

## Research Article

# Thermoelectric Properties of a Single Crystalline Ag<sub>2</sub>Te Nanowire

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Silver chalcogenides have received much attention in potential thermoelectric materials research because of high carrier mobility and low effective mass. Among them, in Ag<sub>2</sub>Te, it was reported that the phase transition from monoclinic to cubic phase occurs at relatively low temperatures, so that extensive research for effective application using this material has been aroused. In this work, we investigated how 1-dimensional nanostructure affects the thermoelectric properties through as-synthesized single crystalline Ag<sub>2</sub>Te nanowires. Adopting well-defined thermoelectric MEMS device structure and transferring an individual Ag<sub>2</sub>Te nanowire, we measure electrical resistance and Seebeck coefficient as a function of temperature. When the phase changes from monoclinic to cubic, the resistance increases, while absolute Seebeck coefficient value decreases. These results are compared with previous reports for Ag<sub>2</sub>Te bulk and film, suggesting the increased density of states of the carriers due to nanowire structure.

## 1. Introduction

Silver chalcogenides (Ag<sub>2</sub>X; X = S, Se, and Te) have attracted much attention due to low effective mass and high carrier mobility [1–3]. Ag<sub>2</sub>Te among silver chalcogenides has aroused significant interests in the field of topological insulators field through recent reported quantum phenomena [4–6]. Most thermoelectric materials having topological insulating properties due to strong spin-orbit coupling have been studied for the improvement of thermoelectric properties prior to establishing the concept of topological insulators. Ag<sub>2</sub>Te reveals the phase transition from narrow-gap semiconductor to superionic conductor at 130–150°C [7]. In addition, the monoclinic structure of Ag<sub>2</sub>Te changes into face-centered cubic and body-centered cubic phase at 145 and 802°C, respectively [8]. Such structural phase transition can induce the modification of electrical transport and lattice volume, closely related to thermoelectric properties.

For the future energy industry, thermoelectric technology, which converts thermal waste to electrical power, plays an indispensable role for efficiently supplying energy that can be lost in devices [9, 10]. The performance of thermoelectric devices is determined by the dimensionless figure of merit

(*ZT*), expressed in  $ZT = S^2T/\rho\kappa$ , where *S* is the Seebeck coefficient, *T* is the absolute temperature,  $\rho$  is the electrical resistivity, and  $\kappa$  is the total thermal conductivity.  $\kappa$  consists of electron ( $\kappa_e$ ) and lattice thermal conductivities ( $\kappa_L$ ), respectively. These values strongly depend on the electronic structures and physical parameters of carriers, often adversely affecting each physical parameter [11]. For semiconductors with wide bandgap, the Seebeck coefficient can increase, but the electrical resistivity decreases due to low carrier concentration. Moreover, if metallic materials are used for lower resistivity, thermal conductivity increases because temperature difference between end sides of the materials is similar. However, when downsizing as the nanoscale, the thermal conductivity is significantly reduced owing to smaller dimensions than the mean free path of the phonons and enhancement of density of states via quantum confinement effect [12–14]. Several research groups have focused on the low-dimensional systems of well-known thermoelectric materials for improvement of the performance like increasing *ZT*.

In this work, we investigated the thermoelectric properties of an individual Ag<sub>2</sub>Se nanowire around phase transition temperature. High quality single crystalline Ag<sub>2</sub>Te nanowires are synthesized by simple chemical vapor transport method.

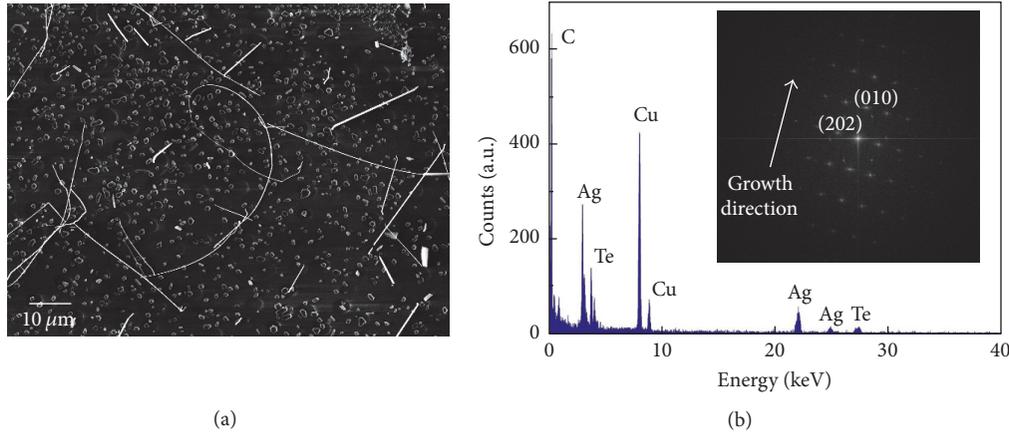


FIGURE 1: (a) SEM image of as-grown  $\text{Ag}_2\text{Te}$  nanowires on a sapphire substrate. (b) TEM-EDS spectrum of single  $\text{Ag}_2\text{Te}$  nanowire. C and Cu peaks are originated from a TEM grid. Inset: FFT of lattice resolved TEM image, indicating high quality single crystalline nature and (010) growth direction.

By a home-made nanomanipulator, individual nanowire is transferred to prepatterned microelectromechanical systems (MEMS) membrane. Applying the current into nanoheater electrodes, we observe drastic changes of both resistance and Seebeck coefficient around phase transition temperature ( $130^\circ\text{C}$ ).

## 2. Materials and Methods

Single crystalline  $\text{Ag}_2\text{Te}$  nanowires were synthesized in a horizontal one-zone furnace with a 1-inch diameter quartz tube.  $\text{Ag}_2\text{Te}$  powder was used as the starting precursor in the synthesis of  $\text{Ag}_2\text{Te}$  nanowires. Sapphire ( $\alpha\text{-Al}_2\text{O}_3$ ) substrates were placed at  $\sim 12$  cm from the boat containing the precursors. Prior to loading the quartz tube was degassed and purged with high purity argon gas, supplied with a flow-rate of 30 standard cubic centimeters per minute (sccm) under 10 Torr. The furnace was set to  $980^\circ\text{C}$  for 30 minutes with  $40^\circ\text{C}/\text{min}$  ramping rate. Morphology and structural analyses of the grown  $\text{Ag}_2\text{Te}$  nanowires were performed using a combination of scanning electron microscope (SEM) and transmission electron microscope (TEM).

Thermoelectric measurement was carried out using a membrane-type microdevice, which was fabricated on Si wafer coated with 500 nm thick SiN layer by e-beam lithography technique. A microelectrode pattern, which consists of a symmetric pair of Pt nanoheater electrodes, Pt thermometers, and current carrying electrodes, was defined in the center of the microdevice using a UV stepper. The microelectrode was formed by e-beam evaporation ( $\text{Pt}/\text{Ti} = 40 \text{ nm}/10 \text{ nm}$ ). The backside Si beneath the micropattern was etched away in a KOH solution so that the microdevice has a SiN membrane structure to enhance the measurement sensitivity. An individual  $\text{Ag}_2\text{Te}$  nanowire was transferred from the sapphire substrate to the microdevice using a nanomanipulator. We adopted sputter deposition of  $\text{Pt}/\text{Ti} = 90 \text{ nm}/5 \text{ nm}$  to make electrical and thermal contacts between the electrodes and the sample.

## 3. Results and Discussion

The SEM image of Figure 1(a) shows as-grown  $\text{Ag}_2\text{Te}$  nanostructures including nanowires and nanoplates on a sapphire substrate. In spite of low density,  $\text{Ag}_2\text{Te}$  nanowires are several tens of micrometers long and 100 to 200 nm in diameter. For  $\text{Ag}_2\text{Te}$  nanoplates, the geometry with approximate  $3 \mu\text{m}$  width and  $10 \mu\text{m}$  length is observed.

The elemental composition of as-synthesized  $\text{Ag}_2\text{Te}$  nanowires was examined by energy dispersive X-ray spectrometry (EDS) using TEM. Figure 1(b) reveals the TEM-EDS spectrum taken from a single  $\text{Ag}_2\text{Te}$  nanowire. The atomic ratio of Ag and Te elements is approximately 2 : 1 within an instrumental accuracy, confirming that the nanowire is composed of only Ag and Te. Except C and Cu from the TEM grid, no impurities and metal catalyst are observed. As shown in the inset of Figure 1(b), the two-dimensional fast Fourier transform (FFT) of the lattice resolved image obtained from the TEM shows a regular spot pattern, indicating high quality single crystalline nature. Furthermore, the spots pattern can be fully indexed to the monoclinic  $\text{Ag}_2\text{Te}$  phase via lattice spacing and demonstrate that the nanowire growth is along the [010] direction down the [100] zone axis. All the structural and compositional results are consistent with prior reports [6].

Figure 2 shows the representative SEM image of thermoelectric device and single  $\text{Ag}_2\text{Te}$  nanowire that we used for our experiments. To prevent heat from diffusing out to the Si substrate, the device platform is suspended. The electrical resistance of the  $\text{Ag}_2\text{Te}$  nanowire is measured by 4-probe method. In addition, the thermal gradient across the nanowire is detected by 4-point thermometry electrodes. During thermal expansion,  $\text{Ag}_2\text{Te}$  nanowire is clamped on the electrode using Pt, avoiding the change in the contact area caused by the movement of the nanowire (inset of Figure 2).

In Seebeck coefficient measurement, Pt nanoheater electrodes are used to generate a temperature gradient  $\Delta T$  across

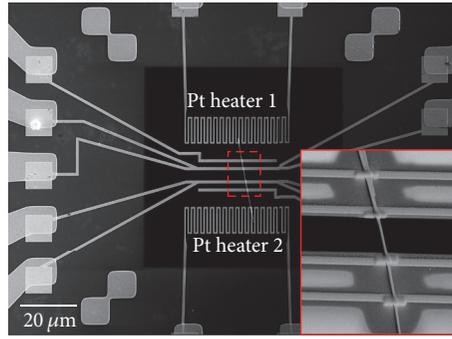


FIGURE 2: SEM image of thermoelectric MEMS structure with  $\text{Ag}_2\text{Te}$  nanowire. Inset: magnified SEM image of the red square box in the main Figure 2.

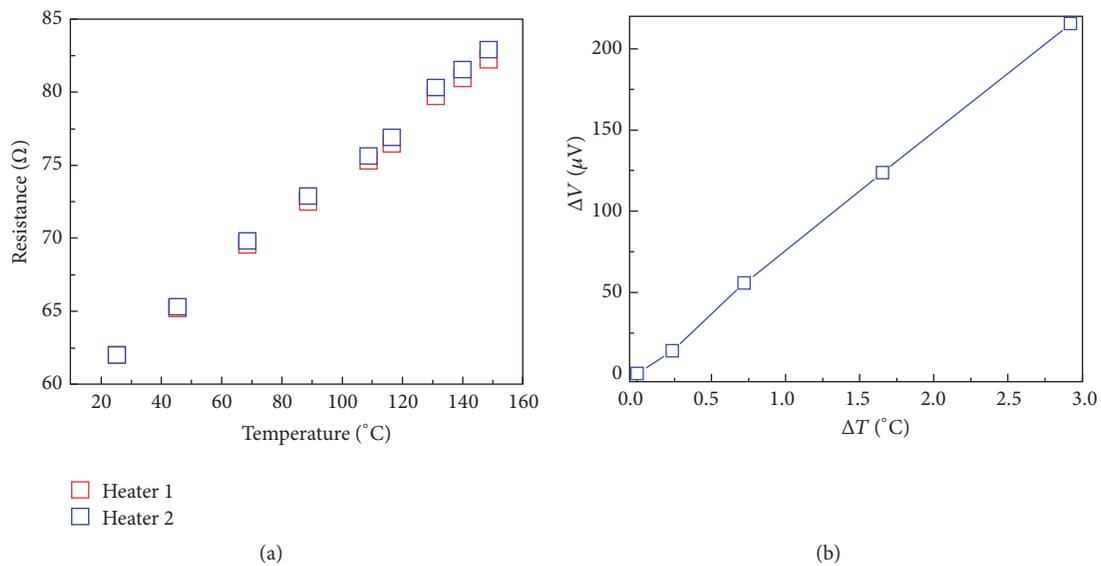


FIGURE 3: (a) Resistance of Pt nanoheaters 1 and 2. The resistance values of each nanoheater are converted into temperatures at hot and cold sides. (b) Thermoelectric voltage ( $\Delta V$ ) as a function of temperature gradient ( $\Delta T$ ) across the sample.

the sample. The thermoelectric voltage  $\Delta V$  in the sample is measured using a nanovoltmeter. The temperature difference  $\Delta T$  between the hot and cold sides is determined by the change of resistance at the Pt thermometers using lock-in amplifiers. To convert the resistance change into  $\Delta T$ , the temperature coefficients of resistance (TCR) of the Pt thermometers are estimated for each of samples. By linear fitting of the resistance as a function of the temperature, the TCR is typically calculated to be about  $0.1636 \Omega/^{\circ}\text{C}$  as seen in Figure 3(a).  $\Delta V$  is measured at different heater powers, leading to the Seebeck coefficient taken from a linear fitting of the  $\Delta V/\Delta T$  plot in Figure 3(b). Indeed, a switching module (scanner relay) is adopted to make the Pt thermometers play the dual roles of a thermometer and a voltage probe without signal interference: two lock-in amplifiers read  $\Delta T$  in the relay-on state while the nanovoltmeter measures  $\Delta V$  in the relay-off state.

Figure 4 shows current-voltage ( $I$ - $V$ ) curves of  $\text{Ag}_2\text{Te}$  nanowire measured by 4-probe method at various temperatures, exhibiting ohmic behavior. The resistance of  $\text{Ag}_2\text{Te}$

nanowire at  $120^{\circ}\text{C}$  is about  $6.43 \text{ k}\Omega$ . When the temperature of the thermal heater membranes increases over  $130^{\circ}\text{C}$ , the resistance of  $\text{Ag}_2\text{Te}$  nanowire suddenly increases to be about  $27.35 \text{ k}\Omega$ . As shown in the inset of Figure 4, the resistance at the temperatures over  $130^{\circ}\text{C}$  slightly decreases with increasing temperature, owing to larger band gap in the cubic phase.

Figure 5 shows the Seebeck coefficient as a function of the temperature measured from  $\text{Ag}_2\text{Te}$  nanowire. As previous studies reported that  $\text{Ag}_2\text{Te}$  is  $n$ -type semiconductor [1, 6, 15], the Seebeck coefficient reveals negative values, confirming that the majority of charge carriers are electrons. The Seebeck coefficients also drastically change at  $130^{\circ}\text{C}$ , similar trend of the resistance. Such results indicate that there is a phase transition of  $\text{Ag}_2\text{Te}$  nanowires between  $120^{\circ}\text{C}$  and  $130^{\circ}\text{C}$ , consistent with reported results [16–18]. The Seebeck coefficient is strongly related to the carrier density. In general, the absolute Seebeck coefficient values increase with decreasing carrier density in semiconductors [19], similar with previous reports in  $\text{Ag}_2\text{Te}$  pellets and  $\text{Ag}_2\text{Te}$  film [17, 18].

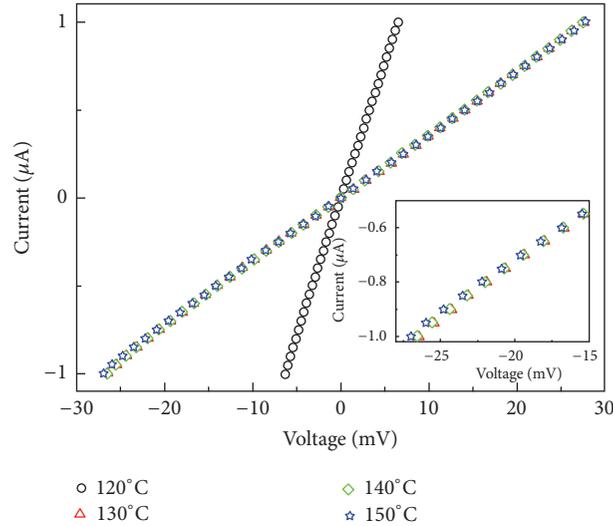


FIGURE 4:  $I$ - $V$  characteristics of individual  $\text{Ag}_2\text{Te}$  nanowire measured at various temperatures. Inset: enlarged  $I$ - $V$  curves ranged from  $-1.0 \mu\text{A}$  to  $-0.5 \mu\text{A}$ .

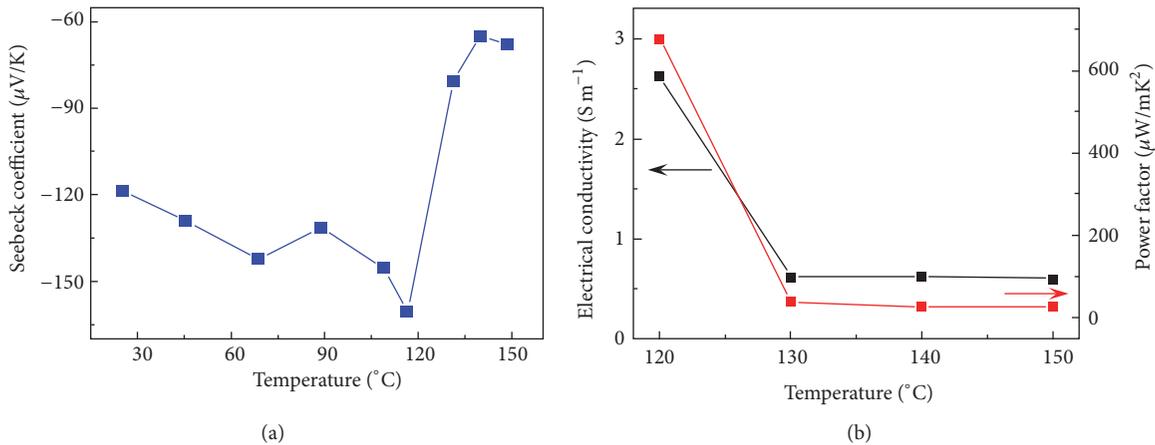


FIGURE 5: (a) Temperature dependent Seebeck coefficient of  $\text{Ag}_2\text{Te}$  nanowire. (b) Electrical conductivity and power factor of the same  $\text{Ag}_2\text{Te}$  nanowire.

However, our results for the single  $\text{Ag}_2\text{Te}$  nanowire show that, with increasing temperature, the absolute Seebeck coefficient values slightly decrease followed by sudden increment before phase transition, but electrical conductivity simultaneously decreases in cubic phase regime as shown in Figure 5(b). Such opposite feature may stem from increased density of states of the electrons when the structural dimension is reduced like nanowires. Figure 5(b) shows the thermopower factor,  $S^2\sigma$ , where  $S$  and  $\sigma$  are Seebeck coefficient and electrical conductivity, respectively. When the phase of  $\text{Ag}_2\text{Te}$  changes into cubic, power factor suddenly decreases up to  $25 \mu\text{W}/\text{mK}^2$ , but larger value than other polycrystalline  $\text{Ag}_2\text{Te}$  results [20, 21]. Our  $\text{Ag}_2\text{Te}$  nanowires with single crystalline nature grown by vapor-solid method result in the benefit to higher electrical

conductivity, expecting excellent performance like high  $ZT$  for thermoelectric applications.

#### 4. Conclusions

We have successfully synthesized high quality single crystalline  $\text{Ag}_2\text{Te}$  nanowires by simple chemical vapor deposition. Adopting the fabrication of MEMS device for measuring thermoelectric properties and transferring individual  $\text{Ag}_2\text{Te}$  nanowire, we observe the temperature dependent resistance and Seebeck coefficient, where the resistance and absolute Seebeck coefficient decrease with increasing temperature over phase transition temperature, different trend from other dimensional  $\text{Ag}_2\text{Te}$  structures. These results pave the way

to explore not only the thermoelectric properties of silver chalcogenide topological insulators, but also the potential application for the efficient thermoelectric devices using nanostructures.

## Conflicts of Interest

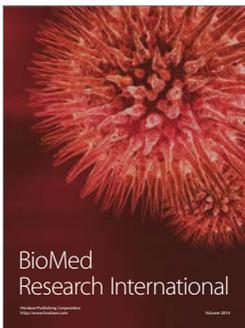
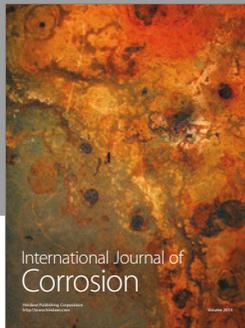
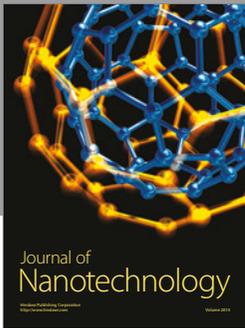
The authors declare that they have no conflicts of interest.

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