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Research Article

Preparation of Mg₂FeH₆ Nanoparticles for Hydrogen Storage Properties

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Magnesium (Mg) and iron (Fe) nanoparticles are prepared by thermal decomposition of bipyridyl complexes of metals. These prepared Mg-Fe (2:1) nanoparticles are hydrogenated under 4 MPa hydrogen pressure and 673 K for 48 hours to achieve Mg_2FeH_6 . Their structural analysis was assessed by applying manifold techniques. The hydrogen storage properties of prepared compound were measured by Sieverts type apparatus. The desorption kinetics were measured by high pressure thermal desorption spectrometer (HP-TDS). More than 5 wt% hydrogen released was obtained by the Mg_2FeH_6 within 5 min, and during rehydrogenation very effective hydrogen absorption rate was observed by the compound.

1. Introduction

The demand for an efficient and clean alternative fuel has increased in the recent years due to energy crisis and growing environmental concerns [1]. The use of hydrogen as an alternate fuel is a forward step to give up noninvasive and nontoxic emission, which is spoiling our surroundings [2]. The increasing relevance of hydrogen economy has generated interest in finding a safe way to transport hydrogen. Magnesium (Mg) is considered as attractive material for hydrogen storage economy due to light weight, low cost, and high storage capacity (7.6 wt%) [3, 4]. But slow kinetics and high temperature limit its practical use. Many efforts have been made to improve its absorption rate by reducing size or by adding transition metals [5, 6], creating defects by ball milling [7–10], synthesizing composite of metal hydrides [11, 12], and adding catalyst [13-18]. Mg-based 3d transition metal hydrides have attracted much interest for the future hydrogen economy, due to their great abundance, high reversible storage capacity, and better kinetics [19-21]. Mg₂FeH₆ is considered as attractive/gorgeous materials in particular due to its high gravimetric capacity 5.4 wt% and superior volumetric density 150 kg H₂ m⁻³. However, Mg and

Fe do not form stable binary compound; therefore synthesis of Mg₂FeH₆ is very difficult [22, 23].

First report for the preparation of Mg₂FeH₆ compound was published by Didishein et al. [24], by sintering process of Mg and Fe powder under high hydrogen pressure. By applying high hydrogen pressure (12 MPa), elevated temperature (793 K) and after several days, the yield of complex hydride was only 50 wt. (%). To improve the yield (%), a processing rout that is, mechanical alloying of precursory materials before sintering was employed [25]. Another approach for the synthesis of Mg₂FeH₆ compound was also performed by reactive milling (RM) under high hydrogen pressure [26]. This mechanically activated method can reduce the particle size to nanolevel and enhance the hydrogen sorption kinetics of the hydrides. However, the obtained compound still contained unreacted MgH₂ and Fe. Alternatively, Raman et al. [23] milled the 2Mg + Fe mixture under 1 MPa hydrogen pressure. Moreover, the compounds often oxidized and got polluted by sintering, mechanical alloying, and reactive milling methods.

Progress regarding the yield has been made by using optimized sintering or RMA conditions, and more recently, by hydriding combustion synthesis [27] or sintering of metal

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nanoparticles produced by a hydrogen plasma metal reaction [28]. However, one drawback of the processing methods (with yields over 90%) is the long processing time involved, which is usually 10 h. Additionally, the Fe is usually retained in the synthesized material; for this reason, a purification process is necessary.

In this work, Mg and Fe pure nanoparticles are prepared by the thermal decomposition of bipyridyl complex of Mg and Fe. This is an alternate approach for the preparation of pure Mg and Fe nanoparticles. (2:1) Mg-Fe nanoparticles were transferred into reactor to form Mg_2FeH_6 under 4 MPa hydrogen pressure. Hydrogen storage behavior of the Mg_2FeH_6 compound having yield more than 90% was also investigated.

2. Experimental

The reagents and chemicals used are magnesium chloride (Sigma Aldrich), iron chloride (Sigma Aldrich), 2,2′ bipyridine (Fluka), 1-propanol, and diethyl ether. Scheme 1 is give brief illustration for the preparation of pure metal nanoparticles.

2.1. Preparation of [Mg(Bipy)₃]Cl₂. 2.56 gm (15 mml) solution of 2,2′ bipyridine and 4.75 g (50 mml) solution of magnesium chloride were prepared in 1-propanol. Both solutions were mixed through dropping funnel in a conical flask with constant stirring at a temperature of 323–333 K. Bipyridyl complex of magnesium did not precipitate by adding the bipyridine solution to the magnesium chloride solution, but the color of the mixture changed from transparent to orange. The solution was concentrated to two-third of the original through rotary evaporator and then kept for crystallization. Crystallization occurred very slowly and took six days for completion. The crystals of the bipyridyl complex were separated through filtration, washed with diethyl ether, and dried under IR lamp.

Melting point, IR, NMR, and TGA results of the compound are given below.

442 K, IR (KBr, cm $^{-1}$): ν (C–H_{arom}) 3050, 2927; ν (C–N_{arom}) 1600; ν (C–C_{arom}) 1501, 1453, 900–600; ν (Mg–N) 415. HNMR (DMSO, ppm): 8.687 (dd, 2H, bipyH2,2'); 8.384 (dd, 2H, bipyH5,5'); 7.95 (ddd, 2H, bipyH4,4'); 7.458 (ddd, 2H, bipyH3,3'). TGA: 460–580 K (80.32% weight loss), 606–722 K (18.22% residual mass).

2.2. Preparation of Fe[(Bipy)₂]Cl₂. 2.97 g (15 mmol) tetrahydrated ferrous chloride (FeCl₂-4H₂O) and 7.05 g (45 mmol) 2,2′-bipyridine dissolved in solvent (2-propanal) separately. Both solutions were mixed with HPLC pump in flask with constant stirring at room temperature, keeping flow rate very slow (0.02 ml/sec) to get the homogenous mixture. Yellow precipitates were abstained, which were washed with tetrahydroformide to remove the impurities. The complex was crystallized from methanol solution in open air.

Melting point, IR, HNMR, and TGA results of the compound are given below.

$$[M(bipy)_{2}]Cl_{2} \xrightarrow{\Delta} [M(bipy)]Cl_{2}$$

$$MCl_{2} \xrightarrow{\Delta} M^{0} + m + Cl_{2}$$

General scheme for the preparation of nanoparticles

SCHEME 1

432 K, IR (KBr, cm⁻¹): ν (C–H_{arom}) 3059, 2905; ν (C–N_{arom}) 1593; ν (C–C_{arom}) 1502, 1442, 900–600; ν (Fe–N) 413. ¹HNMR (DMSO, ppm): 7.798 (s, 6H, bipyH_{4,4'}); 7.301 (s, 6H, bipyH_{5,5'}); 8.441 (s, 6H, bipyH_{6,6'}) 8.224 (s, 6H, bipyH_{3,3'})TGA: 469–510 Kn (28.12% weight loss) 606–718 K, (38.41% weight loss), residual mass 25.13%.

Keeping in view all these observations, following pathway has been proposed for thermal degradation of $M_x[\text{bipy}]_y\text{Cl}$ to metal. Bipyridine was removed from the metal complex in two or three steps. This shows coordination on metal centre paving the way for H-elimination exclusion form pyridine ring followed by the reduction, elimination, and finally formation of metal nanoparticles.

2.3. Preparation of Mg-Fe Hydride Nanoparticles. 1.5 gm of as-prepared pure (2Mg:Fe) nanoparticles was minced in gloves box to get the homogeneous mixture. This mixture was then transferred into sample tube of the chamber. First the chamber was flushed with argon to remove the oxygen contents and evacuated up to 10⁻³ Pa. In the second step the chamber was heated at 673 K with the heating rate of 5 K/min under 4 MPa hydrogen pressure. After 24–26 hours Mg was converted into MgH₂, but no reaction with Fe nanoparticles occurred. After prolonging hydrogenation for 48–50 hours MgH₂ reacted with Fe and converted into Mg₂FeH₆. The structural properties of the compound were analyzed by PANanlytical X'Pert Pro powder X-rays diffractometer (XRD). The details of apparatus are given in our previous report [18].

Morphology and particle size of compound were determined by Amray-1910FE field emission scanning electron microscope (FESEM). Thermal desorption spectra were measured by Setaram Labsys apparatus under an argon flow in an ${\rm Al_2O_3}$ crucible in the range of 300–673 K at heating rates 2, 5, 10, and 15 K/min. Additionally, hydrogen absorption and desorption behavior was studied by using Sieverts type apparatus. The results show that very promising hydrogen storage behavior was observed by the ${\rm Mg_2FeH_6}$ nanoparticles.

3. Results and Discussions

XRD patterns of the Mg-Fe compound are shown in Figure 1. The pattern in Figure 1(a) belongs to pure Mg (PDF-04-0770) and pure Fe (PDF-34-0529) nanoparticles. Figure 1(b) shows

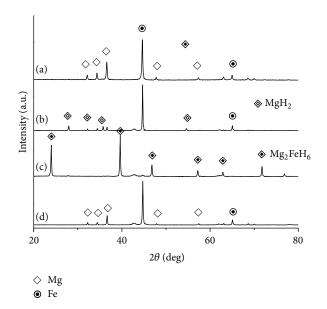


FIGURE 1: XRD patterns of Mg-Fe compound; (a) mixture of Mg and Fe, (b) after 24-hour hydrogenation, (c) after 48-hour hydrogenation, and (d) after dehydrogenation under initial 200 Pa hydrogen pressure.

the mixed pattern of $\mathrm{MgH_2}$ (PDF-12-0697) and Fe after 24-hour hydrogenation. Figure 1(c) confirms the pure phase of Mg-Fe hydride (PDF-38-0843) where Figure 1(d) represents again the pure elemental phase of Mg and Fe nanoparticles after dehydrogenation.

The XRD results also show that, after hydrogenation sharpness of the Mg-Fe hydride peaks increased, with the insertion of hydrogen atom, the surface lattice volume of the compound is increased. This expansion and contraction causes cracks in the particles, resulting in the decrease of particle size [33].

Figure 2 shows the morphology and particle size of Mg-Fe compound after hydrogen absorption and desorption. Figure 2(a) shows the mixed phase of Mg and Fe nanoparticles. It also shows that particles are homogenously dispersed, but the mixture has different particle sizes. The larger particles are surrounded by the smaller particles. Broadness of the peak of Fe from XRD pattern illustrates that smaller particles are of iron, where larger particles belong to magnesium.

Figure 2(b) again shows the mixed phase of $\mathrm{MgH_2}$ and Fe after 24-hour hydrogenation under 4 MPa pressure at 673 K. SEM results also show some aggregation effect due to higher temperature. After 48 hours of hydrogenation, the compound completely converted into Mg-Fe hydrides as conformed from Figure 1(c). Figure 2(c) shows that the compound is elongated. This elongation might be due to the distortion of lattice after the insertion of hydrogen atom [34].

Figure 2(d) shows the SEM micrograph of regenerated Mg and Fe nanoparticles after dehydrogenation under 200 Pa hydrogen atmosphere. Significant decrease in particle size was observed after dehydrogenation, due to the elimination of hydrogen from the compound. Hydrogen desorption results are illustrated in Figure 3 at different temperatures

under initial 200 Pa hydrogen pressure, which indicates that up to 5 wt% hydrogen desorption occurs within 3–5 min. Results also show that the rate of desorption is effectively increased by increasing the temperature. At 573 K, more than 5 wt% hydrogen was released by the system and 4.4 wt% and 3.5 wt% hydrogen desorption occurred within 5–10 min at temperatures of 523 and 473 K, respectively.

More than 5 wt% release of hydrogen also justifies the purity of Mg₂FeH₆ nanoparticles.

The hydrogen absorption of 2Mg: Fe compound at different temperatures, under 4 MPa hydrogen, is presented in Figure 4. Within 30–45 sec, 4.5% hydrogen was absorbed by the system at 573 K. This high absorption rate is due to the catalytic effect of Fe nanoparticles. The absorption rate becomes slow due to the reaction of MgH₂ with H₂ and Fe, resulting in the formation of Mg₂FeH₆. Very promising hydrogen absorption 3.7 wt% and 4 wt% was also observed by the compound at 473 K and 523 K, respectively. This high absorption rate is also due to the decrease in particle size after dehydrogenation and optimal morphology, which provide the shorter diffusion distance for hydrogen atoms during the solid-solid and gas-solid reactions compared with microsize compound to diffuse into the surface lattice [35].

Figure 5 represents the thermal desorption spectrum of $\mathrm{Mg_2FeH_6}$ compound with the different heating rates of 2 K/min, 5 K/min, 10 K/min, and 15 K/min. Desorption peaks at 648 K, 630 K, 625 K, and 610 K corresponds to heating rates of 15 K/min, 10 K/min, 5 K/min, and 2 K/min. The desorption activation energy of the compound can be estimated by using Kissinger equation [36]:

$$\frac{\left(\ln\left(\beta/T_p^2\right)\right)}{d\left(1/T_p\right)} = -\frac{E}{R},\tag{1}$$

where T_p is the peak desorption temperature, β is the heating rate, E is activation energy, and R is the gas constant.

The activation energy of pure Mg-Fe hydrides (Mg₂FeH₆) nanoparticles compound calculated from Kissinger plot as shown in the inset of Figure 5 is 159.8 \pm 6.7 kJ/mol. These results are closely matched to the 165.7 \pm 8.4 kJ/mol and 174 \pm 36 kJ/mol demonstrated by Gennari et al. and Zhang et al. [8, 29], respectively, whereas their synthesis procedure is quite different as compared to this report. In this process, some contents of Fe remained unreacted, which played as catalyst, due to the lower activation energy which they have reported.

The pressure-composition-temperature (P-C-T) curves shown in Figure 6 were obtained during the first hydrogen desorption/absorption cycle without any activation process at 473, 523, and 573 K. Only one plateau is observed during dehydrogenation and rehydrogenation of compound. Hydrogen desorption contents are 3.8 wt% at 473 K, 4.3 wt% at 523 K, and 4.95 wt% at 573 K. Whereas during rehydrogenation 4.7 wt% at 573 K, 4.13 wt% at 523 K, and 3.4 wt% at 473 K were absorbed by the compound. Desorption and absorption plateau pressure of hydrogen is given in Table 1. The hydrogen storage capacity and the equilibrium pressure in both absorption and desorption processes decrease as the temperature falls. This fact is due to the metal atom

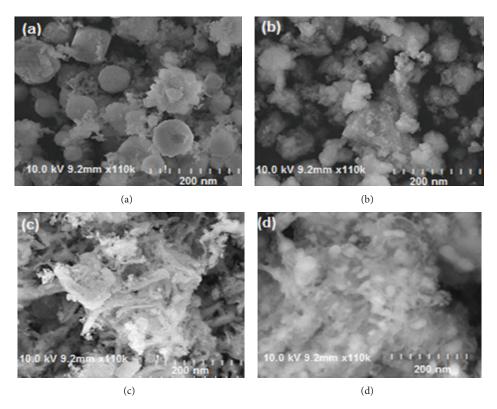


FIGURE 2: SEM images of Mg-Fe system: (a) mixture of Mg and Fe nanoparticles, (b) after 24 hr hydrogenation, (c) after 48 hr hydrogenation, and (d) after dehydrogenation.

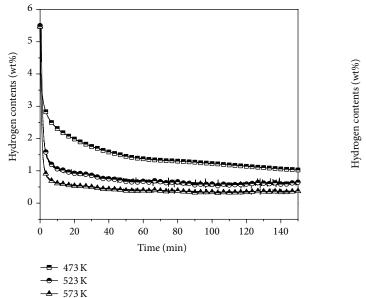


FIGURE 3: Hydrogen desorption properties of Mg₂FeH₆ nanoparticles at different temperatures under initial 200 Pa hydrogen pressure.

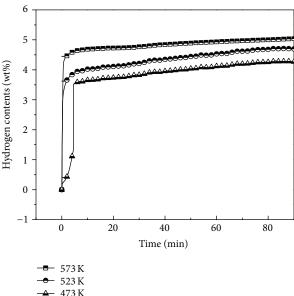


FIGURE 4: Hydrogen absorption properties of Mg-Fe hydride nanoparticles at different temperatures.

diffusion mechanism involved in absorption process, which is enhanced by rise in temperature [22]. Both P-C-T curves of absorption and desorption can reach more than 4 wt% hydrogen content, which suggests the excellent reversibility.

The thermodynamic information for $\mathrm{Mg_2FeH_6}$ hydriding/dehydriding was determined by plotting the logarithmic dissociation pressures against the reciprocal temperature (van't Hoff plot) as shown in Figure 6. The entropy (ΔS) and

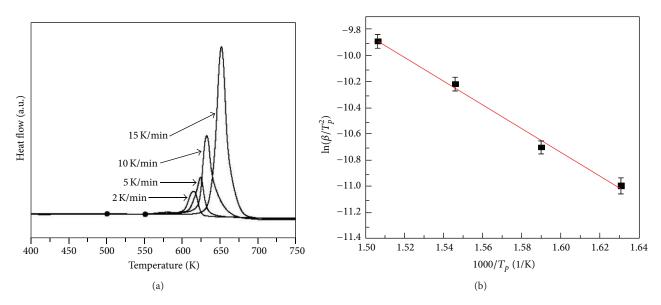


FIGURE 5: (a) Thermal desorption spectrum of Mg-Fe hydride nanoparticles with different heating rates. (b) Represents the Kissinger plot for activation energy.

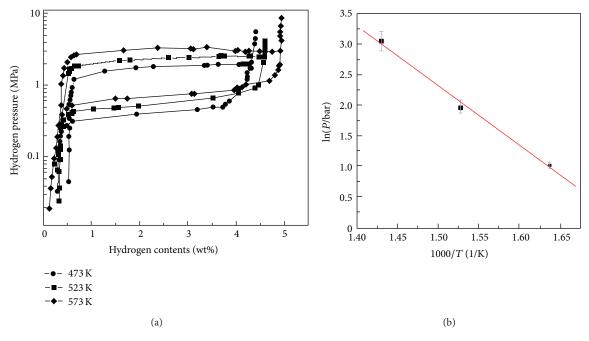


FIGURE 6: (a) Pressure-composition-temperature curves of hydrogen desorption and absorption of the samples at 473, 523, and 573 K, (b) van't Hoff plot of Mg_2 FeH₆ compound constructed from desorption isotherms at 473, 523, and 573 K.

enthalpy (ΔH) values per mole of hydrogen of Mg₂FeH₆ compound were calculated by the data given in Table 1. Comparison of thermodynamic properties of this work with previous reports is given in Table 2.

4. Conclusions

In this work, pure Mg₂FeH₆ with the yield more than 90% was successfully prepared after the hydrogenation of (2:1)

Mg-Fe nanoparticles under 4 MPa hydrogen pressure. The synthesized compound has grain size about 60–70 nm. More than 5 wt% hydrogen released was observed within 5 min under 200 Pa hydrogen pressure at 573 K.

Results show that hydrogen absorption was achieved in two parts. In first step, within 1 min more than 4 wt% hydrogen absorption is attributed to the catalytic effect of Fe nanoparticles. In second step absorption rate is slow due to reaction of Fe with MgH_2 to form Mg_2FeH_6 . This effective dehydrogenation/hydrogenation rate is also due to effective

Temperature (K)	Hydrogen desorption (wt%)	Plateau pressure desorption (MPa)	Hydrogen absorption (wt%)	Plateau pressure absorption (MPa)
473	3.8	0.32	3.4	1.8
523	4.3	0.41	4.1	2.2
573	4.95	0.72	4.7	3.1

TABLE 1: Pressure-composition-isotherm (PCI) characteristics of Mg-Fe hydride nanoparticles at different temperatures.

Table 2: Entropy and enthalpy values for formation and decomposition of Mg₂FeH₆ compound calculated from van't Hoff plot.

No.	Entropy (ΔS)	Enthalpy (ΔH)	Reference
(1)	-145 ± 9	-89 ± 6	This work
(2)	-140.2 ± 8.7	-82.4 ± 5.8	[29]
(3)	147 ± 15	-87 ± 3	[30]
(4)	-137 ± 13	-80 ± 28	[31]
(5)	-147 ± 9	-86 ± 6	[32]

decrease in activation energy which was calculated from Kissinger plot. High absorption and desorption behavior of Mg₂FeH₆ compound at nanoscale is a forward step for the hydrogen storage materials for mobile applications.

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References

- [1] T. Massalski, H. Okamoto, P. Subramanian, and L. Kacprzak, *Binary Alloy Phase Diagrams*, ASM, Materials Park, Ohio, USA, 2nd edition, 1990.
- [2] F. Barbir, T. N. Veziroğlu, and H. J. Plass Jr., "Environmental damage due to fossil fuels use," *International Journal of Hydro*gen Energy, vol. 15, no. 10, pp. 739–749, 1990.
- [3] R. Zidan, D. K. Slattery, and J. Burns, "Study of chemically synthesized MgMgH₂ for hydrogen storage," *International Journal of Hydrogen Energy*, vol. 16, no. 12, pp. 821–827, 1991.
- [4] J. H. Kim, B. G. Kim, and Y. M. Kang, "Synthesis and enhanced hydrogen desorption kinetics of magnesium hydride using hydriding chemical vapor synthesis," *Journal of Alloys and Compounds*, vol. 529, pp. 102–107, 2012.
- [5] G. Liang, J. Huot, S. Boily, A. van Neste, and R. Schulz, "Catalytic effect of transition metals on hydrogen sorption in nanocrystalline ball milled MgH₂-Tm (Tm = Ti, V, Mn, Fe and Ni) systems," *Journal of Alloys and Compounds*, vol. 292, no. 1-2, pp. 247–252, 1999.
- [6] G. Y. Liang, "Synthesis and hydrogen storage properties of Mgbased alloys," *Journal of Alloys Compounds*, vol. 370, no. 1-2, pp. 123–128, 2004.
- [7] A. Zaluska, L. Zaluski, and J. O. Ström-Olsen, "Nanocrystalline magnesium for hydrogen storage," *Journal of Alloys and Compounds*, vol. 288, no. 1-2, pp. 217–225, 1999.
- [8] F. C. Gennari, F. J. Castro, and G. Urretavizcaya, "Hydrogen desorption behavior from magnesium hydrides synthesized by

- reactive mechanical alloying," *Journal of Alloys and Compounds*, vol. 321, no. 1, pp. 46–53, 2001.
- [9] J. Huot, M.-L. Tremblay, and R. Schulz, "Synthesis of nanocrystalline hydrogen storage materials," *Journal of Alloys and Compounds*, vol. 356-357, pp. 603–607, 2003.
- [10] J.-L. Bobet, S. Desmoulins-Krawiec, E. Grigorova, F. Cansell, and B. Chevalier, "Addition of nanosized Cr₂O₃ to magnesium for improvement of the hydrogen sorption properties," *Journal of Alloys and Compounds*, vol. 351, no. 1-2, pp. 217–221, 2003.
- [11] S. R. Ovshinsky and M. A. Fetcenko, "Development of high catalytic activity disordered hydrogen-storage alloys for electrochemical application in nickel-metal hydride batterie," *Applied Physics A*, vol. 72, no. 2, pp. 239–244, 2001.
- [12] J. Huot, J. F. Pelletier, L. B. Lurio, M. Sutton, and R. Schulz, "Investigation of dehydrogenation mechanism of MgH₂-Nb nanocomposites," *Journal of Alloys and Compounds*, vol. 348, no. 1-2, pp. 319–324, 2003.
- [13] J. Charbonnier, P. de Rango, D. Fruchart et al., "Hydrogenation of transition element additives (Ti, V) during ball milling of magnesium hydride," *Journal of Alloys and Compounds*, vol. 383, no. 1-2, pp. 205–208, 2004.
- [14] J. J. Vajo, F. Mertens, C. C. Ahn, R. C. Bowman Jr., and B. Fultz, "Altering hydrogen storage properties by hydride destabilization through alloy formation: LiH and MgH₂ destabilized with Si," *Journal of Physical Chemistry B*, vol. 108, no. 37, pp. 13977– 13983, 2004.
- [15] N. Hanada, T. Ichikawa, and H. Fujii, "Catalytic effect of nanoparticle 3d-transition metals on hydrogen storage properties in magnesium hydride MgH₂ prepared by mechanical milling," *Journal of Physical Chemistry B*, vol. 109, no. 15, pp. 7188–7194, 2005.
- [16] M. Y. Song, D. R. Mumm, S. N. Kwon, S. H. Hong, and J. S. Bae, "Hydroazgen-storage properties of Mg-10 wt.% (Fe₂O₃, Ni, MnO) alloy prepared by reactive mechanical grinding," *Journal of Alloys and Compounds*, vol. 416, no. 1-2, pp. 239–244, 2006.
- [17] G. Barkhordarian, T. Klassen, and R. Bormann, "Catalytic mechanism of transition-metal compounds on Mg hydrogen sorption reaction," *Journal of Physical Chemistry B*, vol. 110, no. 22, pp. 11020–11024, 2006.
- [18] N. A. Niaz, I. Ahmad, K. S. Waheed, and S. Tajammul Hussain, "Synthesis of nanostructured Mg-Ni alloy and its hydrogen storage properties," *Journal of Materials Science and Technology*, vol. 28, no. 5, pp. 401–406, 2012.
- [19] M. Dornheim, S. Doppiu, G. Barkhordarian et al., "Hydrogen storage in magnesium-based hydrides and hydride composites," *Scripta Materialia*, vol. 56, no. 10, pp. 841–846, 2007.
- [20] S. Orimo and H. Fujii, "Materials science of Mg-Ni-based new hydrides," *Applied Physics A*, vol. 72, no. 2, pp. 167–186, 2001.
- [21] B. Sakintuna, F. Lamari-Darkrim, and M. Hirscher, "Metal hydride materials for solid hydrogen storage: a review," *International Journal of Hydrogen Energy*, vol. 32, no. 9, pp. 1121–1140, 2007.

[22] M. Polanski, T. Plocinski, I. Kunce, and J. Bystrzycki, "Dynamic synthesis of ternary Mg₂FeH₆," *International Journal of Hydro*gen Energy, vol. 35, no. 3, pp. 1257–1266, 2010.

- [23] S. S. Sai Raman, D. J. Davison, J. L. Bobet, and O. N. Srivastava, "Investigations on the synthesis, structural and microstructural characterizations of Mg-based K₂PtCl₆ type (Mg₂FeH₆) hydrogen storage material prepared by mechanical alloying," *Journal* of Alloys and Compounds, vol. 333, no. 1-2, pp. 282–290, 2002.
- [24] J.-J. Didishein, P. Zolliker, K. Yvon, P. Fisher, J. Shefer, and M. Gubelmann, "Dimagnesium iron(II) hydride, Mg₂FeH₆, containing octahedral FeH₆⁴⁻ anions," *Inorganic Chemistry*, vol. 23, no. 13, pp. 1953–1957, 1984.
- [25] J. Huot, H. Hayakawa, and E. Akiba, "Preparation of the hydrides Mg₂FeH₆ and Mg₂CoH₅ by mechanical alloying followed by sintering," *Journal of Alloys and Compounds*, vol. 248, no. 1-2, pp. 164–167, 1997.
- [26] F. C. Gennari, F. J. Castro, and J. J. Andrade Gamboa, "Synthesis of Mg₂FeH₆ by reactive mechanical alloying: formation and decomposition properties," *Journal of Alloys and Compounds*, vol. 339, no. 1-2, pp. 261–267, 2002.
- [27] Q. Li, J. Liu, K. C. Chou, G. W. Lin, and K. D. Xu, "Synthesis and dehydrogenation behavior of Mg-Fe-H system prepared under an external magnetic field," *Journal of Alloys and Compounds*, vol. 466, no. 1-2, pp. 146–152, 2008.
- [28] H. Shao, T. Liu, Y. Wang, H. Xu, and X. Li, "Preparation of Mg-based hydrogen storage materials from metal nanoparticles," *Journal of Alloys and Compounds*, vol. 465, no. 1-2, pp. 527–533, 2008.
- [29] X. Zhang, R. Yang1, J. Qu et al., "The synthesis and hydrogen storage properties of pure nanostructured Mg₂FeH₆," *Nanotech*nology, vol. 21, no. 9, Article ID 095706, 2010.
- [30] J. A. Puszkie, P. A. Larochette, and F. C. Gennari, "Thermodynamic-kinetic characterization of the synthesized Mg₂FeH₆-MgH₂ hydrides mixture," *International Journal of Hydrogen Energy*, vol. 33, no. 13, pp. 3555–3560, 2008.
- [31] J. A. Puszkiel, P. A. Larochette, and F. C. Gennari, "Thermodynamic and kinetic studies of Mg-Fe-H after mechanical milling followed by sintering," *Journal of Alloys and Compounds*, vol. 463, no. 1-2, pp. 134–142, 2008.
- [32] P. Selvam and K. Yvon, "Synthesis of Mg₂FeH₆, Mg₂CoH₅ and Mg₂NiH₄ by high-pressure sintering of the elements," *International Journal of Hydrogen Energy*, vol. 16, no. 9, pp. 615–617, 1991.
- [33] H. Wagner and H. Homer, "Eastic interaction and the phase transition in coherent metal-hydrogen systems," *Advances in Physics*, vol. 23, no. 4, pp. 587–637, 1974.
- [34] R. W. P. Wagemans, J. H. Van Lenthe, P. E. De Jongh, A. J. Van Dillen, and K. P. De Jong, "Hydrogen storage in magnesium clusters: quantum chemical study," *Journal of the American Chemical Society*, vol. 127, no. 47, pp. 16675–16680, 2005.
- [35] Y. Li, Y. Cheng, Y. Yinghua, and R. Shen, "Supplement on applicability of the Kissinger equation in thermal analysis," *Journal of Thermal Analysis and Calorimetry*, vol. 102, no. 2, pp. 605–608, 2010.
- [36] H. E. Kissingerz, "Reaction kinetics in differential thermal analysis," *Analytical Chemistry*, vol. 29, no. 11, pp. 1702–1706, 1957.

















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