Micro- and Nanofibrillated Cellulose (MNFC) from Pineapple (Ananas comosus) Stems and Their Application on Polyvinyl Acetate (PVAc) and Urea-Formaldehyde (UF) Wood Adhesives

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Micro- and nanofibrillated cellulose (MNFC) was extracted from pineapple stems by acid hydrolysis, then characterized and tested in two concentrations (0.5 and 1.0 wt%) in polyvinyl acetate (PVAc) and urea-formaldehyde (UF) adhesives. The modified adhesives were used to glue three tropical wood species (Vochysia ferruginea, Cordia alliodora, and Gmelina arborea), and their corresponding bond strength resistance was determined. MNFC and the correspondent adhesives were characterized by TGA, FTIR, SEM, TEM, AFM, and viscosity determination. The TGA analysis of MNFC showed three decomposition reactions. The SEM, TEM, and AFM evaluations demonstrated the presence of micro- and nanosized dimensions of particles after the acid hydrolysis of pineapple stems. Adding 1 wt% MNFC to PVAc and UF adhesives increased their thermal stability in similar manner. Viscosity diminished in both modified adhesives with MNFC; however, this reduction did not affect the adhesion properties in the tropical wood tested. MNFC added to PVAc and UF adhesives improved shear strength (SS) of the glue line in the three tropical species tested. The highest SS increase was obtained when adding 0.5 wt% MNFC to PVAc in V. ferruginea, and 1 wt% MNFC concentration in the case of C. alliodora and G. arborea. In the case of UF adhesives, the application of 1 wt% MNFC produced significant differences in SS for the three tropical species studied.

1. Introduction

In recent years, and considering their abundance, annual crops have been used as raw materials to extract micro- and nanofibrillated cellulose [1, 2]. Several authors have studied their extraction from corn, wheat, rice, coconut, soy [3], jute, and pineapple leaves [1, 2].

The annual crop pineapple (Ananas comosus), with a world production of around 24.8 million tons per year [2, 4], produces large amounts of residues in the form of leaves, stems, crowns, and fruit peel that are not used [5]. In Costa Rica for instance, this crop was introduced in 1970, and by 2014, there were approximately 37 660 ha of pineapple-sown fields available [6]. From these fields, nearly 300 tons of stubble are produced [6] which can be an important source of high-value materials [7]. One possible utilization for the pineapple wastes has been the extraction of natural fibers for rope and textile manufacturing [7] or pulp production [8].
Considering pineapple residues’ availability, the aim of this work was to produce and characterize nanocellulose from pineapple stems and determine the effect of adding this material on two commercial woody adhesives: urea-formaldehyde (UF) and polyvinyl acetate (PVAc). While the behavior of adhesives with nanocellulosic material from crops has been explored with certain success in temperate countries, there is no information regarding their performance in tropical countries [9, 10] or in general tropical wood adhesion [11]. The adhesives were applied to the following tropical species: *Vochysia ferruginea*, *Cordia alliodora*, and *Gmelina arborea* (main wood produced in Costa Rica from fast-growth plantation), and their shear strength was evaluated. The rationale to perform this work was based on preliminary research which demonstrated that adding nanocellulose to traditional wood adhesives (UF and PVAc) improves the bondability between woody elements [12–15].

Mahrdt et al. [12] demonstrated that even though the incorporation of microfibrillated cellulose (MFC) increased the viscosity of the adhesive and delayed the adhesive’s cure, the interaction of UF with MFC leads to a shift in the size distribution of adhesive particles towards larger particles for the UF-MFC compared to UF alone. Considering this fact, a higher fraction of adhesive is available for bond-line formation, and a larger part of the wood particles is covered with adhesive. Therefore, it was proposed that the change in the adhesive distribution together with improved adhesive toughness was the main contributions to the improved board strength. Ayrilmis et al. [13] combined microfibrillated cellulose (MFC) with urea-formaldehyde adhesive and evaluated the adhesive performance on laminated veneer lumber (LVL). In general, the authors’ main findings indicated that the cellulosic filler improved the bonding performance which was quantified through the tensile shear strength determination of the LVL products.

2. Materials and Methods

2.1. Materials. Pineapple (*Ananas comosus*) stems were provided by Reuti-piña, which planted pineapple crops in the northern part of Costa Rica. The plantations had a density of 70,000 plants per hectare, and their age was 18 months old.

Two water-based adhesives were tested in combination with the micro- and nanofibrillated cellulose (MNFC) produced in this research from pineapple stems. Adhesive 1: polyvinyl acetate (PVAc) (Resistol M.R. 850, produced by Henkel Capital S.A.—54.5–55.0% solid content and 1600–2000 cPs viscosity). Adhesive 2: urea-formaldehyde (UF) (Resina CR-560 U-F, manufactured by Química Centroamericana Quibor S.A.—48% solid content and 650–900 cPs viscosity). The technical description of the adhesive 2 indicated that it is prepared with 4 components: resin (51%), water (26%), wheat flour as binder (20%), and ammonium sulfate as catalyst (3%).

Three tropical species were used to evaluate the modified adhesives. These were botarrama (*Vochysia ferruginea*), laurel (*Cordia alliodora*) and gmelina (*Gmelina arborea*). They were selected because of their commercial importance in the manufacture of pallets in Costa Rica [16]. The specific gravity or basic density of these species are 0.44 for *V. ferruginea*, 0.40 for *C. alliodora*, and 0.41 for *G. arborea* [16].

2.2. Micro- and Nanofibrillated Cellulose (MNFC) Production. Figure 1 presents the process used in this research to extract MNFC from pineapple stems. The general process was performed as described by Oun and Rhim [17], but with slight modifications related to acid concentrations and reaction times.

2.2.1. Fibers from Pineapple Stems. Pineapple stems were cut into 2 cm long pieces and then dried at 80°C until reaching 4% moisture content. Then, they were ground to 300 μm size. A 2.5% sodium hydroxide (NaOH) solution was prepared and poured into the ground pineapple stem. The mixture was then heated up to 100°C for 4 h. Afterwards, the material was washed with distilled water until it reached neutral pH and then with bleached with 4% sodium hypochlorite (NaClO) for 2 h. Next, the material was washed with distilled water again to obtain neutral pH.

2.2.2. MNFC from Pineapple Fibers. Fibers from Section 2.2.1 were firstly treated with an acid hydrolysis process using 37% hydrochloric acid (HCl (aq.)) under reflux at 90°C for 30 min. The hydrolyzed microcrystalline cellulose followed subsequent filtration and neutralization with distilled water until it reached neutral pH [18]. It is important to clarify that after this first hydrolysis, a simple vacuum filtration was performed. There was an intrinsic loss of material, but at this point, it is assumed that mostly microfibers, not much nanofibers, were available.

A second acid hydrolysis step was performed using 64% sulfuric acid (H₂SO₄) at ambient temperature (ca. 22°C) for 1 h. MNFC final suspensions were washed with ultrapure water, and successive dilution and centrifugation process at 1200 rpm were applied until the suspensions reached a pH between 3 and 5. After that, the suspensions were purified by a dialysis process. A final mechanical treatment using ultrasound for 10 min at an amplitude of 7 Hz was performed to get MNFC hydrogel (0.033 g ml⁻¹).

2.3. Preparation of Adhesives with MNFC. MNFC suspensions were added to polyvinyl acetate (PVAc) and urea formaldehyde (UF) in two concentrations: 0.5 wt% and 1.0 wt%. The addition was performed based on the solid content of each adhesive, i.e., ca. 55% for PVAc and 48% for UF. The denomination of the adhesives used for further test and analysis is presented in Table 1.

The preparation of the adhesives listed in Table 1 was performed as follows: 150 g of PVAc was stirred with a 45° inclined blade agitator at 1600 rpm. Then, a proportion of MNFC gel (approximately 13.75 ml of gel or 0.75 g of MNFC for 0.5% concentration and 27.5 ml of gel or 1.5 g of MNFC for 1.0% concentration) was slowly added into the adhesive and under continuous stirring for 10 minutes. As for UF, the same general procedure was applied, but low amount of MNFC was added considering the lower solid content of this adhesive. For 0.5% MNFC concentration, 12 ml of MNFC gel was added onto 76.5 g of the UF adhesive (ca. 0.3925 g of MNFC). For the 1% MNFC
concentration, 24 ml of MNFC gel was added onto 76.5 g of the UF adhesive (ca. 0.765 g of MNFC).

2.4. Characterization of MNFC and the Modified Adhesives. MNFC was characterized using Atomic Force Microscopy (AFM), Scanning Electron Microscopy (SEM), and Transmission Electron Microscopy (TEM), for distribution and morphology. Also, MNFC was analyzed using Thermogravimetric Analysis (TGA) and Fourier-Transform Infrared Spectroscopy (FTIR). The pristine adhesives and their mixtures with MNFC were characterized using Thermogravimetric Analysis (TGA), FTIR, and bond strength resistance.

2.4.1. Thermogravimetric Analysis (TGA). Thermal stability of MNFC material and the modified adhesives were analyzed in triplicate using a TGA 5000 (TA Instruments). Thick films from the samples under study were prepared, following the next steps. A sample of approximately 5 g of the adhesives with MNFC was poured on a glass plate, and it was allowed to dry at room temperature (ca. 22°C) for two days. Then, the film formed was detached from the glass and cut to a 1 cm diameter circular shape. Small subsamples of 6-7 mg from the films were analyzed in the TGA under nitrogen atmosphere (90 ml min⁻¹ flow rate and 25°C min⁻¹ heating rate). The experiments run from ambient temperature up to 1000°C. The analysis of data, which includes determination of initial degradation temperature (Tᵢ), maximum degradation temperature (Tₘᵢₓ), final degradation temperature (Tᵣ), percentage of remaining mass at initial temperature (MRᵢ), remaining mass at maximum temperature (MRₘᵢₓ), and remaining mass at final temperature (MRᵣ), was performed using the TA Instruments Universal Analysis Q500 software. Afterwards, the thermal stability, which measured the stability of resin with temperature, activation energy (E), and entropy factor (A) were determined. The final data were modelled (Equations (1)–(4)) with the Gaussian curve (Equation (1)) and MagicPlot software to determine its variables (Equation (2)). With a normal distribution and the constant “c” equal to the standard deviation, the values of the initial and final temperatures correspond to x₀ ± 3c, considering 99.7% of the population.

\[ f(x) = A e^{-\frac{(x - x_0)^2}{2c^2}}, \]  
\[ f(x) = A e^{-\frac{x(x-a)^2}{2c^2}}, \]  
\[ -\frac{1}{2c^2} = \ln \left(\frac{2}{2}\right), \]  
\[ 2c^2 = \frac{\ln (2)}{\ln (2)}. \]
adhesives: determine the following parameters for each reaction in both Doyle model (Equation (5)) [19]. The model was applied to temperature. Kinetics in each reaction was calculated by means of the Doyle model (Equation (5)) [19]. The model was applied to determine the following parameters for each reaction in both adhesives: $E$, which is the energy needed for each reaction to occur, and $A$, which is specific for each reaction [20].

\[
\ln [-\ln (1 - \alpha)] = \ln \left(\frac{AE}{BR}\right) - 1.0518 \frac{E}{RT} - 5.33
\]  

Where $\alpha$ is the degraded mass, $A$ the entropy factor, $E$ the activation energy, and $T$ the temperature.

2.4.2. Fourier-Transform Infrared (FTIR) Spectra. Functional groups present in the MNFC samples and modified adhesives were determined using FTIR analysis. Before the analysis, a 50 ml of MNFC gel sample was placed in an oven during 24 h at 60°C. In parallel, 2 g of the modified adhesives was placed in an oven at 103°C for 72 h, and then the material was pulverized with a mortar. The FTIR spectra of the samples were acquired using a Thermo Scientific Nicolet 6700 equipped with an Attenuated Total Reflectance accessory within the range of 3800-800 cm$^{-1}$, a resolution of 4 cm$^{-1}$, and 16 scans. All spectra were analyzed using the software OMNIC.

2.4.3. Morphology Analysis Using TEM, AFM, and SEM. For TEM observations, dry samples of MNFC were diluted at 1.9 vol% in distilled water. The samples were disintegrated using an ultrasonic bath for 15 min at a frequency of 40 kHz. After that, a 5-drop sample was placed on a TEM grid (covered with formvar), and it rested for 5 minutes for the grid to retain the sample. Then, the grid was left to dry for 2 days, and the samples were observed in a TEM (Jeol JEM 2100) at an acceleration voltage of 200 kV. A drop of the mixture of each modified adhesive was placed on paraffin film on top of two slides to obtain a fine sheet of the material to observe MNFC dispersion in the adhesives through the TEM.

For AFM observations, 1 ml of the 1.9 vol% MNFC suspension was placed on a mica surface, and it was left there to air-dry for 2 days. The surface topography of the MNFC samples was observed by using an atomic force microscope (Asylum Research) operated in the air-tap mode. Silicon probes were used (model Tap150Al-G, rear part of the overhang covered with Al) with resonance frequency of 150 kHz and 5 N m$^{-1}$ force constant.

For SEM observations, one drop of the 1.9 vol% solution was placed on a glass slide with carbon tape, and it was left there to air-dry for 3 days. Then, the dry sample was coated with gold-palladium before analysis. A SEM Jeol JSM-5900LV at 20 kV acceleration voltage was used. To measure MNFC using TEM and SEM micrographs, the software of the corresponding system (Gatan Digital-Micrograph) was used.

2.4.4. Viscosity of the Adhesives. The viscosity of the modified adhesives was measured at 20°C using a Discovery HR-3 hybrid rheometer with a parallel disc geometry of 25 mm diameter, 500 μm between discs, applying shear rates between 0 and 200 s$^{-1}$. Four repetitions for each sample were performed.

2.4.5. Bond Strength Resistance. The bond strength of the modified adhesives was tested using the glue line test (SS) according to ASTM D905-98 [21], after their application on wood samples of three tropical species (Vochysia ferruginea, Cordia alliodora, and Gmelina arborea). Ninety shear tests were carried out in total (30 samples per concentration x 3 concentrations) per each species. Wood samples were dried at 12% moisture content for one week. The adhesive was applied according to manufacturer’s specifications, which recommended 100 g adhesive per square meter wood surface. After applying the adhesive, the samples were pressed in a hydraulic press at 0.20 N mm$^{-2}$ for 24 hours. After that, the samples were conditioned to 20°C and 60% relative humidity for two weeks prior the bond strength test. A Tinus Olsen hydraulic test machine with 50 kN capacity was used for testing the adhesives performance on tropical woods.

2.5. Data Analysis. Compliance of the variables determined with the assumptions of the normal distribution, variance homogeneity and the presence of outliers, was verified. Subsequently, a variance analysis was performed to verify the effect of the MNFC adhesion (three levels: 0.0%, 0.5%, and 1.0%) on the values of $E$, $A$, the different temperatures, and remaining masses in the various reactions of the decomposition kinetics, in addition to the bond strength resistance. The Tukey test at 99% confidence level was applied to determine the statistical difference among averages. The InfoStat software (InfoStat Company, version 2017, Buenos Aires, Argentina) was used for statistical analysis.

3. Results and Discussion

3.1. Thermal Stability Analysis. TGA analyses for pristine MNFC and for the modified adhesives with three concentrations of MNFC (0.0, 0.5%, and 1.0%) are shown in Figures 2(a) and 2(b). As expected, MNFC loses water in the range temperature of 25-100°C (ca. 30%) because it was prepared in aqueous environment. Two inflexions appear in MNFC, at 200°C and at 350°C (Figures 2(a) and 2(b)) which corresponds to degradation temperatures. These results agreed with values presented by Kargarzadeh et al. [22], where temperatures of 230°C and 350°C are reported, and of Kumar et al. [23], who reported values of 220°C and 330°C for MNFC extracted by acid hydrolysis from sugarcane and kenaf, respectively.

With the TGA data and using Equation (5), it was possible to obtain 3 fitting curves that correspond to 3 reactions (Figure 3). Reaction 1 occurred between 169°C and 252°C, reaction 2 occurred between 199°C and 282°C, and reaction 3 occurred along the whole decomposition process, from 37°C to 693°C. Initial degradation temperature (Ti) of MNFC samples was lower than the ones reported by other studies;
however, the methodologies of nanocellulose extraction were also different [1, 24, 25]. The decrease in MNFC thermal stability may be due to shorter MNFC chains promoted by extensive acid hydrolysis [23, 26].

Nanocellulose production includes cellulose degradation that involves processes of dehydration, depolymerization, decomposition of glycoside bonds, and carbonization [23]. Nanocellulose produced by acid hydrolysis incorporates surface sulfate groups to the cellulosic chains [27]. In this research, degradation of the MNFC sulfate groups can be observed at the maximum point of reaction 1 (at 210°C) and in reaction 2 which started after Tmax at 241°C [23, 26]. Roman and Winter [28] support these results when they analyzed the nanocellulose extracted by acid hydrolysis from bacterial cellulose. These authors mention that initial nanocellulose decomposition corresponds to sulfate groups added to the amorphous zones of the nanocellulose during hydrolysis, producing a first peak in the DTG curve. They also mention that sulfate groups adhere to the nanocellulose crystal chains, causing a second decomposition peak. The decomposition of sulfate group allows greater exposure of cellulose nanocrystals and its degradation, giving way to a third process of decomposition, which occurred from 300°C to 400°C, with decomposition Tmax at 365°C (Figure 3).

Complete MNFC decomposition in this study took place between 338°C and 350°C (Figure 3). Studies conducted by Abraham et al. [24], Costa et al. [29], and Draman et al. [2] agreed with these results.

The analysis of the TGA curves of the MNFC 0.5-PVAc adhesive shows a slight increase in the decomposition temperature in relation to MNFC 0-PVAc (Figure 2(a)). Moreover, the thermal stability of both concentrations (Figure 2(b)) differ in UF adhesives with or without MNFC (Figure 2(b)). The first PVAc inflexion appeared around 346°C, and the second inflexion at 448°C, while in UF adhesive, the inflexion occurred at 263°C approximately. Greater mass loss occurred between 300°C and 400°C in both adhesives in all MNFC concentrations (Figures 2(c) and 2(d)).

DTG fitting curves are shown in Figures 2(c)–2(f) for the two modified adhesives. Figures 2(c) and 2(d) show the fitted curves corresponding to the inflexions found in the TGA for PVAc, and Figures 2(e) and 2(f) the equivalent for UF. The second decomposition curve appears starting at 209°C with a maximum peak close to 290°C.

For modified PVAc adhesives, reaction 1 occurred between 297°C and 396°C (Figure 2(c)), which is attributed to formation of volatile components and loss of acetate groups (deacetylation) [28]. Reaction 2 occurred between 342°C and 553°C (Figure 2(d)) and is attributed to the decomposition of the main structure of the residues of the polyene polymer that turn into carbon and hydrocarbons [30, 31].
Reaction 1 in the UF adhesive (Figure 2(e)), occurred between 239°C and 285°C, and it refers to the degradation of the polymer because of the partition of the polymeric chains and formaldehyde emission from the dimethylene ether groups [32]. Reaction 2 occurred within a higher range of temperatures, between 207°C and 380°C (Figure 2(f)) and is attributable to mass loss during fragmentation of the polymer residues [33] in which methylene, methylol, dimethyl ethers, methylene glycols (formaldehyde), and components with carbonyl groups, among others, are freed [34].

Relative to MNFC effect on the temperatures of reaction, a slight increase in Tmax of the MNFC 0.5-PVAc adhesive was observed for PVAc adhesive, as in MNFC 1-UF (Figures 2(c)–2(f)) in both reactions. This is congruent with reports from Kaboorani et al. [22], who added MNFC to PVAc adhesive to determine its mechanical and thermal effects in the adhesive. Increase in reaction Tmax when adding MNFC to the adhesive was determined (Figures 2(c)–2(f)).

An important aspect to highlight considering the curves of the modified adhesives and the pristine MNFC curves is that the decomposition Tmax of the two main reactions of pure MNFC occurred before the reactions of the adhesives: Tmax was 347°C and 448°C on average for reactions (1) and (2), respectively, in PVAc adhesive, and in the UF adhesive, 262°C and 292°C for the reactions (1) and (2), respectively. This behavior was also observed in a study where degradation at lower temperature of MNFC extracted by acid hydrolysis from commercial microcellulose with respect to polymer (polylactide and polyvinyl acetate) occurred. This is because pure MNFC appears in chains with broader surface area and nanoscale sizes that allow its rapid decomposition [35], while adhesives present much stronger bonds between particles that make its decomposition more difficult [31].

Ti and Tf in reaction 1 of UF adhesive exhibit small variations in the three concentrations (Figures 2(e)–2(f)). Tmax increased in MNFC 1-UF, whereas in MRi, MRmax, and MRf, it tended to diminish as MNFC increased. Similar Ti were observed in reaction 2, whereas Tmax and Tf tended to increase proportionally to MNFC percentage and were higher in MNFC 1-UF. Conversely, the remaining masses of the three decomposition points tended to decrease the higher the MNFC. Ash content was lower with 1% MNFC in both adhesives. In both adhesives, the increase in temperatures and diminution of the remaining mass when adding the MNFC was due to disordered network or particle agglomeration containing a crystalline structure formed by the MNFC that restricts mobilization of the material and slows down its decomposition [14, 31].

Table 2 shows the decomposition kinetics parameters $A$ and $E$ for PVAc and UF wood adhesives (with different concentrations of MNFC), and pure MNFC. These parameters help to measure the reinforcement effect of MNFC on the adhesives. For PVAc, the energy reaction 1 needs for decomposition to occur increased with MNFC concentration; however, $E$ values oscillated between 211 and 213 kJ mol$^{-1}$. On the other hand, in reaction 2, MNFC 0.5-PVAc exhibited the smallest value, while the MNFC 0-PVAc adhesive needed the highest amount of energy to carry out decomposition (Table 2). The evaluation of the UF adhesive showed that $E$ in reactions (1) and (2) presented the lowest values for MNFC 1-UF. MNFC 0.5-UF presented the highest $E$ value for reaction 1 (Table 2).

$E$ is the energy necessary for reactants to interact and cause the reaction [36]; therefore, it is also an indicator of the stability of the material [37]. MNFC 1-PVAc and MNFC 0.5-UF adhesives presented better stability in reaction 1 because it has a higher $E$. The increase of stability in adhesives produced by the MNFC is due to the rigid network between the fibers resulting from hydrogen bonds of MNFC, increasing the resistance of the polymer to degradation [38].

$A$ is an indicator of the rate of material decomposition [36, 39]. It exhibits similar values in reaction 1 of PVAc adhesive in the three MNFC concentrations, while in reaction 2 of such adhesive, MNFC 0.5-PVAc showed the lowest value, and MNFC 0-PVAc presented better dynamics during the
reaction. For UF adhesive, in reaction 1, MNFC 1-UF exhibits the lowest $A$ value, while MNFC 0.5-UF shows the highest $A$ value, suggesting higher stability in MNFC 1-UF regarding this parameter. No differences were observed among the three concentrations in reaction 2 of adhesive UF.

3.2. Fourier-Transform Infrared (FTIR) Spectra. Figure 4 shows the FTIR spectra of the samples under study. The FTIR spectrum of MNFC showed the presence of aqueous and organic components (Figure 4(a)). The signal at 3334 cm$^{-1}$ corresponds to the stretching of H-bonded O-H bonds present in water molecules and polysaccharides, while at 1635 cm$^{-1}$, it shows the bending mode of vibration of water molecules [37]. On the other hand, at 2914 cm$^{-1}$, it shows the stretching of C-H of cellulose [37, 40]. The lack of signals in 1500-1600 cm$^{-1}$ that attributed to aromatic ring vibrations suggests the absence lignin in MNFC [41]. Other characteristic signals of cellulose occurred at 1053 cm$^{-1}$ corresponding to the C-O-C of the structure of the pyranose ring [40] and 896 cm$^{-1}$ with ring stretching in glucose units. According to Johar et al. [40] and Poletto et al. [42], the signals at 1053 cm$^{-1}$ and 896 cm$^{-1}$ indicate the presence of amorphous cellulose. The signal at 1428 cm$^{-1}$ is mainly attributed to crystalline cellulose, as it is strong for crystalline cellulose and weak for amorphous cellulose [40, 42–44]. Its presence in the FTIR spectrum of MNFC (Figure 4(a)) suggests the presence of crystalline cellulose. Thus, both crystalline and amorphous celluloses are present in the MNFC extracted from pineapple.

Pristine PVAc adhesive (Figure 4(b)) presented signals at 1725, 1430, 1377, and 1245-1275 cm$^{-1}$ related to the carbonyls, methyl groups, methylenes, and esters, respectively [45]. Other characteristic signals of this adhesive, usually used to identify it, are presented at signals at 1225 cm$^{-1}$ and 896 cm$^{-1}$ [45, 46]. In addition, the peaks at 3300 and 1660 cm$^{-1}$ discussed above increased for higher contents of MNFC (Figures 4(b)–4(d)).

Figures 4(e)–4(g) show the spectra of the UF and modified UF. The signal at 1000 cm$^{-1}$ corresponds to the NH$_3$ group. The peak shown at 760 cm$^{-1}$ corresponds to N-H groups, and the signals 1145 and 1375 cm$^{-1}$ to C-O and C-N bonds, respectively [33]. In addition, in 1650-1550 cm$^{-1}$, characteristic resin functional groups were detected, such as

![Figure 4: Fourier-Transform Infrared (FTIR) spectra of MNFC (a), PVAc (b), modified PVAc ((c, d) for 0.5 and 1.0 wt% of MNFC), UF (e), and modified UF ((f, g) for 0.5 and 1.0 wt% of MNFC).](image-url)

### Table 2: Activation energy and entropy factor for the decomposition kinetics of PVAc and UF wood adhesives (with different concentrations of MNFC), and pure MNFC.

<table>
<thead>
<tr>
<th>Adhesive</th>
<th>Reaction</th>
<th>MNFC (%)</th>
<th>Activation energy (kJ mol$^{-1}$)</th>
<th>Entropy factor (s$^{-1}$)</th>
<th>Correlation coefficient ($R^2$)</th>
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<tbody>
<tr>
<td>PVAc</td>
<td>1</td>
<td>0.0</td>
<td>211.45</td>
<td>2.87 x 10^{17}</td>
<td>0.95</td>
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<td></td>
<td></td>
<td>0.5</td>
<td>212.59</td>
<td>1.21 x 10^{17}</td>
<td>0.95</td>
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<tr>
<td></td>
<td></td>
<td>1.0</td>
<td>212.67</td>
<td>2.01 x 10^{17}</td>
<td>0.95</td>
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<tr>
<td></td>
<td>2</td>
<td>0.5</td>
<td>70.49</td>
<td>6.02 x 10^{4}</td>
<td>0.90</td>
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<tr>
<td></td>
<td></td>
<td>1.0</td>
<td>83.90</td>
<td>3.52 x 10^{6}</td>
<td>0.93</td>
</tr>
<tr>
<td>UF</td>
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<td>0.0</td>
<td>336.91</td>
<td>7.43 x 10^{35}</td>
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<tr>
<td></td>
<td></td>
<td>0.5</td>
<td>345.64</td>
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<td></td>
<td></td>
<td>1.0</td>
<td>323.27</td>
<td>5.06 x 10^{34}</td>
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<tr>
<td></td>
<td>2</td>
<td>0.5</td>
<td>117.03</td>
<td>8.75 x 10^{9}</td>
<td>0.94</td>
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<tr>
<td></td>
<td></td>
<td>1.0</td>
<td>115.42</td>
<td>5.22 x 10^{9}</td>
<td>0.94</td>
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<tr>
<td>Pure MNFC</td>
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<td></td>
<td>24.36</td>
<td>4.53 x 10^{2}</td>
<td>0.78</td>
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<tr>
<td></td>
<td>2</td>
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<td>3</td>
<td></td>
<td>15.20</td>
<td>2.14 x 10^{10}</td>
<td>0.91</td>
</tr>
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</table>
amide and C=O [47]. Also, an increase in the concentration of MNFC resulted in an increase in the signals at 2935 cm⁻¹ and 3300 cm⁻¹ due to an increase in the presence of C-H and O-H bonds [25, 35].

3.3. Analysis of Microscopy Images. Microscopy images revealed existing variation in the fibers dimension and therefore the micro- and nanometric nature of the MNFC produced. SEM analysis showed fibers of micrometric sizes over 200 μm in length and between 4 and 20 μm in diameter (Figure 5(a)). Moreover, TEM micrographs showed a fiber network around 30 nm in diameter and far longer than 200 nm long (Figure 5(b)), which was confirmed in the AFM (Figures 5(c) and 5(d)). In general, an acid hydrolysis process generates fibers with diameters between 5 and 60 nm and lengths of 200-300 nm. The process applied in this research produced fibers with aspect ratios (length/diameter) similar to those obtained by others. Abraham et al. [29] and Dos Santos et al. [1] extracted and evaluated MNFC from pineapple leaves by acid hydrolysis; the resulted dimensions were between 5 and 50 nm in diameter and 190 and 250 nm in length. Cherian et al. [48] evaluated the production of MNFC from pineapple leaves using steam explosion. They got MNFC with diameters between 5 and 60 nm and lengths between 200 and 300 nm. Johar et al. [40] produced MNFC from rice using acid hydrolysis; they reported diameters between 10 and 15 nm and lengths between 50 and 100 nm. Finally, Li et al. [49] used acid hydrolysis to obtain MNFC between 25 and 30 nm in diameter and 400 and 500 nm in length from branches and bark of white mulberry (Morus alba L.).

3.4. Viscosity. Figure 6 shows how the viscosity of the adhesives is affected after the incorporation of MNFC. As for MNFC in PVAc (Figure 6(a)), the viscosity diminished with increasing MNFC concentration, varying between 40 and 15 Pa s at the starting point. The incorporation of MNFC creates paths for humidity, allowing the presence of moisture. Adhesives with MNFC concentrations of 0.5% and 1% showed little variation because the most noticeable changes (increase) in viscosity in PVAc occurred at concentrations above 1% [14]. This is because the joints formed between MNFC particles create a stronger network. Such behavior was also observed for Chaabouni and Boufi [14], where they added MNFC to PVAc adhesive and tested it on Eucalyptus globulus. Same viscosity reduction is observed for UF adhesive, although the variation was less pronounced than for PVAc (Figure 6(b)). The change in the rheological behavior of the adhesive with the addition of MNFC comes from addition of water to the matrix, since the MNFC gel is made up of 97% water. Likewise, the shear force in the viscosity test breaks the polymer network, releasing trapped water [14] and thus decreasing the viscosity.

3.5. Bond Strength Resistance. Shear strength (SS) improved after the addition of MNFC for both PVAc and UF adhesives, in all the tropical species tested (Figure 7). The higher SS increase (35%) occurred for V. ferruginea when PVAc was modified with 0.5% or 1% MNFC. No significant difference was observed in this case for the increase from 0.5% to 1% MNFC. For the other two tropical species (G. arborea and C. alliodora) treated with modified PVAc, no significant differences in SS were found due to the presence of MNFC.
For UF adhesive, only the incorporation of 1% MNFC produced significant differences with respect to control samples (Figure 7(b)) for all three tropical species treated. The maximum increase for such adhesive was for *C. alliodora* (31%).

The results presented above agree with previous reports, which demonstrated that small amounts of MNFC can improve interactions within a water-based adhesive [12, 50–62]. For instance, Veigel et al. [50] and Zhang et al. [51] added 5% and 1.5% MNFC concentration to UF resins, respectively. Their results showed an important increase in the internal strength of the UF adhesive. Chaabouni and Boufi [14], Geng et al. [52], and Pracella et al. [53] added MNFC on PVAc adhesive. Their MNFC concentrations, ranging from 1% to 10%, increased considerably the adhesive shear strength. Kwon et al. [15] found that the addition of 3% MNFC in a UF adhesive increased the SS. Kwon et al. and Mantanis et al. [54] mentioned that the increase of SS was due to cellulosic fibers in the glue line filling the wood cavities together with the adhesive, allowing greater penetration, and therefore, much more strength is required to separate two pieces of wood. Likewise, the reinforcing effect of the wood-
adhesive interface results from the formation of a network between the MNFC and polymer chains of the adhesive [55]. Almeida et al. [56] explain that MNFC’s broader contact area, high resistance to traction, and rigidity allow the formation of a network of crystals that interact with the adhesive, increasing its mechanical strength.

4. Conclusions

Micro- and nanocellulose (MNFC) material was effectively obtained from pineapple stem by acid hydrolysis. The Thermogravimetric Analysis showed three decomposition events for the MNFC. The first two events correspond to degradation of sulfate groups, while the third is due to decomposition of the cellulosic material. After the pineapple acid hydrolysis, microscopic analyses confirm the presence of cellulosic material in the micro- and nanoscale.

The incorporation of MNFC on PVAc and UF adhesives improved their thermal stability. The viscosity of these adhesives was reduced after the addition of MNFC, mainly because of the addition of water contained in the MNFC gel. This fact was more evident in the PVAc adhesive where the initial viscosity was reduced in ca. 90% after the incorporation of 0.5 or 1% MNFC.

Significant increase of the shear strength of the glue line (SS) for both modified adhesives, PVAc and UF, was determined when 1% MNFC was incorporated. The highest increases were 35% and 31%, for PVAc and UF, respectively.

In summary, the material extracted from pineapple crops in this research resulted in a combination of micro- and nano-fibrillated cellulose that improved the adhesive performance of UF and PVAc. Bondability differences, through shear strength measurements, were found depending on the tropical species evaluated and the adhesive concentration used.

Data Availability

The nanocellulose data used to support the findings of this study are available from the corresponding author upon request. The following author’s email can be contacted: rimoya@itcr.ac.cr.

Conflicts of Interest

The authors declare no conflict of interest.

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