through a grant to the Berkeley Education Alliance for Research in Singapore (BEARS) for the Singapore-Berkeley Building Efficiency and Sustainability in the Tropics (SinBerBEST) Program. The authors also wish to acknowledge Professor Wenk at University of California at Berkeley who helped them prepare STXM samples and Timothy Teague at University of California at Berkeley with his help on sample preparation. The Advanced Light Source is supported by the Director, Office of Science, Office of Basic Energy Sciences, of the US Department of Energy under Contract no. DE-AC02-05CH11231.

References


