

Research Article

Morphology Evolution of Mn-Si Composition Gradient Micro/Nanomaterials Prepared by Oxygen Assisted Chemical Vapor Deposition

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The micro/nanostructure of manganese silicide (Mn-Si) compounds with various morphologies (nanowires, films, particles, and polyhedron shape structure) has been synthesized through oxygen assisted chemical vapor deposition by changing the stacking geometry of manganese powder. Polyhedrons prepared in the Mn-Si contact area were identified to be chemical composition gradient functionally graded materials which were verified by analyzing atomic ratio of Mn/Si from top to bottom. Evolution of morphology greatly depended on the stacking shape correlated distance from precursor to the substrate, resulting in distinctive growth mechanisms. Main structures on the substrate have been verified to be Mn_5Si_3 and Mn_4Si_7 with different Mn stacking in bumps comparing to sole Mn_4Si_7 with flat surface.

1. Introduction

Recently, manganese silicide nanostructures which have unusual magnetic properties are considered to be potential candidates in spintronic applications [1, 2]. Among them, the most heavily studied phase is Mn-Si which undergoes transitions from helimagnetic to magnetic skyrmion phase near 30 K [3–5]. Skyrmion shows a great potential to be used as a computer device with a much faster computing speed, larger storage capability, and much less energy consumed [6, 7]. The Mn₄Si₇ as a weak itinerant system which orders below 40 K nanowires has several application as optoelectronic material and thermoelectric material [8-11]. With a hexagonal structure, Mn₅Si₃ is another interesting material which has the potential for a high magnetocrystalline anisotropy with novel spin-electronic properties, while the nanocrystal behaved like a ferromagnet due to size effect [12-14]. A diluted magnetic semiconductor Mn_xSi_{1-x} (x = 0.03–0.05) showed a ferromagnetic transition point at 250 K with a method to manipulate the charge and spin [15, 16]. Mn₃Si was studied to find out the unique electronic and magnetic properties [17, 18]. Mn_5Si_2 which has a ferromagnetic ordering with a T_C of 300 K [19] does have a potential in the microelectronics.

2. Methods

The chemical vapor deposition (CVD) system was set up based on a single zone Lindberg tube furnace with maximum temperature of 1200°C. High purity Argon gas (99.999%) was used as the carrier gas. The system allows precise control of growth temperature, gas flow rate, gas pressure, and distance from substrate to precursor. In this home designed CVD system, temperature was set to 850°C when flow rate of Ar gas was varied from 140 to 300 sccm. The pressure was always kept at 760 Torr and oxygen was brought into the system for not fully sealing the tube intentionally. A clean Si (001) wafer was placed above the Mn powder as substrate inside a ceramic crucible located in the center of the tube. The Mn powder (99.9%) was purchased from CHANGSHA TIANJIU Company. The Mn powder was laying in different shape, such as flat, cone, and multiple-cones. The silicon wafer (001) was purchased from HEFEI KEJING Materials technology CO., LTD. Silicon (100) chips were cut into $1 \text{ cm} \times 1 \text{ cm}$ in size, and they are put in deionized water sonicating for 10 min. Then they are washed by ethanol sonicating for 10 mins and dried with N₂ gas.

Scanning electron microscope (FEI Nano230) (SEM) equipped with an energy dispersive X-ray spectrometer



FIGURE 1: Various laying shapes of Mn powder: (a) flat, (b) cone, and (c) multiple-cones. (d) Calculation of distances from source to substrate for various products in cone laying shape of Mn powder.

(EDX) was used in this study to verify the morphology and composition of the nanostructures. Sample was attached by conductive tape on the sample holder. The voltage was adjusted from 5 kV to 20 kV in order to fulfill higher resolution and X- ray spectrum. X-ray diffraction (RIGAKU 2550, Japan) (XRD) was employed to confirm the primary phase of the samples with 2Theta scanning from 5 to 90°. Magnetic properties of the synthesized material were characterized with a magnetic property measurement system (Quantum Design MPMS3) (MPMS) capable of applying magnetic field as large as 7 T. Cooling by liquid helium, temperature dependent magnetization was measured from 5 to 300 K with magnetic field set to be 1000 Oe.

3. Results and Discussion

To investigate the effect of the constant distance between manganese (Mn) powder and silicon substrate on the morphology of nanostructures, Mn powder was stacked flat and smooth underneath the Mn powder (Figure 1(a)). Then two supporting silicon wafers were placed directly on the Mn powder. One silicon substrate was placed on the supporting wafer leaving a gap between substrate and powder. The gap between the substrate and Mn powder is $500 \,\mu$ m which is the thickness of one piece of supporting wafer. Direct contact of Mn powder and silicon substrate was avoided. After 80 min deposition, the substrate was covered with a thick layer of film whose Mn/Si composition ratio was verified to be approximately 4/7 by EDX. Few Mn atoms were evaporated and captured by silicon substrate due to the longer mean free path. Forming a growth site on the surface of substrate, Mn-Si atomic cluster began to absorb nearby silicon atoms leading to growth of Si rich phase.

Constant distance gave rise to an unchanged morphology; it was quite easier to distinguish the deposits by laying manganese powder to a cone-like shape (Figure 1(b)) in order to find the effect of various distance on the morphological evolution of Mn-Si compounds. After deposition, four different nano/microstructures including dispersed particles, rods, nanowires, and polyhedron were found on the substrate. Each of them was growing in a ring like growing zones (Figure 2(a)) distributing from edge to center of substrate. In the center of substrate, many polyhedron structures with 10 μ m in width spread evenly (Figure 2(b)). Those polyhedron were a well-shaped object composed of several folding layers. The width of each layer was about hundreds of nanometers.



FIGURE 2: Micro/nanostructure prepared by cone laying shape of Mn powder: (a) growing bands from center to edge of substrate; (b) polyhedron structures; (c)-(d) quasi-polyhedron structures with nanowires growing in edges; (e) nanowires; (f) nanorods; (g)–(i) Mn and Si element mapping of substrate. (j) X-ray diffraction pattern of nanostructure prepared by various stacking shape of Mn powder.

In multiple-cone shape of Mn powder (Figure 1(c)), similar situation happened.

The region of growth is specialized. Considering the width of band, with the angle of bump 15.57°, the range of distance from substrate to powder is divided into four

regions: (1) polyhedrons structures grow at area where Mn powder contacts with Si substrate; (2) nanowires grow at distance from 0 μ m to 189 μ m; (3) nanorods grow at a narrow range of distance from 189 μ m to 216 μ m; (4) particles grow at a distance from 216 μ m to 500 μ m (Figure 1(d)). The

distance might not be that precise because in the boundary between those regions, usually two forms of nanostructure were found.

By examining the outer growing zone near polyhedron, randomly oriented nanowires were growing. Surprisingly additional quasi-polyhedron structures were prepared in the location between nanowires and polyhedron (Figure 2(c)). Those surfaces of quasi-polyhedrons were oxidized leading to several holes and inflated bumps. Quasi-polyhedrons with both features of polyhedron and nanowires revealed a clear sign of distance dependent morphological evolution. The transitional morphology was growing due to the various growth mechanisms. In this process, temperature and oxygen both played an important role. The higher concentration of Mn atoms was an essential condition for growth of polyhedron. The direct contact of Mn powder and silicon substrate led to the growth of polyhedrons in the area where oxygen content is relatively low. In the places very near to the center of the substrate, the distance from Mn powder to silicon substrate was so small; polyhedron firstly grew with relatively low temperature due to surrounding high concentration of Mn atoms. As temperature increased, polyhedron was oxidized which resulted in irregular shape forming in the surface. Then diffusion of silicon atoms was enhanced through the oxide inducing the growth of Mn-Si nanowires.

All of them were well-shaped layered structures with very thin nanowires growing in the edges (Figure 2(d)). The nanowire was straight with tens of micrometers in length and 100–200 nm in width.

Nanowires found growing separately near the quasipolyhedron structures were 500-800 nm in width and $40-80 \,\mu$ m in length covered with a thick layer of silicon dioxide (Figure 2(e)).

Next to nanowires band there were many nanorods distributed around (Figure 2(f)). Those nanorods were short and thinner with length of $10 \,\mu\text{m}$ and width of $100 \,\text{nm}$. The distribution of nanorods was not as dense as that of nanowires. The outermost band was well-shaped particles.

In the whole surface of substrate, it was found that composition varies (Figure 2(g)). Element analysis was conducted by comparing only manganese Figure 2(h) and silicon mapping Figure 2(i). Manganese was distributed more evenly both in the polyhedron and nanowires, while silicon was more concentrated in the surface of polyhedron.

X-ray diffraction patterns were shown in Figure 2(j). Peaks largely depended on the laying shapes of manganese powder. With the flat laying shape of Mn powder, all the peaks matched Mn_4Si_7 solely. However, for samples growing in other two manganese powder stacking shapes, peaks of Mn_5Si_3 began to appear. The Mn_5Si_3 signals came from the polyhedron structures. Peaks became intensively distributed for more polyhedron structure resulting from larger contact areas in multiple-cones laying shape of Mn powder.

To further examine the polyhedron in the center area, atomic composition measured at five points chosen from top layer to bottom layer of the polyhedron structure was shown in Figure 3(a). Manganese mapping and silicon mapping were compared separately in Figures 3(b) and 3(c). By plotting the

element concentration in Figure 3(d), composition of Mn : Si was found at the top of polyhedron structure close to 5 : 3; then Mn : Si atomic ratio increased to nearly 4 : 1 at the bottom of polyhedron (Figure 3(e)). Besides tiny peaks of carbon and oxygen, the EDX spectrums measured on points 1, 3, and 5 were compared in Figures 3(f), 3(g), and 3(h). It showed a gradual decrease in the height of Si peak. That means the composition of silicon dropped. Element ratio was similar to Mn-Si chemical formula illustrating that polyhedron was multiple-phases composition gradient material.

Element mapping revealed that morphological evolution depends on whether oxygen was involved in the growth process. In the center where manganese powder directly contacts with silicon substrate, the main mechanism is a solid-solid reaction. But a layer of manganese has been found underneath the polyhedrons indicating that Mn particles touched the surface of silicon substrate forming a thin layer of high manganese compounds in absence of oxygen at low temperature; then silicon began to react with manganese to form polyhedron structure (Figure 3(e)) because EDS data showed no oxygen in the polyhedron structures.

In the border of contact area, high manganese polyhedrons began to grow at relatively low temperature first. Then as temperature increases, assisted by higher concentration of oxygen in the gap between silicon and manganese, Mn-Si nanowires on the top of polyhedron structure began to grow. The growth in this area depended on oxygen assisted vaporsolid growth. The supersaturation ratio depending on oxygen concentration plays an important role in the composition and structure of nanowires [5].

Then in the area where Mn particle did not contact with silicon substrate, oxygen first reacted with silicon substrate at lower temperature; then when the temperature increased, Mn vapor directly reacted with SiO_2 forming different micro/nanostructures. In this study, nanowires showed a composition ratio of Mn : Si which equals 1:1, 5:3, and 4:7 due to uneven thickness of oxide forming on silicon substrate.

By applying magnetic field (~1000 oe) parallel to the substrate surface, it is clear that various Mn powder stacking shape resulted in different magnetic signals (Figure 4). It is obvious that black curve was smooth and has only one transition near 40 K which corresponds to X-ray pattern showing mostly Mn₄Si₇ on sample. When Mn powder was stacking in a shape with one bump, the manganese element will be deposited on the substrate in shorter time with less oxygen involved. So Mn-Si, Mn4Si7, and Mn5Si3 nanowires contributed to all signals. The relative amount of Mn₄Si₇ was reduced leading to a weaker magnetic signal of 40 K transition point (Figure 4 inset (a)). In Figure 4 inset (b), two new transitions were observed near 75 K and 220 K. Sample prepared with Mn powder stacking in several cones shows stronger ferromagnetic signal near 220 K than that prepared with Mn powder stacking into one cone, while on the contrary, sample prepared with Mn powder stacking in several cones shows weaker antiferromagnetic signal near the transitions at approximately 75 K than that prepared with Mn powder piling up with one cone. The antiferromagnetic transition point at 75 K relies on the Mn₅Si₃ nanowires and bulk Mn₅Si₃ which came from the area where Mn powder



FIGURE 3: (a)–(c) Mn and Si element mapping of composition graded polyhedron materials prepared by direct reaction of Mn particles with silicon wafer; (d) change of Mn and Si concentration via location points from top to down; (e) typical polyhedron structures; (f)–(h) measured spectrum of elements Mn and Si at points 1, 3, and 5.

contacts with silicon substrate. Another ferromagnetic transition point at 220–250 K relies on the diluted magnetic semiconductor $Mn_x Si_{1-x}$ (x = 0.03-0.05). This diluted magnetic semiconductor form basing on Mn atom was incorporated into the silicon lattice. Liu et al. prepared this material by sputtering single crystalline Si partially covered with Mn chips on silicon substrate [15, 16]. So it is estimated that Si contacting with Mn chips directly might be the reason for preparing this diluted magnetic semiconductor successfully. Thus in sample prepared with stacking Mn powder in several bumps, more Mn-Si contact points brought about more Mn_xSi_{1-x} (x = 0.03-0.05) leading to stronger ferromagnetic signal at 220 K.

4. Conclusion

Direct contact of Mn powder and silicon substrate gave rise to the solid-solid reaction of Mn and Si, leading to the growth of polyhedrons (a special type of chemical composition gradient functional graded materials). Nanowires, nanorods,



FIGURE 4: Magnetic properties showing various transition temperatures correlated to different nano/microphases. Inset figures magnetic transitions in the temperature ranges (a) 20–50 K; (b) 50–300 K.

and particles were prepared by oxygen assisted vapor-solid growth on the area where Mn particle did not contact with Si wafer. The spatial morphological evolution influenced by distance from precursor to substrate clearly revealed trend of varying the supersaturation ratio. Finally, magnetic signal not only shows transitions of Mn-Si, Mn₄Si₇, and Mn₅Si₃, but also gives a ferromagnetic signal of diluted magnetic semiconductor Mn_xSi_{1-x} (x = 0.03-0.05).

Conflicts of Interest

The author declares that there are no conflicts of interest regarding the publication of this paper.

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