

## Research Article

# Influence of Hydrophobic Coating on Freeze-Thaw Cycle Resistance of Cement Mortar

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Due to the porous characteristics of cement-based materials, they are often corroded by salt solutions, which results in decreased durability, especially against damage under freeze-thaw cycles (FTCs). Improving surface properties is an effective way to improve the durability of these materials. In this study, a hydrophobic coating was applied to the surface of cement mortar by chemical modification of low surface energy materials. Fourier transform infrared spectroscopy (FT-IR) showed that low surface energy substances are linked to hydration products through chemical bonds. A water contact angle test indicates that the surface of cement mortar changed from hydrophilic ( $\theta=14^\circ$ ) to hydrophobic ( $\theta=140^\circ$ ) after chemical modification. The cumulative water uptake of hydrophobic samples decreased by 90%. Meanwhile, the wear resistance of the hydrophobic coatings was excellent. Compared with the baseline sample, mass loss rate, flexural strength, and compressive strength of hydrophobic coating samples increased several-fold in the FTC test. Microstructural changes of the mortar were characterized by scanning electron microscopy. The results show that a hydrophobic coating can significantly improve the freeze-thaw resistance of cement-based materials. The formation of a hydrophobic layer on the surface of cement-based materials can improve their durability. The research results not only have applications in civil engineering but will also have great impact in the restoration of historic structures.

## 1. Introduction

Concrete is a type of building material based on cement with highly effective mechanical properties; it is widely used as a structural material for buildings, bridges, undersea tunnels, etc. In ancient times, cementitious materials made from calcium hydroxide and clay were often used to build what have become today's world-famous historical buildings, such as the Pantheon in Rome. However, both modern and historical buildings are usually corroded by salt solutions, by which water penetrates the concrete, which is a factor contributing to concrete degradation. The freeze-thaw cycle (FTC) in severely frozen regions will cause sustained damage to concrete due to osmotic pressure, water expulsion, and in-pore crystallization during the FTC process [1–6]. Researchers have proposed many methods to improve the FTC resistance of cement-based materials, such as adding air-

entraining substances [7–10], pozzolanic minerals, or fiber admixtures [11–24]. The first method can relieve crystallization pressure in the FTC, while the latter method can improve the compactness of the concrete. However, the aforementioned methods lead to negative impacts on the concrete, such as deteriorating mechanical properties, difficult workability, and increased drying shrinkage.

Super-hydrophobic phenomena exist widely in nature [25–32]. Previous studies have pointed out that two basic requirements must be met for the surface of a solid material to be super-hydrophobic: (1) microscale and nanoscale rough structures and (2) lower surface free energy. Researchers have accurately expressed this theory through the Wenzel model [33] and Cassie–Baxter model [34].

Based on the above research, super-hydrophobic coatings have been applied to concrete surfaces for waterproofing, deicing, and self-cleaning [35–40]. Super-

hydrophobic coatings can be prepared by bonding low surface energy materials to the concrete surface. Materials such as polytetrafluoroethylene (PTFE), polyether ether ketone (PEEK), and silanized diatomaceous earth (DE) are bonded to the concrete surface by epoxy resin to obtain a super-hydrophobic surface [41]. Besides, super-hydrophobic surfaces can also be obtained by bonding super-hydrophobic rice husk ash [42], paper sludge ash [43], or nanosilica gel [34] to the concrete surface. Another way to obtain super-hydrophobic surfaces is the template method, where the features of micropillared molds made of polydimethylsiloxane (PDMS) are replicated immediately after demolding, and then siloxane-based compounds are sprayed to form a low energy surface [44].

Due to the excellent waterproofing effect of super-hydrophobic coatings, the water absorption of concrete decreases significantly, but the durability of such coatings is insufficient and they can easily fall off. Up to now [37], there has been a lack of research on the mechanical stability of super-hydrophobic coatings; therefore, the application of super-hydrophobic coatings in engineering is limited. To solve this problem, a vacuum impregnation process was adopted in this study. Such technology is more suitable for performance improvement of prefabricated concrete structures, similar to the anticorrosion treatment of steel structures. Through this technology, low surface energy materials (iso-octyltriethoxysilane) can penetrate the cement mortar and combine with the cement hydration products, such as calcium hydroxide and ettringite, to form a continuous self-assembled molecular film layer. This molecular film layer reduces the surface energy of the mortar, thus achieving chemical modification of the rough surface of the mortar to form a hydrophobic coating. The wetting property was characterized by a water contact angle (WCA) test. A water absorption test and FTC resistance test were used to evaluate the protective effect of the hydrophobic coating on mortar blocks. Building structures are often subjected to external forces, which may result in surface wear and tear. Thus, the wear resistance was tested by sandpaper polishing under a certain pressure, after which the change of water absorption was tested. The microstructure of the cement mortar was characterized by scanning electron microscopy (SEM). Interface chemical reactions were characterized by Fourier transform infrared spectroscopy (FT-IR).

## 2. Materials and Methods

**2.1. Materials.** Ordinary Portland cement (OPC) was used as the binding material in all mortar specimens. The chemical composition of the OPC is shown in Table 1. The aggregates were acquired from Xiamen ISO Standard Sand Co., Ltd., with particle diameters ranging from 0.5 to 2.0 mm. Iso-octyltriethoxysilane was acquired from Wacker Chemicals. Tap water was used in the preparation of mortar samples. Anhydrous ethanol was acquired from Cormio Inc, China. To ensure permeability, iso-octyltriethoxysilane was used as a low surface energy material for surface treatment with a concentration of 2%, with the remainder composed of ethanol (28%) and water (70%).

**2.2. Preparation.** The proportions used to prepare the mortar samples and their properties after 28 days of curing are shown in Table 2. To ensure the uniformity of all mortar blocks, the following mixing process was adopted: (1) 450 g of cement and 225 g of water were added to the mixer and stirred for 60 s at slow speed; (2) 1350 g of sand was added evenly for 30 s; (3) the mixture was stirred at high speed for 30 s; (4) the mixer was stopped for 90 s for manual mixing; and (5) the mixture was further stirred for 60 s at high speed. After the concrete mixture was mixed, it was horizontally poured into a cuboid mold (40 mm × 40 mm × 160 mm) and a cubic mold (40 mm × 40 mm × 40 mm). After molding, all specimens were wet cured for 24 h (RH = 100% and  $T = 21 \pm 1^\circ\text{C}$ ) and then demolded and cured for 28 days in water ( $21 \pm 1^\circ\text{C}$ ). The cuboid specimen was used for the FTC test, and the cubic specimen was used for water absorption and wear resistance tests.

To ensure the effective penetration of modifiers, the hydrophobic coating on the surface of the mortar specimen was prepared using an osmosis vacuum degassing device (Figure 1). After curing for 28 days, the sample was dried to a constant weight and placed in the vacuum tank. When the vacuum was below 20 kPa, the iso-octyltriethoxysilane solution was slowly infused into the vacuum tank until it inundated the entire sample. Then, the hydrophobic coating was applied to the mortar surface after drying at  $60^\circ\text{C}$  for 12 h.

**2.3. Test Methods.** The WCA was measured using a contact angle tester (KRUSS, K100, Germany). It was determined using deionized water (2.5  $\mu\text{L}$  deposited with a micropipette) and by calculating the average of three measured values on the surface.

The capillary water absorption test was used to quantify the ability of the concrete to absorb water by capillary suction. To evaluate the hydrophobicity, the baseline samples and hydrophobic samples were placed in water under atmospheric pressure (101 kPa) and vacuum (20 kPa) for two days each. The weights of the samples were recorded before and after immersion to calculate the mass uptake of water. An MPD-2 metallographic polisher (Shanghai Zhongyan Instrument Co., Ltd., China) was used to determine the wear resistance of the mortar. As the polisher rotates, the sandpaper begins to slide and creates friction with the surface of the sample under pressure, such that the microstructure of the sample surface will be destroyed. After polishing, the mass loss rate was calculated and the mass uptake of water was measured. 240-grit sandpaper was fixed on the turntable of the polisher, and the turntable rotated at 500 rpm. The longer the polishing time, the higher the wear degree of the test block. In this study, the lengths of polished mortar blocks indicate the degree of wear.

The KDR-A rapid freeze-thaw circulator (Beijing Kangluda Test Instrument Co., Ltd., China) was used to determine the FTC resistance (Figure 2). The sample was immersed in water for two days and then put into the rubber sleeve of the circulator, which was filled with water. The temperature cycle consisted of freezing and heating stages

TABLE 1: Chemical composition of ordinary Portland cement (P. O42.5R).

	Chemical composition (%)									
	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	CaO	Fe <sub>2</sub> O <sub>3</sub>	MgO	K <sub>2</sub> O	TiO <sub>2</sub>	Na <sub>2</sub> O	SO <sub>3</sub>	Others
Cement	19.26	4.33	65.46	3.06	1.60	0.77	—	0.13	4.45	0.94

TABLE 2: Mix proportions and mechanical properties of the mortar.

Water-cement ratio	Raw material (g)			Density (kg/m <sup>3</sup> )	Mechanical parameter at 28 d (MPa)	
	Cement	Water	Sand		Compressive strength	Flexural strength
0.5	450	225	1350	2300	52.4	12.8

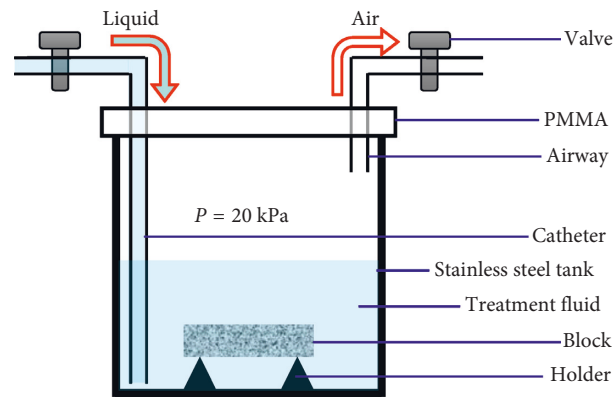


FIGURE 1: Schematic of osmosis vacuum degassing device.

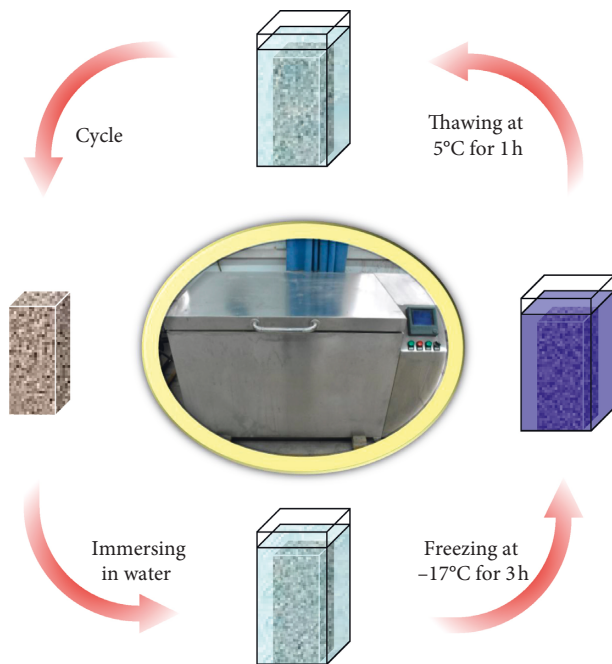


FIGURE 2: Schematic of the freeze-thaw cycle (FTC) test.

and took approximately four hours in total. During the freezing stage, the water temperature dropped from 5°C to -17°C after 2.5 h. During the melting stage, the water temperature rose from -17°C to 5°C after 1.5 h. With an

increase in the FTCs, the damage degree of the samples increased. After FTC damage, the mass loss rates and flexural and compressive strengths of the samples were measured to evaluate their degrees of damage.

The microscopic morphologies of the samples were observed by SEM (MAIA3, TESCAN, Czech Republic). Fourier transform infrared spectroscopy (FT-IR) spectra were acquired in the range of 400–4000 cm<sup>-1</sup> with an IR spectrophotometer (380FTIR, Thermo Fisher Scientific, America). The flexural and compressive strengths of the mortar were evaluated on the same testing machine (SANS CMT5105, Shenzhen, China) at a loading rate of 2400 ± 200 N/s.

### 3. Results and Discussion

**3.1. Wetting Properties and Water Contact Angle.** Without changing the surface microstructures of the mortar specimens, the surface of the mortar shows good hydrophobicity only through iso-octyltriethoxysilane solution modification, as shown in Figure 3(b). This phenomenon can be explained by the Wenzel theory that a hydrophobic surface can be obtained by modifying the rough mortar sample with low surface energy materials. First, iso-octyltriethoxysilane is hydrated to produce silanols (Si-OH). Secondly, silanol is combined with quartz sand, hydrated C-S-H gel, ettringite, and calcium hydroxide through -OH group reactions. Finally, the two -OH groups of iso-octyltriethoxysilane form Si-O-Si bonds by condensation while

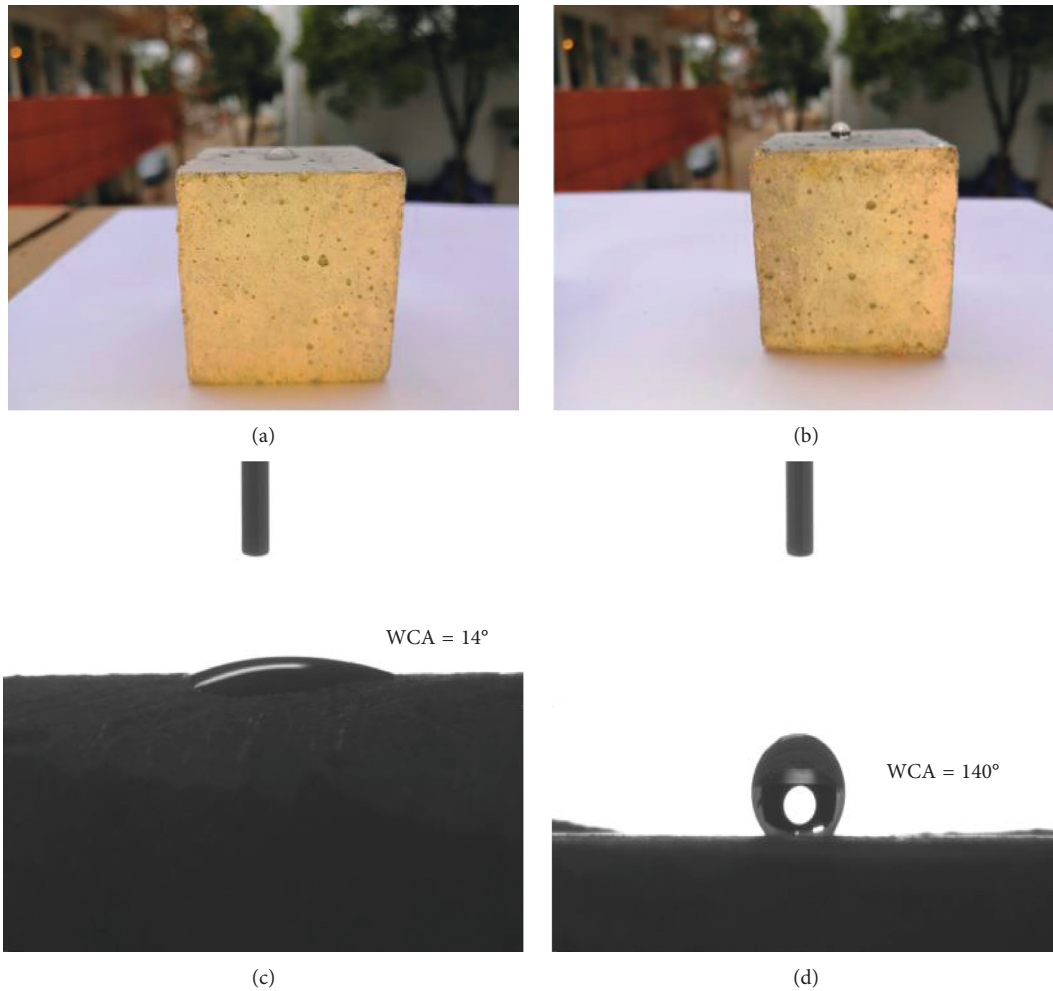


FIGURE 3: Water contact angles on the surface of mortar specimens: (a) baseline sample, (b) hydrophobic sample, (c) WCA of the baseline sample, and (d) WCA of the hydrophobic sample.

releasing water. After the above reaction, iso-octyltriethoxysilane forms a continuous self-assembled molecular film on the surface of the hydrated products. Iso-octyltriethoxysilane contains a  $-CH_3$  group and  $-CH_2$  group which effectively reduce the surface energy of the cement mortar. Therefore, the rough surface structure modified by low surface energy materials exhibits excellent hydrophobic properties.

Figure 3(c) shows that the contact angle of the baseline sample is approximately  $14^\circ$ , indicating that the porous, rough surface of the mortar belongs to the hydrophilic surface. Figure 3(d) shows that the WCA of the modified mortar surface increases to  $140^\circ$ , which proves that a hydrophobic surface can be obtained by modifying the rough hydrophilic surface with low surface energy materials. The WCA of the modified mortar surface did not reach a super-hydrophobic state ( $\theta > 150$ ) because the surface roughness of the mortar itself did not conform to the Cassie–Baxter model. The fragile micro/nanostructure is not conducive to the wear resistance of the coating itself. Therefore, the obtained WCA ( $\theta = 140^\circ$ ) is sufficient to improve the waterproof performance of the mortar.

**3.2. Water Absorption.** The influence of the hydrophobic coating on the water absorption of the mortar samples is shown in Figure 4. The results show that the cumulative water uptake of the baseline sample increased gradually from the beginning to achieve equilibrium and remained at a stable level thereafter, while the water absorption of the hydrophobic sample remained at a low level. After 15 days of immersion, the cumulative water uptake of the hydrophobic samples was reduced by 90%. This excellent waterproofing effect is equivalent to the waterproofing effect of a nano-composite waterproof coating [45].

The microscale rough structure of the mortar surface is modified by the low surface energy material to reach a Wenzel state, which shows an excellent waterproofing effect. The unmodified samples retain the hydrophilic properties of the cement-based materials.

**3.3. Wear Resistance and Thickness of the Hydrophobic Coating.** Fragile micro/nanostructures on hydrophobic surfaces are susceptible to damage, leading to degradation of the hydrophobic properties. In this research, to test the wear resistance of the hydrophobic coating, the mortar sample



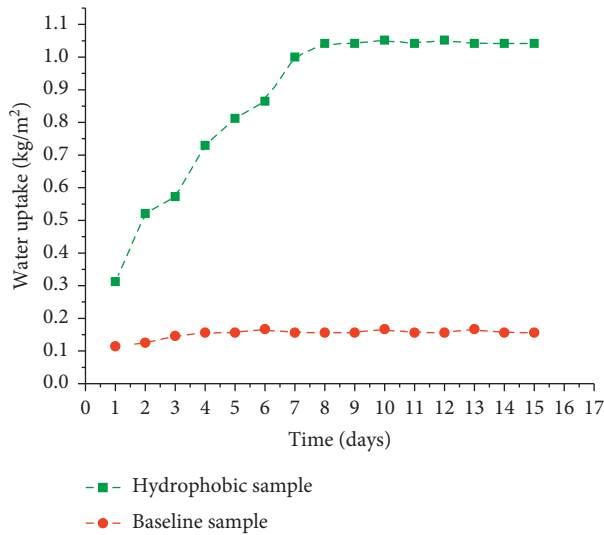


FIGURE 4: Water absorption results for the baseline and hydrophobic samples.

was placed on a burnisher and polished with 240-grit sandpaper at 500 rpm, and then the water absorption was tested. The lengths of the mortar blocks after polishing indicate their degrees of wear. Table 3 lists the length reductions corresponding to different polishing times.

As shown in Figure 5(c), after polishing, the WCA ( $\theta = 77^\circ$ ) of sample 5 decreased but remained between that of the baseline sample as shown in Figure 5(a) and that of the unpolished hydrophobic sample as shown in Figure 5(b). Because the thickness of the chemically modified mortar reaches the range of 1–3 mm, the hydrophobic sample remains hydrophobic even after the rough surface structure is destroyed. In the next section, the mass loss and water absorption after polishing are discussed. Figure 6(f) shows that the mass loss rate of the samples increases with an increase in length reduction. The mass loss rate of the sample is the largest, and its WCA ( $\theta = 77^\circ$ ) is still significant. Samples 1, 3, and 5 were immersed in water for 15 days with the baseline sample (Figure 6(a)) and the unpolished hydrophobic sample (Figure 6(b)). When the samples were immersed in water, many bubbles were observed on the surface of the baseline sample (Figure 6(a)). In contrast, we could see only a few bubbles on the surface of the hydrophobic sample (Figure 6(b)), even if it was reduced by over 10 mm by the sandpaper (Figure 6(e)). Surface bubbles are formed when water enters the sample and displaces air from the sample. A few bubbles on the surfaces of the polished samples show that the hydrophobic coating maintains an excellent waterproofing performance even after polishing. In the cumulative water uptake test, it was also proved that the hydrophobic coatings have excellent wear resistance, as shown in Figure 7.

In Figure 7, the water absorption curve shows that the cumulative water uptake of the polished hydrophobic samples remained at a low level. This phenomenon shows that the wear resistance of the hydrophobic coating is outstanding. The surface of the hydrophobic coating is worn after polishing, which leads to decreased WCA, but it still

maintains excellent waterproof properties. This phenomenon can be explained by Figure 8.

Figure 8 illustrates the thickness of the hydrophobic coating by wetting the cross section with water. In this figure, the light hydrophobic coating can be observed continuously around the perimeter of the dark central area, which indicates that a continuous hydrophobic coating was formed on the surface of the sample by vacuum impregnation. The thickness of the hydrophobic coating is within the range of 1–3 mm. This is a reasonable explanation for the hydrophobic properties of the polished hydrophobic samples decreasing after polishing but the waterproof properties remaining excellent. When iso-octyltriethoxysilane penetrates the mortar sample, self-assembled membranes will form on the surface of the hydrated particles. As the interior of the mortar is rough and porous, a stable hydrophobic network structure with a certain thickness is obtained. In this way, even if the surface of the microscale rough structure is destroyed, the network structure can still play a perfect waterproofing role.

**3.4. FTC Resistance Analysis.** As shown in Figure 9, steep curves for the mass loss rate and flexural and compressive strengths were observed for the baseline sample after the FTC tests. On the contrary, the corresponding curves for the hydrophobic samples change more smoothly. This phenomenon shows that the baseline sample was severely damaged after the FTCs, while the hydrophobic sample was much less damaged due to the protection of its hydrophobic coating. After 36 FTC tests, the mass loss rate of the baseline sample was approximately 48.0 wt.%. Meanwhile, the flexural and compressive strengths of the baseline sample were reduced to 0.3 MPa and 11.0 MPa, respectively. After 36 FTC tests, the mass loss rate and flexural and compressive strengths of the hydrophobic sample were 0.8 wt.%, 7.5 MPa, and 38.2 MPa, respectively. After 48 FTC tests, the baseline sample lost its original morphology and size (Figure 10) because its mass loss rate was over 62 wt.%. In contrast, the original morphology and size of the hydrophobic sample remained after 72 FTC tests (Figure 10). The test results show that the mass loss rate and flexural and compressive strengths of the hydrophobic sample are 10.4 wt.%, 1.0 MPa, and 16.5 MPa, respectively, which are very close to the values of the baseline sample after 24 FTCs. The test results indicate that the hydrophobic coating not only has an excellent waterproofing effect but it also has excellent anti-FTC performance. Due to the protective effect of the hydrophobic coating, water cannot impregnate the sample, thus alleviating damage from FTCs. Exfoliation of the surface coating of the hydrophobic sample occurred at the 36<sup>th</sup> cycle. This is because the sample had been immersed in water at below  $0^\circ\text{C}$ . The sample was encapsulated by external ice, which produced certain stresses and destroyed the hydrophobic coating. It is predicted that increasing the hydrophobic coating thickness will effectively improve the FTC resistance of the samples. This will be further explored in future research. The improvement of frost resistance was also observed in concrete modified with metakaolin and

TABLE 3: Length reduction of the modified sample.

Sample number	Radius of turntable (mm)	Burnisher revolutions per minute	Polish time (min)	Polish distance (km)
1			0.5	0.3
2			4.5	2.8
3	100	500	8.5	5.4
4			12.5	7.9
5			16.6	10.4

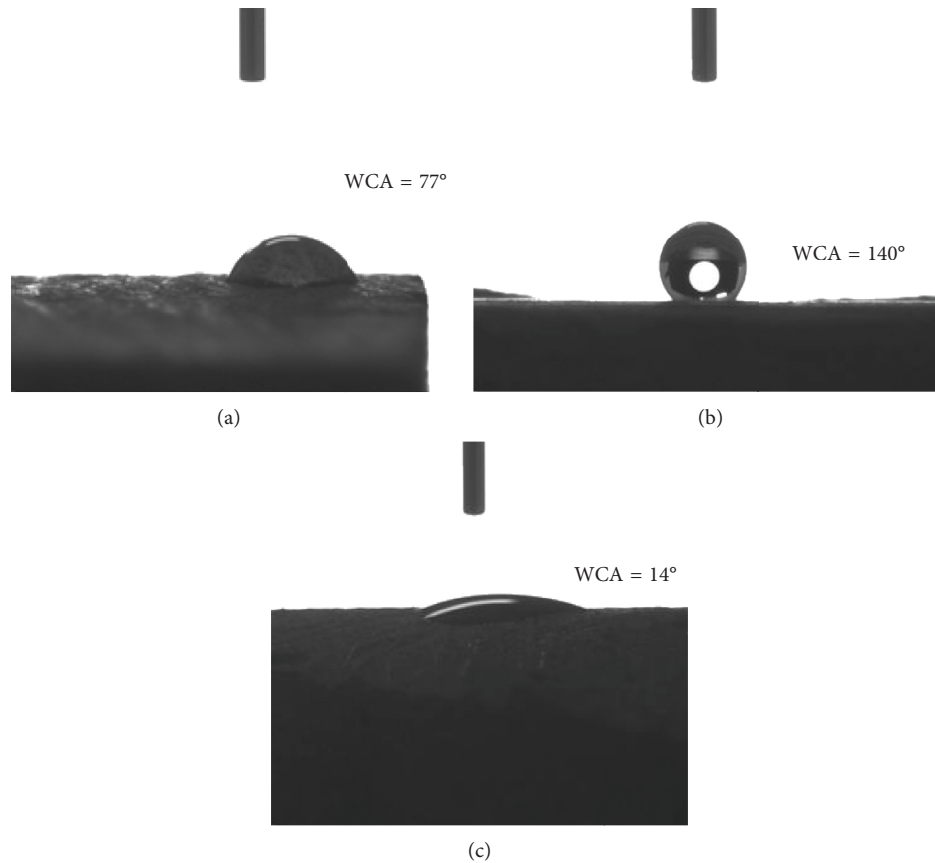


FIGURE 5: WCA of the (a) baseline sample, (b) hydrophobic sample, and (c) polished hydrophobic sample.

nanoparticles [46]. This is an entirely different technical concept from the hydrophobic coating, but it can be combined to achieve better frost resistance in future research.

### 3.5. Microscopic Analysis and Chemical Characterization.

To study the effect of the FTC on the internal structure of the mortar, the sectional microstructures of the hydrophobic and the samples were compared after enduring FTC damage (Figure 11). As shown in Figure 11(a), after the 12<sup>th</sup> FTC test, a visible crack could be seen on the baseline sample. Moreover, the crack was further extended due to the continuous damage from FTC after the 36<sup>th</sup> FTC test. The width of the crack increased from 2.1  $\mu\text{m}$  to 4.3  $\mu\text{m}$  (Figure 12(b)). Significantly, only cracks less than 1  $\mu\text{m}$  (Figure 11(c)) could be observed on the hydrophobic sample after the 12<sup>th</sup> FTC test. After the 48<sup>th</sup>

FTC test, a crack becomes evident with a width of approximately 1  $\mu\text{m}$  (Figure 11(d)). As exhibited in Figure 11(e), the width of the crack is still below 4  $\mu\text{m}$  after the 72<sup>nd</sup> FTC test. Figure 11(f) displays the microstructure of the hydrophobic coating on the mortar block after enduring 72 FTCs, indicating that the hydrophobic coating could effectively reduce the damage to the mortar caused by FTCs. Based on this, we can confidently predict that the FTC resistance of the mortar block would be improved by increasing the thickness of the hydrophobic coating.

As shown in Figure 12(a), the hydrophobic coating appeared after the sample section was wet with water. In Figure 12(b), by magnifying by 20,000 times, the SEM images show that the interior of the mortar is filled with acicular or flaky hydration products. This micron-scale rough structure is one of the conditions for the formation of the hydrophobic coating. The chemical modification of the

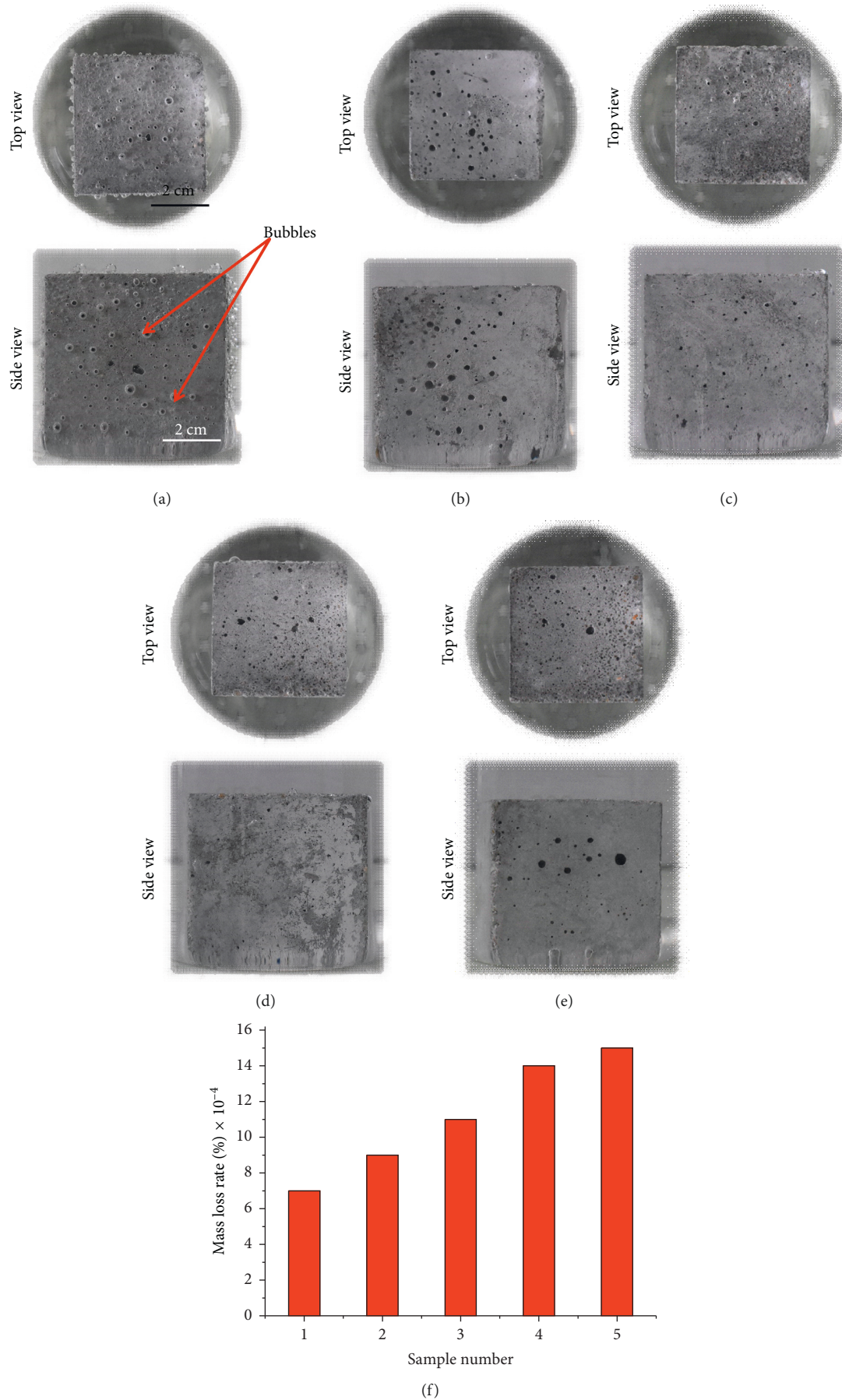


FIGURE 6: Photograph of the mortar block immersed in water: (a) contrast sample and (b) hydrophobic sample and after abrasion with 240-grit sandpaper for (c) 0.3 mm, (d) 5.3 mm, and (e) 10.4 km. (f) Mass loss rate after polishing.

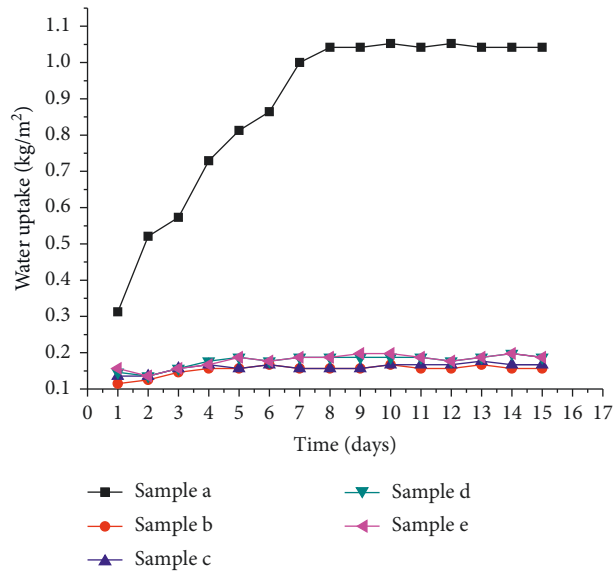


FIGURE 7: Effect of length reduction on water absorption of the hydrophobic sample.

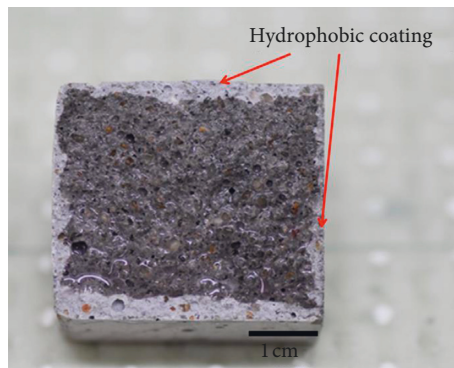


FIGURE 8: Photograph of the hydrophobic coating's thickness.

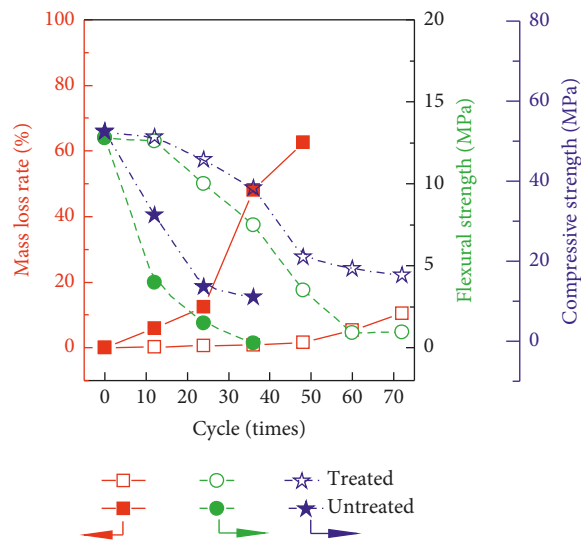


FIGURE 9: Mass loss rate (■ and □) and flexural (● and ○) and compressive (★ and ☆) strengths of the hydrophobic sample (□, ○, and ☆) and baseline sample (■, ●, and ★) measured after FTC tests.



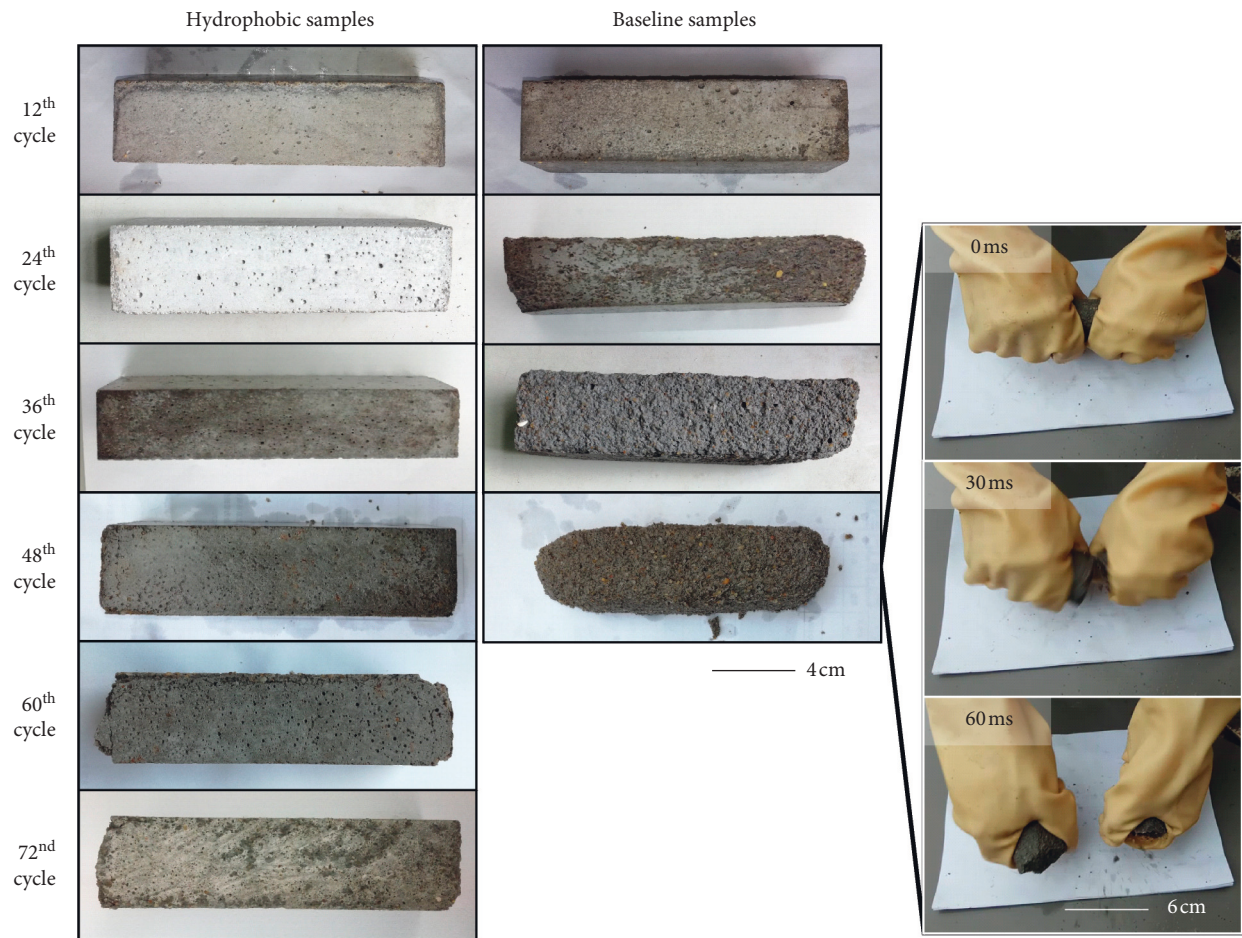


FIGURE 10: Photographs of the sample after different FTC tests.

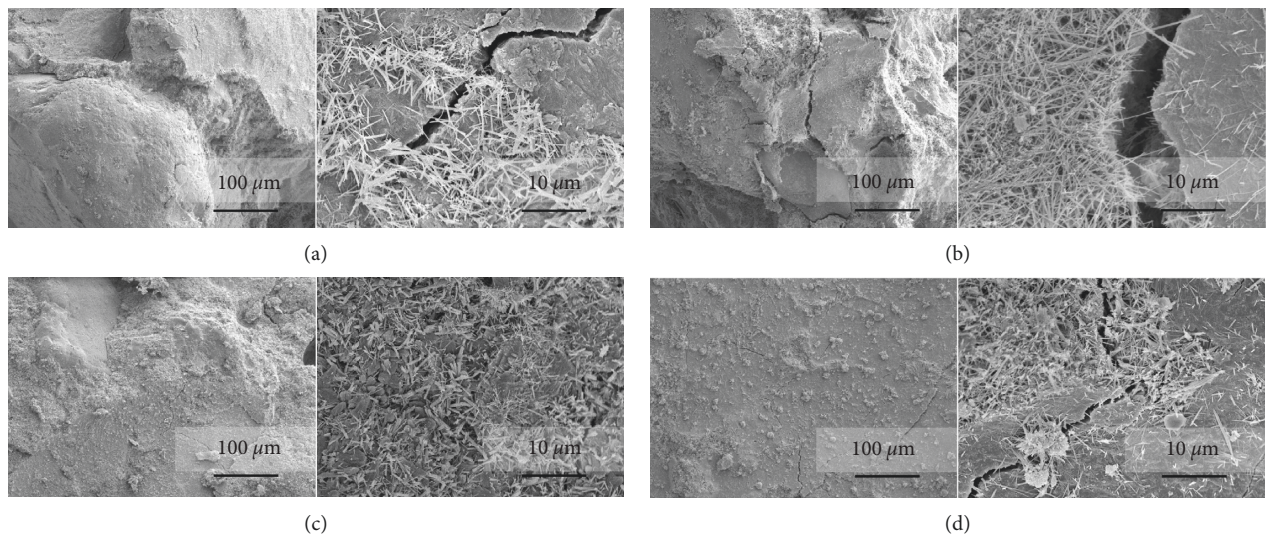


FIGURE 11: Continued.

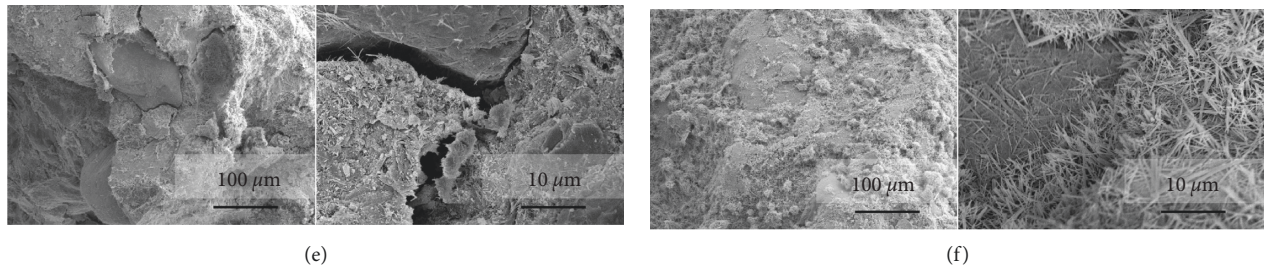


FIGURE 11: Micrographs of the mortar sample. Sectional microstructure of the baseline sample at the (a) 12<sup>th</sup> and (b) 36<sup>th</sup> FTCs. Sectional microstructure of the hydrophobic sample at the (c) 12<sup>th</sup>, (d) 36<sup>th</sup>, and (e) 72<sup>nd</sup> FTCs. (f) Microstructure of the hydrophobic coating on the hydrophobic sample at the 72<sup>nd</sup> FTC.

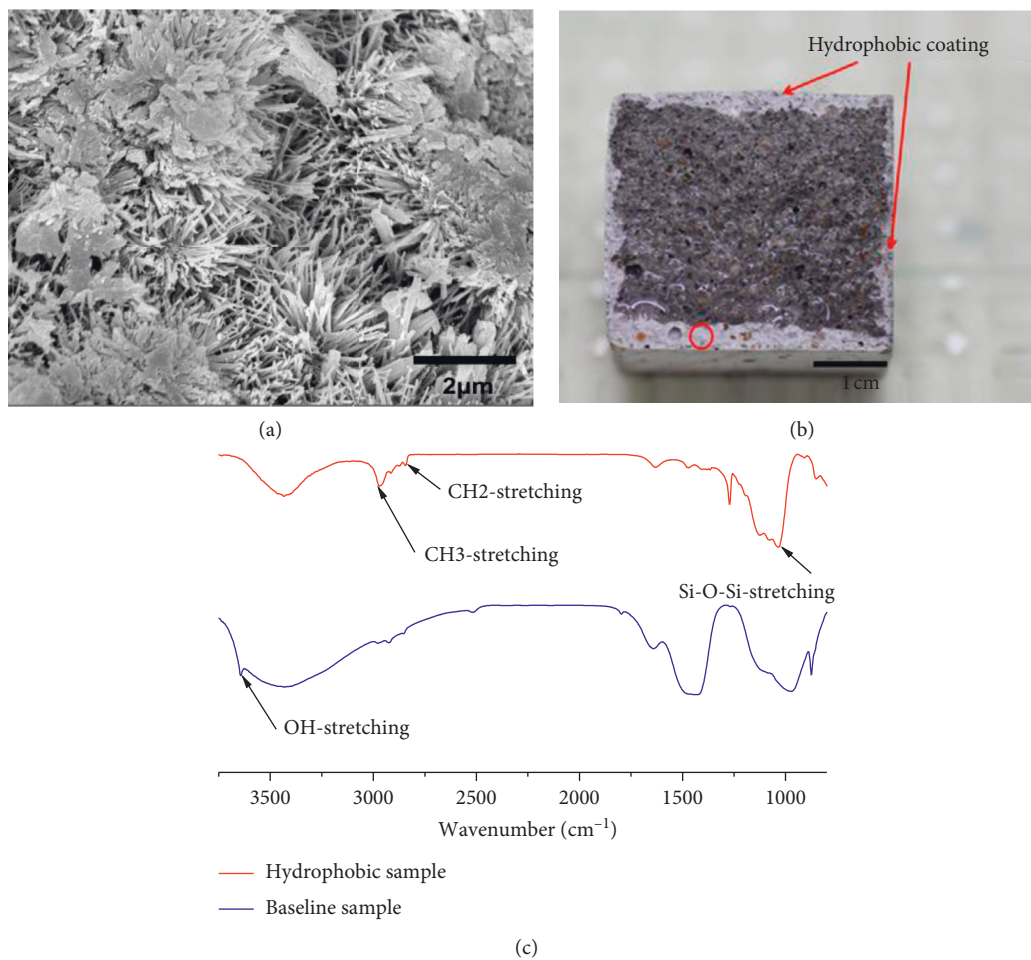


FIGURE 12: Microscopic analysis and chemical characterization of the hydrophobic coating: (a) hydrophobic sample, (b) SEM photograph of the hydrophobic coating, and (c) FT-IR spectra of the sample.

hydration products was characterized by the FT-IR. As shown in Figure 12(c), the FT-IR wavenumbers ranged from  $3750\text{ cm}^{-1}$  to  $1000\text{ cm}^{-1}$ . The absorption peak at  $3644\text{ cm}^{-1}$  was attributed to the stretching vibration of  $-\text{OH}$  from  $\text{Ca}(\text{OH})_2$ . The  $-\text{OH}$  stretching vibration peaks of  $\text{Ca}(\text{OH})_2$  were only observed in the baseline sample. This indicates that hydroxyl groups are consumed in the reaction of calcium hydroxide with iso-octyltriethoxysilane. Absorption peaks at

$2970\text{ cm}^{-1}$  and  $2920\text{ cm}^{-1}$  were observed in the hydrophobic coating, corresponding to  $-\text{CH}_3$  and  $-\text{CH}_2$  groups, respectively, implying that chemical bonds are formed between the coating and cement hydration products. The hydrophobic coating peak at  $1130\text{ cm}^{-1}$  corresponds to the Si-O-Si group, which shows that continuous self-assembled molecular films are formed on the surface of hydration products.



## 4. Conclusions

This study aimed to transform a porous, hydrophilic cement mortar surface into a hydrophobic surface by chemical modification. Through hydrolysis and condensation, iso-octyltriethoxysilane forms continuous self-assembled molecular films on the surface of hydrated products, thus producing a hydrophobic coating with a thickness of 1–3 mm on the surface of the mortar. The WCA of the hydrophobic coating was 140°, and it had good waterproofing and wear resistance. A water absorption test showed that the cumulative water uptake of the hydrophobic samples decreased by 90%. Compared with a baseline sample, the mass loss rate and flexural and compressive strengths of the hydrophobic sample increased several-fold in the FTC test stages. Chemical bonding between iso-octyltriethoxysilane and cement hydration products ensures excellent wear resistance of the hydrophobic coating. In conclusion, the hydrophobic coating prepared by vacuum impregnation has an excellent protective effect on cement-based materials, and this technology has a wide range of applications in the building industry. Future work will focus on the preparation of hydrophobic coatings with higher thicknesses and better waterproofing performances via a simpler process.

## Data Availability

All the data in this study are original.

## Conflicts of Interest

The authors declare that they have no conflicts of interest.

## Acknowledgments

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