Synthesis of Carbon Nanomaterials from Rice Husk via Microwave Oven

Muhammad Asnawi, Saman Azhari, Mohd Nizar Hamidon, Ismayadi Ismail, and Intan Helina

Institute of Advanced Technology (ITMA), Universiti Putra Malaysia (UPM), 43400 Serdang, Selangor, Malaysia

Correspondence should be addressed to Muhammad Asnawi; muhdasnawi92@gmail.com

Received 21 November 2017; Accepted 4 January 2018; Published 27 February 2018

1. Introduction

Nanocarbons including fullerene, graphene, nanodiamonds, nanotubes, and activated carbon have unique physical and chemical properties leading the extensive research interest and exhibit promising applications [1–4]. Among them, graphene and carbon nanotubes are the most interactive stars in the field of nanotechnology. These carbon structures have proven strong potentials in revolutionizing wide range of applications globally. Basically, they can be classified as zero-dimensional (0D) fullerenes, one-dimensional (1D) carbon nanotubes, two-dimensional (2D) graphene, and three-dimensional (3D) graphite which consist of carbon as the building block [5, 6]. CNTs are made of hexagonal network of sp² hybridization of carbon atoms that is similar to graphene [1]. CNTs are the strongest materials due to carbon–carbon σ bonds and can be mainly categorized into single-wall carbon nanotubes (SWCNTs) and multiwall carbon nanotubes (MWCNTs) [7]. They consist of honeycombed lattice carbon sheets with interlayer spacing of 0.32–0.35 nm and 2 to 50 sheets tubules range of wall thickness [8].

Novel structure of CNTs has led to their remarkable properties including high chemical stability, excellent electrical conductivity, and high mechanical properties [9]. Compared to CNTs, graphene possesses special features, for instance, high reactivity of graphene edge [10] and high light transmittance [11]. Exceptional electrical, mechanical, thermal, and electrochemical properties are exhibited by CNTs; it is widely used in semiconductor, sensor, superconductive electrode, drug delivery, and other forms of electronics [12]. So far, various techniques for synthesis of CNTs had been reported: laser ablation [13], chemical vapor deposition [14], arc-discharge [15], flame synthesis [16], and high-pressure carbon monoxide (HiPCO) [17].

Recently, the synthesis of CNTs using waste product has received attention from researchers. In this study, a residue from paddy called rice husk (RH) is used to synthesize CNTs using microwave oven (MO) technique [18]. RH is rich in cellulose and lignin which qualifies it as a good and economic source of carbon precursor [19].

Raman spectroscopy and field emission scanning electron microscope (FESEM) as well as thermal gravimetric analysis (TGA) have confirmed the presence of carbon nanomaterials consisting of CNTs and other structures. On the basis of our experimental result combined with previous works a possible growth mechanism in conversion of RHs into carbon nanomaterials has been proposed [20].
2. Experimental Details

2.1. Synthesis of Carbon Nanomaterials. Rice husks were washed and cleaned by distilled water from unwanted materials using sonication; RH was then dried for 24 hours at 65°C. The dried RH was mechanically ground into powder by commercial blender machine and sieved using laboratory test sieve of ~20 μm. Ferrocene, Fe(C₅H₅)₂, as catalyst was dissolved into ethanol and sonicated for 20 minutes at room temperature. 120 mg of dried RH powder and 80 mg of Fe(C₅H₅)₂ solution were mixed together and placed on 2.5 cm × 2.5 cm aluminum sheet as substrate. Aluminum sheet was then covered with a handmade aluminum casing, Ø 40 mm (diameter) × 10 mm (height). The sample was then placed at the center of the quartz tube inside the microwave oven (Samsung M539 MAN200405W), 600 W power and 2.45 GHz frequency (household microwave frequency). Prior to microwave plasma irradiation, the quartz tube was evacuated into base pressure of 1.3 × 10⁻¹ mbar. Deposition time was employed for 38 minutes. Even though plasma forms shortly after switching on the microwave, it will not react rapidly to form CNTs. This is because, within 5–30 minutes, production of amorphous carbon is more likely to occur. It has also been reported that under plasma radiation amorphous carbon forms if the process exceeds 40 minutes [20]. The tube pressure was controlled by a rotary pump and a leak valve. No additional heater was installed in the system for substrate heating; however the substrate temperature could increase with increasing the growth time. During the growth process, quartz tube pressure initially increased to 3 × 10⁻¹ mbar and finally decreased to 1 × 10⁻¹ mbar. Thermocouple was placed in direct contact with the substrate holder to measure the substrate temperature. After irradiation, the sample was cooled down naturally to room temperature. After the plasma processing, the top of aluminum foil case was covered by soot-like appearance as shown in Figure 1(b).

2.2. Material Characterization. The microstructure morphological analysis was performed using a field emission scanning electron microscope (FESEM) operated at 10 kV. The crystalline structure was investigated via Raman spectroscopy using 532 nm laser. To determine the amount of catalyst impurities thermal gravimetric analysis (TGA) under air was utilized.

3. Result and Discussion

Figures 1(a) and 1(b) show the schematic diagram of MO technique for derivation of carbon nanomaterials from RH. The synthesis process as explained in experimental section shows that synthesis pressure plays a crucial role in synthesizing carbon nanomaterials from RH. The RH powders were irradiated for maximum time of 38 minutes. The maximal temperature and pressure were 750°C and 3 × 10⁻¹ mbar, respectively. Figures 2(a) and 2(b) show the growth of CNTs which was investigated using FESEM. CNTs growth has received much attention but yet the growth mechanism remains unclear. From these results it could be hypothesized that, due to strong van der Waals interaction between RH and catalyst, catalyst coalesces at nucleation sites, stimulating root growth. Figure 2(b) shows a weak interaction between catalyst-RH and substrate (metal has an acute contact angle with the substrate); it seems like that hydrocarbon decomposes from the top surface of RH. It could be assumed that the carbon diffuses down through the RH, and CNTs precipitate out across the metal bottom pushing the RH particle off the substrate. As long as the RH top is open for fresh hydrocarbon decomposition, the concentration gradient exists in the metal allowing carbon diffusion and CNTs continue to grow longer. Once the RH is fully covered with excess carbon, its catalytic activity ends and the CNTs growth stopped. This process is called tip growth mechanism.

Figure 2(c) shows the surface morphology of carbon nanocluster and CNTs. The images from FESEM show the formation of CNTs structures. Surface morphology indicates twisted and randomly oriented web-like network, several micrometers long, showing that the reduced powders retain the shape of the reduction vessel. Some of the particles may be onion-like carbon nanostructures. Depending on the nature and amount of the catalyst, some differences are revealed in FESEM images.

Figure 3 shows the corresponding Raman spectrum of the RH derived carbon nanomaterials with the two prominent D and G bands. The D peak observed at 1335 cm⁻¹ with FWHM of 105.35 and area of 64.75, obtained from Gaussian curve fitting, is disordered induced band that is often observed in defective graphitic structures and disordered sp³ hexagonal carbon atom. The G peak observed at 1601 cm⁻¹ with FWHM of 45.88 and area of 28.92, obtained from Gaussian curve fitting, is associated with the doubly degenerated phonon mode of the Brillouin zone center and indicates the level of order in sp³ hexagonal carbon atom; it is associated with the first-order Raman scattering. The ratio of D band to G band (I_D/I_G) is utilized to estimate defects present on carbon nanomaterials. The ratio of (I_D/I_G) for the synthesized nanomaterials was calculated to be 1.013. The D and G band intensities (I_D/I_G) indicate lower quality sp³ carbon structures when (I_D/I_G) ratio is high. Lower quality may arise due to amorphous carbon and structural defects in carbon nanomaterials. Although radial breathing modes (RBM) are the peaks found at the lower wavenumber between 120 cm⁻¹ and 350 cm⁻¹ which depend directly on the nanotubes diameter indicating the presence of SWCNTs the possibility of catalyst particles should not be neglected. On one hand these peaks do not appear in MWCNTs which suggest the presence of SWCNTs; on the other hand these peaks positions are similar to the peaks observed and reported previously as hematite and magnetite [21, 22]. The presence of these peaks suggests the reaction of ferrocene under microwave condition to form various arrangement of iron oxide. The small peak observed at 1488 cm⁻¹ can be attributed to semicircle stretching of carbon atoms in single aromatic ring [23]. Based on this result we may hypothesize that during the synthesis and plasma ionization ferrocene dissociates and elemental iron forms, which then reacts with oxygen dissociated oxygen atoms from RH. This explains the presence of iron(II, III) oxide peaks in the Raman spectrum. It has already been shown...
Figure 1: ((a) and (b)) Schematic diagram of MO technique for carbon nanostructure growth from RHs. (c) The conversion process of RHs soot-like material on the top of aluminum case surface including spherical and tubular structures.

by other groups that the presence of oxygen during CNTs synthesis is crucial [24]. This result indicates that during this process the CNTs growth could be a continuous process if RH surface remains exposed.

The TGA result shown in Figure 4 exhibits 10 wt% weight reduction at 343°C which may be due to the presence of amorphous carbons and structural defects; the sample then begins to decompose rapidly from 90 wt% to 4.6 wt% beginning from 343°C to 490°C, respectively; this suggests the presence of carbon nanomaterials. The DTGA result displays a major peak observed at 421.5°C at 50 wt% which does not seem high enough for CNTs in comparison with results obtained by other research groups; however, this result is justifiable since it has been shown that the presence of iron oxide in CNTs sample may result in lower decomposition temperature [25]. Based on this result we may presume the presence of carbon nanomaterial as a main outcome of the experiment [26]. The residual of the sample at 750°C was about 3.7 wt%; this indicates a good conversion rate of ferrocene to carbon nanomaterials.

4. Conclusion

We have demonstrated MO process to fabricate carbon nanostructure, specifically CNTs, from waste RH powders. It is necessary to use carbon source, catalyst, and commercial microwave oven to induce plasma. The plasma enhances and speeds up the catalytic decomposition of RH in presence of ferrocene. In addition, the results suggest the conversion of ferrocene to iron(II, III) oxide with notable conversion rate. CNTs offer great promises for applications such as electrochemical electrodes, due to their special features including high surface area and specific capacitance. This experiment shows that waste biomass can potentially be used as feedstock
Figure 2: ((a), (b), (c), and (d)) FESEM images of the RH derived CNTs by MO technique for 38 min.

Figure 3: Raman spectrum of the prepared sample.

Figure 4: TGA and DTGA result obtained from synthesized sample.

Conflicts of Interest

The authors have no conflicts of interest to declare.

Acknowledgments

The authors thank Dr. Ismayadi Ismail for many helpful discussions about Raman scattering in graphite and also for...
help with Raman measurement used in Figure 3. In addition, they thank Intan Helina for useful suggestion of FESEM. The work was supported by the Nanomite, Department of Higher Education, Ministry of Higher Education.

References


