

## Research Article

# Effect of the Side Chains and Anode Material on Thermal Stability and Performance of Bulk-Heterojunction Solar Cells Using DPP(TBFu)<sub>2</sub> Derivatives as Donor Materials

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An optimized fabrication of bulk-heterojunction solar cells (BHJ SCs) based on previously reported diketopyrrolopyrrole donor, ethyl-hexylated DPP(TBFu)<sub>2</sub>, as well as two new DPP(TBFu)<sub>2</sub> derivatives with ethyl-hexyl acetate and diethyl acetal solubilizing side-chains and PC<sub>60</sub>BM as an acceptor is demonstrated. Slow gradual annealing of the solar cell causing the effective donor-acceptor reorganization, and as a result higher power conversion efficiency (PCE), is described. By replacing a hole transporting layer PEDOT:PSS with MoO<sub>3</sub> we obtained higher PCE values as well as higher thermal stability of the anode contact interface. DPP(TBFu)<sub>2</sub> derivative containing ethyl-hexyl acetate solubilizing side-chains possessed the best as-cast self-assembly and high crystallinity. However, the presence of ethyl-hexyl acetate and diethyl acetal electrophilic side-chains stabilizes HOMO energy of isolated DPP(TBFu)<sub>2</sub> donors with respect to the ethyl-hexylated one, according to cyclic voltammetry.

## 1. Introduction

Currently polymer and small molecules (SMs) bulk-heterojunction (BHJ) organic solar cells (SCs) exceeding 11% power conversion efficiency (PCE) [1] and lifetimes above 1000 hours under continuous illumination [2, 3] attracted particular interest due to their flexibility, color tunability, and predominantly potentially low production cost. Considering SMs as a donor material, despite the recent successful results [4, 5], they still lag behind the polymer BHJ solar cells. However small-molecules BHJ SCs do not depend on a polymer weight distribution; thus they do not suffer from batch-to-batch differences [6]. One of the most promising structural classes of the donors in BHJ SCs is diketopyrrolopyrrole (DPP) derivatives, which show excellent photovoltaic performance as both a part of polymers [7–10] and dimers [11–13], because of a low band gap, enabling them to absorb a large portion of the solar spectrum, as well as an ability to be

further functionalized to match the parameters required for the photovoltaic applications. Among the DPP monomers, DPP(TBFu)<sub>2</sub> ethyl-hexyl alkylated derivative (compound (A) on Figure 1) showing high power-conversion efficiency (PCE) up to 4.4–4.8% with PC<sub>70</sub>BM [14, 15] and 4.0% with PC<sub>60</sub>BM [16] represents the most successful design. Applying a solvent annealing for the DPP(TBFu)<sub>2</sub>:PC<sub>70</sub>BM blend 5% of PCE can be exceeded [17], which is, up to now, the highest reported PCE among the monomeric DPP derivatives. Consequently, relatively easily (as compared to, e.g., bis-DPPs) obtainable DPP(TBFu)<sub>2</sub> is often used as a benchmark compound in the current research dealing with new SM donors [18–20], acceptors [21], or tandem cells [22]. The attempts to modify DPP(TBFu)<sub>2</sub> structure using different heteroatoms brought only lower PCE [15, 20]. DPP(TBFu)<sub>2</sub> belongs to SM donors, for which the relationships between the structure [15, 23, 24] and various aspects of the photovoltaic activity such as hole mobility [25] were understood to the most detailed extent.

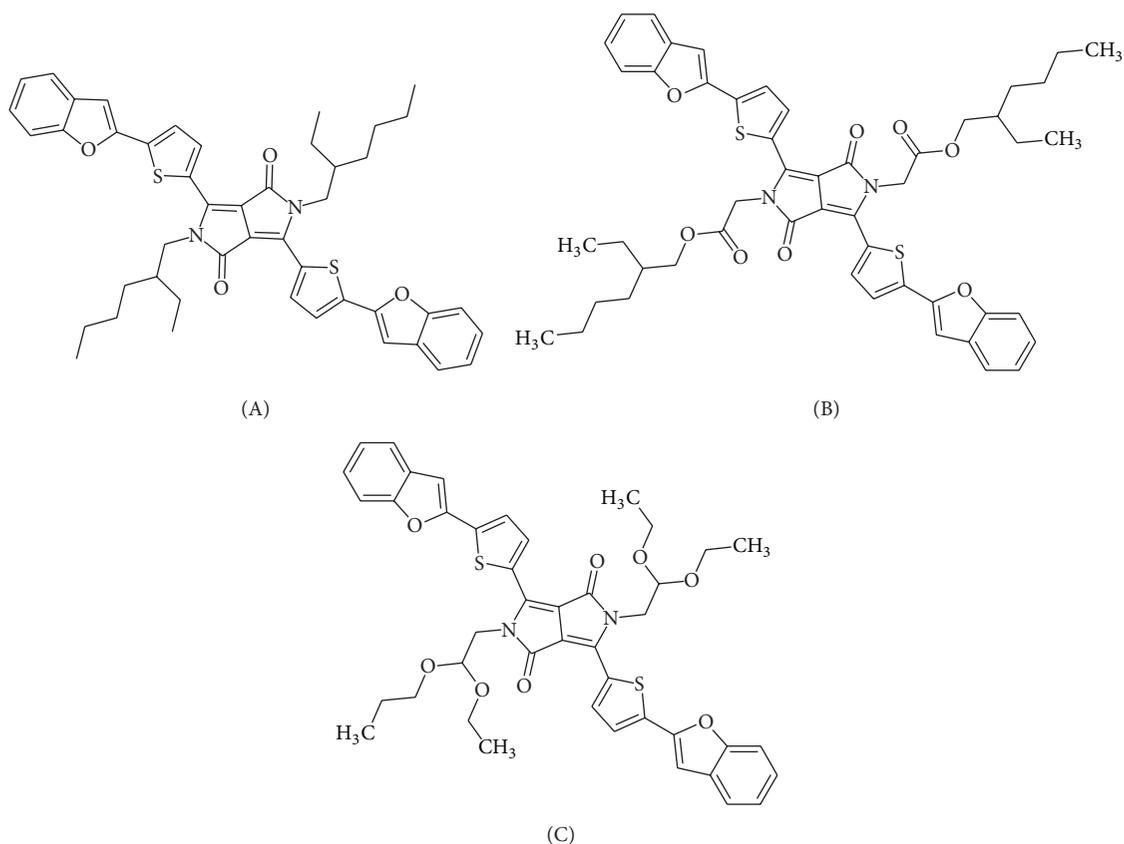


FIGURE 1: DPP(TBFu)<sub>2</sub> derivatives substituted with ethyl-hexyl (A), ethyl-hexyl acetate (B), and diethyl acetal (C) solubilization groups.

PCE of BHJ SCs using DPP(TBFu)<sub>2</sub> as a donor is known to be extremely sensitive to annealing at (constant) temperature [14]. Furthermore, to the best of our knowledge, PEDOT:PSS is the only hole-selective (transporting) contact (HSC) reported in the previous studies. Here we report a new method of the gradual annealing, causing the progressive reorganization of the donor-acceptor bulk-heterojunction, improving the solar cells performance. It is known that the MoO<sub>3</sub> used as HSC [26, 27] instead of PEDOT:PSS can improve in some cases PCE of BHJ SCs [28], so we studied its effect in detail, including the thermal stability of the contacts. Moreover, we have tested the new DPP(TBFu)<sub>2</sub> derivatives with ethyl-hexyl solubilization groups substituted by ethyl-hexyl acetate [19] and diethyl acetal [29] groups (Figure 1 compounds (B) and (C)). It was expected that the electron-accepting effect of these substituents could stabilize HOMO level [30] of a donor and thus improve the open-circuit voltage of the cell.

## 2. Experimental

**2.1. Materials.** The synthesis and analysis of DPP donors (A)–(C) are described in Supporting Info (SI).

PC<sub>60</sub>BM (Solenne, 99%), chloroform (Aldrich, 99.9%), MoO<sub>3</sub> (Aldrich, 99.98%), and Al (Aldrich, 99.99%) were used as received without further purification. Prior to dissolving,

materials for the active layer were kept in vacuum at 60°C overnight and then transferred to the nitrogen atmosphere. The solution for the active layer films was prepared from a DPP:PC<sub>60</sub>BM blend (1.5:1 by weight for all DPP(TBFU)<sub>2</sub> derivatives) in chloroform with a total concentration of 20 mg/mL following the sonication for 5 minutes until complete DPP dissolution. All manipulations were carried out in a glovebox under a nitrogen atmosphere unless otherwise stated.

**2.2. Device Fabrication.** For the regular device architecture glass substrates with patterned ITO were covered by PEDOT:PSS by spin coating at 5500 rpm for 60 seconds with consequent annealing for 10 minutes at 150°C in the air and then annealed in the inert nitrogen atmosphere for another 10 minutes at 150°C to remove residual moisture. As an alternative hole-selective contact 10 nm thick MoO<sub>3</sub> layers were thermally evaporated in the vacuum chamber on the pixelated glass/ITO substrates. DPP(TBFu)<sub>2</sub>:PC<sub>60</sub>BM heterojunction layer was created by dynamic spin coating of 25 μL of prepared solution at 2500 rpm for 40 seconds in nitrogen atmosphere. Thickness of all the layers measured by the profilometer was in a range of 100 nm. Consequently gradient thermal annealing was applied in the glovebox starting from 50°C and speed 1°C/minute up to the notable degradation of the cell. After every 10 degrees efficiencies

TABLE 1: The first half-wave ( $E_{1/2}$ ) oxidation (Ox1) and reduction (Red1) potentials obtained from rotating disk voltammetry recomputed [31] to frontier orbital energy levels. Absorption maxima  $\lambda_{\text{MAX}}$  of 0-0 and 0-1 vibronic subbands of the longest wavelength band (absolute maxima are in bold and relate to molar absorptivities  $\epsilon$ ).

Compound	$E_{1/2}$ (ox1) [V]	$E_{1/2}$ (red1) [V]	$E_{\text{HOMO}}$ [eV] <sup>a</sup>	$E_{\text{LUMO}}$ [eV] <sup>b</sup>	$E_{\text{HL}}$ [eV] <sup>c</sup>	$\lambda_{\text{MAX}}$ [nm]	$\epsilon$ [L/mol-cm]
A	0.91	-0.88	-5.26	-3.47	1.79	<b>628</b> , 584	65153
B	1.01	-0.86	-5.36	-3.49	1.87	<b>622</b> , 578	69754
C	0.95	-0.88	-5.30	-3.47	1.83	614, <b>578</b>	60386

<sup>a</sup> $E_{\text{HOMO}}$  [eV] =  $E_{1/2}$  (ox1) [V] + 4.35 [32].

<sup>b</sup> $E_{\text{LUMO}}$  [eV] =  $E_{1/2}$  (red1) [V] + 4.35 [32].

<sup>c</sup> $E_{\text{HL}}$  [eV] =  $E_{\text{HOMO}}$  -  $E_{\text{LUMO}}$  = electrochemical HOMO - LUMO gap.

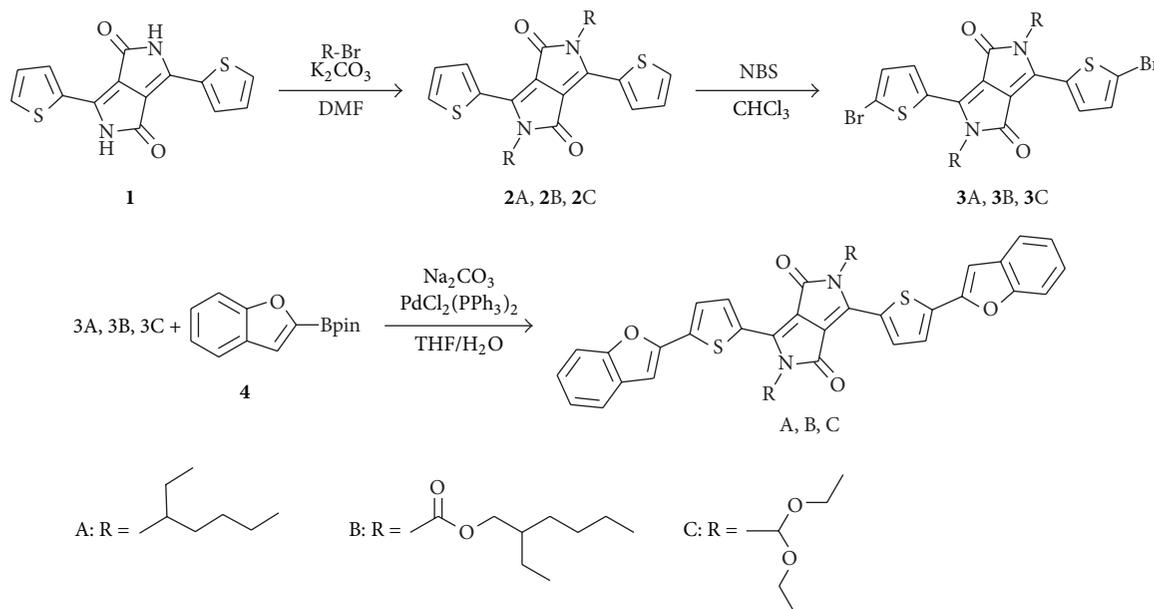


FIGURE 2: Reaction scheme.

of the solar cells were measured in ambient atmosphere at room temperature. Al top electrode was deposited by thermal vacuum deposition; active area mask was 6 mm<sup>2</sup>.

**2.3. Solar Cell Characterization Techniques.** Current density-voltage measurements were carried out under illumination using an Abet Sun 2000 solar simulator with an air mass (AM) 1.5 G filter. The simulated light intensity was adjusted to 1000 Wm<sup>-2</sup> by using a NREL-calibrated Si solar cell.

### 3. Results and Discussion

**3.1. Synthesis and Molecular Properties.** All three DPP derivatives (A)–(C) were synthesized according to scheme in Figure 2. While the yields of the bromination and cross-coupling reactions always exceeded 80% and 70%, respectively, the alkylation reactions proved to be more problematic. Alkylation of moderately soluble substrate **1** with 1-bromo-2-ethylhexane (or bromoacetaldehyde diethyl acetal) yielded half amount of the desired product compared to that obtained when 2-ethylhexyl 2-bromoacetate was used. This is mainly caused by a competitive *O*-alkylation [33] in the former case,

which was not observed in the latter one by TLC, as in previous studies [29, 32]. Furthermore, the latter reaction was finished at considerably shorter reaction time. Slight modification of the original Suzuki-Miyaura cross-coupling reaction [14] utilizing benzofuran pinacol boronic acid instead of the corresponding boronic acid and PdCl<sub>2</sub>(PPh<sub>3</sub>)<sub>2</sub> instead of Pd<sub>2</sub>(dba)<sub>2</sub> catalyst increased the yield of the last step from 67 to 81% for (A).

Spectral and electrochemical data are summarized in Table 1 and the absorption spectra in DCM are shown in Figure S1 (in Supplementary Material available online at <http://dx.doi.org/10.1155/2015/734917>). It is quite interesting that the HOMO and the LUMO levels of (A), derived from two electrochemical ionization reactions in solution (-5.26 eV and -3.47 eV), are only about 0.06–0.07 eV lower than the values recorded in the solid-state. The latter values were derived from a combination of the ionization reaction, estimated by photoelectron spectroscopy, and the absorption onset (-5.20 eV and -3.40 eV) [12, 14]. These two measurements gave almost identical gaps (1.79 eV versus 1.80 eV). On assumption of a similar solid-state arrangement of (A) and (B), the decrease of the HOMO from -5.26 eV to -5.36 eV should lead to an increase of the open-circuit voltage of BHJ

TABLE 2: Main parameters of the fabricated solar cells.

Donor material	Anode contact	Optimum annealing $T$ , °C	PCE, %	$V_{oc}$ , mV	$J_{sc}$ , mA/cm <sup>2</sup>	FF, %
(A)	PEDOT:PSS	110	3.75	865	9.39	46
(A)	MoO <sub>3</sub>	130	4.52	911	10.06	49
(B)	PEDOT:PSS	100*	2.04	823	5.11	48
(B)	MoO <sub>3</sub>	140	3.29	856	7.90	49
(C)	PEDOT:PSS	110	1.49	550	6.57	41
(C)	MoO <sub>3</sub>	140	2.28	797	6.59	44

\*100°C in case of constant temperature and up to 160°C in case of gradual annealing (PCE > 2.50%).

SC by about 0.1 eV as the  $V_{oc}$  depends primarily on the energy difference between the HOMO of the SM donor and the LUMO of the (fullerene) acceptor [34].

**3.2. Gradient Annealing Effect and Anode Contact Thermal Stability.** Firstly, the solar cells containing ethyl-hexyl alkylated sample (A) were fabricated using standard procedure reported in [14] with donor/acceptor ratio 60 : 40 and dynamically spin-coated at 2500 rpm for 60 seconds. PEDOT:PSS was used as an anode hole-selective contact and Al as a top cathode. Annealing at 110°C for 10 minutes was applied prior to Al deposition. After annealing PCE of the devices was in a range of  $3.6 \pm 0.2\%$ , which is comparable with previously published results [15, 20] when PC<sub>60</sub>BM was used as an acceptor. Fabrication of the BHJ SCs based on the samples (B) and (C) was optimized using various donor/acceptor concentrations (50 : 50, 60 : 40, and 70 : 30) and spin coating conditions; however similarities of the donor structures lead to the same preparation conditions as for the reference (A) sample. For the above mentioned series of devices annealing at 110°C for 10 minutes was applied. Best PCE of the SCs based on the materials (B) and (C) was obtained as 2.04% and 1.49%, respectively (Table 2), that is, lower than for (A) sample. Main losses were observed in the  $V_{oc}$  and especially in  $J_{sc}$ .

As an alternative anode material MoO<sub>3</sub> was used. Annealing of the devices for 10 minutes at constant temperature 110°C possessed similar PCE values in comparison with the devices with PEDOT:PSS. However using gradual increase of the annealing temperature it was discovered that above mentioned solar cells can not only sustain higher temperatures but also possess higher PCE values when temperatures up to 140°C were applied (Figure 3). On the contrary, for the PEDOT:PSS based devices annealing temperatures higher than 110°C resulted in the PCE decrease when direct annealing is applied (Figure S2), which relates to the previously reported results [14]. Gradient annealing for PEDOT:PSS based devices possesses better thermal stability at higher temperatures. In the case of compound (A) maximum efficiency reached at 80°C was preserved till 150°C with consequent degradation at higher temperatures. Devices based on compound (B) and PEDOT:PSS anode have shown the best thermal stability among all the PEDOT:PSS based devices; maximum efficiency was reached at 160°C with consequent degradation.

Interestingly, for the materials (A) and (C) during the gradient annealing maximum short-circuit current was achieved

at the temperature above 100°C; further PCE augmentation was determined by the  $V_{oc}$  and FF increase; notably the highest  $V_{oc}$  (954 mV) for the material (A) was measured prior to annealing; however after  $V_{oc}$  decrease up to 100°C one can observe its slight augmentation up to 911 mV, which is even exceeding maximum  $V_{oc}$  previously reported using PC<sub>70</sub>BM. In the case of donor (C) gradual  $V_{oc}$  augmentation is observed through the whole annealing process; however at temperatures above 140°C solar cells based on material (C) degraded rapidly.  $V_{oc}$  increase at the constant short-circuit current can be associated with the reorganization of the donor-acceptor stacking in the bulk heterojunction. In case of the material (B) maximum current can be observed even without annealing procedure, and no significant improvement in the  $V_{oc}$  was observed. In case of compound (C) the highest  $V_{oc}$  (800 mV) was observed at 150°C (752 mV at 110°C), in case of MoO<sub>3</sub> anode; however in case of PEDOT:PSS the highest  $V_{oc}$  is observed prior to annealing (~700 mV), with a decrease to 550 mV at 110°C and 400 mV at higher temperatures (Figure S3). Thus we assume that the anode interface is not thermally stable in case of compound (C) and PEDOT:PSS. In fact, among all the considered materials sample (B) showed the highest PCE at nonannealed samples (Figure 2(A)). Increase of the PCE during the annealing is observed due to the FF increase, which is influenced by the series resistance decrease. This observation can be related to the changes in contact/BHJ interface population during the annealing process. The substitution of widely used ethyl-hexyl with ethyl-hexyl acetate solubilization groups thus enables a formation of an appropriate bulk-heterojunction without the thermal annealing. Irrespective to the lower PCE, such behavior can be favorable if thermally unstable materials are used in a fabrication of SCs. This fact, together with above mentioned better reaction yields, makes the ethyl-hexyl acetate side-chain promising for further studies.

Analyzing the results of the above described gradient annealing process, we can observe that for the considered materials optimum annealing temperature lies between 130 and 140°C (Table 2), when MoO<sub>3</sub> is used as a hole-selective electrode. As it has been already mentioned devices were degrading, when PEDOT:PSS layer was used instead of MoO<sub>3</sub>. As far as PEDOT:PSS is known to be stable even at higher temperatures [35, 36], we associate this phenomenon with the donor-acceptor morphology and PEDOT:PSS/DPP(TBFu)<sub>2</sub> interface degradation, which can be circumstantially identified by slight increase of the series

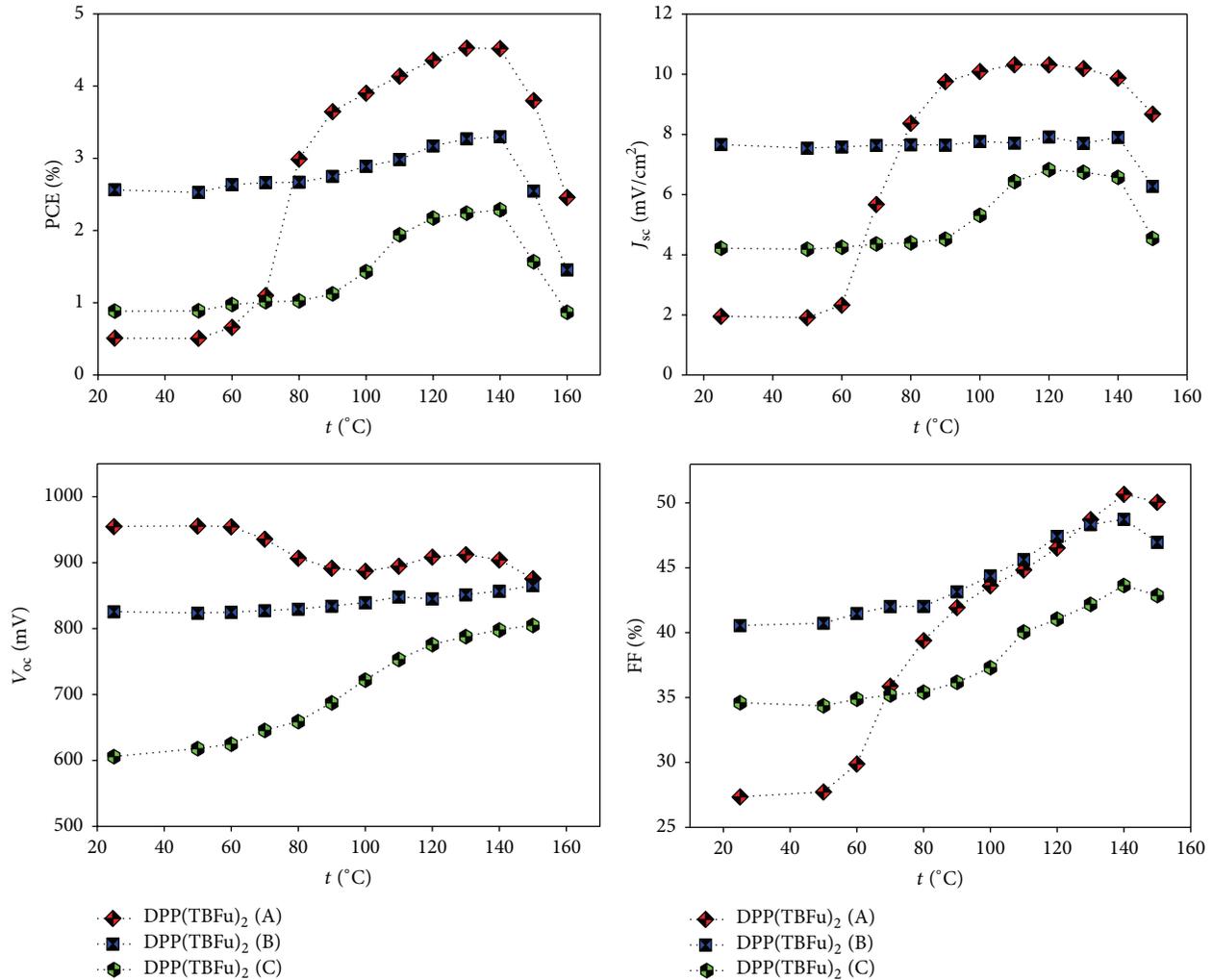


FIGURE 3: Main photovoltaic parameters of the gradually annealing of the devices with MoO<sub>3</sub> anode layer.

resistance when samples were annealed at temperatures above 110 °C.

AFM images depicted in Figure 4 (material (A)) and Figure S3 (material (B)) clearly show the changes in the crystallinity during the annealing in case of material A. The rms roughness has decreased from 2.6 nm to 0.9 nm. Larger crystalline size is observed when the layer was annealed at 140 °C (rms roughness 1.5 nm); however at 160 °C notable degradation of the morphology can be noticed; at the same time rms roughness has increased to 3.6 nm. In case of the material (B) high crystallinity is observed even at nonannealed samples and significant changes in the layer morphology during the annealing were not observed.

An evolution of the solid-state absorption spectra upon annealing (Figure 5) shows the trends corresponding to the ones detected through morphology changes. In accordance with previous results for (A) [14], the 0-0 vibronic band of a solid-state absorption dramatically decreases due to annealing and the vibronic structure becomes sharper. The former effect relates to the change of an excitonic coupling from joint J- and H-aggregation [37] to dominant H-type

[38], while the latter one implies the better ordered solid-state structure [39]. In the case of DPP generally [40] and compound (A) especially [15, 23] these changes relate to a shift of the  $\pi$ - $\pi$  stacked molecules along the long molecular axis towards the shorter center-to-center distance. As the solid-state absorption of compound (B) shows very similar shape even as-cast with no effect of annealing, we can conclude that an optimal donor phase arrangement is formed immediately during casting, improving thus a phase separation observed by AFM.

In summary we fully associate PCE decrease of the PEDOT:PSS containing (A) DPP(TBFu)<sub>2</sub>:PC<sub>60</sub>BM devices with the degradation of the PEDOT:PSS/BHJ interface when annealing at  $T > 110$  °C is applied. On the other hand at temperatures  $T > 160$  °C BHJ morphology degradation is the main issue. However material B possesses high crystallinity even at as-cast layers; PEDOT:PSS based devices have shown PCE augmentation up to 160 °C and morphology degradation was observed only at 170 °C. Therefore it is correct to suppose that the PEDOT:PSS/BHJ interface is more thermally stable in case of the material (B).

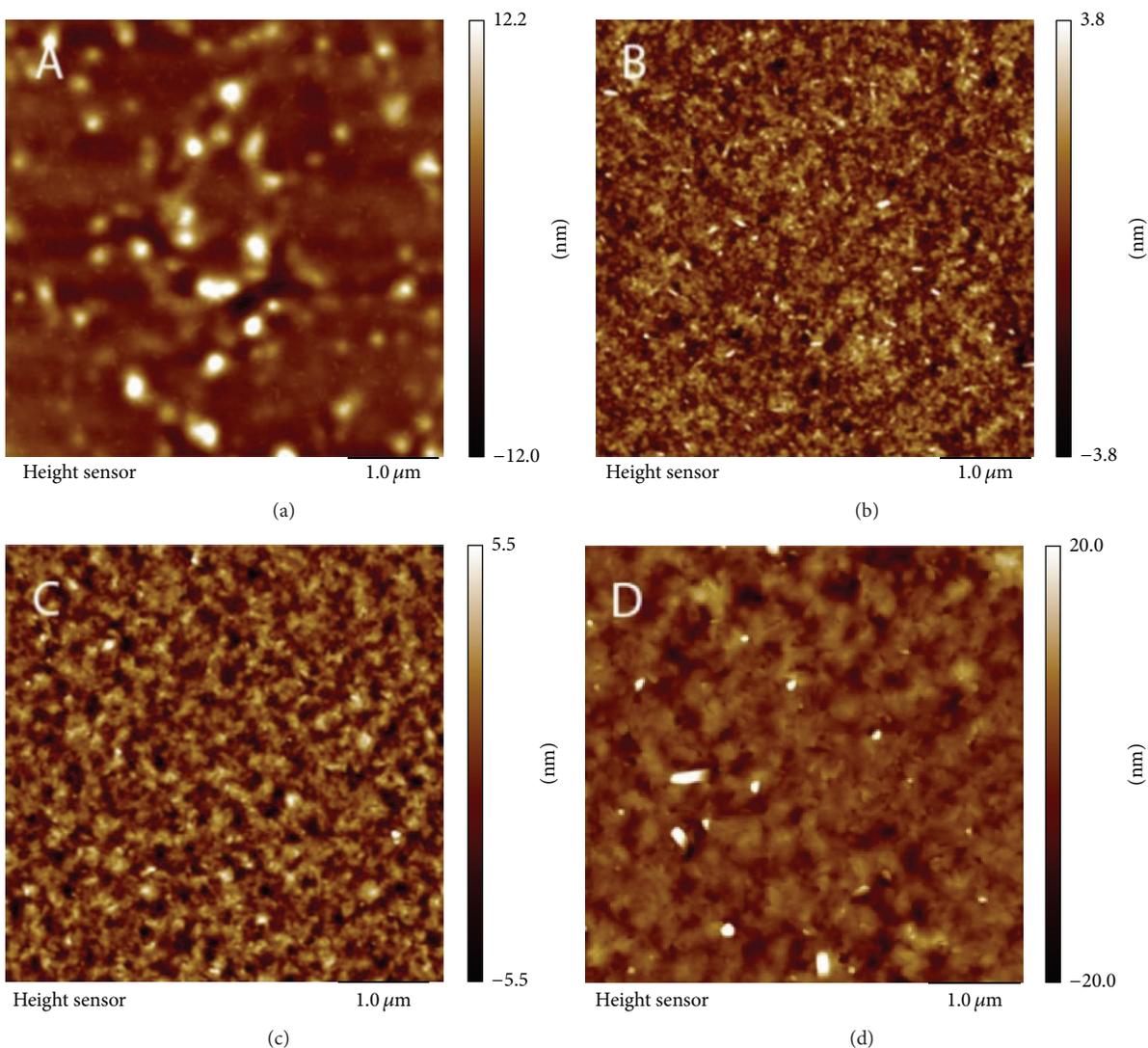


FIGURE 4: AFM images of the ethyl-hexyl alkylated (compound (A)) DPP(TBFu)<sub>2</sub>:PC<sub>60</sub>BM blend: as-cast (A) and annealed at 110°C (B), 140°C (C), and 160°C (D).

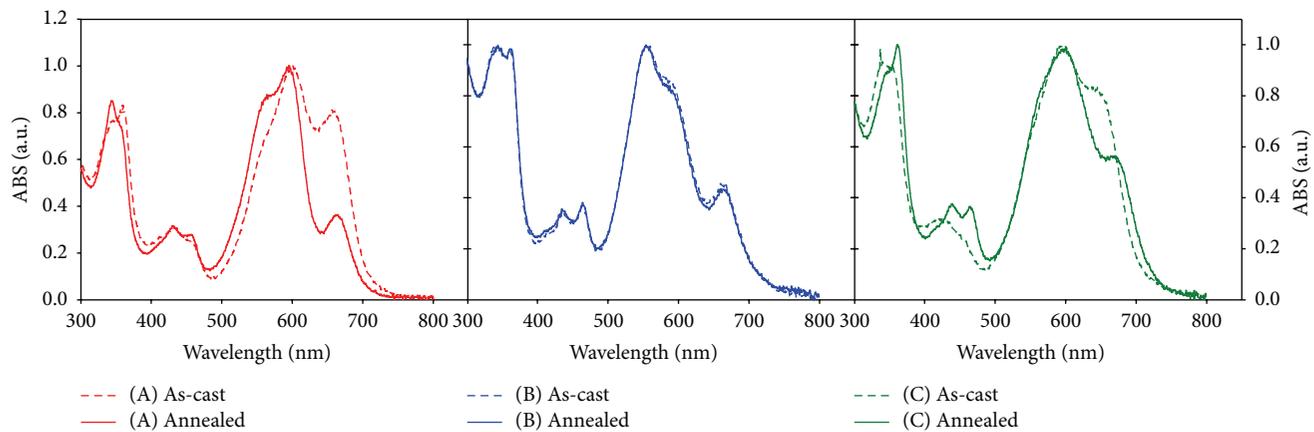


FIGURE 5: Normalized UV-Vis absorption spectra of the as-cast (dashed lines) and annealed at 140°C (solid lines) DPP:PC<sub>60</sub>BM blend layers.

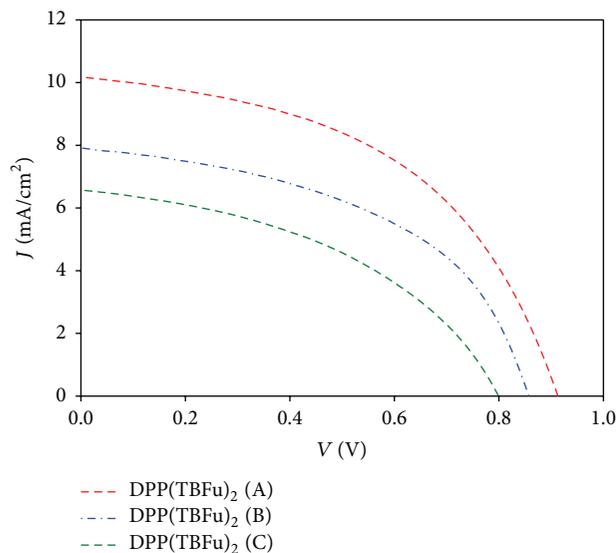


FIGURE 6:  $J$ - $V$  curves of the gradually annealed (at highest PCE) solar cells with  $\text{MoO}_3$  as an anode material.

As a result of the annealing and anode contact stability study, using  $\text{DPP}(\text{TBFu})_2:\text{PC}_{60}\text{BM}$  we attained efficiencies (PCE = 4.52%), with the values of FF,  $V_{\text{oc}}$ , and  $J_{\text{sc}}$  almost identical with the ones containing  $\text{PC}_{70}\text{BM}$ , described in original report [14]. New materials used in the present research possessed lower PCE values, even though material (B) shows a moderate PCE above 3.2%. We see the main reason of PCE decrease when going from A to B in the lower  $V_{\text{oc}}$ , that is, the trend on the contrary to the expectations coming from HOMO energies of isolated (A) and (B) derivatives, obtained from CV (Table 1). The decrease of  $J_{\text{sc}}$  is considered as a consequence of the higher rate of charge recombination in the CT state of B. These effects will be further studied on an enlarged set of DPP donors. Final  $J$ - $V$  curves of the considered materials at the best annealing conditions are depicted in Figure 6. It has to be noted that good reproducibility was observed in case of all materials used. Statistics was done on 12 electrodes for each configuration and 80% of the electrodes were possessing PCE in a range of  $\pm 0.2\%$  when PEDOT:PSS was used as an anode material. In case  $\text{MoO}_3$  cathode reproducibility was even better 90% of devices have shown PCE deviation of  $\pm 0.2\%$ .

**3.3. Thermal Stability of Donor Materials.** It is known that DPP derivatives, especially poorly soluble pigments, usually possess exceptional thermal stability [41, 42]. TGA measurements in air have shown temperatures 356.47°C, 317.08°C, and 300.55°C for samples (A), (B), and (C), respectively. Thus the main issue of the thermal degradation was expected to be the bulk-heterojunction morphology and contact/BHJ interfaces. Thermal degradation of the considered materials was carried out at 80°C and 110°C on both PEDOT:PSS and  $\text{MoO}_3$  based devices for 60 hours with several intermediate device characterizations to monitor the dynamics of the processes ongoing during the thermal annealing. Devices based

on the above mentioned materials are showing exceptional stability at 80°C possessing no notable PCE losses after 60 hours of annealing in ambient atmosphere. However at 110°C after 24 hours degradation was observed, which is associated with devices decapsulation. As it was mentioned morphology of the bulk-heterojunction degrades at temperatures above 150°C for the materials (A) and (C). Material (B) was found to be the most stable regarding the BHJ morphology thermal stability; PCE decrease was observed at temperatures above 170°C.

## 4. Conclusion

One of the main achievements of the present paper is the study of the  $\text{MoO}_3$  anode contact advantages in case of the OSCs based on  $\text{DPP}(\text{TBFu})_2$ , where more thermally stable interface of active layer and anode contact allows using higher annealing temperatures resulting in higher efficiencies in comparison with PEDOT:PSS. It has also been shown that the gradual annealing is more advantageous and results in slightly higher PCE of the considered devices. As a result PCE of organic photovoltaics based on  $\text{DPP}(\text{TBFu})_2:\text{PC}_{60}\text{BM}$  BHJ is approaching the one based on  $\text{DPP}(\text{TBFu})_2:\text{PC}_{70}\text{BM}$  and exceeding previously reported PCE values for  $\text{DPP}(\text{TBFu})_2:\text{PC}_{60}\text{BM}$ . Although PCE of the reference ethyl-hexyl alkylated  $\text{DPP}(\text{TBFu})_2$  was not exceeded by the materials with electrophilic side chains, the acceptable values of the efficiencies were obtained, which along with higher reaction yields in the case of alkyl esters make considered materials relatively prospective. It is also worth mentioning that the modification of  $\text{DPP}(\text{TBFu})_2$  with ethyl-hexyl acetate caused maximum values of  $V_{\text{oc}}$  and  $J_{\text{sc}}$  even without annealing, which can be identify as-cast appropriate BHJ morphology as well as higher thermal morphology stability.

## Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

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