

## Research Article

# **Electrochemical Meaning of Cumulative Corrosion Rate for Reinforced Concrete in a Tropical Natural Marine Environment**

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The objective of this paper is to discuss the meaning of cumulative corrosion rate ( $i_{CCR}$ ) of reinforced concrete in a tropical marine microclimate of the Yucatan Peninsula identifying four stages that correspond to passivation, beginning of depassivation, breakdown and formation of subsequent corrosion layers, and nucleation and development of cracks. Sixty Portland cement concrete cylinders were exposed in a tropical marine environment at 50 m from the seashore. One-half of the samples had a reinforcing bar embedded at the center of the sample (corrosion measurements) and the other half was made with plain concrete (chloride measurements). Five water/cement (w/c) ratios and three times of curing (CT) were tested representing the common practices of this region. The corrosion rate was monitored using the linear polarization resistance technique ( $R_p$ ) which enables calculating the apparent and cumulative corrosion rate. Representative results indicated that  $i_{CCR}$  was effective not only to detect the beginning and duration of the reported stages but also to find the right influence of CT and w/c ratios on the corrosion performance of reinforced concrete.

## 1. Introduction

Corrosion of reinforcing steel is one of the most important causes of reinforced concrete structures degradation [1-3]. The environment and climate change effects contribute also to the acceleration of this deterioration. The humidity, rainfall, and different concentrations of sea chloride ions in the environment enhance its deposition on the concrete surface and the degradation rate significantly depends on the seashore distance [4]. In the case of structures exposed in a marine environment, chloride ions are deposited on the surface of the concrete and penetrate until they reach the surface of the reinforcing steel, causing the breakdown of the passive layer [5]. Once reinforcing bar corrosion take place, corrosion products start to build up around the reinforcing bar, causing volumetric expansions that will result in concrete cover cracking after subsequent breakdown and formation of corrosion layers [6-8]. Apparent corrosion rate  $(i_{app})$  obtained through  $R_p$  has been possibly the most useful technique to obtain information about depassivation of steel and propagation of corrosion.

Andrade and González [9] were the first authors to establish a range of corrosion rate  $(0.1-0.5 \,\mu\text{A/cm}^2)$  where depassivation of reinforcement starts. However, sensibility of equipment, cell configurations, and natural data dispersion have made the interpretation of corrosion data before and after the depassivation state (initiation and propagation periods) difficult. With these limitations, the depassivation can only be recognized once corrosion has really started instead of when first data denote the beginning of the depassivation state. Then, formation and breakdown of subsequent corrosion layers cannot be easily detected. Actually, corrosion techniques have improved due to the digital age but precision depends mainly on mathematical treatment. With this in mind, cumulative corrosion rate has become a useful tool to know details of initiation and propagation periods. This tool has been reported formerly in this field by Andrade et al. [10-13] and subsequently by Troconis De Rincón et al. and Castro-Borges et al. [14, 15]. The reported information discussed by all of these authors contains not only the use of cumulative corrosion but also its interpretation in both natural and accelerated tests. However, it would be interesting to improve



FIGURE 1: Specimen configuration.

the interpretation of  $i_{CCR}$  through more exposure conditions from different type of materials, but specially on identifying the exact initiation of the depassivation, the breakdown and formation of subsequent corrosion layers, and the nucleation and development of cracks, all of which are an important part of this paper. The objective of this paper is, then, to discuss the electrochemical meaning of  $i_{CCR}$  behavior of reinforced concrete exposed for about five years in a tropical marine environment.

#### 2. Experimental Procedure

2.1. Materials and Sample Dimensions. ASTM Type I ordinary Portland cement (OPC) and crushed limestone aggregates were used for fabrication of concrete cylinders (diameter ~ 75 mm, height ~ 150 mm) (Figure 1). Plain concrete cylinders were used to determine 28-day compressive strength and chloride penetration during the experimental period. Typical number 3 deformed steel rebar (~0.4 wt% carbon content) was used to simulate reinforcement. Rebar sections were weighed before cylinder casting and section ends were protected with epoxy and adhesive tape. This limits the exposed area inside the concrete specimen and isolates any rebar areas directly exposed to the environment. Rebar sections were positioned axially within the cylinders and extended the entire length (Figure 1). A 50 mm activated titanium rod (ATR), characterized elsewhere, was used as a reference electrode for electrochemical measurements [16, 17]. For this study, it was calibrated versus a saturated calomel electrode (SCE). An epoxy coating layer was applied to the cylinder's ends to delimit concrete and steel exposure area (Figure 1). Concrete mixture proportions and physical/mechanical properties of the hardened concrete have been published elsewhere [16-19]. A total of 60 cylinders were tested at 50 m from the seashore, in the Port of Progreso having two cylinders from each combination of plain and reinforced concrete for a total of 30 plain concrete and 30 reinforced concrete cylinders.

One of the most important parameters for defining concrete quality is the w/c ratio [20]. A low w/c ratio reduces concrete porosity, and concrete permeability to external agents (i.e., chlorides) decreases when porosity is low, improving its service life [21]. To study the effect of w/c ratio on service life, concrete was made using five w/c ratios: 0.76, 0.70, 0.53, 0.50, and 0.46.

Ratios as high as 0.70 and 0.76 are commonly used in the Yucatan Peninsula region, although public buildings in Mexico (i.e., schools, hospitals) are constructed using lower w/c ratios (0.53 or 0.50). The 0.46 ratio is still not commonly used in this region but was included in this study for comparative purposes since this value is frequently used in the relevant literature. Curing time periods for concrete structures in Mexico range from 1 day to 28 days, although 7 days is by far the most common CT period. This variable was tested at 1, 3, and 7 days, with two cylinders tested per each combination of w/c ratio and CT.

2.2. Environmental Exposure. Specimens had been exposed to the marine environment of Progreso's Port, located at the north coast of the Yucatan Peninsula (21°18′N, 89°39′W), in Mexico, and are characterized as having a tropical humid climate. Specimens were placed about 50 m away from the seashore, on top of a roof (approximately 4 m above ground level). Both specimen types (plain and reinforced concrete cylinders) were oriented vertically. The carbon steel corrosivity category in this particular marine environment is higher than C5 (environment with  $\tau_4$ , time of wetness, or time that relative humidity is  $\geq 85\%$  and temperature > 30°C

and  $\leq 60^{\circ}$ C, between 2,500 h/y and 5,500 h/y; atmospheric chloride concentration  $300 \text{ mg/m}^2 \text{d}$  to  $1,500 \text{ mg/m}^2 \text{d}$ ) with an average annual temperature of 26°C and average annual relative humidity of 79% [22, 23]. Chloride was not added to the concrete mixtures prior to specimen casting to ensure that any chloride contamination was due to natural exposure during the experimental period.

2.3. Corrosion Measurements. Rebar electrochemical properties were measured using an external conductive rubber counter electrode tightly attached to the concrete cylinders with aluminum clamps for each test (Figure 1). Electrochemical measurements were taken with a commercially available potentiostat/galvanostat/ZRA and included corrosion potentials  $(E_{\text{corr}})$  and apparent corrosion rates  $(i_{\text{app}})$  determined by using the linear polarization resistance  $(R_p)$  technique [24, 25].  $R_p$  was measured by applying 10 mV in the cathodic direction from  $E_{\rm corr}$  at a 0.06 mV/s scan rate. Ohmic drop  $(R_s)$  as a result of the high concrete resistivity was measured with a commercially available resistance meter. The adjusted  $R_{\scriptscriptstyle n}$  estimates were then converted to apparent corrosion rate  $(\dot{i}_{app})$  values using the following equation [9]:

$$i_{\rm app} = \frac{B}{R_p - R_s},\tag{1}$$

where B = 0.026 when the rebar exhibited corrosion activity (i.e., when rebar potentials were more negative than -200 mV versus CSE); or B = 0.052 [9, 24] when the rebar exhibited passive behavior (i.e., when rebar potentials were more positive than –200 mV versus CSE); and  $R_p$  or  $R_s$  are in  $\Omega \cdot \text{cm}^2$ (calculated via multiplication by the nominal rebar surface area  $\sim$  33 cm<sup>2</sup> in contact with concrete). These measurements were taken every two to three months during the entire experimental period.

#### 3. Results and Discussion

Figures 2-6 show the average (two specimens)  $i_{app}$  (subsection (a)) and  $i_{CCR}$  (subsection (b) and (c)) values as a function of exposure time for the specimens at 50 m from the seashore exposure site, per each combination of w/c ratio and CTs. After depassivation and as observed, all w/c ratio concrete showed apparent  $i_{app}$  values higher than 0.5  $\mu$ A/cm<sup>2</sup>, indicative of steel active corrosion (DURAR Network, 1997) [26]. Figures 2–6 show clear trends in terms of depassivation for both w/c ratios and CTs. Also, electrochemical results presented well defined trends: higher w/c ratio concrete presented higher  $i_{\rm app}$  values. The higher the CT, the bigger the reinforcing bar depassivation time ( $i_{app} < 0.5 \,\mu\text{A/cm}^2$ ). But additional analysis is required with the data obtained from field measurements in order to obtain other important parameters like, for example, the corrosion initiation times as a function of w/c ratio and distance from the seashore and the corrosion degradation. Because of that, it was decided to observe the possible contributions to the knowledge of the electrochemical behavior that the cumulative corrosion rate can provide and give an interpretation as discussed from here.



FIGURE 2: (a) Average apparent  $i_{corr}$  and (b and c) cumulative  $i_{corr}$ versus exposure time, w/c ratio = 0.76, and three CTs.

Figures 2–6 show the behavior of  $i_{\rm app}$  and  $i_{\rm CCR}$  of the specimens exposed to the natural marine environment for 48 months or more. Each value is the average of 2 specimens. There are some clear facts:

- (a) The lower the w/c ratio, the longer the depassivation period.
- (b) In general, the lower the w/c ratio, the longer the stay of  $i_{app}$  at the transition zone. This means that more drift is detected between the active and the passive zone.
- (c) In general,  $i_{CCR}$  was able to show the contribution of CT to the passivation period: the longer the CT, the longest the depassivation period.
- (d) Despite a concrete cover close to 4 cm, depassivation times are shorter than those normally reported in



FIGURE 3: (a) Average  $i_{app}$  and (b and c)  $i_{CCR}$  versus exposure time, w/c ratio = 0.70, and three CTs.

the literature. This was due to the following reasons [27–29]:

- (i) High w/c ratios
- (ii) High levels of fines content in aggregates
- (iii) High porosity of the construction aggregates from this geographic area
- (iv) High atmospheric aggressiveness (>C5).

The fact that depassivation can occur early but then nothing much happens for a long time (green on the exploded  $i_{CCR}$  plots), as shown in Figure 6, was proposed earlier [30], where the authors also gave physical reasons for this phenomenon.

(e) Figures 2–6 show inflection points on  $i_{CCR}$  that denote a noticeable change of tendency on  $i_{CCR}$ . A number of points maintaining a constant tendency after the inflection point are better observed at lower w/c ratios.



FIGURE 4: (a) Average  $i_{app}$  and (b and c)  $i_{CCR}$  versus exposure time, w/c ratio = 0.53, and three CTs.

- (f) Figures 2–6 show color thickness to denote the stages detected according to  $i_{\rm CCR}$ . This color was maintained in  $i_{\rm app}$  plots just to denote that depassivation begins before the detected one with the  $i_{\rm app}$  criterion.
- (g) Figures 2-6 show four stages as defined below.

Stage 1 (passivation). At early stages of exposure to the marine environment, it is common that any specimen shows initial passivation, denoted by a passive layer. The electrochemical behavior, doing a follow-up of apparent corrosion, is distinguished by consistent and stable values below  $0.1 \,\mu\text{A/cm}^2$ . Passivation is stable if the  $i_{\text{app}}$  values do not increase and remain with a trend below the threshold  $(0.1-0.5 \,\mu\text{A/cm}^2)$  as shown in Figures 2(a)–6(a) with green thickness. This is clear in cumulative corrosion rate where a constant linear increase is observed in subsections (b) and (c) of Figures 2–6 in green color. In practical terms, the observed inflection point denotes the beginning of the depassivation



FIGURE 5: (a) Average  $i_{app}$  and (b and c)  $i_{CCR}$  versus exposure time, w/c ratio = 0.50, and three CTs.



Stage 2 (depassivation). When the passive layer begins to break by action of chlorides, as in this case,  $i_{app}$  values start drifting in the transition zone until a good part of the passive layer is broken. Figures 2(a)–6(a) show this trend with those data in yellow color. However, part of the difficulties on interpreting  $i_{app}$  is recognizing the exact moment in which depassivation occurs. For this reason, it is considered



FIGURE 6: (a) Average  $i_{app}$  and (b and c)  $i_{CCR}$  versus exposure time, w/c ratio = 0.46, and three CTs.

that depassivation occurs when the  $i_{app}$  values already do not return to the area of passivation (below the range 0.1–  $0.5 \,\mu$ A/cm<sup>2</sup>). Although this interpretation is correct, the fact is that it does not allow knowing whether the passivation will happen soon or not. It is therefore difficult to take preventive actions to control complete depassivation. This is where the utility of using cumulative corrosion comes. At the beginning of this stage (yellow dots in the data series in subsections (b) and (c) of Figures 2–6), the cumulative corrosion shows an inflection point from which a series of data following a stable trend that will end when the availability of oxygen is constrained initiates and another inflection point denotes the beginning of another stage (Stage 3).

Trends in stable increments of data can be seen in yellow color in Figures 2–6 (subsections (b) and (c)), noting a point in blue color on all of them that denotes the depassivation according to  $i_{app}$ . As shown, the points in blue were contained somewhere in the data sets in yellow, meaning that the inflection point is denoted the beginning of Stage 2 instead of

the blue point detected with criterion of  $i_{app}$ . The utility, thus far, of cumulative corrosion is detected there on the inflection point before the blue one, which allows taking preventive actions to control depassivation from that moment provided that values of  $i_{app}$  are still below the threshold most widely used (0.1–0.5  $\mu$ A/cm<sup>2</sup>).

Stage 3 (layer 1 of corrosion products). The new point of inflection denotes the beginning of a layer formation of corrosion products as it is limiting the access of oxygen and chloride. Although these and the consequent layers of corrosion products are unstable, the truth is that the decrease of intensity in the cathodic reaction controls the process during some time where the increase in corrosion is stable and constant (series in red; Figures 2–6, subsections (b) and (c)). This process continues until a new inflection point appears that denotes again a new availability of oxygen and chlorides that gives place to the dissolution of the layer 1 of corrosion products.

Stage 4 (nucleation and development of cracks). There are physical and chemical reasons that may influence the breakdown of that layer 1 of Stage 3, some of them being the ease or difficulty of corrosion products migration in the matrix of concrete due to its quality [31], the nucleation and development of cracks, and the effect of other external agents as humidity, temperature, and so forth. [6, 32]. The above reasons can give place to subsequent processes during Stage 3 denoted by new inflection points and other series of stable data. In these cases, an inflection point may suggest the beginning of a crack, produced by stress caused by corrosion products which triggered the cracking and, in other cases, new corrosion layers produced by the permeability of concrete.

 $i_{\rm CCR}$  has been used recently by some authors [10–13, 15]. Previous findings from Andrade related to the utility of this tool were confirmed here with data from a natural marine environment.

One of the benefits of the cumulative corrosion technique is to offer a better definition among behaviors of variables with difficulty of discrimination. Traditionally, the w/c ratio is one of the variables most used to easily compare physical, chemical, and electrochemical behavior among concrete mixtures. However, it is uncommon to find comparisons among different variables such as CT or others. Therefore, in this manuscript, the CT was chosen as a variable of difficult discrimination, which under conditions of apparent corrosion rate does not usually show clear differences in behavior. By using the  $i_{CCR}$  tool, the effect of CT on the depassivation of the reinforcement can be observed (Figures 2–6).

On the other hand, the corrosion rate data in any of its representations, including the cumulative corrosion, are quantitative and kinetic, being thus clear and precise. The influence on them of thermodynamic parameters such as temperature is then implicit. This thermodynamic character, under representations such as the corrosion potential, is of a random nature [33]. From the above discussion,  $i_{CCR}$  can be considered an interesting tool to detect the four mentioned stages at least for the marine environment data reported.

#### 4. Conclusions

The conclusions reported here apply to the specific conditions of the natural environment of Yucatán as well as to the materials tested. Other conditions of exposure and materials could be effective to improve the meaning and discussion of these findings.

CCR tool allowed the detection of inflection points that denote the beginning of the reported stages:

- (a) Passivation
- (b) Depassivation
- (c) Formation and breakdown of corrosion layers
- (d) Nucleation and development of cracks.

Stable series of data maintained among the inflection points correspond to the detected physic-chemical behavior.

 $i_{\rm CCR}$  tool was able to show important differences among CT and w/c ratios at early stages. This means to take cheap preventive actions to avoid depassivation and/or cracks.

 $i_{\rm CCR}$  was also able to show the beginning of the depassivation period.

 $i_{\rm CCR}$  allows predicting/anticipating the depassivation thanks to the first inflection point.

More information is needed to perform predictions after depassivation.

## **Competing Interests**

The authors declare that there is no conflict of interests regarding the publication of this paper.

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