

## Research Article

# Aluminum Silicate Nanotube Coating of Siloxane-Poly(lactic acid)-Vaterite Composite Fibermats for Bone Regeneration

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In our earlier work, a flexible fibermat consisting of a biodegradable composite with soluble silicate species, which has been reported to enhance bone formation, was prepared successfully using poly(L-lactic acid) and siloxane-containing calcium carbonate particles by electrospinning. The fibermat showed enhanced bone formation in an *in vivo* test. In the present work, to improve the hydrophilicity of skeletal fibers in a fibermat, they were coated with nanotubular aluminum silicate crystals, which have a hydrophilic surface that has excellent affinity to body fluids and a high surface area advantageous for pronounced protein adsorption. The nanotubes were coated easily on the fiber surface using an electrophoretic method. In a conventional contact angle test, a drop of water rapidly penetrated into the nanotube-coated fibermat. The culture test using murine osteoblast-like cells (MC3T3-E1) showed that the cell attachment to the nanotube-coated fibermat at an early stage after seeding was enhanced in comparison with that to the noncoated one. This approach may provide a new method of improving the surface of polymer-based biomaterials.

## 1. Introduction

Many types of scaffolds have been investigated for use in bone regeneration [1, 2], which serve as templates for cell interaction and the formation of a bone extracellular matrix (ECM) to provide structural support to newly formed tissues. Ceramics, such as hydroxyapatite (HA) and  $\beta$ -tricalcium phosphate ( $\beta$ -TCP), show good biocompatibility, but they are brittle and thus are difficult to form into various shapes. In contrast, synthetic biodegradable polymers are more easily shaped and show flexibility. However, they may elicit an inflammatory response in the host tissue because of the release of acidic degradation products [3]. Considering these problems, composites have been fabricated using

biodegradable polymers coupled with calcium phosphate ceramics, whose excellent biocompatibility is established [4, 5].

The present authors have developed a novel composite consisting of poly(L-lactic acid) (PLLA), which is a biodegradable polymer that has already been applied clinically, and siloxane-containing vaterite (SiV); the siloxane-PLLA-vaterite hybrid-composite is denoted as SiPVH, hereafter. It can release calcium and silicate ions, which have been reported to enhance the proliferation, differentiation, and mineralization of osteoblasts [6–8]. The cellular compatibility of SiPVH was tested using mouse osteoblast-like cells [9]. A film of SiPVH was prepared by dip-coating a coverglass with the composite solution. The film showed the release

of calcium and silicate ions into the culture medium. Cell proliferation and mineralization on the composite film were enhanced in comparison with those on an undoped vaterite-PLA composite film, which showed no silicon release.

Three-dimensional scaffolds are common in biomaterials. Electrospinning is a process for forming fiber mats with a high porosity and a high flexibility [10–12]. These fiber mats are very promising for the regeneration of tissues and organs [11, 13] owing to the possibility of mimicking some structural features of natural ECM [14]. Numerous studies have shown the enhanced cellular attachment and proliferation on fibrous materials [15–17]. The present authors have successfully prepared a SiPVH fiber mat by an electrospinning method and then performed *in vivo* experiments by placing the fiber mat with 8 mm defects in the front midline of the calvaria of New Zealand rabbits [18, 19]. They found enhanced bone formation on SiPVH fiber mats. PLLA as a matrix phase in SiPVH is relatively hydrophobic. This is a property to be improved for the adhesion of bone-forming cells at the initial stage after implantation in the body.

A naturally occurring hydrated aluminum silicate,  $(\text{HO})_3\text{Al}_2\text{O}_3\text{Si}(\text{OH})$ , which is called imogolite, has a nanotubular structure of  $\sim 2.2$  nm external and  $\sim 1.0$  nm internal diameters [20], with a length of up to several hundreds of micrometers. The imogolite nanotube (INT) is a hydrophilic material owing to numerous hydroxyl groups on its surface and has a high surface area, resulting in a high protein adsorption activity [21–23]. Ishikawa and coworkers reported that the amount of normalized protein per osteoblast-like cell (SaOS-2) cultured on INTs is two times larger than that cultured on a culture dish, and that the cells bond more strongly to INTs than to a culture dish [24, 25]. These results demonstrate the good biocompatibility between cells and INTs; thus, INTs may be useful for various biomedical applications.

In the present work, SiPVH fiber mats were coated with INTs to improve their hydrophilicity to enhance cellular compatibility at the early stage after implantation.

## 2. Materials and Methods

**2.1. Preparation of SiPVH Fiber Mats.** SiV particles of  $1\ \mu\text{m}$  diameter, containing 2.6 wt% silicon, were prepared by a carbonation method using methanol and  $\gamma$ -aminopropyltriethoxysilane (APTES; Momentive Performance Materials, Japan) as the silica precursor, as described in our previous works [9, 18, 19]. One hundred and fifty grams of  $\text{Ca}(\text{OH})_2$ , 60 mL of APTES and 2 L of methanol were mixed by bubbling a  $\text{CO}_2$  gas in the resulting mixture for 75 min at a rate of  $2\ \text{L}\cdot\text{min}^{-1}$ . The resulting slurry was dried at  $110^\circ\text{C}$ , resulting in the formation of spherical SiV particles.

SiPVH was prepared by kneading PLLA (Mitsui Chemicals, LACIA; molecular weight:  $M_w = \sim 140$  kDa) with SiV particles at  $200^\circ\text{C}$  for 10 min in a weight ratio of PLLA/SiV = 7/3.

SiPVH was dissolved in chloroform for electrospinning to prepare 10 wt% PLLA solution. The fiber mats were prepared by electrospinning at a voltage of 20 kV using the conditions and compositions previously found to be optimal

for preparing microfibers of  $\sim 10\ \mu\text{m}$  diameter [18, 19]. The flow rate of the chloroform solution was  $50\ \mu\text{L}\cdot\text{min}^{-1}$ , and the distance between the nozzle and the collector was 150 mm. The drum-shaped collector was rotated at  $2\ \text{m}\cdot\text{min}^{-1}$ , and the syringe filled with the SiPVH solution was traversed at  $70\text{--}100\ \text{mm}\cdot\text{min}^{-1}$  from side to side.

**2.2. Preparation of INT-Dispersed Aqueous Solution.** INTs were synthesized, essentially following a method described by Suzuki et al. [26]. In 500 mL of distilled water (DW), 18.47 g of aluminum chloride ( $\text{AlCl}_3\cdot 6\text{H}_2\text{O}$ ; Wako Pure Chemicals) and 9.2 g of sodium silicate ( $\text{Na}_4\text{SiO}_4\cdot n\text{H}_2\text{O}$ ; Wako Pure Chemicals) were dissolved to yield a Si/Al molar ratio of 0.41.

Sodium hydroxide (NaOH; Wako Pure Chemicals) aqueous solution of 1 M was slowly added to the above-described sodium silicate/aluminum chloride solution at a rate of  $2\ \text{mL}\cdot\text{min}^{-1}$  until the pH of the solution reached 6.8. The sample was separated by centrifugation and the obtained precipitates were rinsed in DW with stirring. After the centrifugation-rinsing process was repeated three times, the resulting aluminum silicate precursors were dispersed again in 12 L of DW. Subsequently, the solution was acidified again by the addition of 12 mL of hydrochloric acid (HCl; Wako Pure Chemicals) with a concentration of 5 M and then heated at  $95^\circ\text{C}$  for 1 or 4 d for INT formation. Consequently, the resulting INTs were dispersed in an aqueous solution with a concentration of 0.087 wt%. The INTs were observed by atomic force microscopy (AFM) in the tapping mode. Figure 1 shows AFM images of the products after heating at  $95^\circ\text{C}$ . The average lengths of the INTs obtained after the heating for 1 and 4 d were 220 and 570 nm, respectively. They are denoted as INT<sub>220</sub> and INT<sub>570</sub>, respectively, hereafter.

**2.3. INT Coating of SiPVH Fiber Mats.** The INT coating of SiPVH fiber mats was achieved using an electrophoretic deposition (EPD) method.

Each fiber mat was cut into  $15\ \text{mm} \times 15\ \text{mm} \times 0.2\ \text{mm}$  dimensions with scissors and then placed on  $15\ \text{mm} \times 60\ \text{mm} \times 0.25\ \text{mm}$  aluminum foil. One hundred microliters of ethanol were dropped on the foil to fix each fiber mat on the cathode electrode.

The objective of the present work is to prepare a hydrophilic coating with an extremely thin layer of INTs on skeletal fibers. Our preliminary work showed the EPD conditions for the preparation. The point of zero charge (PZC) in an INT-containing solution has been reported to be pH 6.0 [27]. The INT-dispersed aqueous solution described in Section 2.2 was adjusted to pH 5.5 with dilute NaOH solution of 0.1 M. The resulting solution was considered to be slightly charged positively. The cathode and anode made of aluminum of  $15\ \text{mm} \times 60\ \text{mm} \times 0.25\ \text{mm}$  dimensions were dipped in 30 mL of the solution, and EPD was carried out for 1 min at 0.1 V of applied voltage. The separation distance between the electrodes was 10 mm. After EPD, the sample was dried at  $50^\circ\text{C}$  for 3 h in air.

**2.4. Scanning Electron Microscopy.** The samples were coated with amorphous osmium using plasma chemical vapor

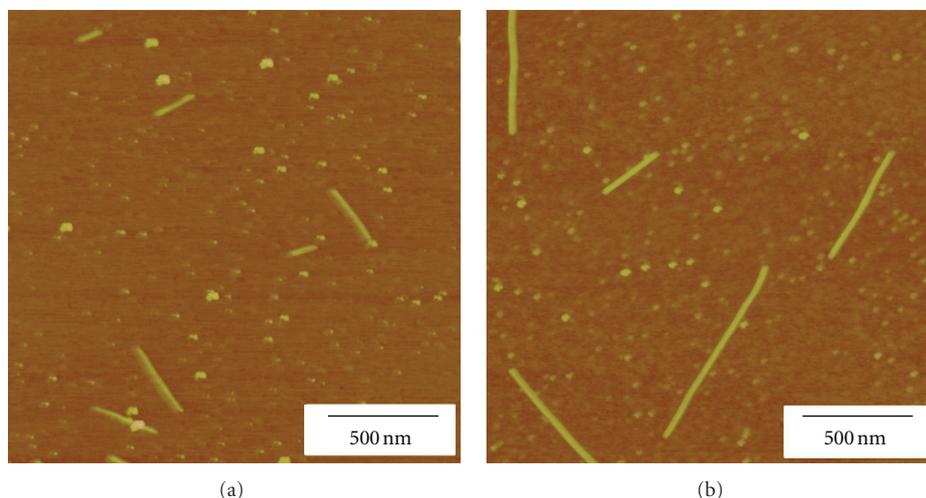


FIGURE 1: AFM images of products after heating at 95°C for (a) 1 d and (b) 4 d.

deposition (CVD) and then observed morphologically using a field emission scanning electron microscopy (SEM) system (JSM-6301F, JEOL, Japan), with an energy-dispersive spectrometer (EDS).

**2.5. Contact Angle Measurement.** The static contact angle of water on the fibermats was measured using a CCD camera and SImage mini ver. 5.01 software. The average contact angle was determined from the measurements at ten random points per sample.

**2.6. Cell Culture Tests.** The fibermats were cut into disk-shaped pieces of 15 mm in diameter with scissors for cell culture tests, and the resulting samples were sterilized using ethylene oxide gas. Murine osteoblast-like cells (MC3T3-E1 cells) were seeded onto the samples in 24-well plates at a density of 50,000 or 80,000 cells·well<sup>-1</sup> to evaluate their proliferation or initial adhesion, respectively, and cultured in alpha minimum essential medium ( $\alpha$ -MEM) containing 10% fetal bovine serum (FBS) at 37°C in 5% CO<sub>2</sub>.

The number of murine osteoblast-like cells (MC3T3-E1 cells) was evaluated using Cell Counting Kit-8 (Dojindo, Japan). The cells were rinsed with  $\alpha$ -MEM and then incubated in the culture medium containing the reagent of the kit at 37°C for 2 h. Cell number was evaluated after measuring the absorbance of the resulting medium at 450 nm. The differences between the samples were determined by Student's *t*-test, with  $P < 0.05$ , which is considered to be statistically significant.

The cells cultured on the samples for 3 h were fixed in 2.5% glutaraldehyde for 40 min at 4°C, dehydrated through a series of increasing concentrations of ethanol, and finally dried with hexamethyldisilazane. The dried samples were observed by SEM after they were coated with amorphous osmium.

To evaluate the initial adhesion of the cells on the fibers, they were cultured on the samples for 3 h and then fixed in phosphate buffer solution (PBS) containing 4% paraformaldehyde for 30 min at 4°C. The cells were then

treated with PBS containing 1% bovine serum albumin (BSA) and 0.1% Triton X for 25 min at 4°C. Finally, they were fluorescence-stained with 50  $\mu$ L of Alexa Fluor 488 Phalloidin (Molecular Probe), which was diluted to a concentration of 2.5% with PBS, for 30 min at 37°C. The samples were observed with a fluorescence microscope (Keyence, BZ-9000).

### 3. Results and Discussion

**3.1. Preparation of INT-Coated Fibermats.** In our earlier work, PLLA fibermats consisting of skeletal fibers of  $\sim 10 \mu$ m diameter and  $\sim 40 \mu$ m spaces between them, on average, were reported to be suitable for cellular in-growth [28]. Figure 2 shows SEM images of the surface morphologies of the noncoated and INT-coated SiPVH fibermats. Microfibers of  $\sim 10 \mu$ m diameter were observed on the surface and found to be intertwined. The sizes of the spaces between the microfibers varied between 10  $\mu$ m and several hundred micrometers, as shown in Figure 2(a). Numerous pores of submicrometer diameter were found on the surfaces of the fibers, which were due to the volatilization of the solvent of the polymer solution during electrospinning [29, 30].

There are almost no significant differences in the appearance between the SiPVH fibermat coated with INT<sub>220</sub> and the noncoated one, as shown in Figures 2(a) and 2(b), respectively. On the other hand, in the fibermat coated with INT<sub>570</sub> shown in Figure 2(c), weblike plates around cross-linked points of the fibers can be observed.

The EDS spectrum in Figure 2(a) shows the peaks of Ca and Si due to SiV. The Si peak is small, since the silicon content of SiV is very small (2.6 wt%). In the spectra of the INT-coated SiPVH fibermats (Figures 2(b) and 2(c)), Al peaks are seen in addition to the peaks of Ca and Si due to SiV. The Si peak intensities are higher than that in Figure 2(a). The Al contents of the fibermats coated with INT<sub>220</sub> and INT<sub>570</sub> were estimated to be  $\sim 2$  and  $\sim 7$  %, respectively. The element originates from INTs on the surface, since no Al is included in SiPVH. This indicates

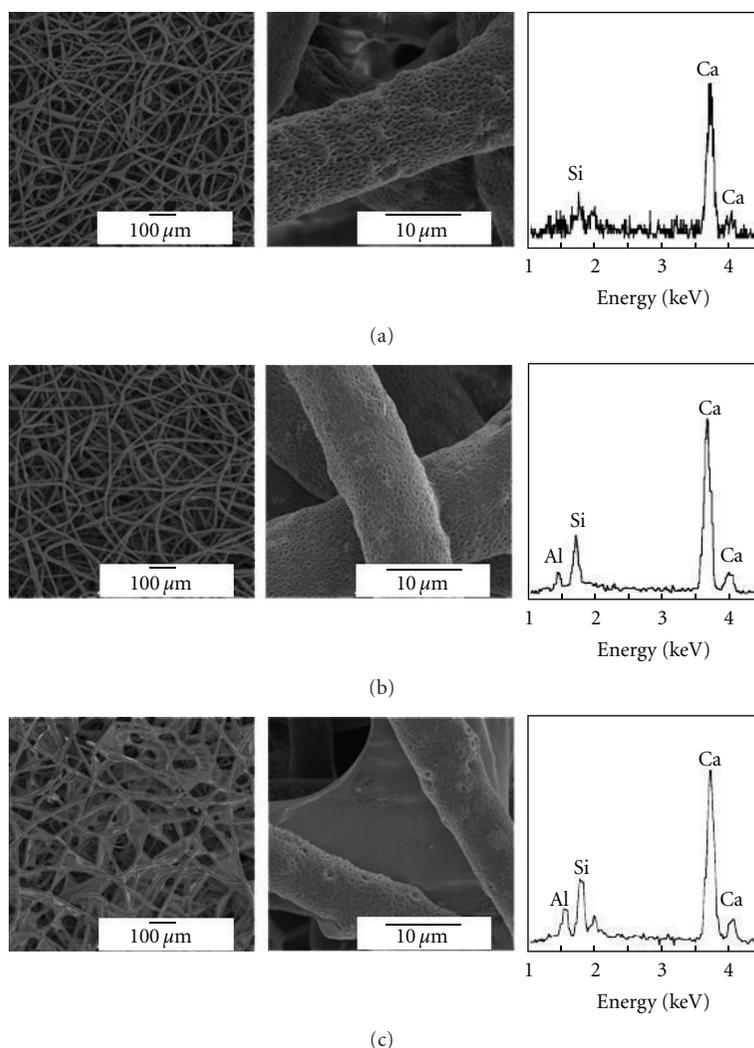


FIGURE 2: SEM images and EDS spectra of (a) noncoated and (b, c) INT-coated SiPVH fiber mats. (b) INT<sub>220</sub>-coated and (c) INT<sub>570</sub>-coated SiPVH fiber mats. Center: magnified views; right: EDS spectra.

that INTs were coated on the fiber mats by EPD. The weblike plates around cross-linked points, as shown in Figure 2(c), were found to consist of aluminum silicates with a Si/Al ratio of  $\sim 0.5$ , which corresponds to the INT composition. That is, the plates were concluded to be an aggregation of INTs. It has been reported that the viscosity of the INT-dispersed aqueous solution increases with increasing INT length and the number or forming net texture of INTs [24]. INTs attracted to an electrode in an electric field would easily aggregate around the surface of the fibers, since the number of INTs in a dispersed aqueous solution increases. Then, the attracted INTs would likely form bridges around cross-linked points. Since INT<sub>220</sub> was shorter than INT<sub>570</sub>, the incidence possibility of aggregation would be lower.

To discuss the EPD effect, a dip-coating method was used: after the SiPVH fiber mat was dipped in the INT-dispersed aqueous solution (0.087%), it was drawn up at a speed of  $\sim 1 \text{ mm} \cdot \text{s}^{-1}$  and then dried. The EDS analysis of the resulting fiber mat showed no Al peak (under the analytical limit). That is, it is difficult to coat the fiber mats with INTs

by dip coating, and thus EPD is concluded to be a powerful method for such coating. Since the INT concentration in the solution is considerably small, almost no INTs would adhere to the skeletal fibers after the dip coating. It is not easy to prepare the dispersed solution containing a large number of INTs owing to some technical problems, such as impurity formation in the solution, and gelation owing to its concentration. The hydrophobic surface of SiPVH fibers may also be one of the causes of the difficulty in dip coating. In the case of EPD, INTs would pass through the porous structure of fiber mats in an electric field and concentrate around the surfaces of skeletal fibers, and then accumulate on them.

The noncoated fiber mat exhibited hydrophobicity of a contact angle of  $121^\circ$ . In contrast, a drop of water immediately penetrated the fiber mats coated with INTs, which indicated that the hydrophilicity of these fiber mats was improved markedly after INT coating. This is because INTs are hydrophilic owing to the hydroxyl groups on their surfaces and have a high water absorbency, attributed to their nanotubular structure. In particular, the fiber mat coated

with INT<sub>220</sub> showed excellent hydrophilicity as well as that coated with INT<sub>570</sub>, although it is difficult to morphologically distinguish them from the noncoated one. A very small number of INTs seem to be sufficient for improving the hydrophilicity of fibermats.

**3.2. Cellular Compatibility.** The cellular proliferation behavior of the INT-coated fibermats in comparison with that of the noncoated one was evaluated using MC3T3-E1 cells. Figure 3 shows the numbers of cells after culturing on them for 3 days. The numbers of cells after culture on INT<sub>220</sub>- and INT<sub>570</sub>-coated fibermats were significantly higher ( $P < 0.05$ ) than those on the noncoated one on both day 1 and day 3. There were no significant differences in the number between the two types of INT-coated fibermat. This might have been due to the effects of the INT coating, which shows high cellular compatibility [24, 25]. Even with coating with a trace amount of INT<sub>220</sub>, excellent improvement in the compatibility on the fibermat was achieved.

The doubling time (DT) of the number of cells for discussing proliferation behavior is determined by

$$DT = \frac{(t_2 - t_1) \log 2}{\log N_2 - \log N_1}, \quad (1)$$

where  $N_1$  and  $N_2$  are the numbers of cells at the culture time points  $t_1$  and  $t_2$  (h), respectively. The DTs of the noncoated, INT<sub>220</sub>-coated, and INT<sub>570</sub>-coated fibermats were estimated to be 33.0 h, 33.7 h, and 33.7 h, respectively, which are not significantly different. The difference in the numbers of cells after culture between the noncoated and INT-coated fibermats might have originated from the adhesion behavior of the cells at an early stage after seeding.

Figure 4 shows the numbers of cells attached to the noncoated and INT-coated fibermats at 3 h after seeding. The numbers of cells cultured on the INT-coated fibermats were significantly higher than that on the noncoated one. INTs were suggested to enhance the initial cell attachment to the fibermats. However, there was no significant difference in the number of cells cultured between INT<sub>220</sub> and INT<sub>570</sub>.

Figures 5 and 6 show the morphologies of the cells attached to the fibermats at 3 h after seeding. The cells adhering to the noncoated fibermat were still spherical and were preferentially attached to the surface around cross-linked points between skeletal fibers, as shown in Figure 5(a). On the other hand, as shown in Figures 5(b) and 5(c), the cells on the INT-coated fibermats were elongated not only on the surface around the cross-linked points but also on individual fibers and actin filaments were aligned to the fibers. The cells were attached over the entire surface of the INT-coated fibermats, as shown in Figures 6(b) and 6(c). These cell culture tests suggested that INTs enhanced cell adhesion and growth.

Watari and coworkers reported that protein adsorption on nanofibrous scaffolds contributes to excellent cell adhesion and growth [21–23]. In our preliminary work, INTs showed a high adsorbability to various types of protein including acidic, neutral, and basic ones, owing to the large specific surface area of the nanotubular structure. Thus, it is proposed that cell-adhesive proteins, such as fibronectin

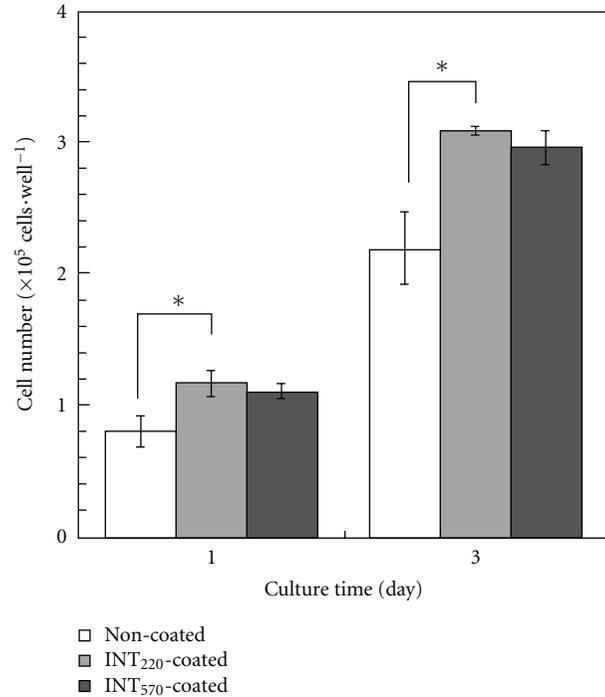


FIGURE 3: Numbers of cells attached to noncoated and INT-coated fibermats after 3 days of culture.

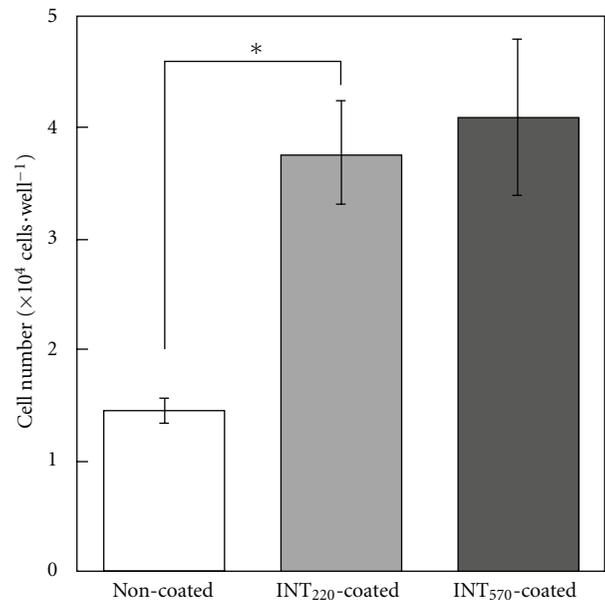


FIGURE 4: Numbers of cells attached to noncoated and INT-coated fibermats at 3 h after seeding.

and vitronectin, could be easily adsorbed on the INT surface [31, 32].

Furthermore, the INT nanostructure may also be related to cell adhesion. Hirata et al. reported that the nanostructure of multiwalled carbon nanotubes might be effective in cell entrapment [33]. The geometrical factor in the INT nanostructure may be beneficial for cell attachment to surfaces.

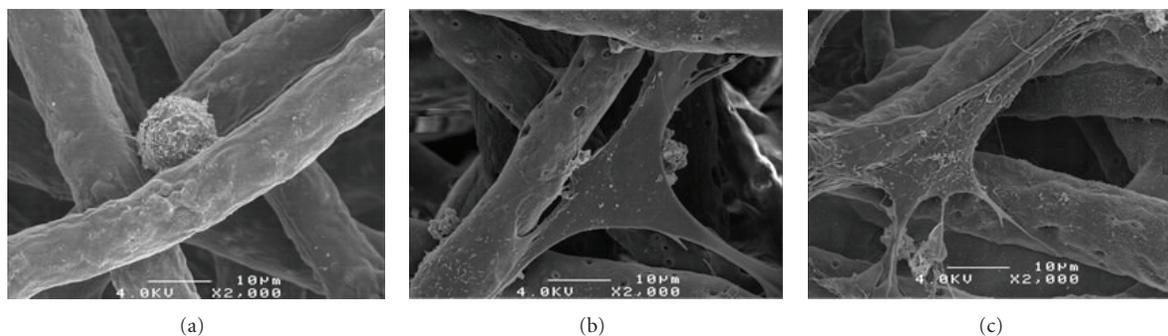


FIGURE 5: SEM images of cells on (a) noncoated and (b, c) INT-coated SiPVH fiber mats at 3 h after seeding. (b) INT<sub>220</sub>-coated and (c) INT<sub>570</sub>-coated SiPVH fiber mats.

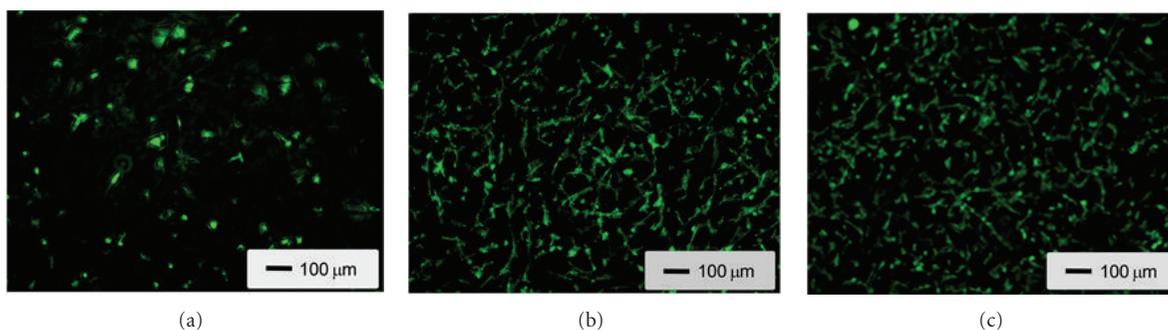


FIGURE 6: Fluorescence micrographs of cells on (a) noncoated and (b, c) INT-coated SiPVH fiber mats at 3 h after seeding. (b) INT<sub>220</sub>-coated and (c) INT<sub>570</sub>-coated SiPVH fiber mats.

In the present work, a trace amount of INT was coated on SiPVH fibers: no existence could be observed by SEM. The measurement of the INT amount on the fibers is not easy and the investigation is in progress. Although there might be almost no need to consider the influence of the Al<sup>3+</sup> ion dissolved from the trace amount of INT on living body, we would have to discuss its behavior in future.

There are almost no reports on the solubility of aluminum silicates in living body or under its simulated environment, to the best of our knowledge. Although imogolite has been reported to dissolve slightly in alkaline solutions of pH > 10 [34] and in acid solutions of pH < 3 [35], almost no information around the neutrality is seen. It would be important to examine the solubility of INTs in detail.

#### 4. Conclusion

Aluminum silicate INTs were coated successfully on an electrospun fiber mat, consisting of poly(L-lactic acid)-based composite microfibers containing siloxane-doped calcium carbonate particles by electrophoretic deposition using an applied voltage of 0.1 V for 1 min. INTs of 220 and 570 nm lengths on average were coated very thinly such that their presence could not be observed by SEM. In the case of coating with INTs of 570 nm length, weblike plates formed around cross-linked points of skeletal fibers owing to their aggregation. Shorter INTs were better for preparing a homogeneous coating on a fiber mat. The hydrophilicity of the fiber mat was improved markedly after INT coating. Cell

culture tests indicated that INT enhanced the attachment and extension of MC3T3-E1 cells. The excellent compatibility might have originated from the protein adsorbability and geometrical factor of the nanostructure. Thus INT coating may be one of the best approaches to improving the hydrophilicity and cellular compatibility of biomaterials for bone regeneration.

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