Research Article

Thermoelectric Characterization of Electronic Properties of GaMnAs Nanowires

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Nanowires with magnetic doping centers are an exciting candidate for the study of spin physics and proof-of-principle spintronics devices. The required heavy doping can be expected to have a significant impact on the nanowires’ electron transport properties. Here, we use thermopower and conductance measurements for transport characterization of Ga0.95Mn0.05As nanowires over a broad temperature range. We determine the carrier type (holes) and concentration and find a sharp increase of the thermopower below temperatures of 120 K that can be qualitatively described by a hopping conduction model. However, the unusually large thermopower suggests that additional mechanisms must be considered as well.

1. Introduction

Self-assembled semiconducting epitaxial nanowires are promising building blocks for field effect transistors [1], sensors [2], and solar cells [3]. An exciting new direction, which has recently been shown to be possible due to successful incorporation of magnetic Mn dopants into epitaxially grown GaAs nanowires (NWs) [4–11], is their use for proof-of-concept spintronics devices [12]. The doping techniques are advancing rapidly, and it has recently been shown that ion beam implantation can produce single crystalline, homogeneously doped GaMnAs NWs [13]. Furthermore, a recent study found that the Curie temperature of GaMnAs nanostrips could be enhanced to 200 K with nanostructure engineering [14], suggesting the possibility for nanowire-based devices to operate at higher temperatures compared to thin films or bulk. In addition to the exciting possibilities for application, from the fundamental point of view, ferromagnetic NWs will provide an opportunity to investigate the spin-Seebeck effect in reduced dimensions [15]. A deeper understanding of how spins and phonons couple thermodynamically could in turn lead to fundamentally new applications, such as spin-based cooling and magnetically sensitive thermoelectrics.

Here, we investigate the thermoelectric properties of Ga0.95Mn0.05As NWs. Combining thermopower and conductance (or resistance) measurements can provide information on carrier density when conventional characterization techniques via the Hall effect and field effect are not possible [16]. We were able to estimate the hole carrier density from thermopower measurements to be $p \sim 10^{17}$–$10^{18}$ cm$^{-3}$ in our NW. In addition, we find a dramatic rise in the resistance and thermopower of the NW below 120 K [17–19]. The resistance versus temperature measurements point to the role of Mott variable range hopping (VRH) transport with activation energy 62 meV at 100 K and hopping lengths of 11 nm. We show that the addition of a term due to Mott variable range hopping (VRH) transport [20–22] can also
qualitatively describe the rise in thermopower observed in these Ga$_{0.95}$Mn$_{0.05}$As NWs. However, our model suggests that a simple parabolic-band picture is not fully adequate, implying that more complex behavior is taking place.

2. Experimental Methods

GaAs NWs of 40 nm diameter were grown by MOVPE and subsequently implanted with Mn ions to doping concentrations of 0.5 to 2.9%, corresponding to Ga$_{1-x}$Mn$_x$As with $x = 0.01$ to 0.058 stoichiometry. Both simulations and transmission electron microscopy (TEM) showed that the NWs are of high crystalline quality after implantation and that the Mn is reasonably homogeneously distributed. In this work, we concentrate on the higher doped wires. The NW growth and implantation techniques were discussed in detail previously [13].

To prepare devices for thermoelectric characterization, NWs were collected from the growth substrate with clean-room tissue paper and then brushed onto Si/SiO$_2$ chips with 110 nm thick oxide. A thick metallic plane was evaporated on the backside of the chip for voltage-gating measurements. We searched optically for suitable NWs for processing and determined the location of the NWs relative to predefined Au alignment marks on the SiO$_2$ chips. Contacts to the NWs were fabricated via standard e-beam lithography processes. Briefly, 950-A5 PMMA was spun onto the sample at 5000 rpm for 60 s and baked at 180 °C for 5 minutes. After the contact pattern exposure, the resistance was developed in MBK:IPA = 1 : 3 for 30 s. To ensure good ohmic contacts to the NWs, we performed the following procedures: the samples were treated in HCl/H$_2$O solution for 15 seconds followed by passivation in (NH$_4$)$_2$S solution at 40 °C for 2 minutes. Then metal contacts of Pd (10 nm)/Zn (10 nm)/Pd (35 nm) were evaporated followed by lift-off [23]. Measurements were done in a Janis Varitemp cryostat between 60 K and 190 K.

3. Results and Discussions

Figure 1 shows a scanning electron microscope (SEM) image of a typical nanowire device with heater and thermometer metallic microstrip lines is shown. In a separate measurement, the contact electrodes can be independently used to probe the voltage drop along the nanowire or conductance. Image is of a device similar to the one that is measured.

![Figure 1: Scanning electron microscope image of a typical nanowire device with heater and thermometer metallic microstrip lines.](image)

Figure 2: Ga$_{0.95}$Mn$_{0.05}$As nanowire resistance as a function of temperature. The inset shows a plot of ln (R) versus $T^{-1/4}$, with the extracted activation energy of 62 meV at 100 K.

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![Figure 3: Ga$_{0.95}$Mn$_{0.05}$As nanowire resistance as a function of temperature.](image)

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The gradient generated along the wire was typically $\Delta T \sim 1 \text{ K}$. A metallic backgate was also available for field effect, but we found that the NW transport did not respond significantly to the applied gate voltages.

We discuss first the electronic transport of the NW as a function of temperature. In Figure 2, we show the resistance calculated from the measured two terminal conductances. The contact resistances obtained from our fabrication procedure were found to be from 500 Ω to 1 kΩ (for wires with 4 terminal contacts), and we conclude that the measured response is due to the wire itself [27]. The NWs show dramatic increase in the resistance for $T < 120 \text{ K}$, and this rise saturates below $T \sim 80–90 \text{ K}$. In order to obtain a better understanding of the transport mechanisms down to 60 K, we plot ln $(R)$ versus $T^{-1/4}$ in the inset of Figure 2. The linearity of such a trace indicates that Mott variable range hopping may likely be in play [20, 27]. Deviation from linearity occurs around $T^{-1/4} = 0.32$ or $T \sim 95 \text{ K}$. From the slope of this trace, we estimate the carrier density to be approximately $3 \times 10^{18} \text{ cm}^{-3}$ at 100 K with hopping lengths.
of 11 nm. This allows a determination of an activation energy of ΔE = 62 meV at 100 K. For larger 80 nm diameter wires, similar behavior was found though the hopping transport was found to persist down to T ~ 50 K [27].

Figure 3 shows the measured thermopower, which is defined as $S = \Delta V / \Delta T$, where $\Delta V$ is the measured voltage drop across the nanowire. The positive sign of the measured thermopower indicates that the transport carriers are holes. This is expected for Mn-doped GaAs since substitutional Mn is an acceptor in GaAs. The thermopower is seen to increase dramatically for temperatures lower than 120 K, which is the temperature where the electrical resistance also begins to sharply increase. In order to understand why the thermopower rises to the rather large values of ~800 µV/K, we first consider the thermopower in the paramagnetic regime (without taking into account hopping transport) consisting of the following two terms:

$$S_{\text{tot}} = S_{\text{diff}} + S_{\text{exch}}.$$  

The first term is essentially the semiclassical Mott relation for diffusive transport [16],

$$S = -\pi^2 k_B^2 T \frac{d}{dE} \ln \sigma \bigg|_{E=E_F} = -\frac{\pi^2 k_B^2 m^*}{(3\pi^2)^2/3} \frac{T}{\hbar^2 |e|}$$  

Here, $\sigma$ is the conductivity, $k_B$ is the Boltzmann constant, $e$ is the electron charge, $\hbar$ is Planck’s constant, and $m^*$ is the carrier density. The hole effective mass for GaMnAs is taken to be $m^* = 0.5 \times m_e$, where $m_e$ is the free electron mass [28]. The second term is from exchange mechanisms

$$S_{\text{exch}} = \frac{S_0 T}{T + T_0},$$  

$$S_0 = \frac{4\pi^2 k_B}{e} D(E_F) I_{pd} V \frac{\rho_{\text{exch}}}{\rho},$$  

where $D(E_F)$ is the density of states at the Fermi energy, $I_{pd}$ is the exchange integral between carriers and magnetic centers, $V$ is the nonmagnetic scattering potential, and $\rho_{\text{exch}} / \rho$ is the ratio between exchange contribution to resistivity to total resistivity. Hence, $T_0$ is a material-dependent parameter with weak temperature and magnetic impurity dependence, and for thin films it is found to be $T_0 = 150$ K [21]. We utilize these functional forms to obtain a fit to the data above 120 K, where the thermopower is relatively flat, as seen in Figure 3 (blue dot dash curve). We find that the parameters that give the best fit to the curve are $p \sim 3 \times 10^5$ cm$^{-3}$ and $S_0 \sim 4$ mV/K, with $T_0 = 150$ K fixed based on the thin film value. Due to the limited number of data points, we believe our estimate for the density to be correct to within an order of magnitude. This is comparable to what has been observed for comparable nanowires [27] and is, within our accuracy, consistent with the estimate obtained from the resistance versus temperature curves above. Note that the value of $S_0$ is unusually large, a point that we will return to below. These results demonstrate that thermopower presents a powerful method for extracting carrier sign and density when conventional Hall effect and field effect techniques are not possible.

Below 120 K, it is clear that additional contributions to the thermopower must be considered to explain the observed data. Taking a cue from the temperature dependence of the resistance, we consider adding a term due to hopping conduction. We see below that this hopping contribution, $S_{\text{hop}}$, can qualitatively describe the observed experimental behavior. We start with a hopping term of the form [21]

$$S_{\text{hop}} = \left( F_{\text{corr}} \ast \frac{k_B}{e} \right) \left( \frac{\Delta E}{k_B T} + A \right),$$  

where $A$ does not have significant temperature dependence and includes an additional factor, $F_{\text{corr}}$, to account for electron correlations. Without $F_{\text{corr}}$, (4) expresses that, in the simple case of single particle, noninteracting transport, the thermopower of a device or material essentially measures the average energy (here assumed to be the activation energy $\Delta E$) where transport takes place. By plotting the experimental data versus $1/T$, and taking the correlation factor, $F_{\text{corr}} \sim 5$, we can extract the activation energy $\Delta E \sim 63$ meV, in good agreement with that seen in the $R(T)$, as seen in Figure 4. Based on this, we can calculate the full theoretical trace including hopping mechanisms as shown in Figure 3 (green-dashed line). We see that inclusion of this hopping mechanism can describe the observed rise in thermopower. At 100 K, the theoretical trace seems to overshoot the experimental data. This is consistent with the fact that the ln($R$) versus $T^{-1/4}$ trace starts to deviate from linearity around 95 K.

Comparing the results from thermopower to that obtained from the electrical resistance, we see that an activation energy of 62 meV at 100 K is deduced from Mott VRH theory [27]. In order for (4) to give a comparable extracted
activation energy, we required $F_{\text{corr}} \sim 5$. This implies strongly that correlation effects play a more significant role here. The fact that $F_{\text{corr}}$ is much larger than one is not so surprising if we consider the unusually large measured thermopower. Indeed, colossal thermopowers have been seen in strongly correlated semiconductors such as FeSb [29], where the large thermopowers are presumed to be a result of the Fe 3d-states and Sb 5p-states hybridizing into a coherent, high mobility state. We note that compared to 2D GaMnAs films, which exhibit hole-mediated ferromagnetic coupling, these GaMnAs nanowires are clearly in the hopping regime, with a relatively low hole concentration and large hopping lengths. These observations are an indication of accep ter compensation from Mn interstitial impurities and As antisites, which were created by the ion beam implantation process. More analysis will be necessary to understand the relative importance of these two contributions. Clearly then, our results (i) highlight the important role that thermopower measurements can play as a characterization tool complementary to conductance measurements, and (ii) that our heavily Mn-doped nanowires exhibit more complex electronic behavior than is apparent from conductance measurements, leaving much room for further investigation.

4. Conclusions

In this work, we studied the thermopower and conductance of Ga$_{0.95}$Mn$_{0.05}$As nanowires. We utilize thermopower measurements as a means to estimate the hole carrier concentration when Hall effect and field effect are not possible. We find from thermopower and from conductance measurements the hole density $p \sim 10^{17}$ to $10^{18}$ cm$^{-3}$. Furthermore, we observe dramatic increases in both thermopower and resistance at low temperature. The resistance versus temperature suggests that the NW is in the regime of variable range hopping transport. From the thermopower, we deduce that additional correlated phenomena must be taking place. These fundamental studies provide critical insight into the transport mechanisms likely to be found in next-generation GaMnAs and ferromagnetic NWs.

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